



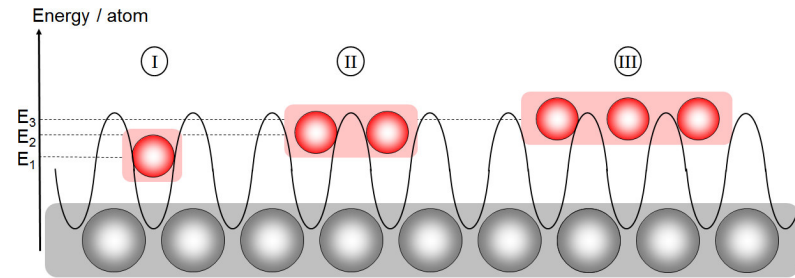
Limitations of Structural Superlubricity: Chemical Bonds versus Contact Size

André Schirmeisen

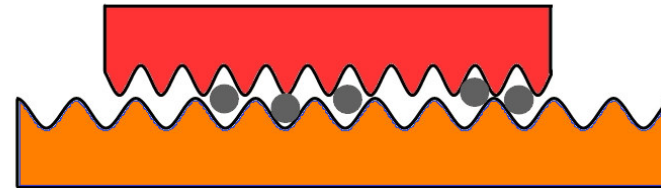
Institute of Applied Physics, University of Giessen, Germany

Structural Superlubricity of Nanoparticles

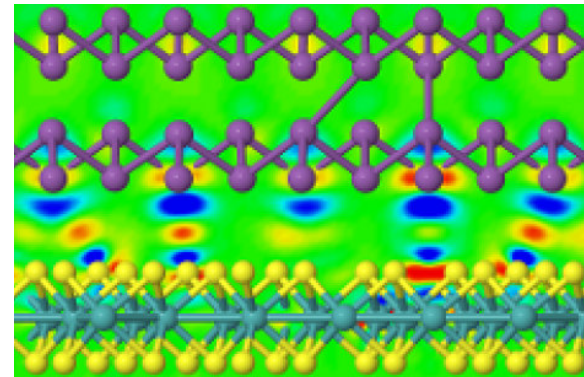
Structural Superlubricity Basics: Concepts and Scaling



Limits of Structural Superlubricity Case a) Mobile interface molecules

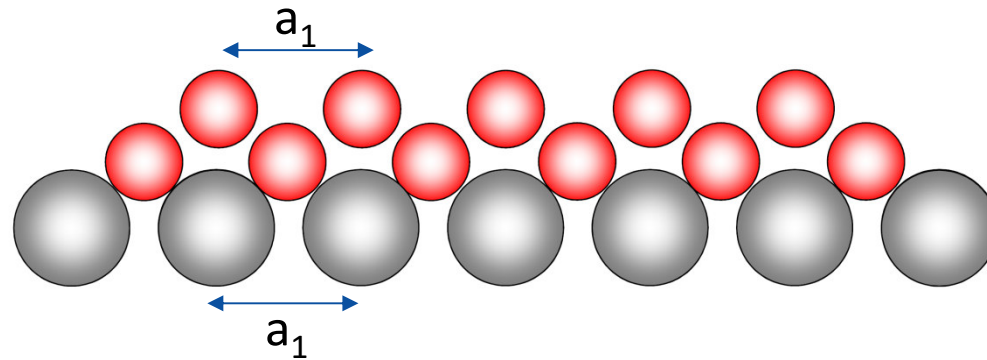


Limits of Structural Superlubricity Case b) Size dependent break-down



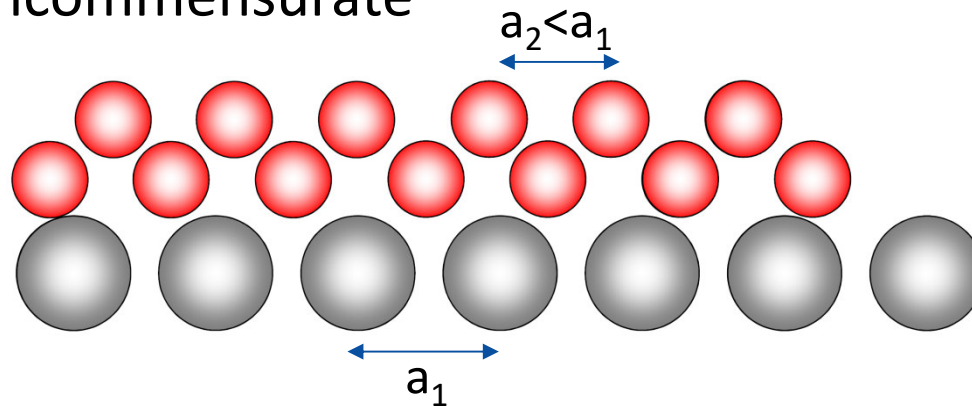
Structural Superlubricity: Commensurate vs Incommensurate

Commensurate



Interlocking of
atomic potentials:
Friction $\propto A_{\text{contact}}$

Incommensurate



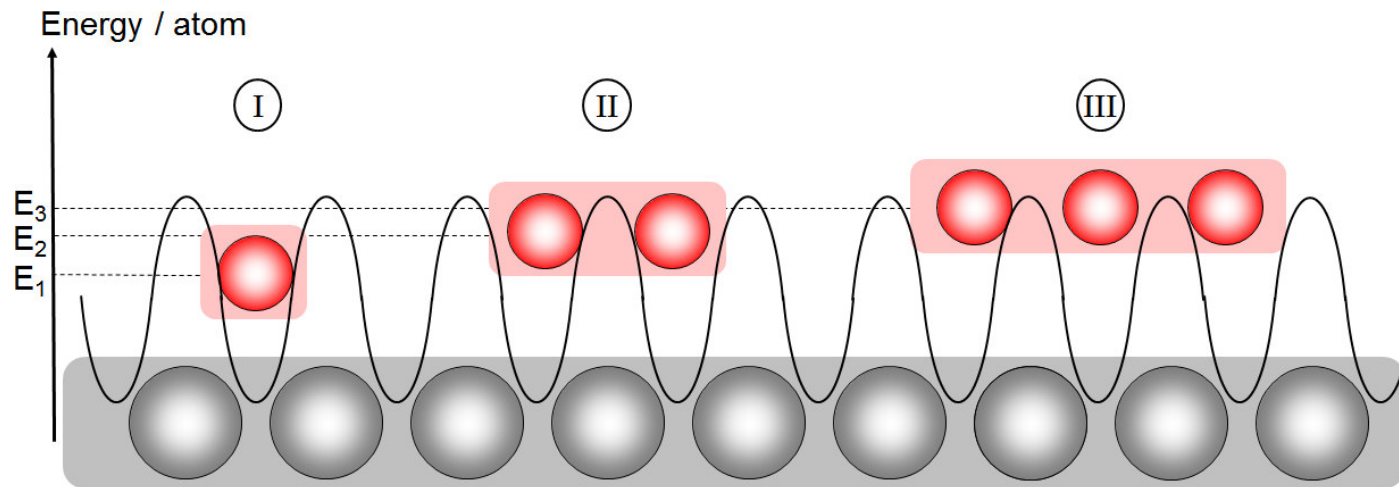
no interlocking of atomic
potentials:
structural lubricity
(‘superlubricity’)

Hirano & Shinjo, PRB 41, 11 837 (1990)
Müser, Wenning, Robbins, PRL 86, 1295 (2001)
Dienwiebel et al., PRL 92, 126101 (2004)

Incommensurate Case: Friction Area Scaling

If particle and substrate have non matching lattice constants ($a \neq b$)

Number of top atoms increases $\rightarrow$ Energy barrier per atom decreases



Theoretical
(calculations / simulations):
Energy barrier vs. particle size
follows sublinear power law



$$\Delta E \propto N_{atom}^{\gamma} \quad (\gamma < 1)$$

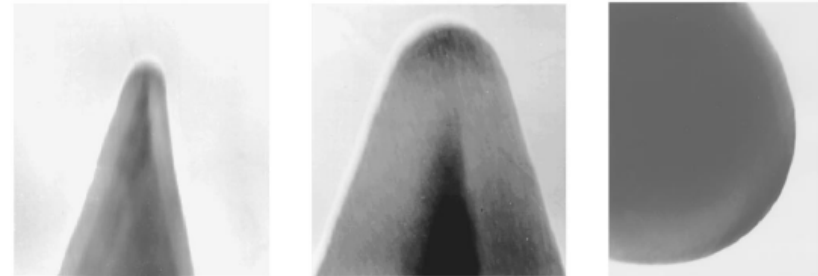
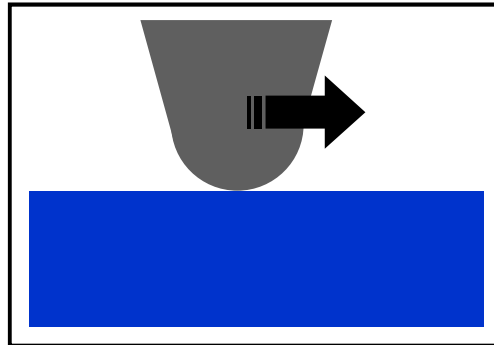
deWijn, Phys. Rev. B. 86, 085429 (2012)

Müser, Wenning, Robbins, Phys. Rev. Lett. 86, 1295 (2001)

M. Müser in "Fundamentals of Friction and Wear at the Nanoscale", Eds: E. Gnecco, E. Meier, Springer (2007)

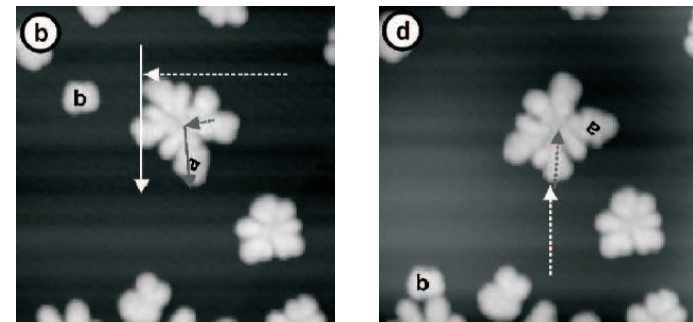
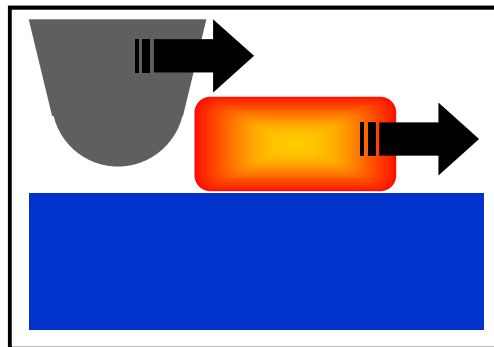
From Friction Force Microscopy to Particle Manipulation

Vary tip apex
by TEM
preparation



Schwarz et al, Phys Rev B 56 (1997) 6987

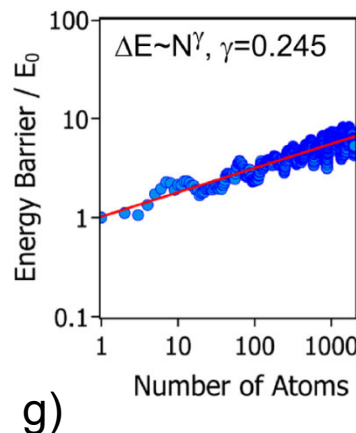
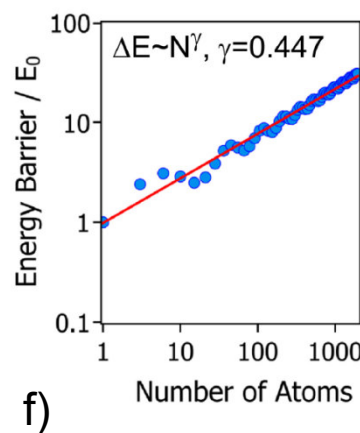
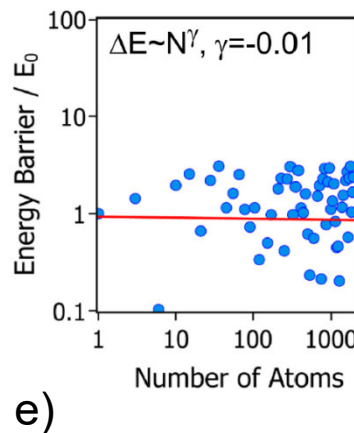
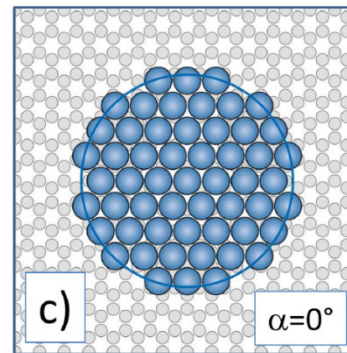
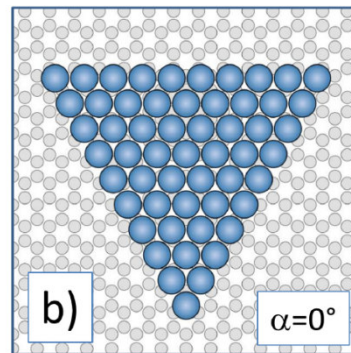
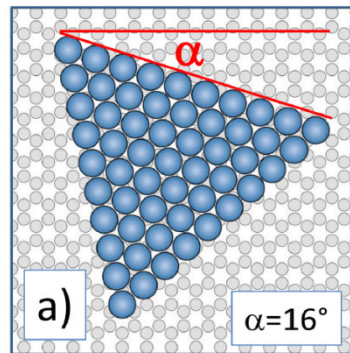
Use tip to push
particles with
varying size



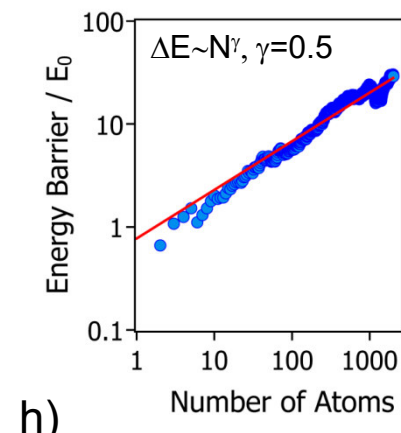
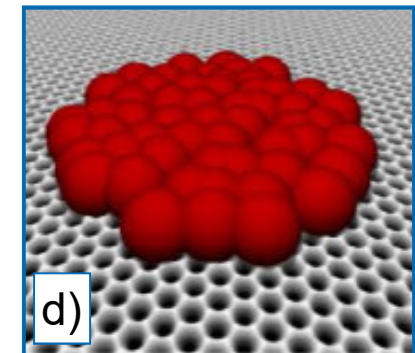
Ritter et al., Phys. Rev. B **71**, 085405 (2005)

Theory: Calculation of Energy Barriers for Nanoparticles

Crystalline Au on HOPG



Amorphous Sb on HOPG

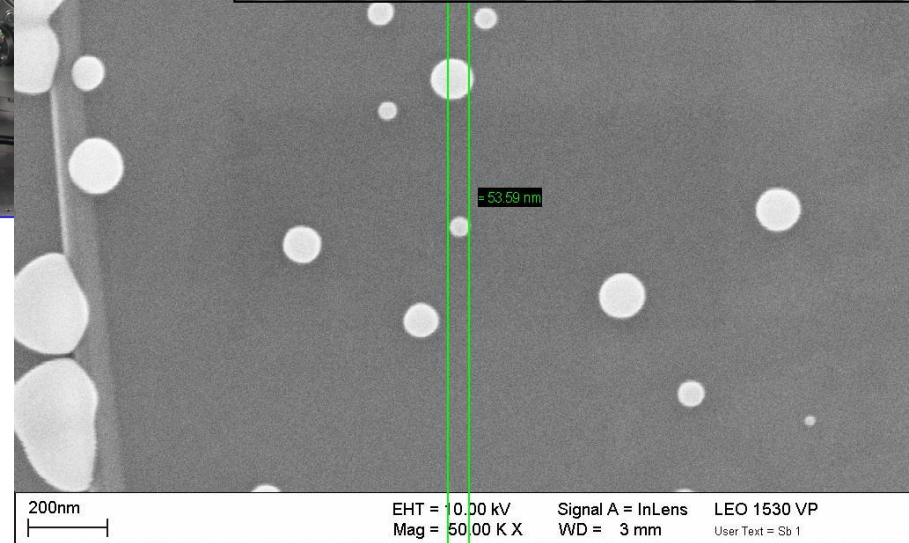
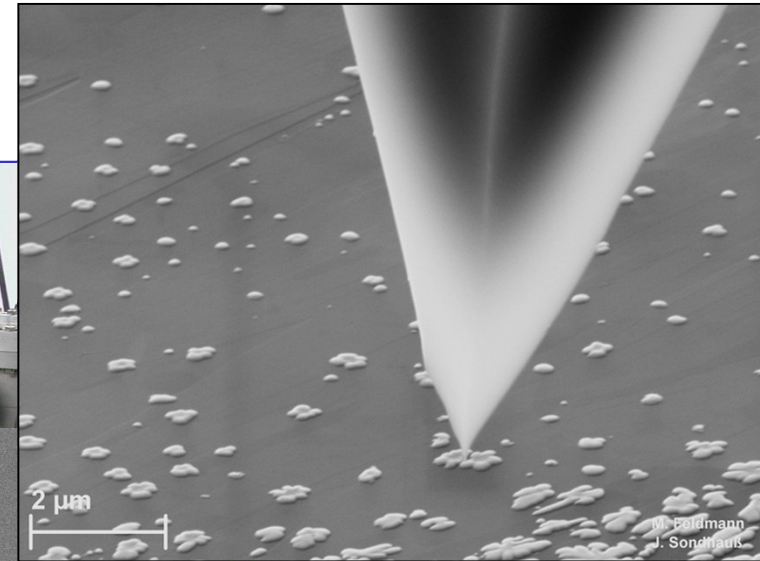
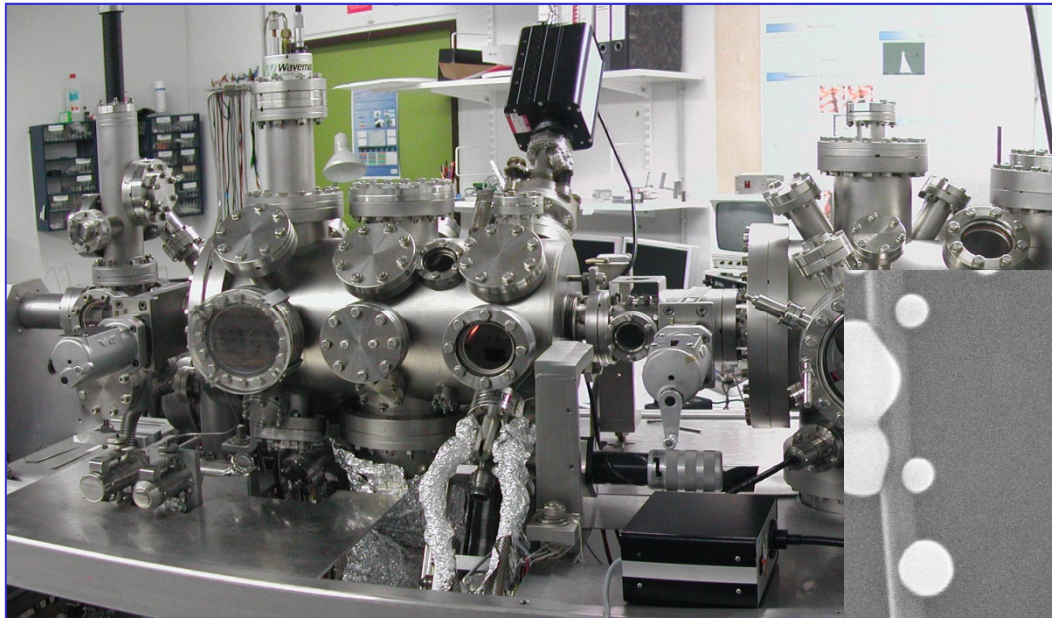


Gold Nanoparticles: Scaling depends on shape and orientation ($0 < \gamma < 0.5$)

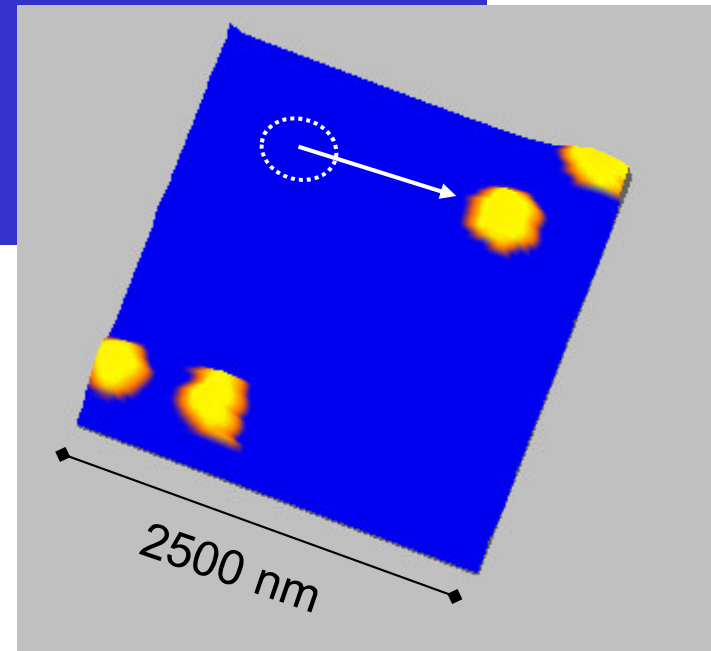
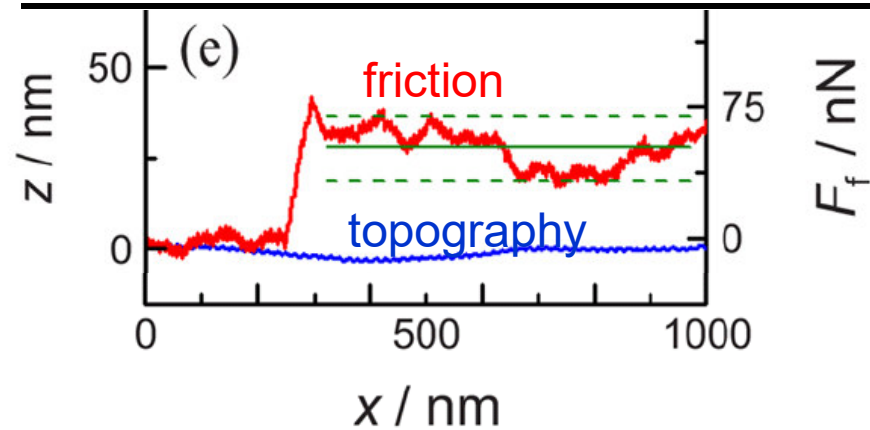
Antimony Nanoparticles: Scaling independent of shape and orientation ($\gamma = 0.5$)

Metallic Antimony nanoscale islands by sublimation

Omicron room temperature
AFM in UHV

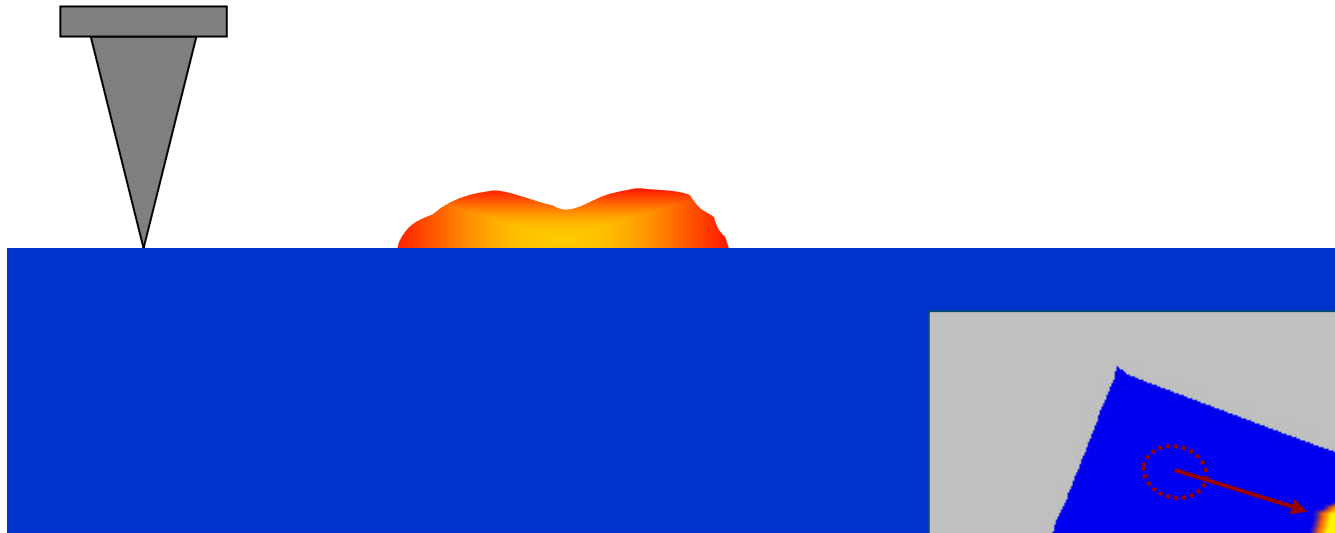


Moving Nanoparticles by Atomic Force Microscopy

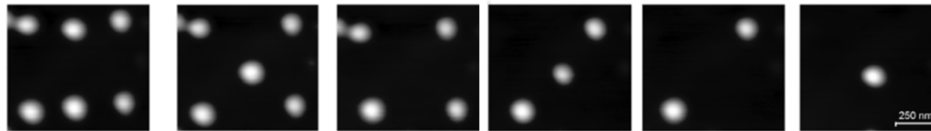


Dietzel, Mönninghoff, Jansen, Fuchs, Ritter,
Schwarz, Schirmeisen, JAP 102 (2007) 084306

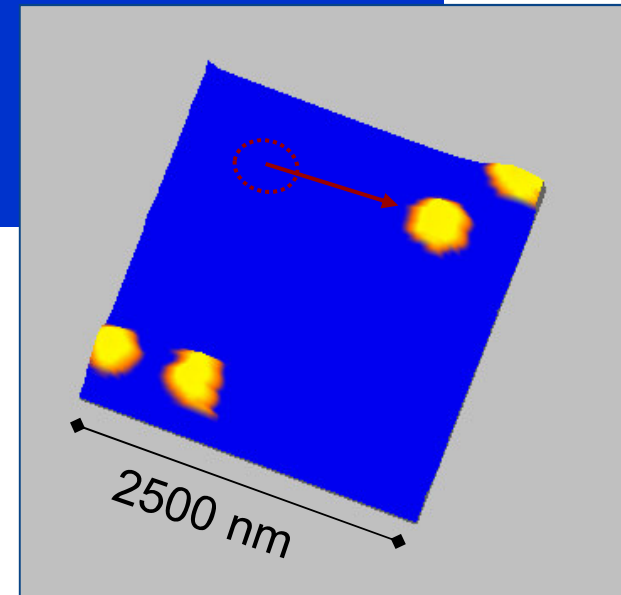
Moving Nanoparticles by Atomic Force Microscopy



Full Control over particle positioning!
(Precision $\approx 30\text{nm}$)

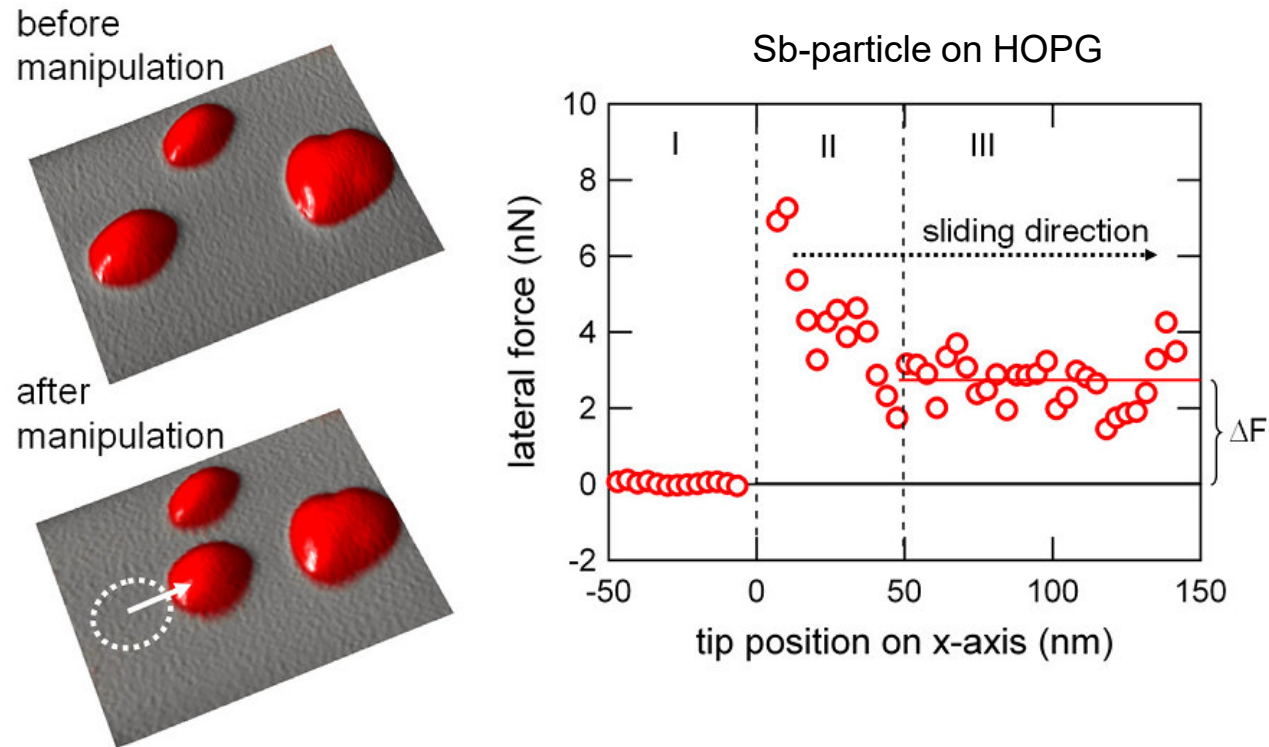


D. Dietzel et. al, Tribology Letters 39, 273-281 (2010)



D. Dietzel et al., J. Appl. Phys. 102, 084306 (2007)

Measuring Friction by Manipulation of Nanoparticles

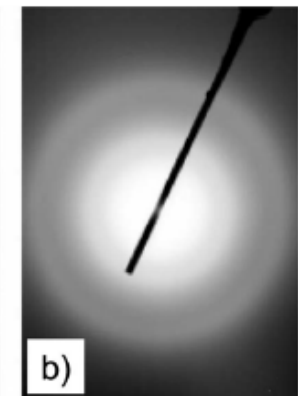
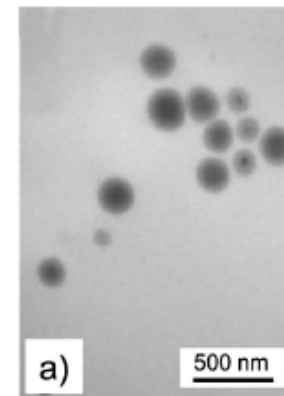
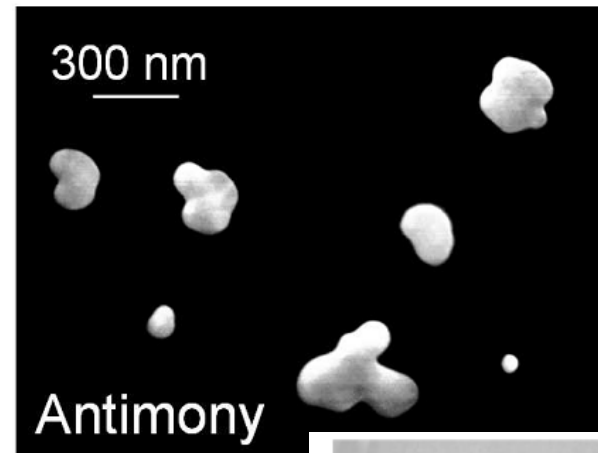
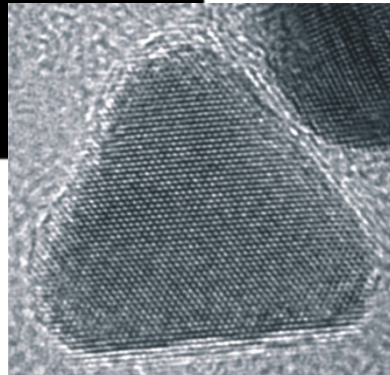
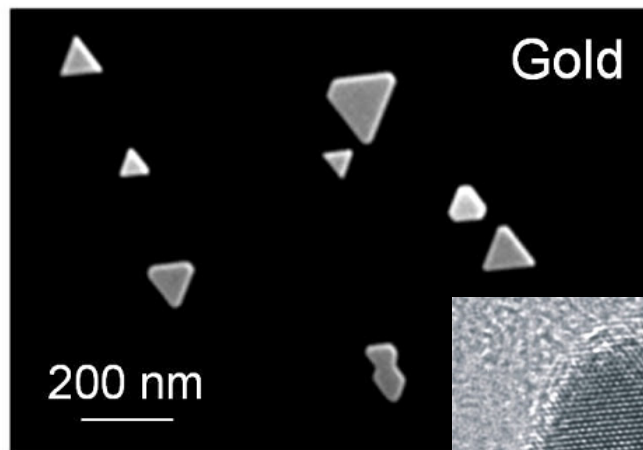


Manipulation by tip pushing from side:

- ⇒ Quantitativ friction can be extracted from manipulation
- ⇒ Fast manipulation of statistically significant number of particles of different size

D. Dietzel et al., J. Appl. Phys. 102, 084306 (2007)
D. Dietzel et al., Phys. Rev. Lett. 101, 125505 (2008)

Metallic Nanoscale Islands by Sublimation

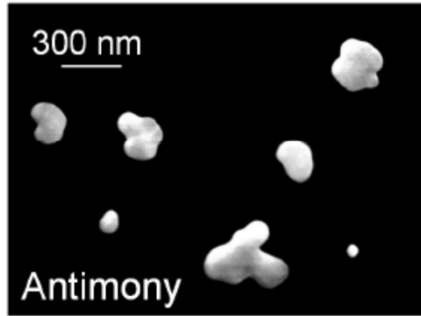


Preparation of metallic nanoparticles by thermal evaporation of Sb and Au under **UHV (ultrahigh vacuum)** conditions

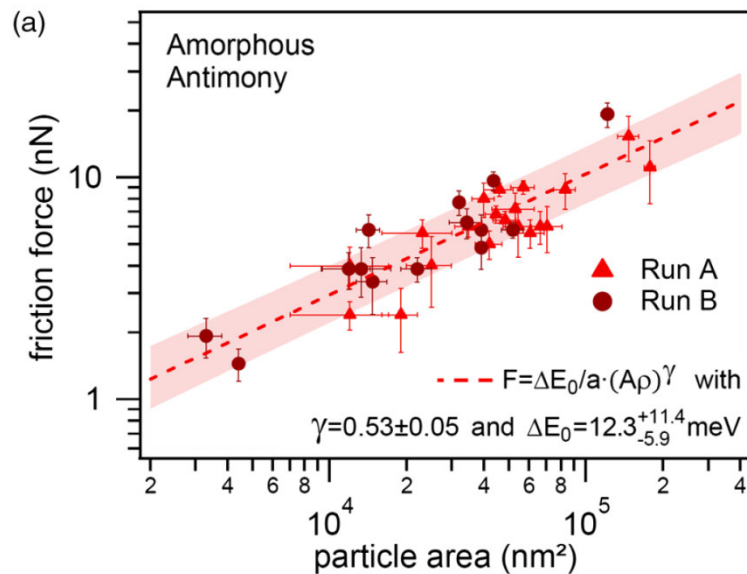
- Substrate: HOPG
- Antimony nanoparticles ⇒ amorphous
- Gold Nanoparticles ⇒ crystalline

Contact Area Dependence of Nanoparticle Friction

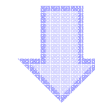
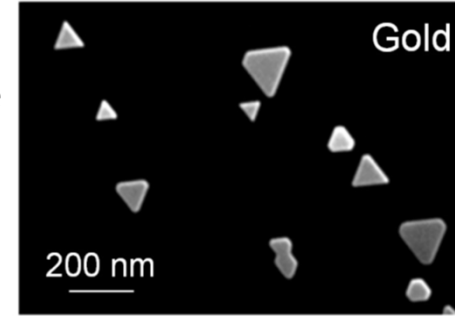
amorphous
antimony



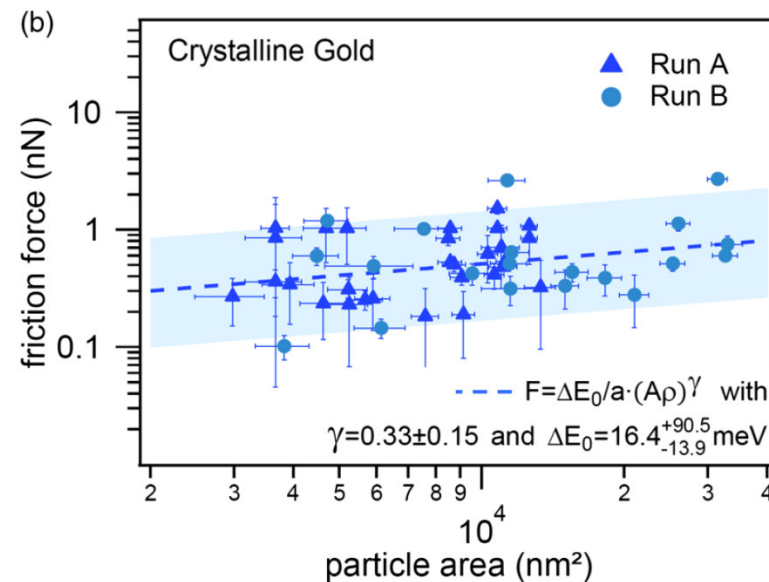
$$F \sim A^{0.53}$$



crystalline
Gold



$$F \sim A^{0.33}$$



Sublinear friction vs. area scaling
is experimentally confirmed!

Dietzel et al., Phys. Rev. Lett. 111, 235502 (2013)

Atomic Diffusion Energy: Upscaling to Nanoparticle Friction

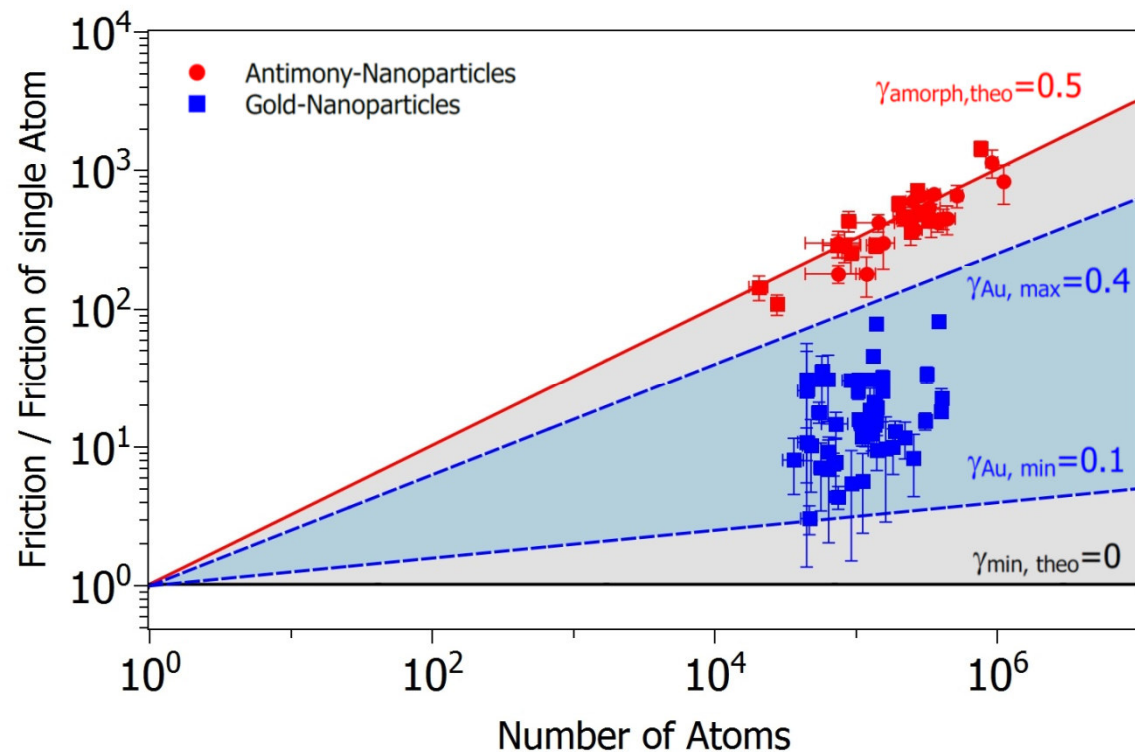
Simple model:
Diffusion energy barrier
yields single atom friction:

$$F_{\text{atom}} = \frac{\Delta E}{a}$$

$$F_{\text{particle}} = \frac{\Delta E}{a} \cdot N^\gamma$$

normalize
↓

$$F_{\text{particle}} / F_{\text{atom}} = N^\gamma$$

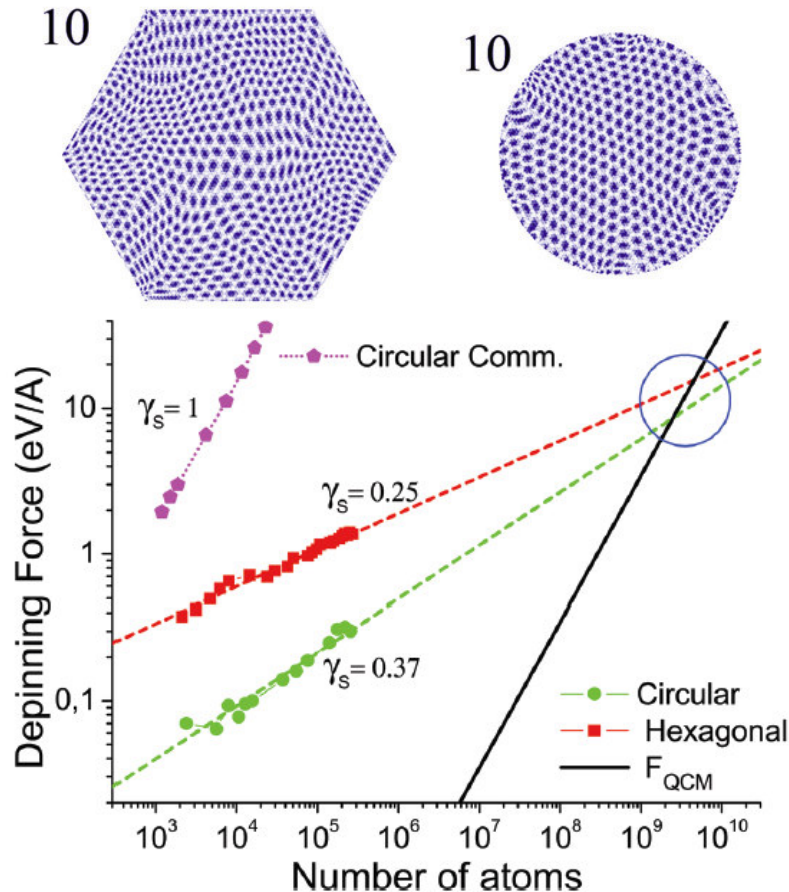


- Amorphous Sb particles: good agreement with theory ($\gamma=0.5$)
- Crystalline Au particles: No Uniform scaling due to variety of particle configurations
- Data can best be described by range of scaling ($0.1 < \gamma < 0.4$)
- No “perfect” triangles => specific expectations for triangle $\gamma=0$ or $\gamma=0.5$ not met

Sub-linear Scaling: A Fingerprint of Superlubricity

Physisorbed islands - Simulation

Varini et al., Nanoscale, 7 (2015) 2093

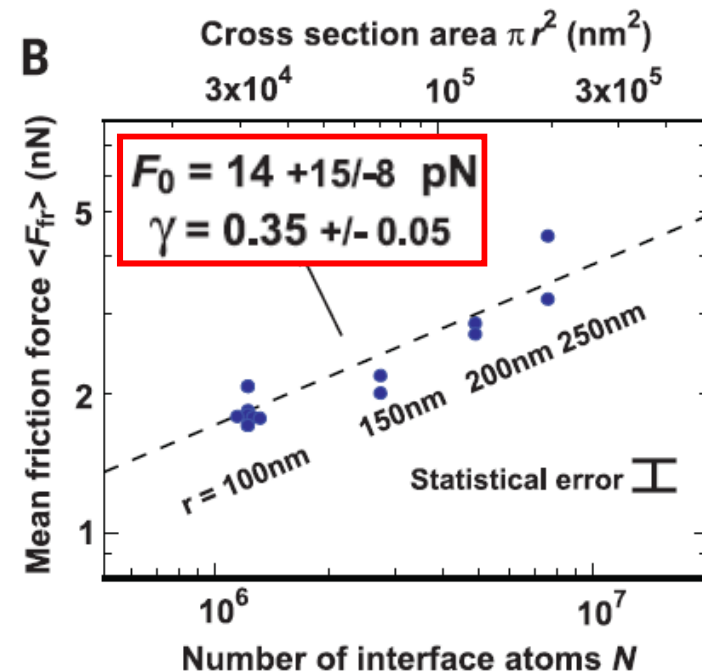
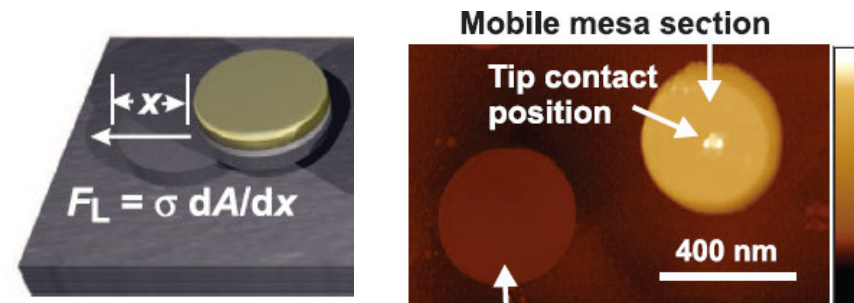


Again, incommensurate islands scale with

$$F \propto N_{atoms}^{\gamma} \quad (\gamma < 0.5)$$

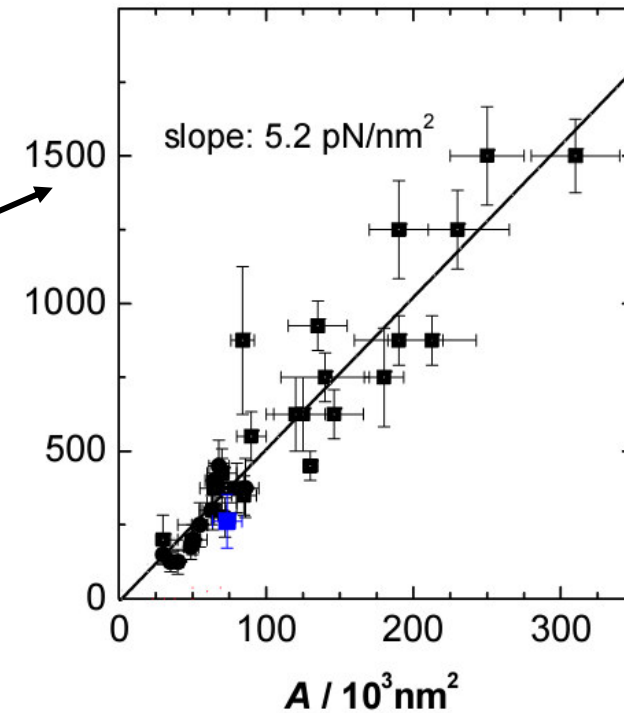
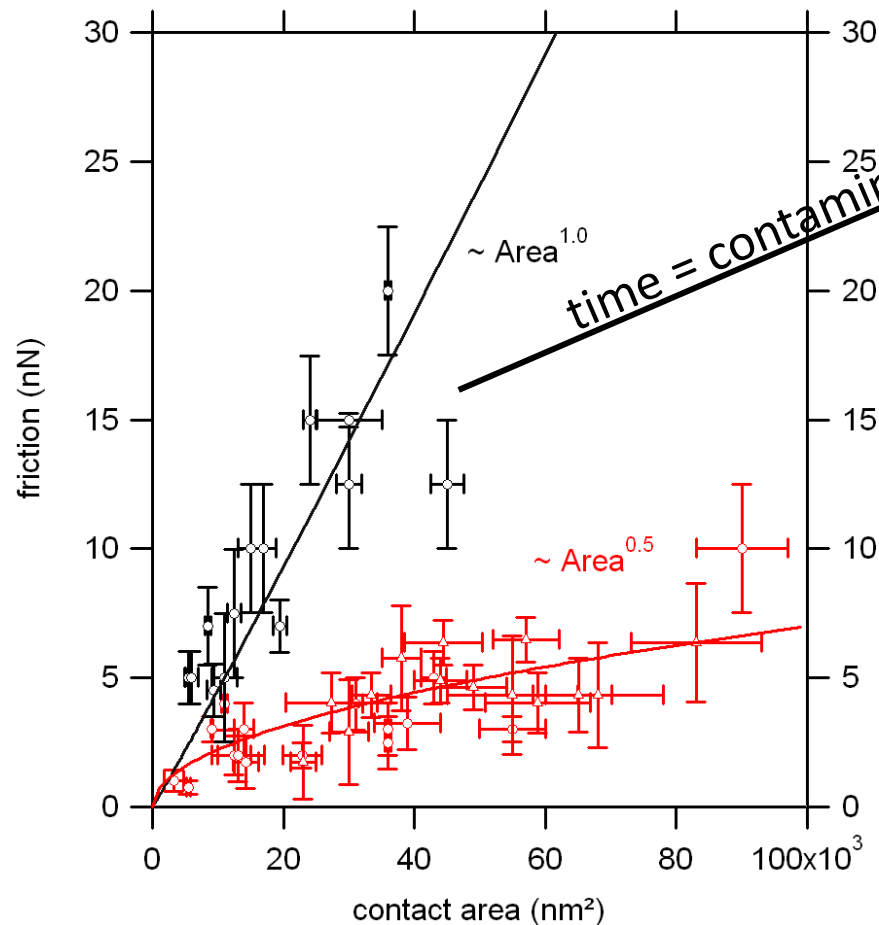
Graphite Interfaces - Experiments

Koren et al., Science, 348 (2015) 679



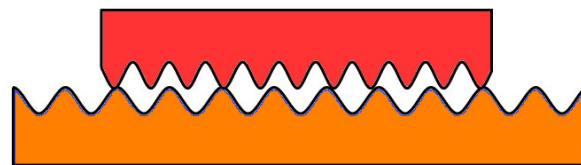
The Break-Down of Superlubricity: a) Interface Contamination

Antimony: Friction versus Area

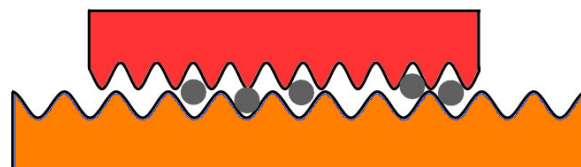


After weeks of storage in
vacuum chamber
 $F = \tau \cdot \text{area} !$
(almost exclusively)

MD Simulation: Friction and Contamination



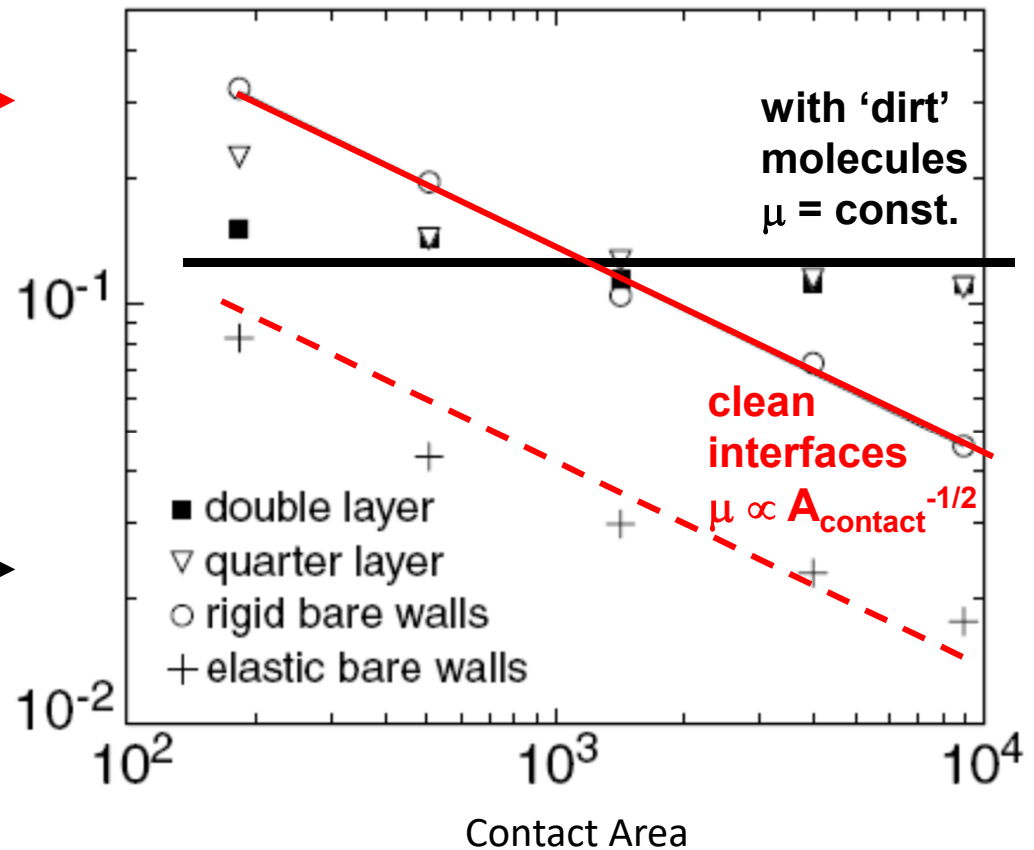
clean interface



contaminated interface

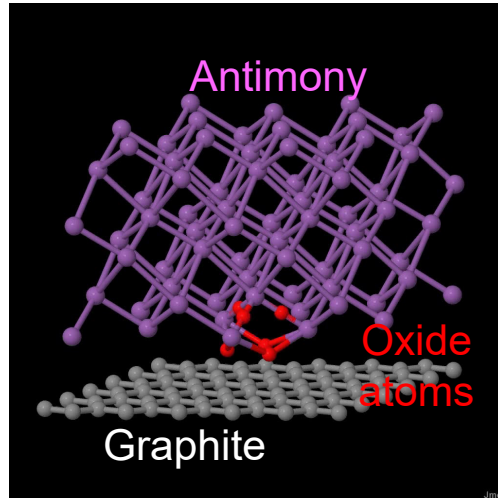


Friction coefficient versus contact area

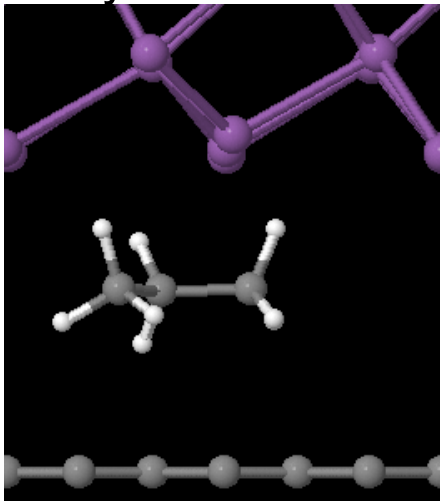


Typical Interface Contaminations – Ab-Initio Simulations

Interface oxidation

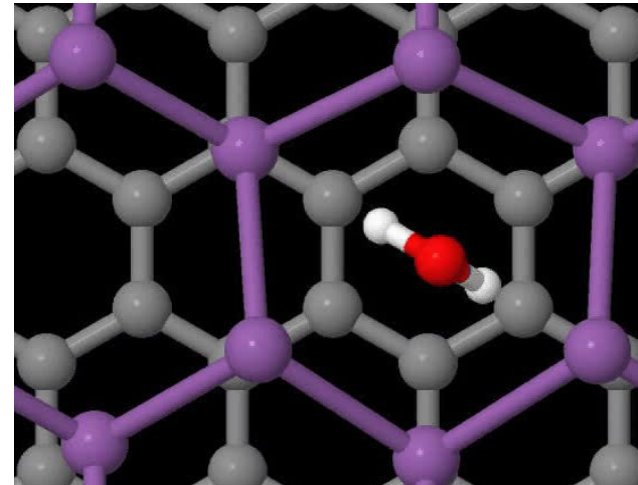


Hydrocarbons

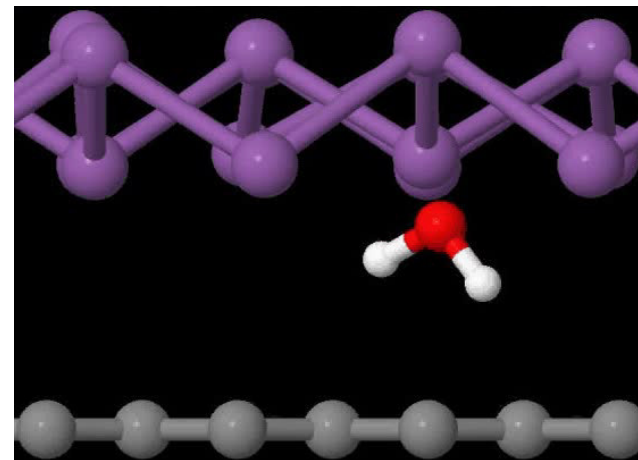


Water molecules

Top view

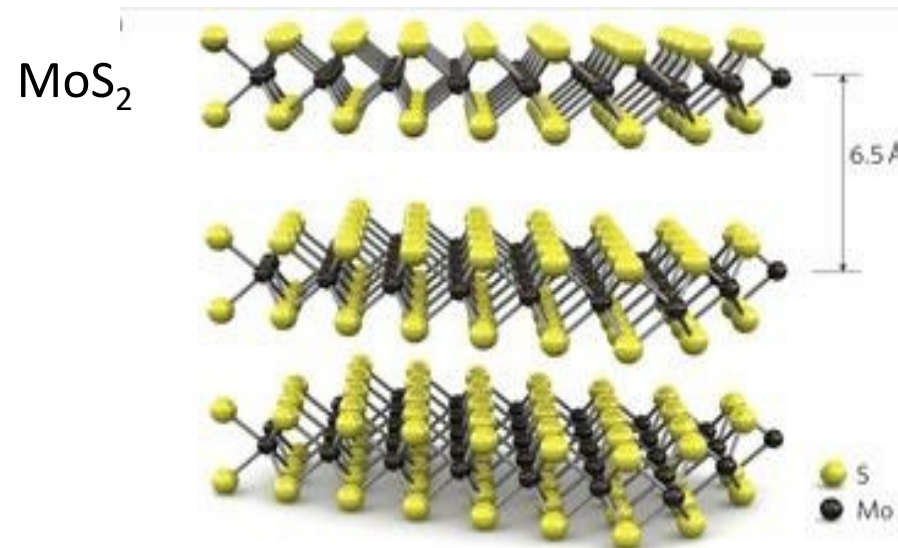
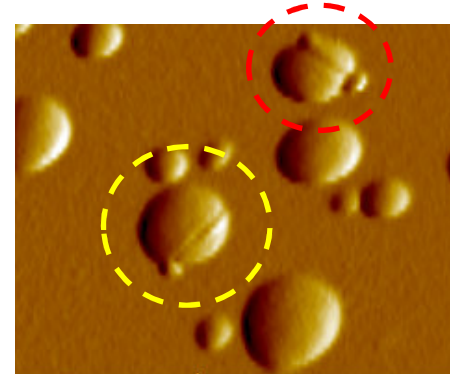
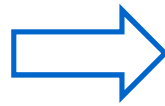
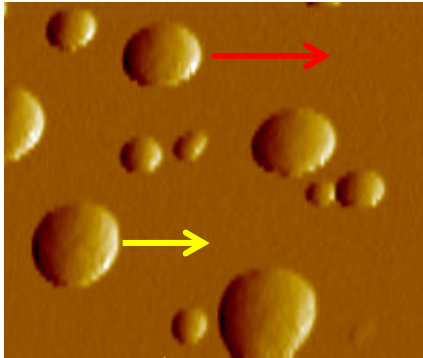


Side view



The Break-Down of Superlubricity: b) Relaxations/Dislocations

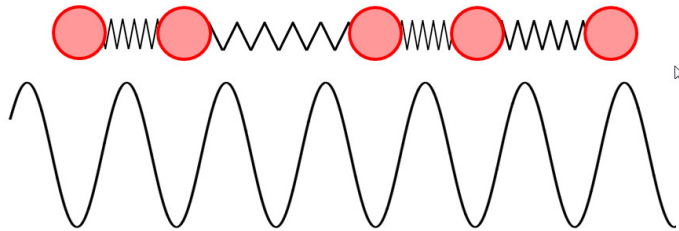
Nanomanipulation Experiments
Sb-particles on MoS₂



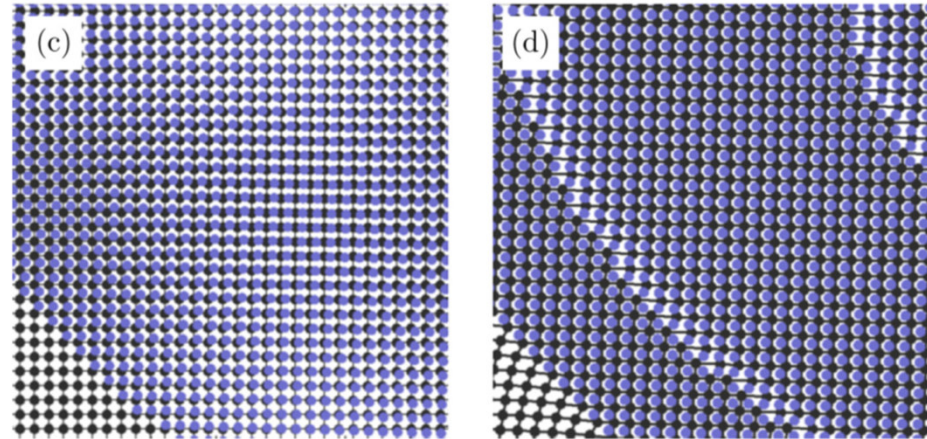
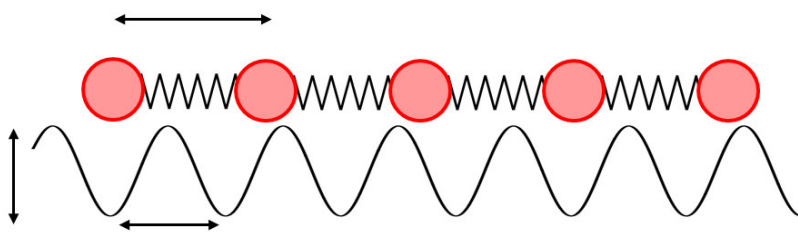
The Break-Down of Superlubricity: b) Relaxations/Dislocations

1D: Frenkel-Kontorova

interface energy $\tau_{\max}$ large:
break-down of superlubricity



interface energy $\tau_{\max}$ small:
superlubricity



For 3d islands: Transition from superlubric to ,break-down' with dislocations, if island exceeds certain size limit b_{core}

$$b_{\text{core}}/d = G/\tau_{\max}$$

lattice
constant

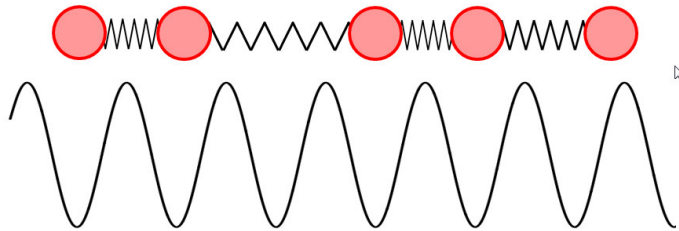
shear
modulus

maximum
shear stress

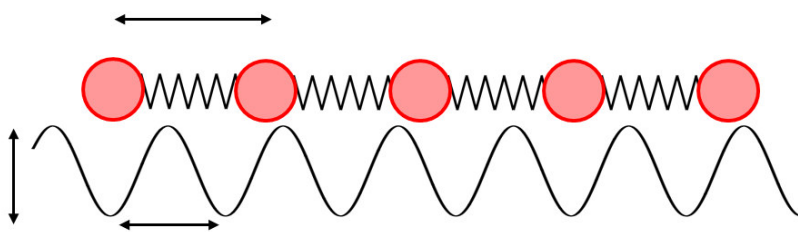
The Break-Down of Superlubricity: b) Relaxations/Dislocations

1D: Frenkel-Kontorova

interface energy $\tau_{\max}$ large:
break-down of superlubricity

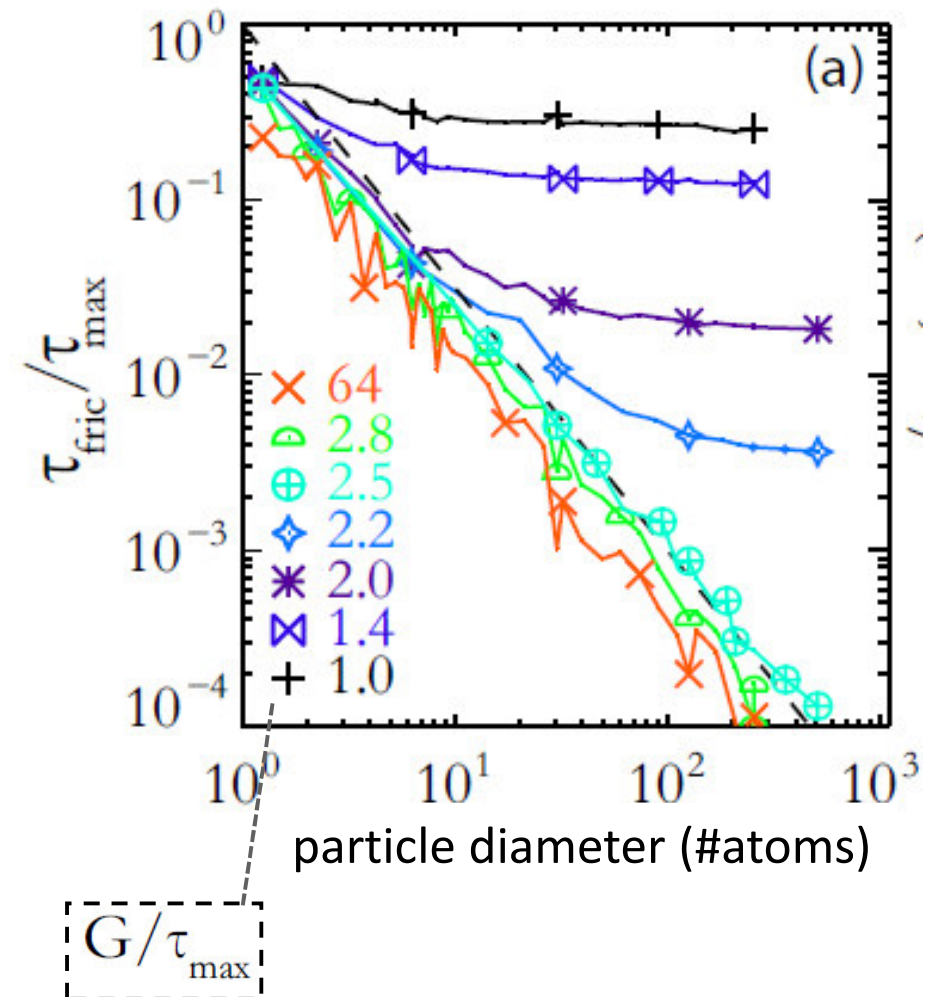


interface energy $\tau_{\max}$ small:
superlubricity



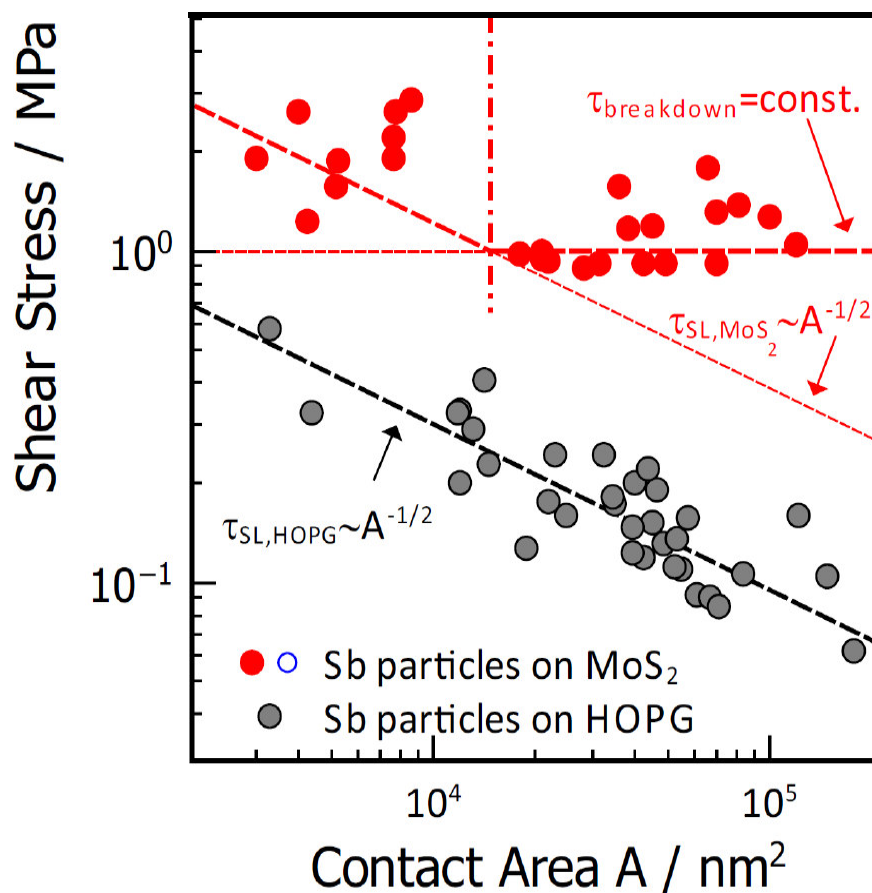
2D: Peierls-Nabarro Transition

3d atomistic simulation: Friction per atom



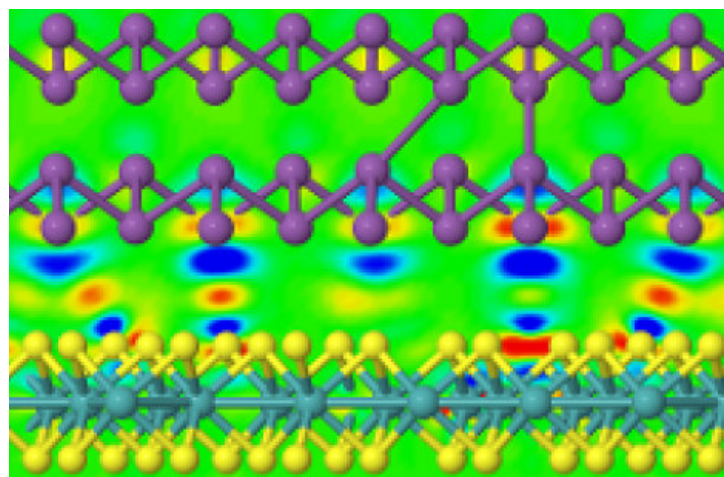
The Break-Down of Superlubricity: b) Relaxations/Dislocations

Expt.: Friction of particles: MoS₂ vs HOPG

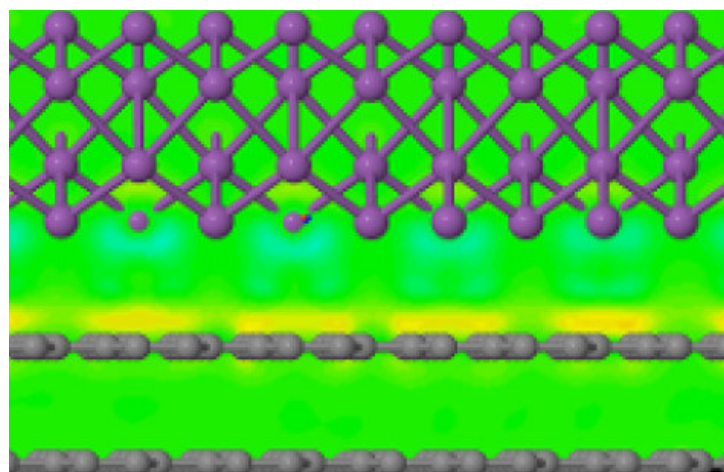


Dietzel, Brndiar, Stich, Schirmeisen,
ACS Nano 11 (2017) 7642

Sb on MoS₂



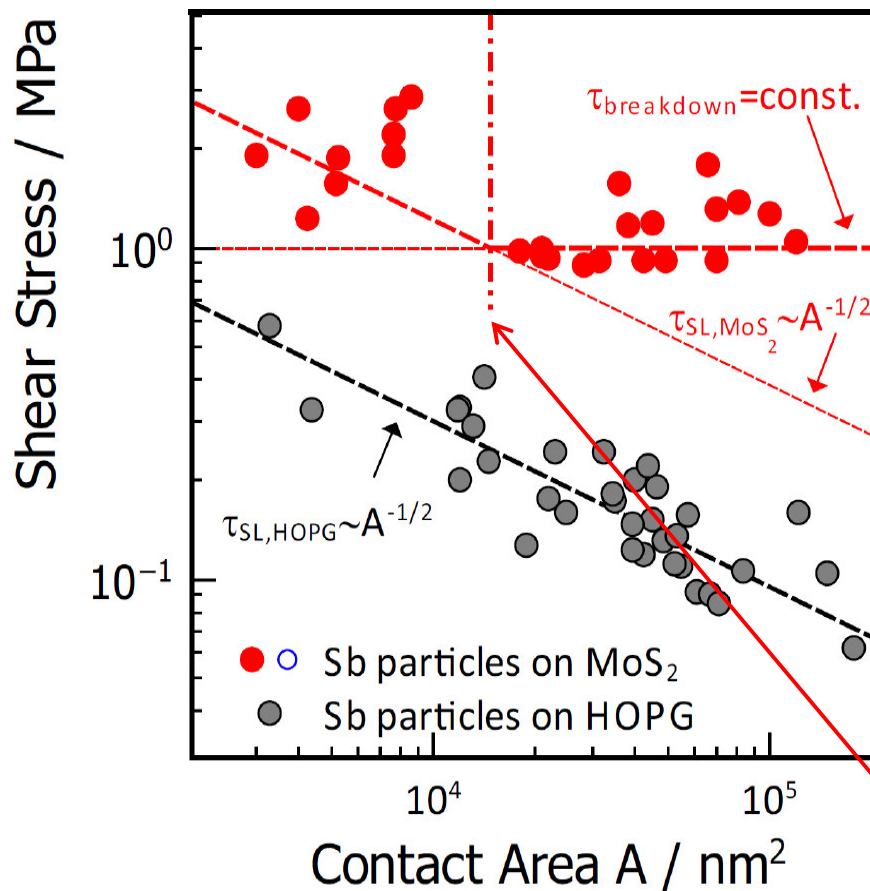
Sb on HOPG



Ab initio simulations: I. Stich, J. Brndiar

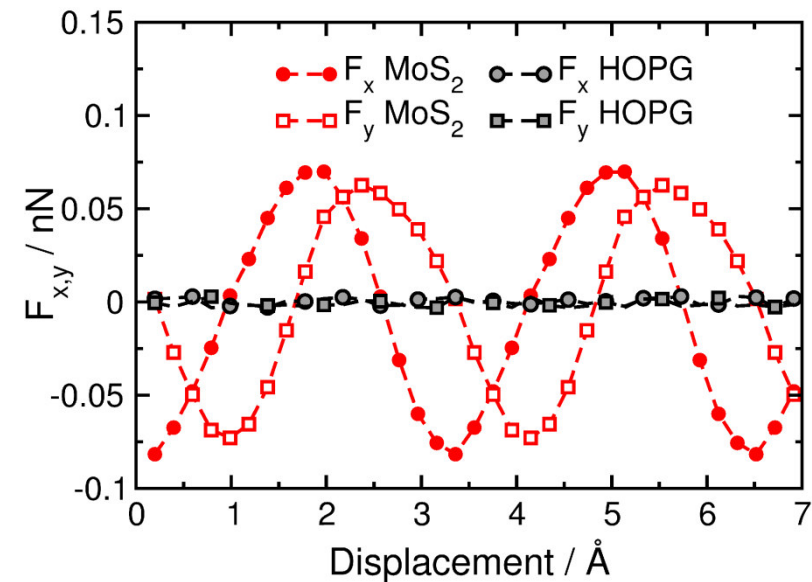
The Break-Down of Superlubricity: b) Relaxations/Dislocations

Expt.: Friction of particles: MoS₂ vs HOPG



Dietzel, Brndiar, Stich, Schirmeisen,
ACS Nano 11 (2017) 7642

Force acting on Sb atom during sliding

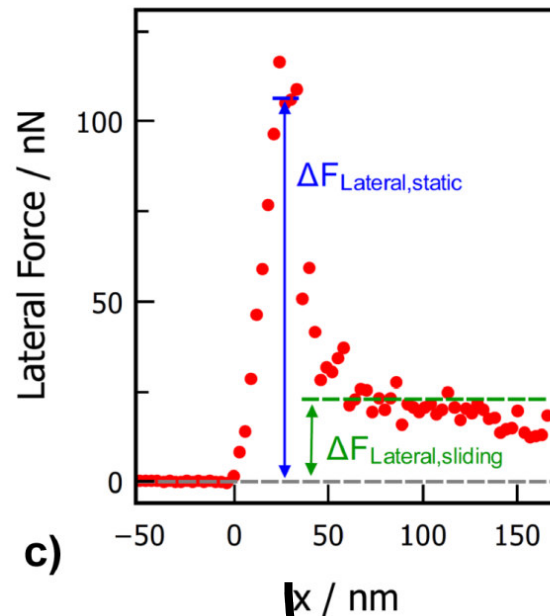
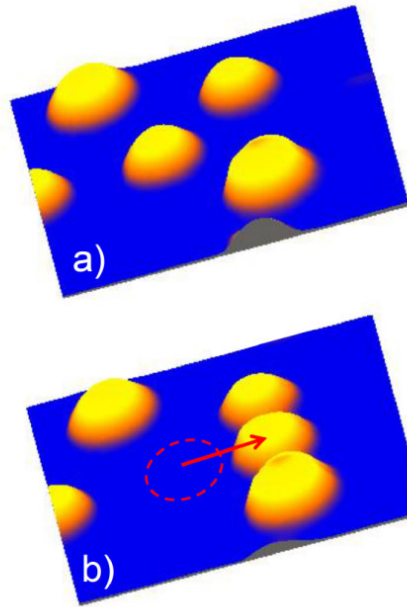


using $b_{\text{core}}/d = G/\tau_{\text{max}}$
(with $\tau_{\text{max}} = 290$ MPa, $G = 20$ GPa)

➡ $b_{\text{core}} = 30 \text{ nm}$

Experiment: $r_{\text{critical}} = 50 - 80 \text{ nm}$

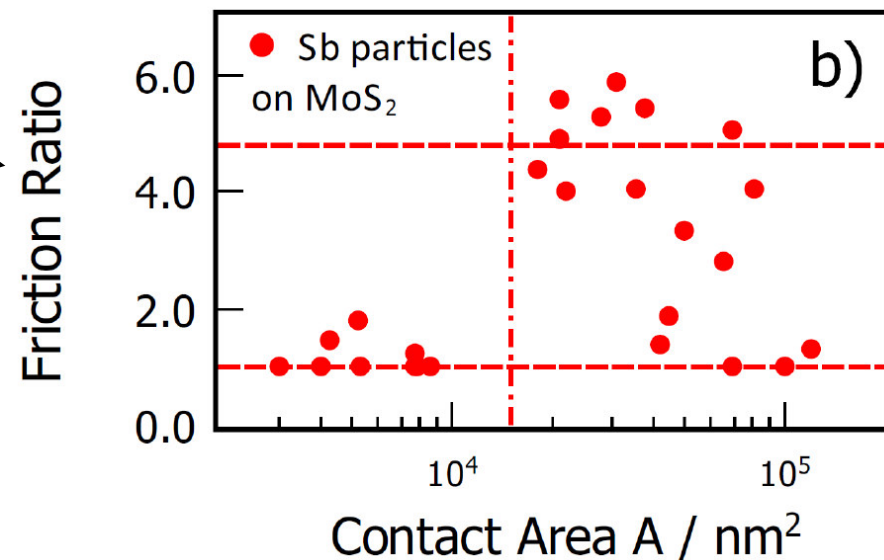
The Break-Down of Superlubricity: b) Relaxations/Dislocations



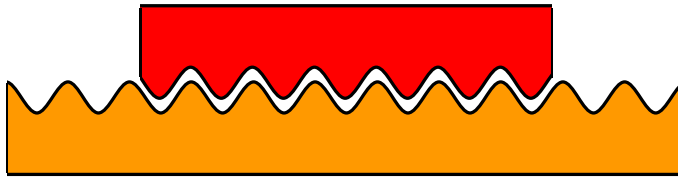
Experiments:

Sb islands on MoS₂ in UHV

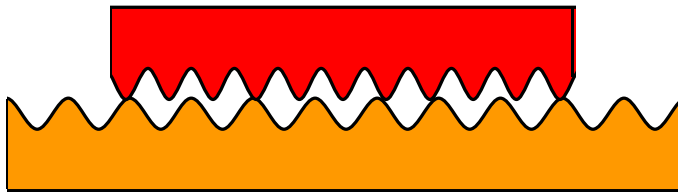
Static/sliding friction ratio depends on island size!



Structural Superlubricity - Categorization



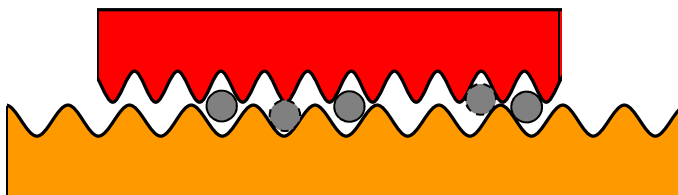
Interlocking of atomic potentials
friction $\propto$ area



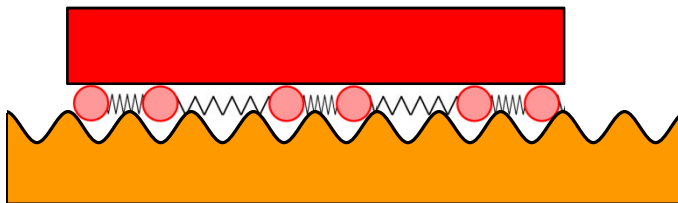
No interlocking of atomic potentials: Low friction
“Superlubricity”

$$F = \Delta E / a (A\rho)^\gamma$$

with $\gamma \leq 0.5$



Mobile molecule mediated interlocking of atomic potentials
friction $\propto$ area



Critical size threshold for interface dislocations
friction $\propto$ area

Acknowledgements

The Nanoparticle Friction Team...



Dirk
Dietzel



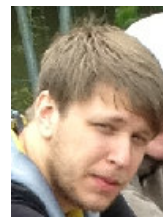
Michael
Feldmann



Matthias
Vorholzer



Carina
Herding



Felix
Harthmut



Petr
Chizikh

Funding:



...and “Nanoparticle” Cooperation Partners



Udo D. Schwarz,
Yale University



Ivan Stich,
Slovak Academy of Sciences



Enrico Gnecco,
FSU Jena, Germany



seit 1558



Peter Grütter,
McGill University, Canada

