

# Electronic Structure and Atomistic Computations for the interpretation of Synchrotron Experiments

*Spectroscopy and microscopy*

*Lecture Focus: Mainly Density Functional Theory (DFT-based) computations*

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# Density-Functional Theory (DFT)

Total energy:

$$E[\rho]$$

*all ground-state properties determined by*

Electronic density:

$$\rho(\vec{r})$$

→  $\rho$  obtained by solving

One-electron Schrödinger equation

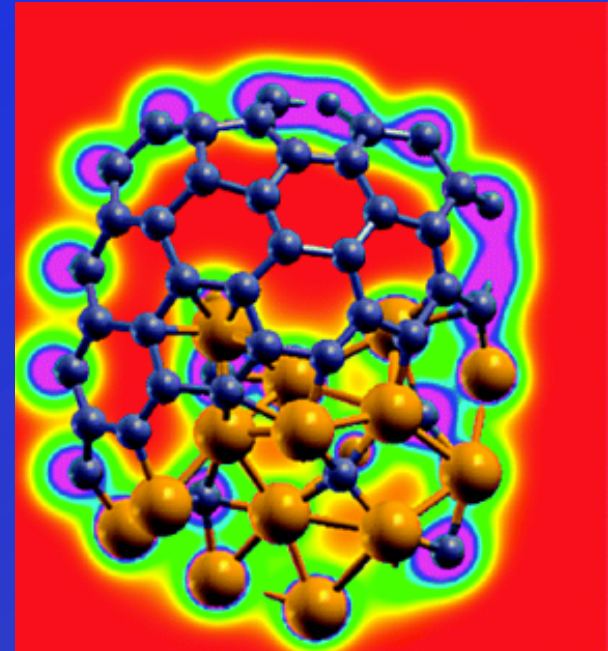
(Kohn-Sham) :

$$H\phi_i(\vec{r}) = \varepsilon_i\phi_i(\vec{r})$$

*For the  $N$  lowest  
energy states*



$$\rho(\vec{r}) = \sum_{\varepsilon_i \leq E_F} |\phi_i(\vec{r})|^2$$



Nanotube cap on Ni<sub>55</sub>C<sub>14</sub> Nanoparticles  
Gomez-Ballesteros and Balbuena, *Phys. Chem. Chem. Phys.*, **17**, 15056 (2015)

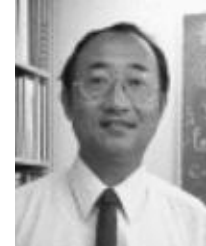
# Density-Functional Theory (DFT)

Kohn-Sham DFT (Physical Review 1965)

(Use one-electron  $\phi_i$  that reproduce true interacting  $\rho$ )



*W. Kohn*  
*Nobel 1998*



*L. J. Sham*

**One-electron Kohn-Sham equation:**

$$\left\{ -\frac{1}{2m_e} \nabla^2 + V_{nuc/ion}(\mathbf{r}) + V_C[\rho(\mathbf{r})] + V_{XC}[\rho(\mathbf{r})] \right\} \phi_i(\mathbf{r}) = \varepsilon_i \phi_i(\mathbf{r})$$

**Periodicity**

3D (crystal)  
2D (surface)  
1D (wire)  
0D (molecule)

**Functional**

beyond.....  
hybrid-GGA  
GGA+U  
GGA  
LDA

TD-DFT  
GW  
GW+DMFT

# Material modeling with DFT

## Input

Atomic Structure or Composition

## Output

Total Energy & Electronic Structure

### Structural properties

atomic forces  
equilibrium coordinates  
atomic vibrations  
phonons  
stresses, pressure

### Electronic properties

density of states  
band structure  
effective mass tensors  
electron charge distribution  
magnetism

### Thermodynamics

internal energy (U)  
enthalpy (H)  
free energy (G)  
activation energies ( $\Delta E$ )

### Excitations

transition intensities  
dielectric functions  
spectroscopy



# *From atomic aggregates to crystals*

With periodic potential



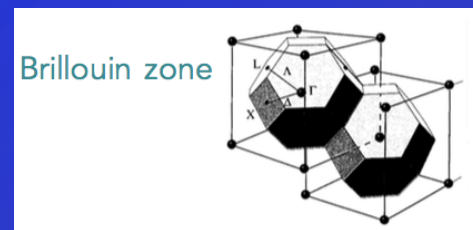
$$\phi_{\vec{k}}(\vec{r}) = u_n(\vec{r})e^{i\vec{k}\cdot\vec{r}}$$

**Bloch state**

F. Bloch  
1905-1983



u: periodic (unit cell) part  
k: wavevector in Brillouin zone



Band structure

$$\epsilon(\vec{k})$$

# *Electronic bulk band structure: Bulk Cu - GW compared to LDA*

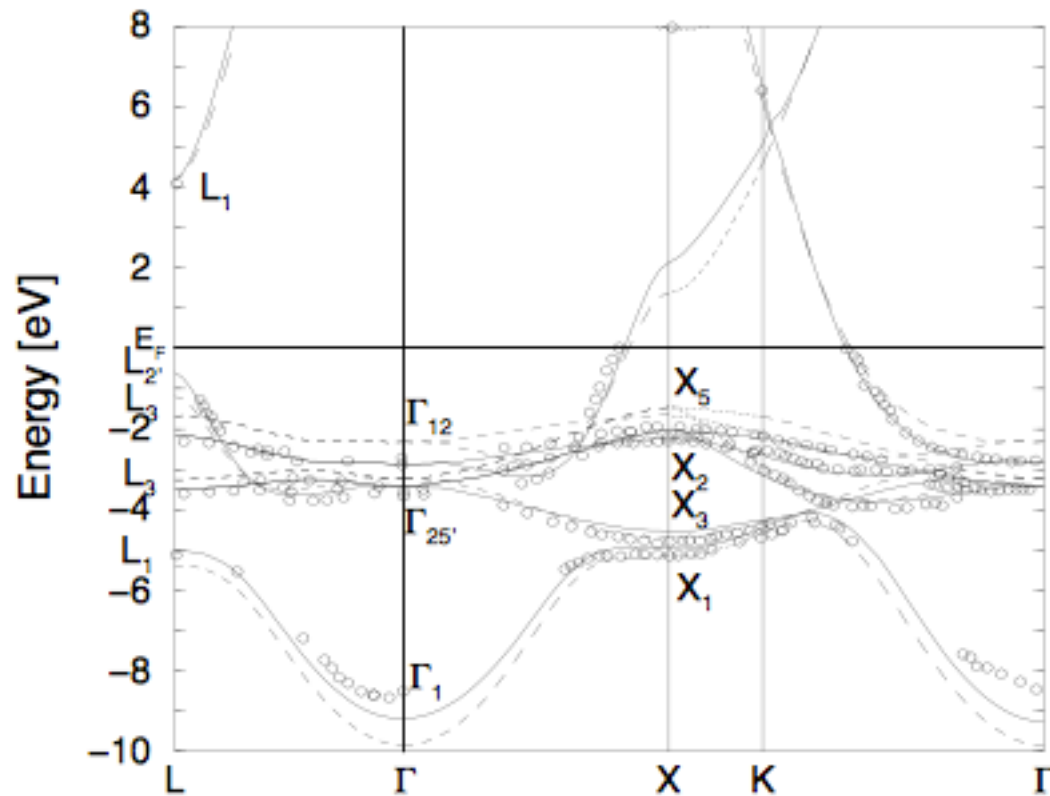


FIG. 2.: Full line: present GW results for the bulk copper band-structure, compared with the DFT-LDA results (dashed line), and with the experimental data reported in reference [16] (circles).

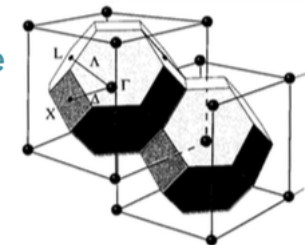
From: Andrea Marini,  
Giovanni Onida, and  
Rodolfo Del Sole, PRL **88**,  
016403 (2001)

GW(solid line) best  
LDA (dashed line) not bad

PS: Not always so:  
 strongly correlated  
materials

PE from:[16] R. Courths and S.  
Hufner, Phys. Rep. 112, 53 (1984)

Brillouin zone

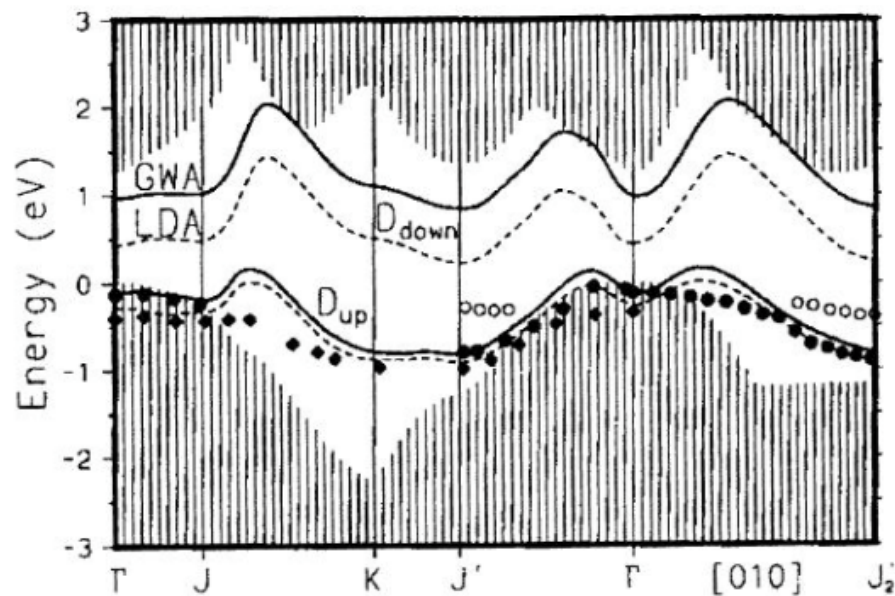


# *Electronic surface band structure*

## *Si surface states – GW versus LDA*

Surface states:

Si(100)-(2x1) surface

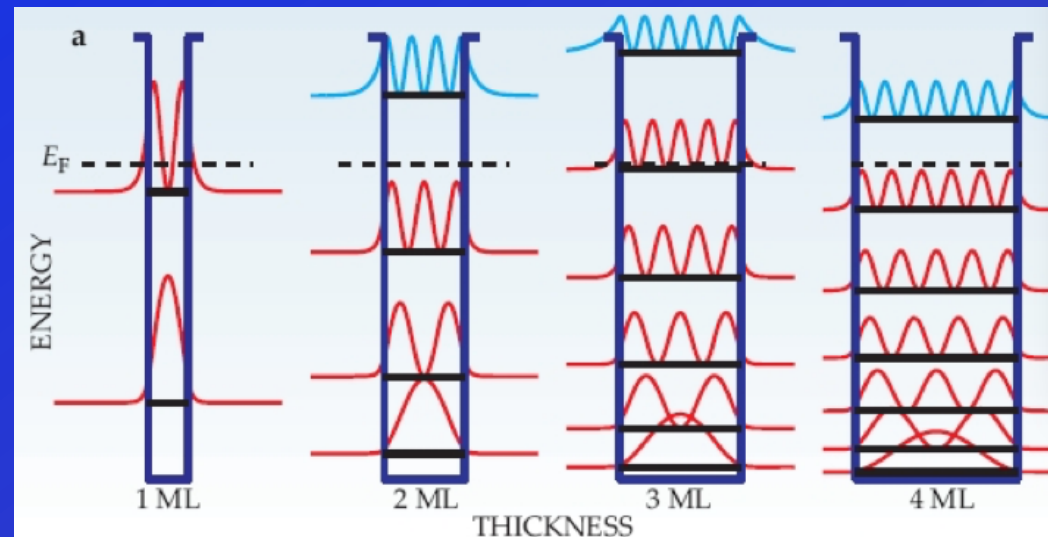


Rohlfing, Krüger, and  
Pollmann, PRB 52, 1905  
(1995)

**Figure 15.** Calculated dangling-bond bands. Full curves, GWA energies; dashed curves, LDA energies. The experimental results are shown by diamonds (Uhrberg *et al* 1981) and circles (full and open) (Johansson *et al* 1990). (After Rohlfing *et al* 1995b).

# Quantum-size effects on surface reactivity

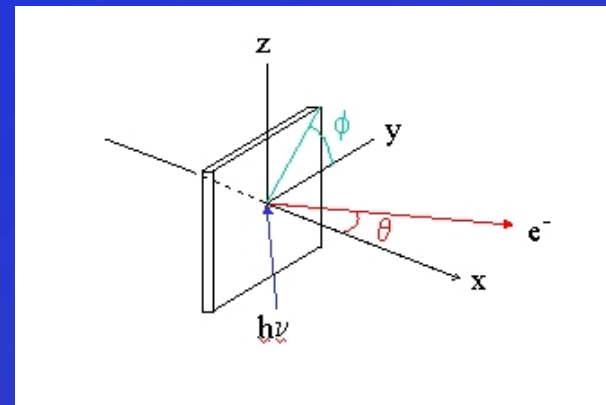
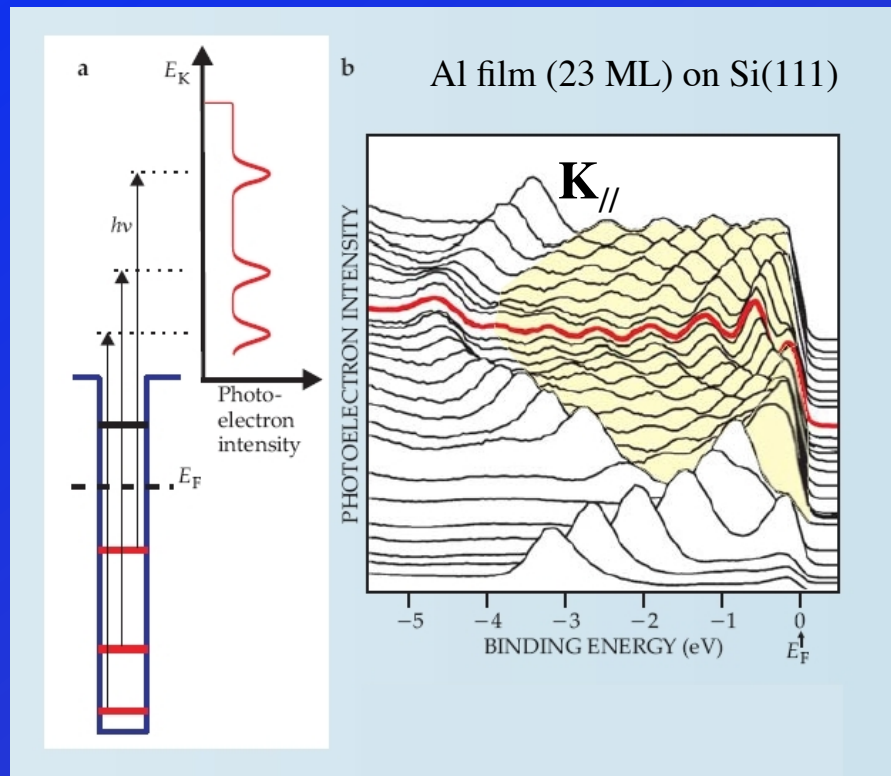
Quantum-size effects are effects related to Quantum Well States (QWS), e.g., in ultrathin epitaxial films



# Observation of quantum well states *in thin metal films*

By photoemission (PE): occupied QWS

## Angle-resolved photoelectron spectroscopy



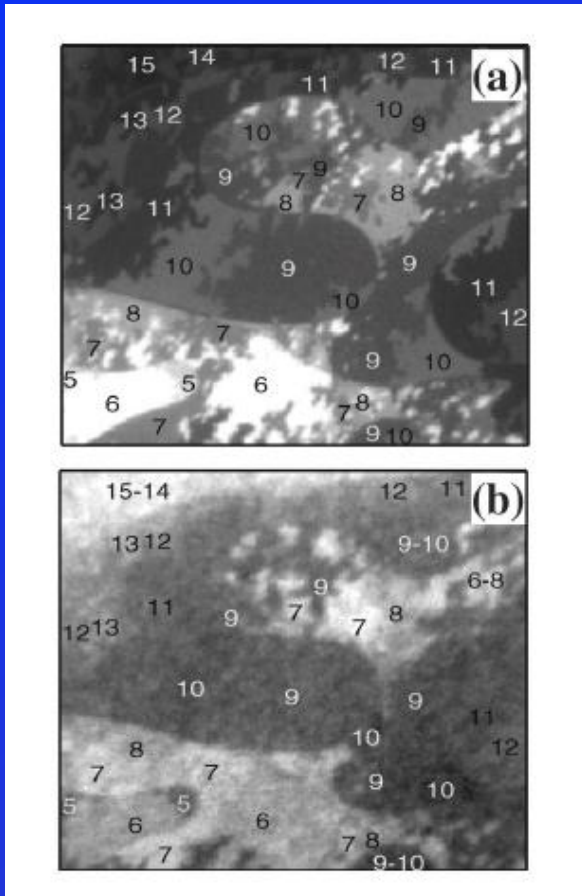
Measures the kinetic energy  $E_K$  and momentum  $\mathbf{p}_{//}$  of electrons emitted from the surface of the film

Tringides, Jalochofski, Bauer,  
Physics Today **60** (no 4), 50 (2007)



# Quantum-size effects on surface reactivity: Mg films

## Spectromicroscopy Experiment:



6 x 5  $\mu\text{m}^2$  images of a Mg (0001) film on W (110) with microregions of different Mg film thicknesses

Aballe, Barinov, Locatelli, Heun, Kiskinova, PRL 93, 196103(2004); Aballe et al., J. Phys.: Cond. Matter 22, 015001 (2010)

### (a) Before oxygen exposure

LEEM (Low-Energy Electron Microscopy)

*Thickness determined from reflectivity changes*

### (b) After oxygen exposure

XPEEM (X-ray photoelectron Emission Microscopy)

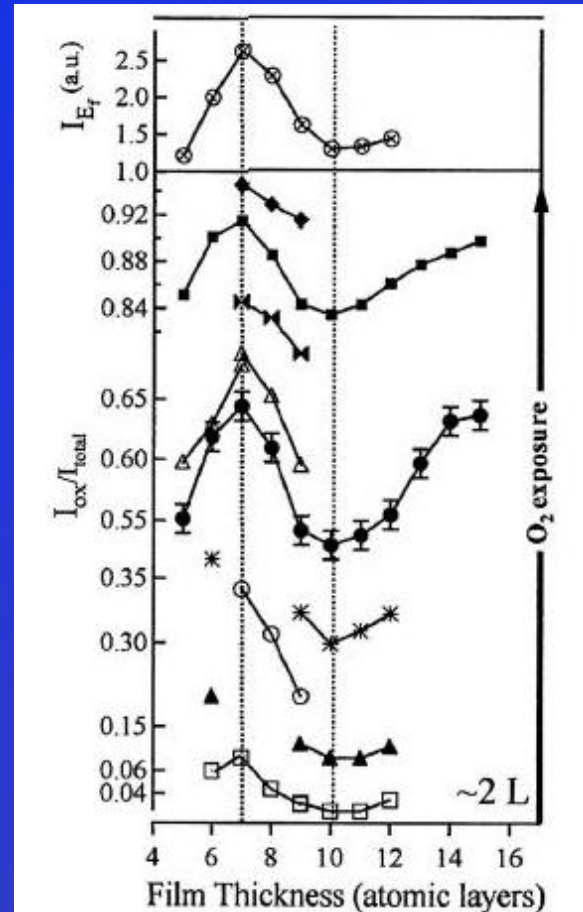
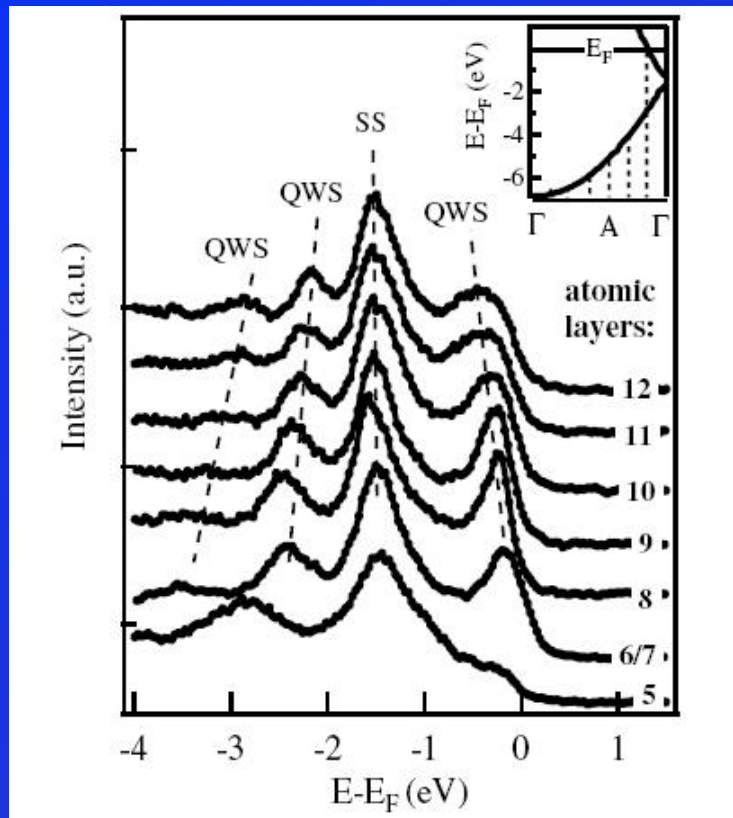
*Contrast corresponds to the extent of Mg oxidation (from the intensity of Mg 2p oxide component)*

➤ **Modulation in the oxidation rate with Mg film thickness**

# Quantum-size effects on surface reactivity: Mg films

Spectromicroscopy: Aballe et al. PRL 93, 196103(2004);  
J. Phys.: Cond. Matter 22, 015001 (2010)

Valence PE near  
normal emission ( $k_{\parallel} \approx 0$ )



$I_{k_{\parallel} \approx 0}(E_F)$

Oxidation rate  
(from intensity  
of Mg 2p oxide  
component)

Drastic (1-2  
orders of  
magnitude)  
changes!

Origin of the correlation between surface reactivity and QWS ?



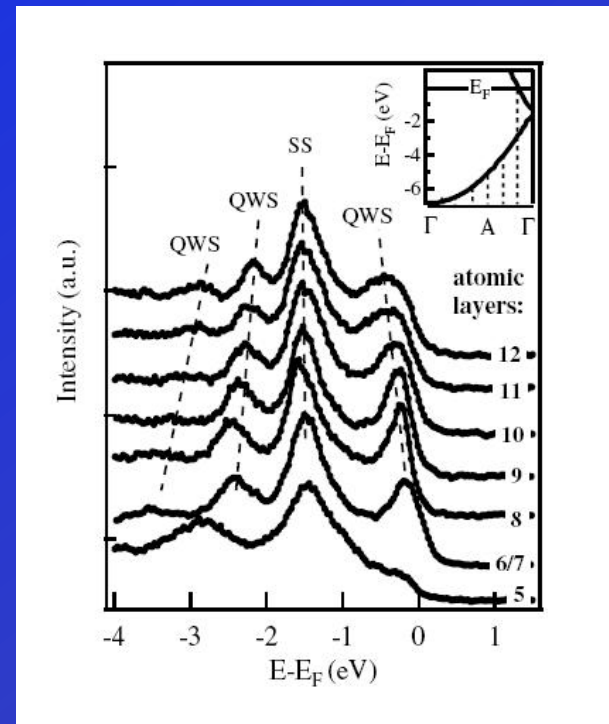
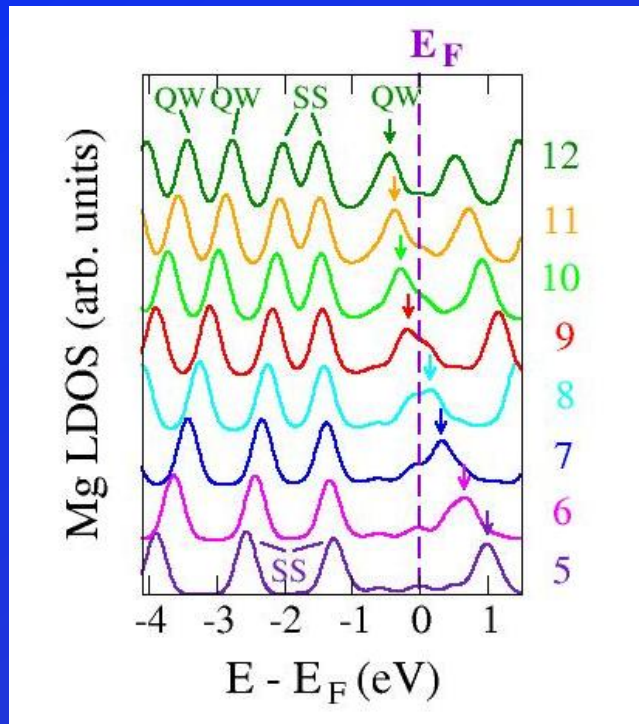
# DFT calculations of Mg thin films on W(110)

Calculated DFT DOS versus measured normal-emission PE spectra

Partial DOS at  $k_{||} = 0$

PE by Aballe *et al.*

Mg(0001)/W

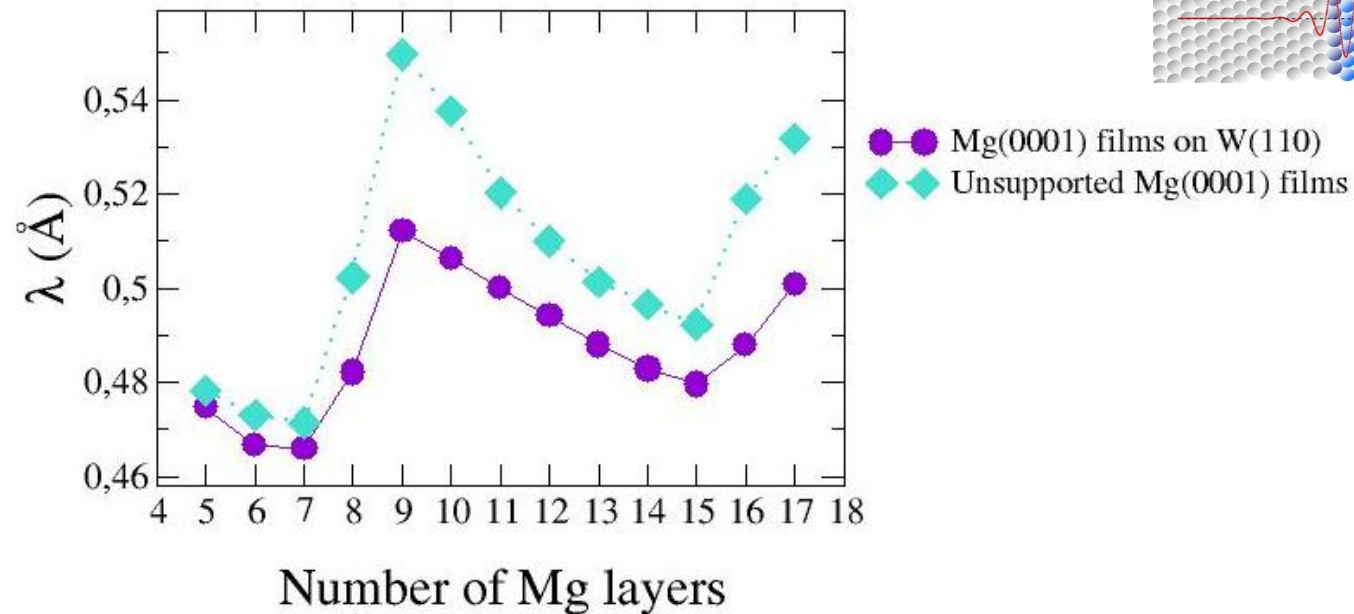
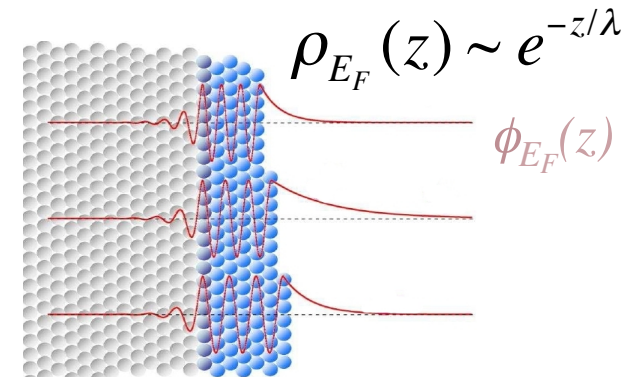


- Good general agreement in the peak positions

Binggeli and Altarelli, Phys. Rev. Lett. 96, 36805 (2006); Phys. Rev. B 78, 35438 (2008)

# DFT calculations for Mg films on W(110)

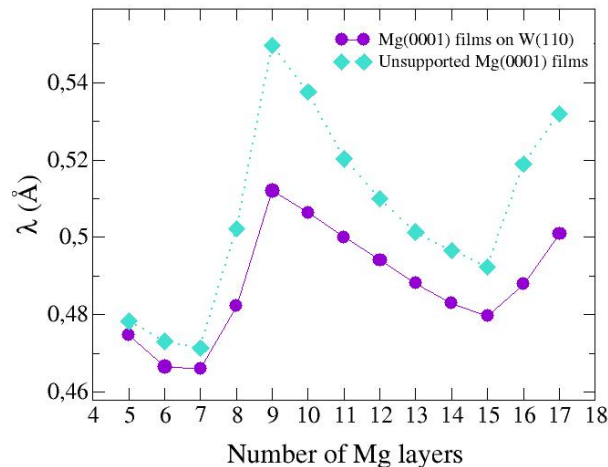
- Decay length in vacuum  $\lambda$  of the metal film electronic local density of states at  $E_F$



- **Substantial oscillations in  $\lambda$**   
substrate reduces the variation from 17% to 10 %

# DFT calculations of Mg thin films on W(110)

## Electronic decay length $\lambda$



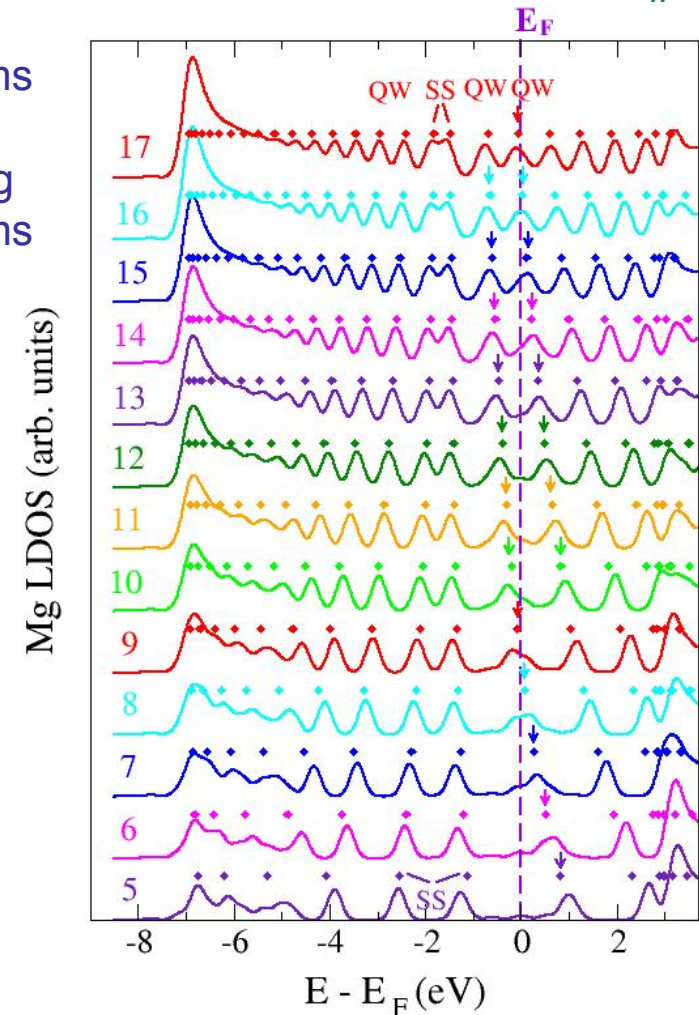
- $\lambda$  is maximum when a Quantum-Well State (QWS) at  $k_{||}=0$  crosses  $E_F$

trend explained by model description for  $\lambda$

$$\lambda \sim 1/\sqrt{-E_M}$$

$E_M$ : Energy of the highest occupied QW state at  $K_{||}=0$  relative to the vacuum level

## Partial density of states ( $k_{||}=0$ )



## Origin of the changes in the surface reactivity ?

Binggeli and Altarelli, Phys. Rev. Lett. 96, 35805 (2006); Phys. Rev. B 78, 35438 (2008)

**The changes in  $\lambda$  are expected to influence the electron transfer process from the  $O_2$  molecule to the metal surface (by tunneling), which is believed to control the initial sticking of  $O_2$  on the metal surface (via the attractive image charge potential on the ionized  $O_2^-$  molecule)**

Hellman et al., Surface Science 532, 126 (2003).

***The changes in  $\lambda$  are expected to have an exponential impact on the electron tunneling rate, and therefore on the initial sticking of  $O_2$***

*A 10% change in  $\lambda$  will produce a 100% change in the transfer rate  $\sim e^{-d/\lambda}$ , at  $d \sim 3.5 \text{ \AA}$ , which is of the order of magnitude of the experimental change in the oxidation rate at low  $O_2$  exposure*

→ **Can explain the order of magnitude change observed experimentally**



# Angle resolved photoemission spectra

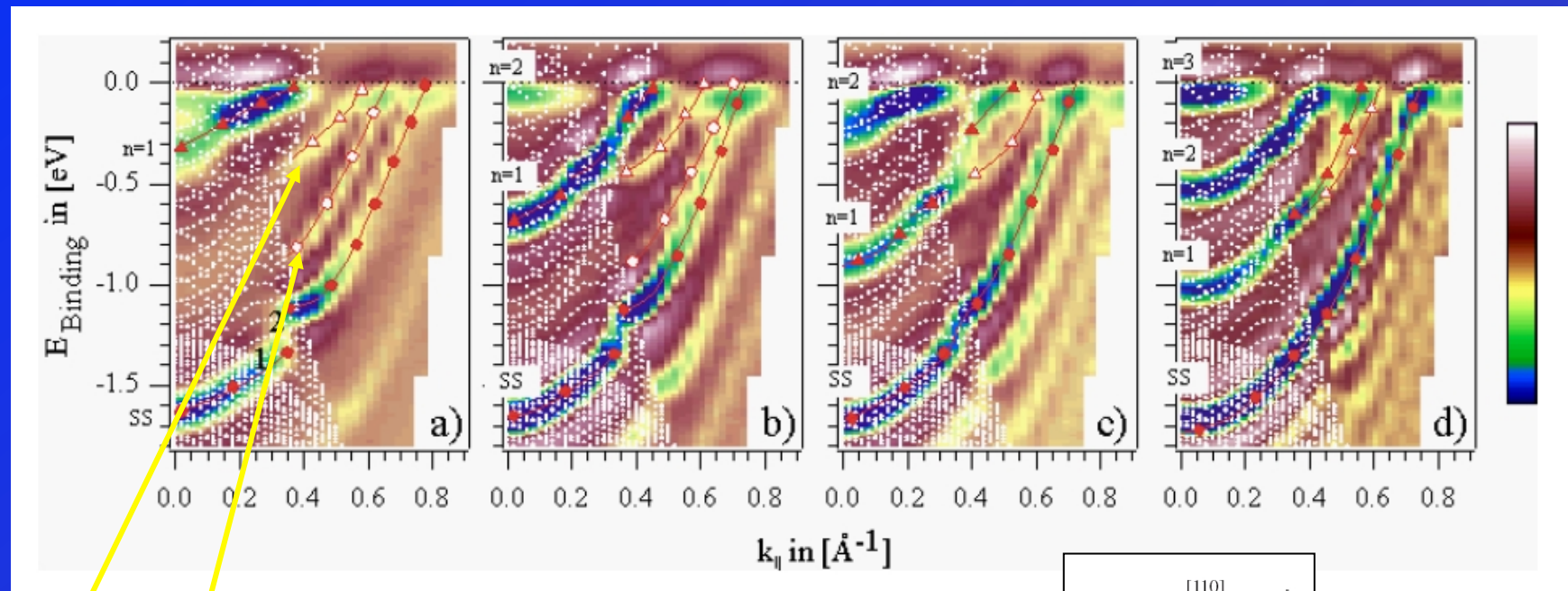
Mg films on W(110)

8 ML

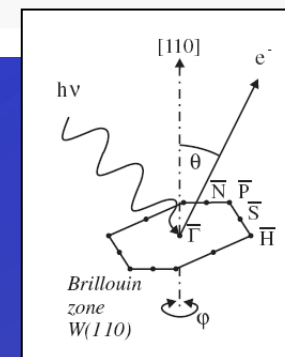
12 ML

16 ML

22 ML



New (split) states

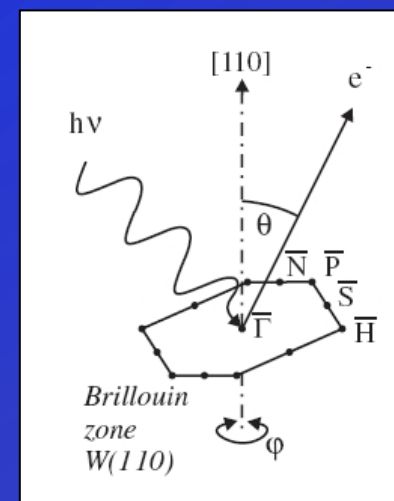
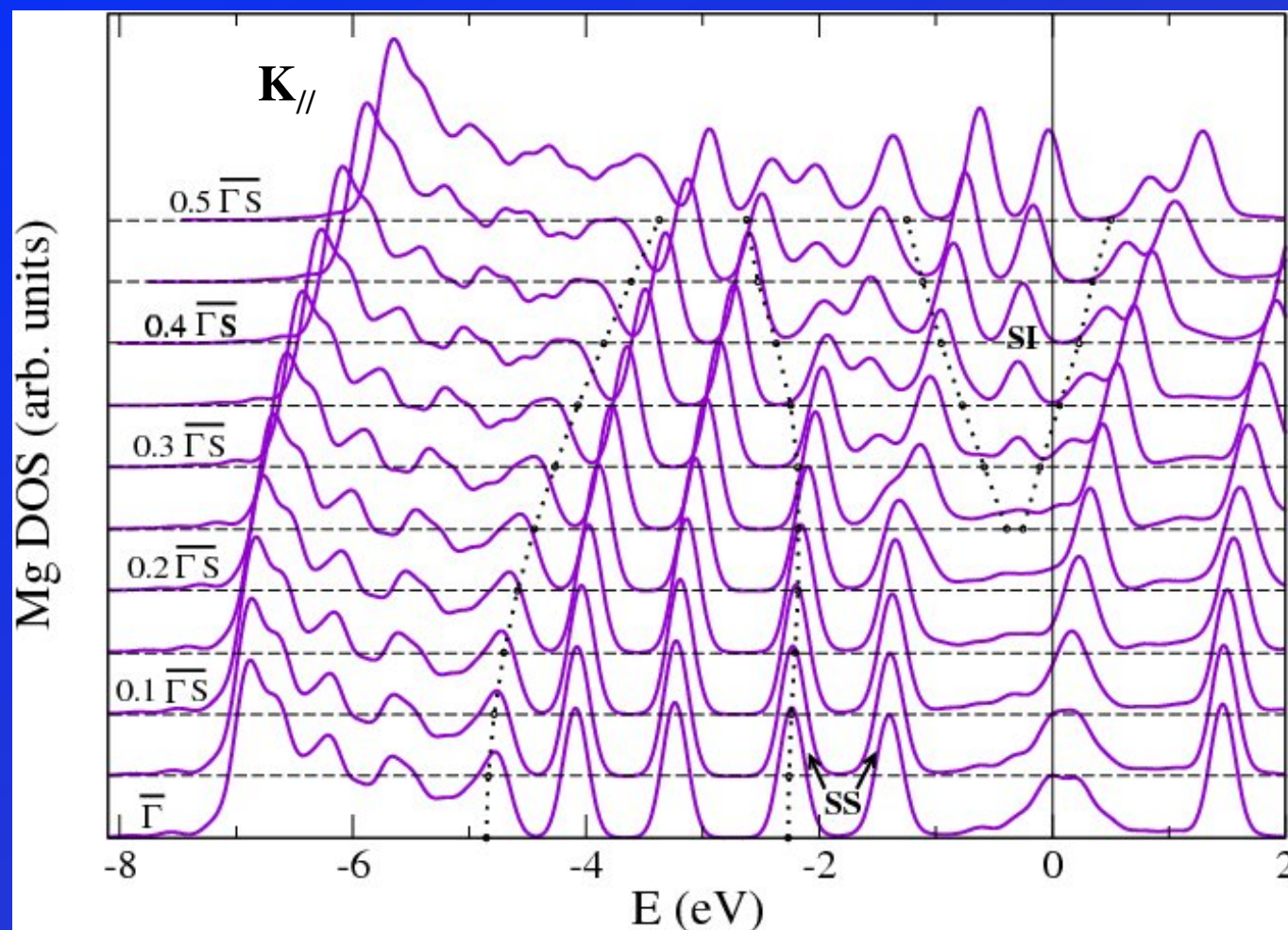


Koitzsch et al., Phys. Rev. Lett. **95**, 126401 (2005)

# DFT results for Mg QWS: 8 Mg ML on W(110)

Binggeli and Altarelli, Phys.  
Rev. B **78**, 35438 (2008)

$K_{//}$  along the  $\Gamma S$  direction



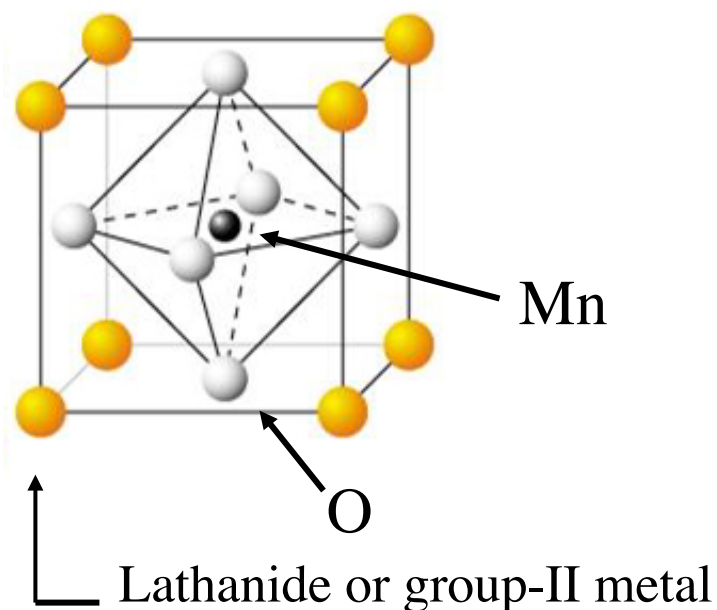
**SI:** These states result from the interaction of the Mg(0001) Shockley surface states with the W(110) surface state at the Interface

# Resonant elastic x-ray scattering to probe orbital order in complex transition-metal compounds

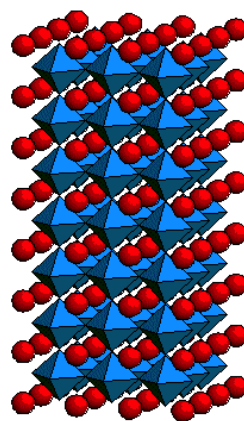
(such as manganites)

- Properties controlled by a **complex interplay** between **structural**, **magnetic**, and valence **electronic charge/orbital** degrees of freedom

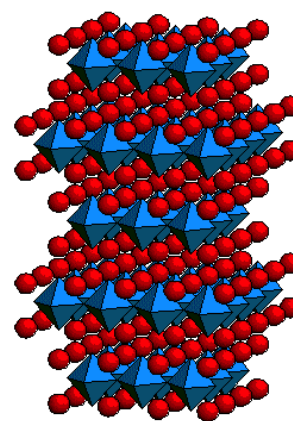
Perovskite structure



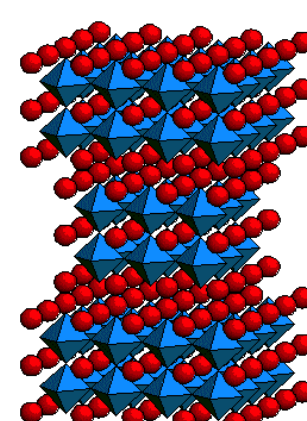
Layered Perovskite structures



$n = \infty$   
 $\text{ABO}_3$   
perovskite



$n = 1$   
 $\text{A}_2\text{BO}_4$   
 $\text{K}_2\text{NiF}_4$



$n = 2$   
 $\text{A}_3\text{B}_2\text{O}_7$   
bilayer



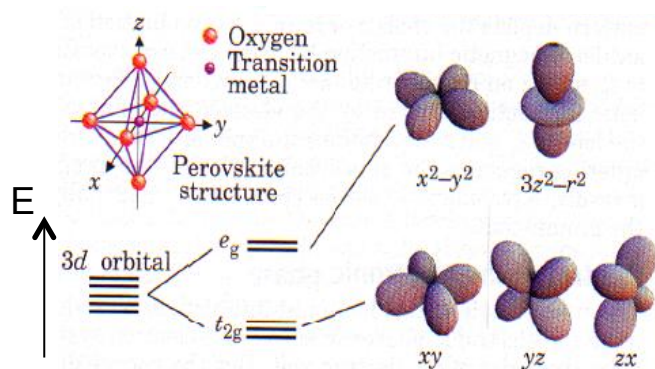
# Resonant elastic x-ray scattering to probe orbital order in complex transition-metal compounds

(such as manganites)

- Properties controlled by a **complex interplay** between **structural**, **magnetic**, and valence **electronic charge/orbital** degrees of freedom

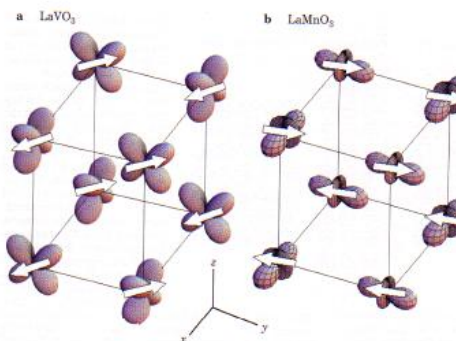
## Orbital and spin ordering in perovskites

### 3d orbital degeneracy



**LaVO<sub>3</sub>**

V (3d<sup>2</sup>)

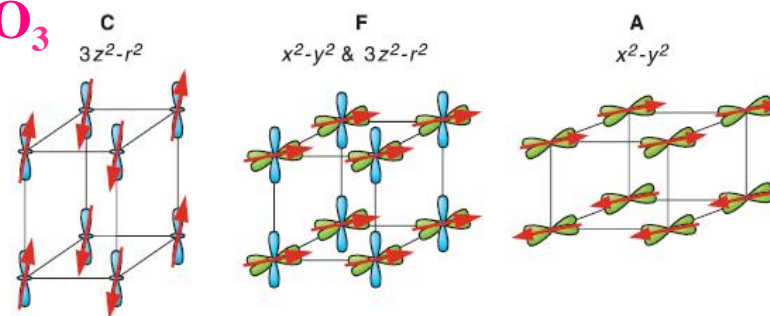


**LaMnO<sub>3</sub>**

Mn (3d<sup>4</sup>)

**La<sub>0.5</sub>Sr<sub>0.5</sub>MnO<sub>3</sub>**  
(Coherently strained)

Mn (3d<sup>3.5</sup>)

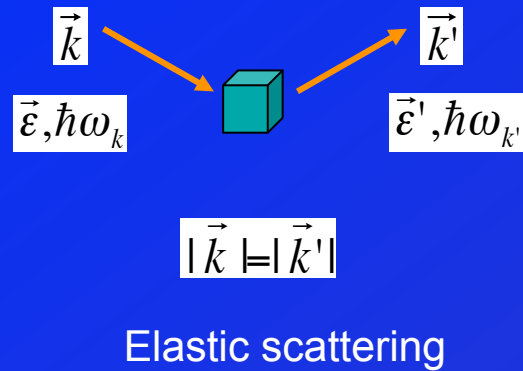


## The orbital order in complex transition-metal compounds

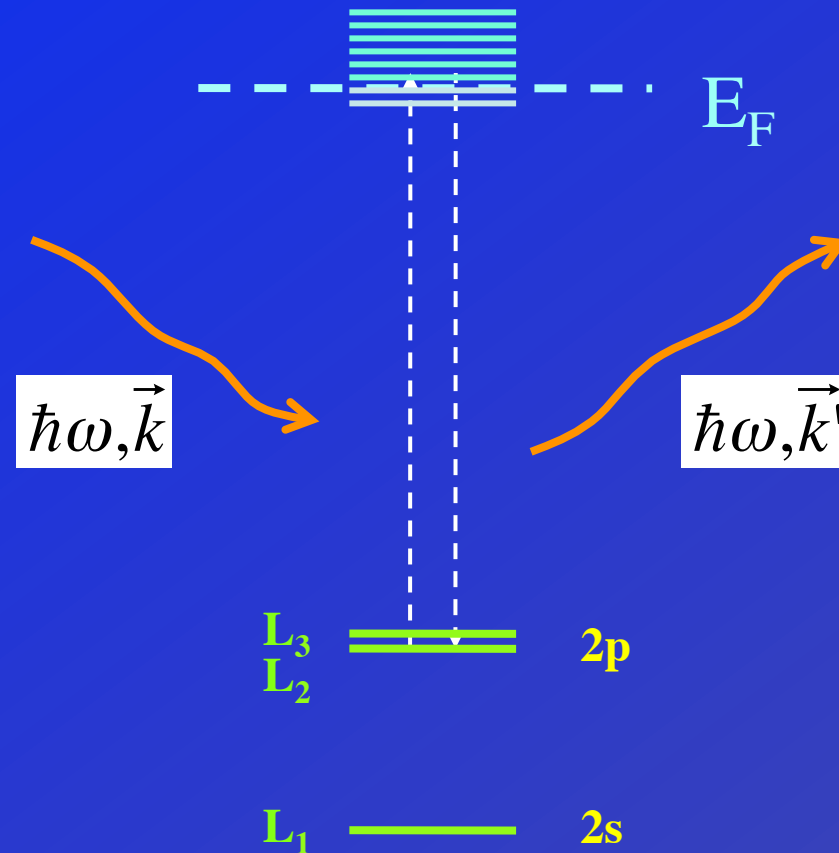
- **The orbital degree of freedom** of the 3d electrons plays a **key role** in the physics of manganites, but the orbital ordering is difficult to observe experimentally (**hidden to most experimental probes**)
- **Resonant elastic X-ray Scattering** (RXS) is a powerful **tool to probe orbital ordering**, but the interpretation of the experimental spectra is often controversial

→ **importance of theoretical predictions**

# Resonant elastic X-ray Scattering (RXS)

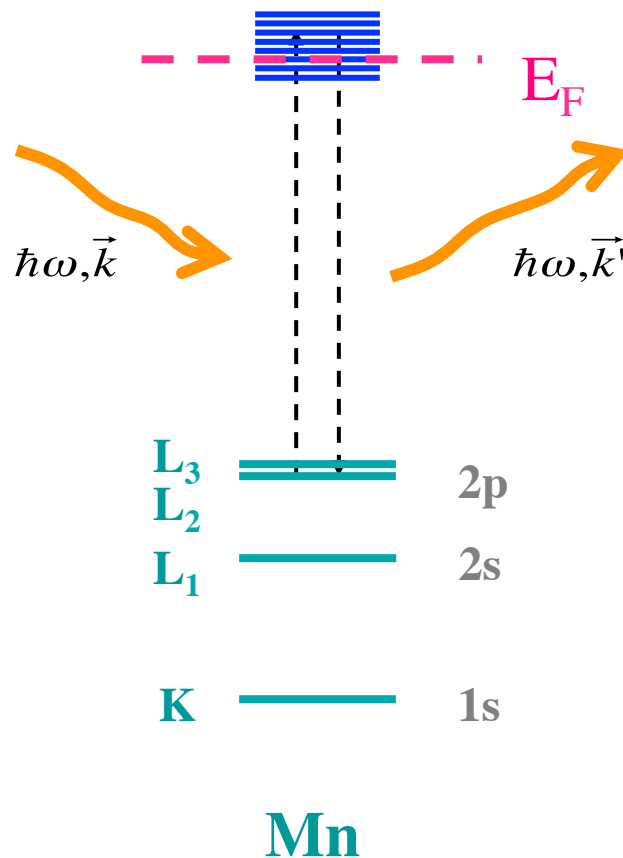


**Photon energy  
resonant with a core  
level absorption edge**



*Resonant elastic x-ray scattering is a second order process in which a core electron is virtually promoted to some intermediate states above the Fermi energy, and subsequently decays to the same core level*

# Resonant elastic x-ray scattering



- **Scattering** for Bragg condition:

$$\vec{k} - \vec{k}' = \vec{G}_{lmn},$$

$\vec{G}_{lmn}$  : reciprocal lattice (Bragg) vector

- **Resonant** scattering amplitude:

$$f(\vec{G}, \hbar\omega) \propto \sum_{I(\text{ions})} e^{i\vec{G}\vec{d}_I} \sum_{\alpha, \beta} F_{\alpha, \beta}^I(\hbar\omega) \varepsilon_{\alpha} \varepsilon_{\beta}'$$

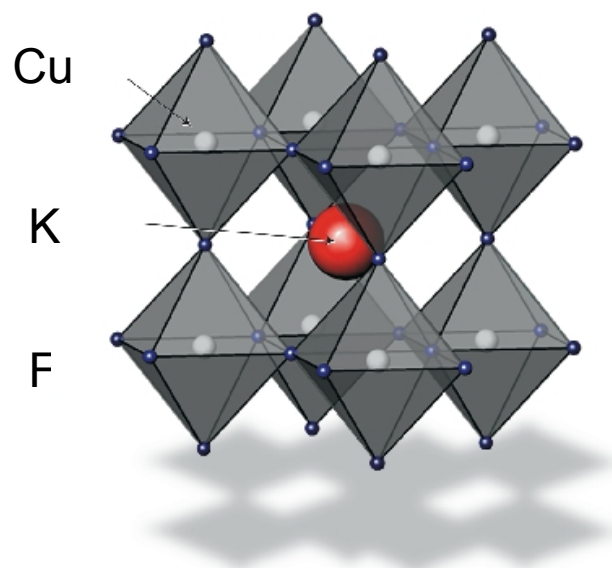
$$F_{\alpha, \beta}^I(\hbar\omega) \propto \sum_n \frac{\langle 0 | R_{\alpha}(I) | n \rangle \langle n | R_{\beta}(I) | 0 \rangle}{\hbar\omega + E_0 - E_n - i\Gamma/2}$$

- selective probe of order/periodicity
- element sensitive
- photon-energy dependent

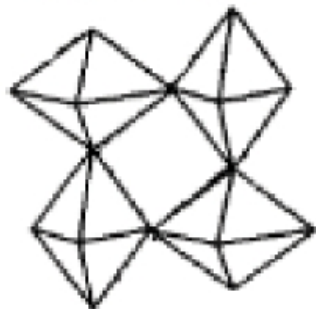
➤ At L<sub>2,3</sub> edge, in principle **orbital-order sensitive** (optical selection rule)

# Resonant x-ray scattering to probe orbital order in $\text{KCuF}_3$ ?

Crystal structure:



Jahn-Teller  
distortion



RXS at the Cu K edge in  $\text{KCuF}_3$

*Cu 1s  $\rightarrow$  4p dipole transitions*

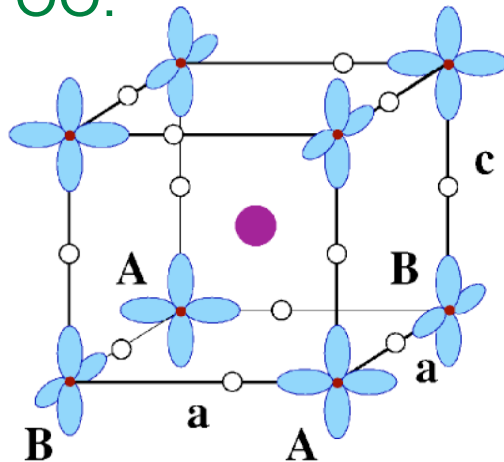
**Exp.:** Paolasini et al., PRL 88, 106403 (2002); Caciuffo et al., PRB 65, 174425 (2002)

- pseudo cubic perovskite
- cooperative Jahn-Teller distortion below 1000 K
- *Neel* temperature  $\sim 38$  K
- $d_{x^2-y^2}$  hole antiferroorbital ordering

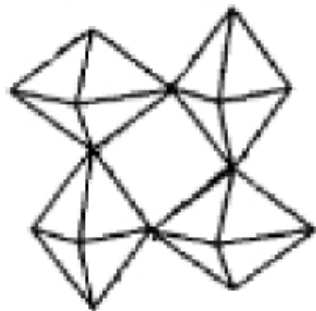
# Resonant x-ray scattering to probe orbital order in $\text{KCuF}_3$ ?

Crystal structure  
and OO:

$\text{Cu } 3d^9$



Jahn-Teller  
distortion



RXS at the Cu K edge in  $\text{KCuF}_3$

*Cu  $1s \rightarrow 4p$  dipole transitions*

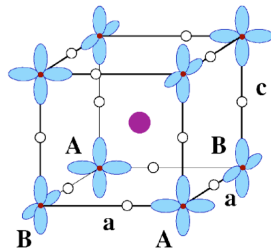
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- pseudo cubic perovskite
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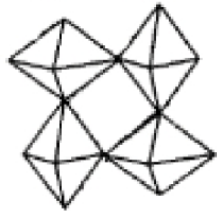
# Resonant x-ray scattering to probe orbital order in $\text{KCuF}_3$ ?

$\text{KCuF}_3$

$\text{Cu } 3d^9$



Jahn-Teller distortion



LDA+U

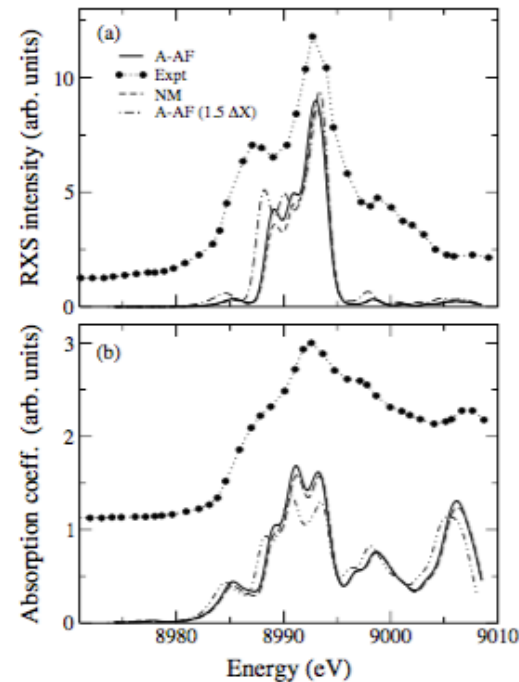


FIG. 9: Orbital RXS intensity (a) and absorption coefficient (b), as a function of photon energy, near the Cu K edge in  $\text{KCuF}_3$ . The calculated spectra of the A-AF (solid line) and NM (dashed line) structures are displayed. The effect of a 50% increase in the quadrupolar distortion,  $\Delta X$ , in the A-AF structure is also shown (dashed-dotted line). The experimental data are from Ref. 7; the measured RXS spectrum corresponds to a  $(3,3,1)$  orbital Bragg reflection and a  $\sigma - \pi'$  polarization.

**Cu K-edge:**

**$1s \rightarrow 4p$  dipole transitions**

**Atomic resonant scattering amplitude**

$$F_{\alpha,\beta}^j(\hbar\omega) = \sum_{\mathbf{k},n} \frac{\langle \psi_0^{(j)} | r_\alpha(j) | \psi_{\mathbf{k},n}^{4p} \rangle \langle \psi_{\mathbf{k},n}^{4p} | r_\beta(j) | \psi_0^{(j)} \rangle}{\hbar\omega + E_0 - E_{\mathbf{k},n}^{4p} - i\Gamma/2}$$

**RXS intensity**

$$I(\mathbf{G}, \hbar\omega) \propto \left| \sum_j e^{i\mathbf{Q} \cdot \mathbf{R}_j} \sum_{\alpha,\beta} F_{\alpha,\beta}^j(\hbar\omega) \epsilon_\alpha \epsilon'_\beta \right|^2 \quad \text{with } \vec{G} = \vec{Q}_{00}$$

$$I_{\text{orb}}(\hbar\omega) \propto |F_{x,x}^A(\hbar\omega) - F_{y,y}^A(\hbar\omega)|^2$$

**Absorption**

**For Orbital Ordering**

$$A(\hbar\omega) \propto \text{Im}[F_{x,x}^A(\hbar\omega) + F_{y,y}^A(\hbar\omega) + F_{z,z}^A(\hbar\omega)]$$

- Good general agreement with exp.
- RXS: controlled by Jahn-Teller distortion  
not sensitive to OO  
not sensitive to spin order



# Resonant elastic x-ray scattering to probe orbital order in manganites ?

RXS at the Mn  $L_{2,3}$  edges in manganites

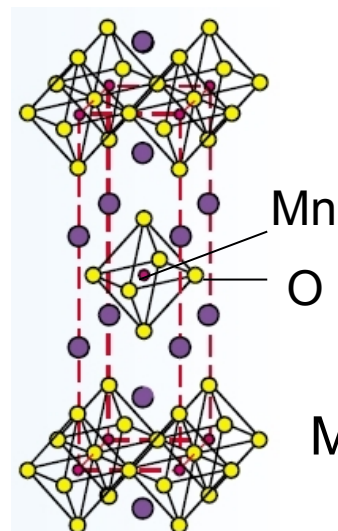
( $L_2:2p_{1/2}$ ,  $L_3:2p_{3/2}$ )

*Mn 2p  $\rightarrow$  3d dipole transitions*

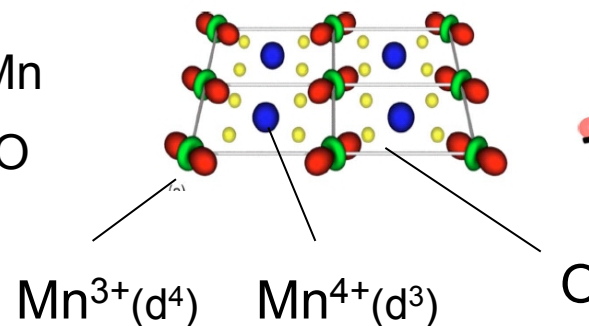
**Exp.:** Wilkins, Spencer, Hatton *et al.*, PRL **91**, 167205 (2003)



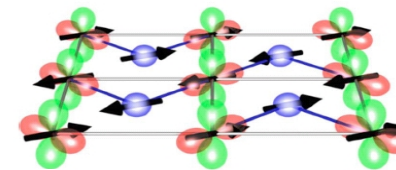
Crystal structure



Mn-3d charge and orbital order

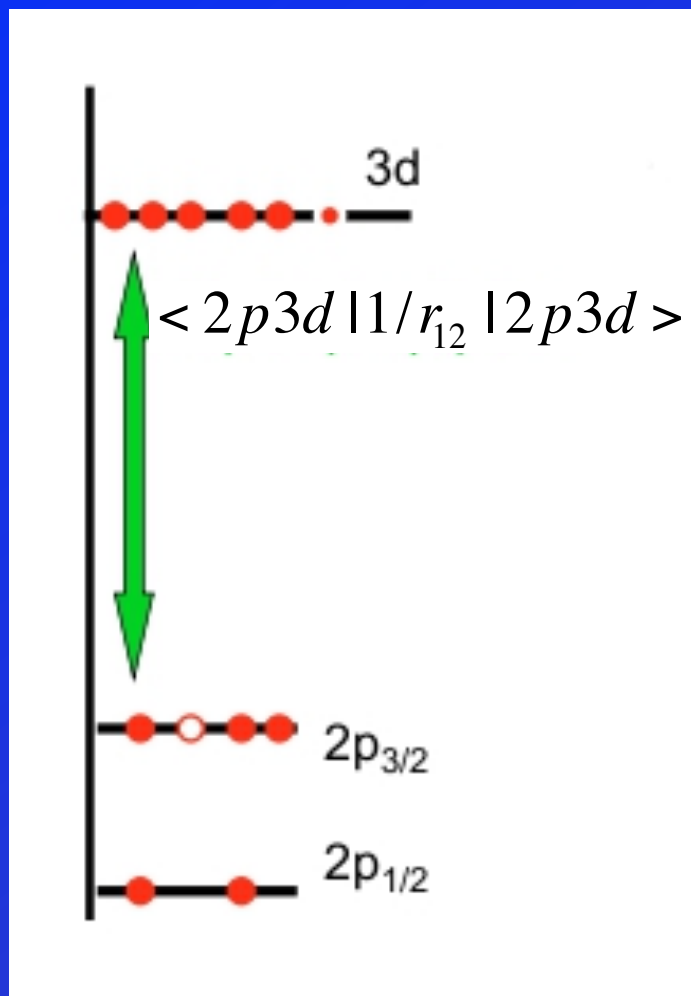


Magnetic order



# $L_{2,3}$ absorption edge

## Multiplet effects in XAS, RXS



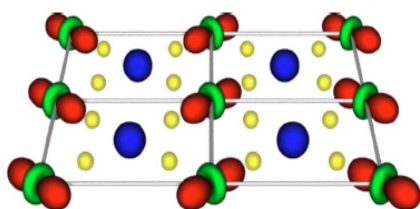
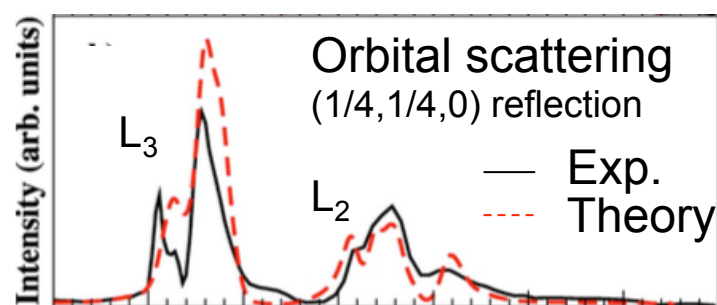
Strong overlap of core and valence wave functions

***2p → 3d dipole transitions***

$\langle 2p3d | 1/r_{12} | 2p3d \rangle$  Coulomb interaction terms cannot be neglected

**Single particle picture breaks down!**

# Resonant x-ray scattering at the Mn $L_{2,3}$ edges in $\text{La}_{0.5}\text{Sr}_{1.5}\text{MnO}_4$



Wilkins, Stojic, Beale, Binggeli *et al.*, Phys. Rev. B 71, 245102 (2005); J. Phys. Condens. Matter 18, L323 (2006).

**Mn<sup>3+</sup> atomic multiplet calculations  
in a crystal field**

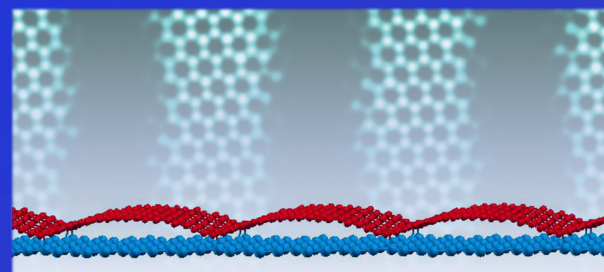
- Good general agreement with experiment
- RXS sensitive to OO and Jahn-Teller distortion

# Nanostructured Graphene

on Ir(001) surface

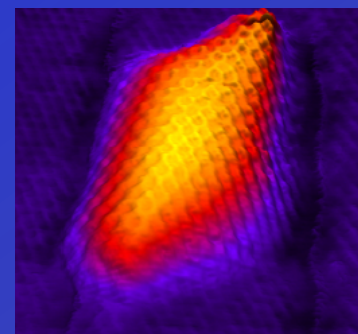
Temperature-driven reversible rippling  
and bonding of a graphene superlattice

A. Locatelli, C. Wang, C. Africh, N.  
Stojic, T. O. Montes, G. Comelli, and N.  
Binggeli, *ACS Nano* **7**, 6955 (2013)



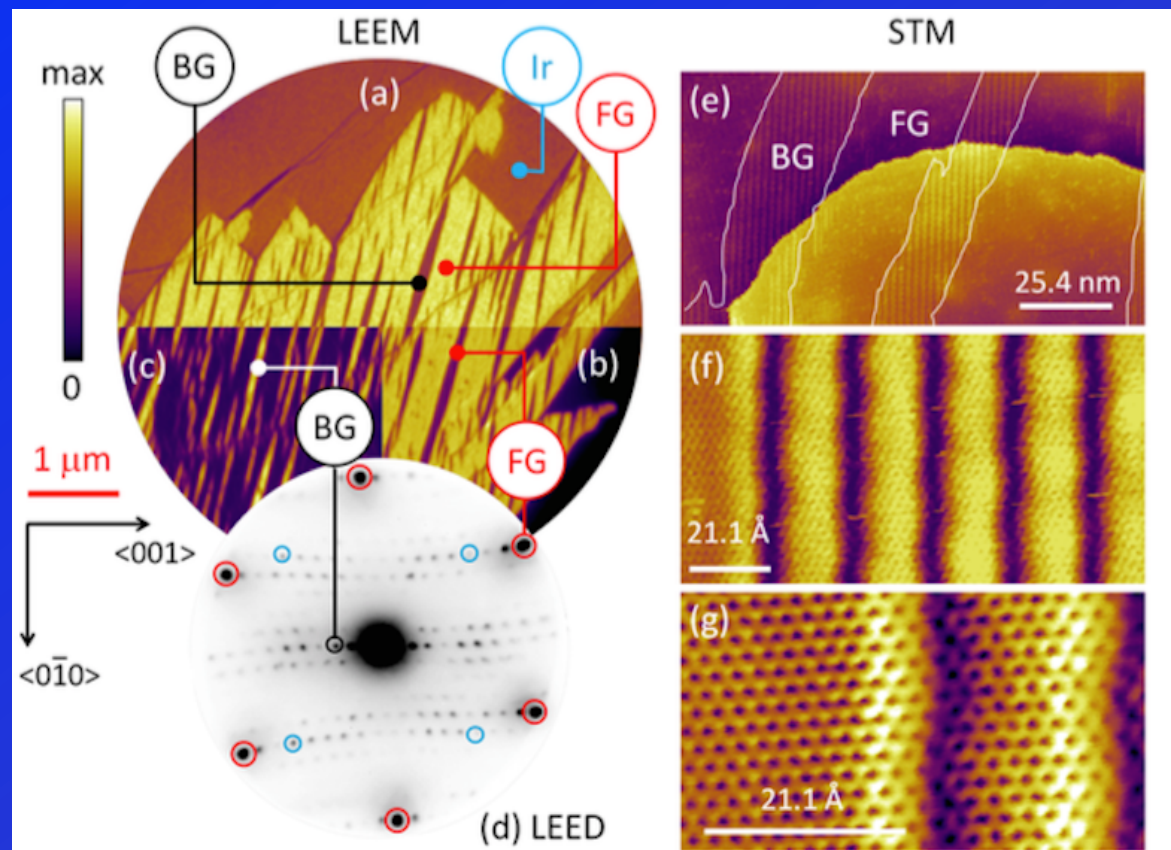
Nanobubbles at GPa Pressure under Graphene

G. Zamborlini, M. Imam, L. L. Patera, T. O.  
Montes, N. Stojic, C. Africh, A. Sala, N.  
Binggeli, G. Comelli, and A. Locatelli, *Nano  
Lett.* **15**, 6162 (2015)



# Graphene nanobuckled phase on Ir(001)

LEEM/  
LEED



STM

On Ir substrate: Buckled phase of Graphene with exceptionally large one-dimensional ripples with regular nanometer periodicity

A. Locatelli, C. Wang, C. Africh, N. Stojic, T. O. Montes, G. Comelli, and N. Binggeli, *ACS Nano* **7**, 6955 (2013)



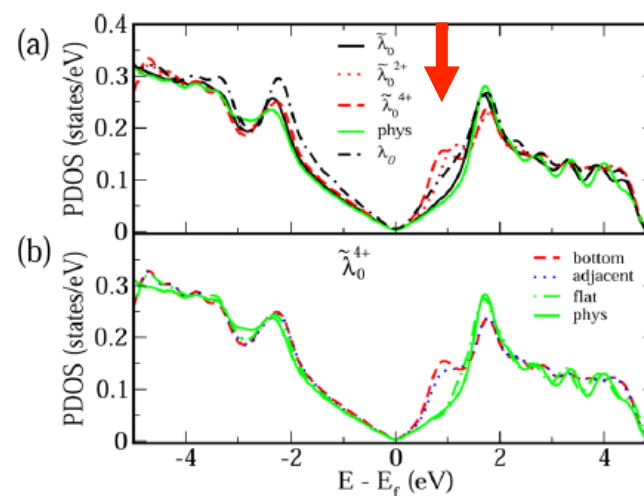
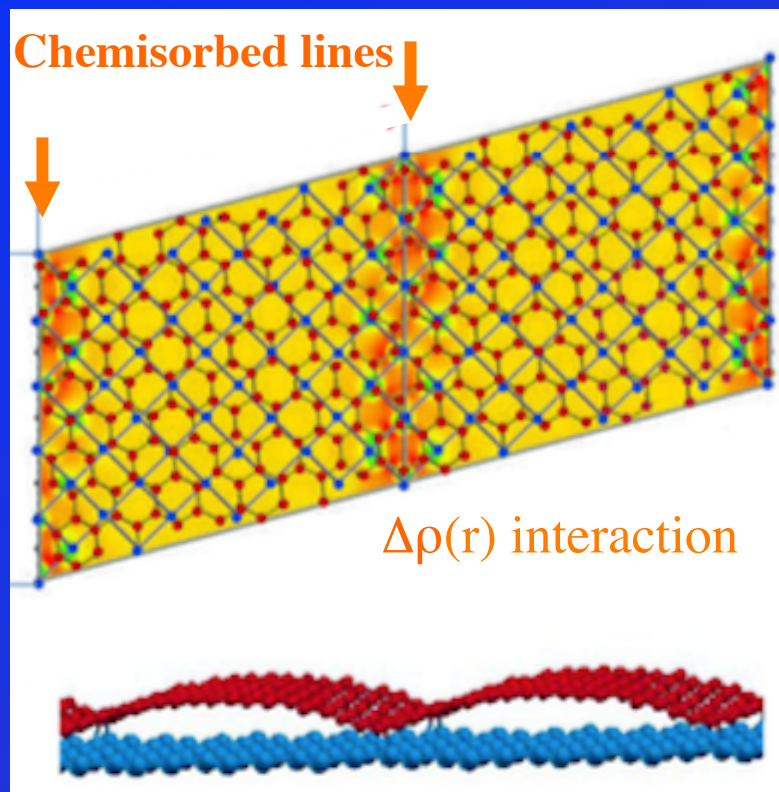
# Graphene nanobuckled phase on Ir(001)

Gr/Ir

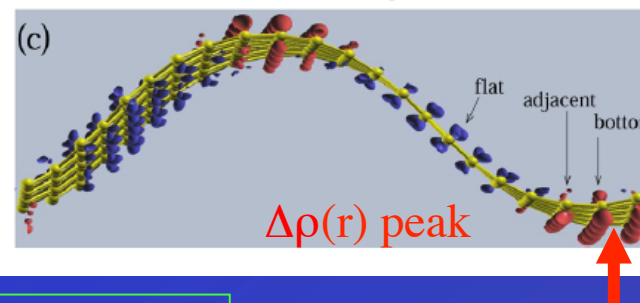
DFT

isolated Gr

Top  
view



DOS  
bottom  
C rows



Side  
view

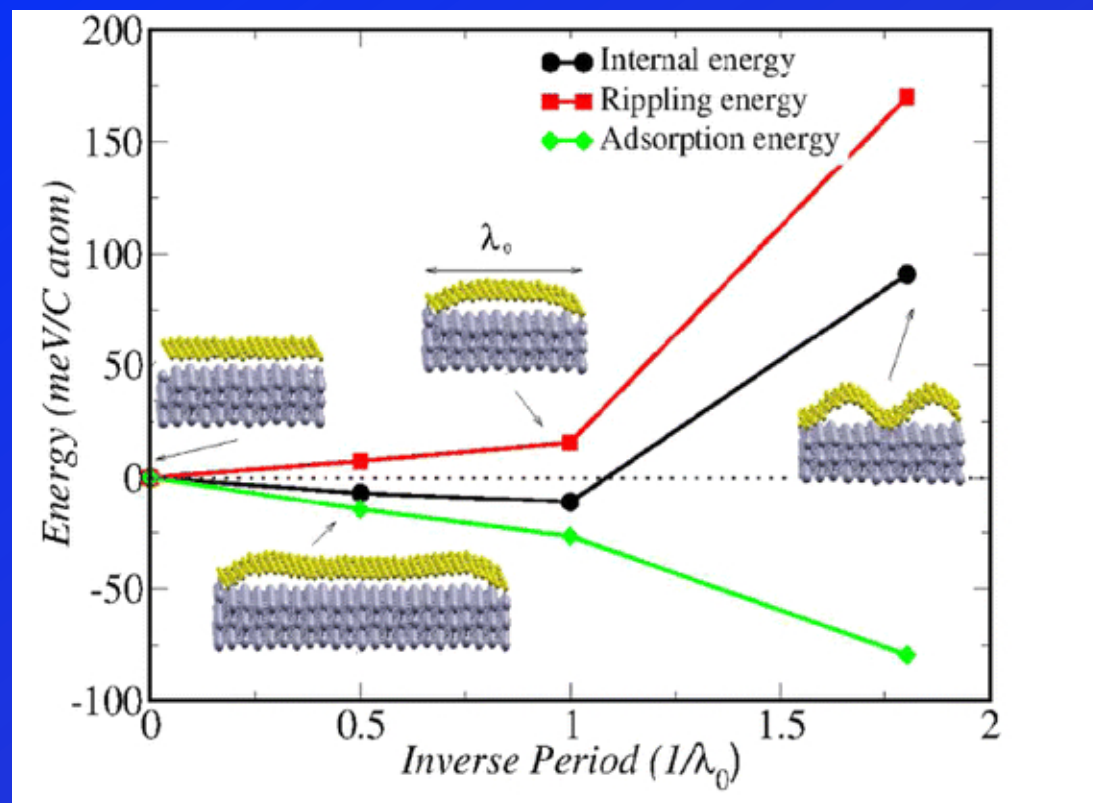
- Chemisorbed line includes  $\sim 3$  rows of C
- BG curvature induces a C-2p<sub>z</sub> DOS feature near  $E_F$
- the corresponding states are found to be largely responsible for strong local chemisorption

M. Imam, N. Stojic, and N. Binggeli, *J. Phys. Chem. C*, **118**, 9514 (2014)



# Graphene nanobuckled phase on Ir(001)

DFT



The nm periodicity is explained by the trends of the rippling and chemisorption energy

The nm period is related to the onset of the non-linear behavior of the rippling energy - which overcomes the chemisorption energy at short period

M. Imam, N. Stojic, and N. Binggeli,  
*J. Phys. Chem. C*, **118**, 9514 (2014)

# Graphene nanobubbles on Ir(001)

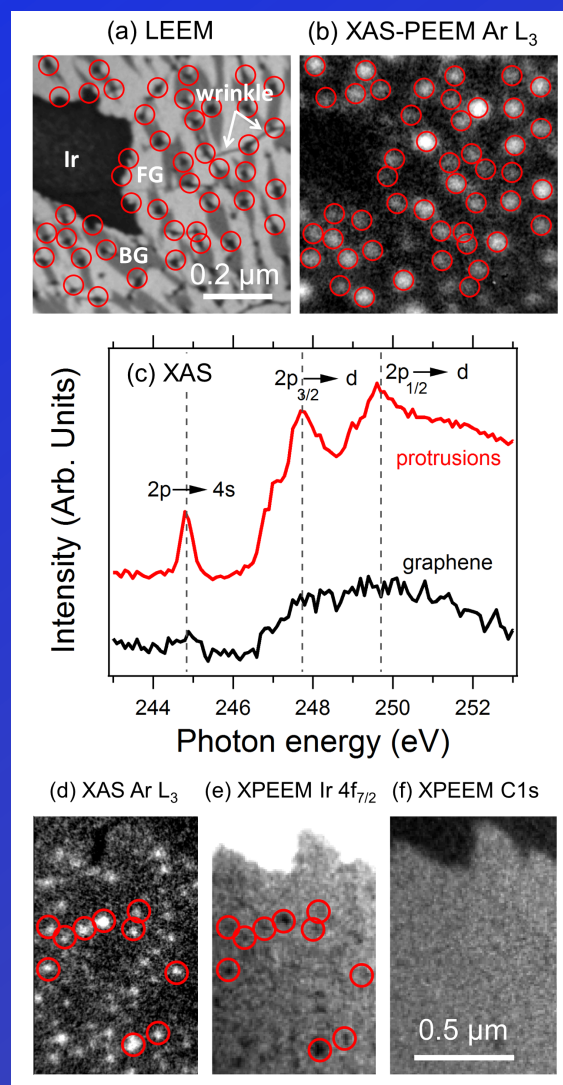
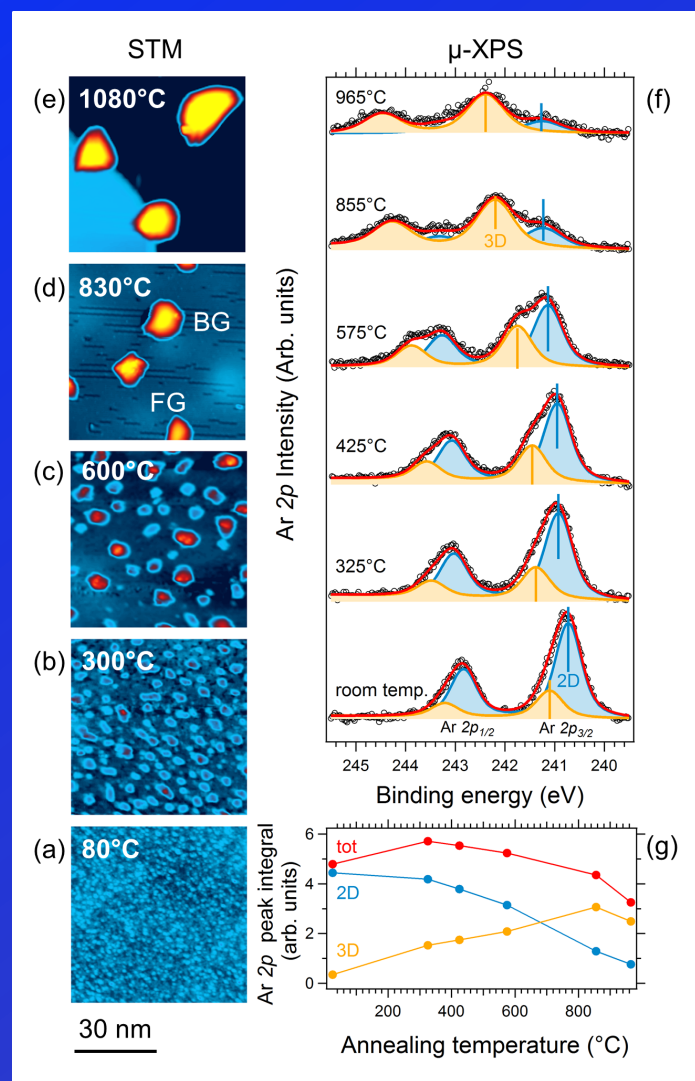
Nanospectroscopy  
group @ Elettra:

Upon irradiation  
with  $\text{Ar}^+$  ions  
(sputt. 0.1-0.4 kV)  
and subsequent  
annealing T:  
80 C, ..., 1080 C

STM @ CNR-IOM  
TASC

Trapped Ar

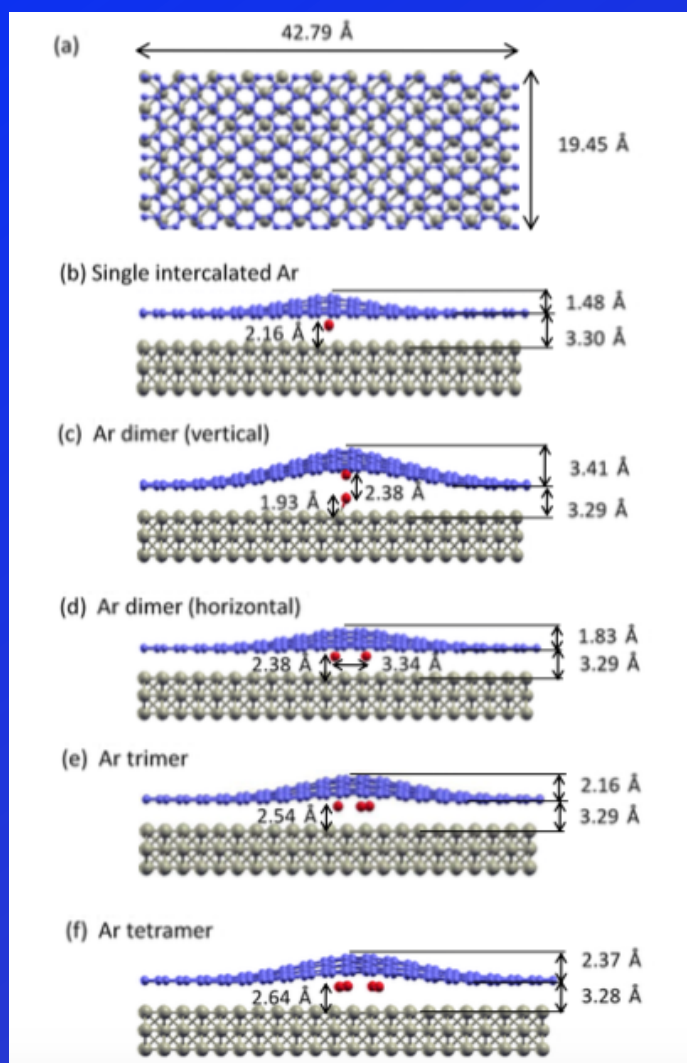
Ripening with T



# Graphene nanobubbles on Ir(001)

DFT

Equilibrium configurations



The Ar-Ar bond in intercalated cluster is strongly contracted with respect to bulk Ar (3.82 Å) or Ar dimer in vac. (3.74 Å)

The corresponding effective pressure experienced by intercalated Ar cluster is in the GPa range (8-25 GPa)

G. Zamborlini, M. Imam  
et al. *Nano Lett.* **15**, 6162 (2015)

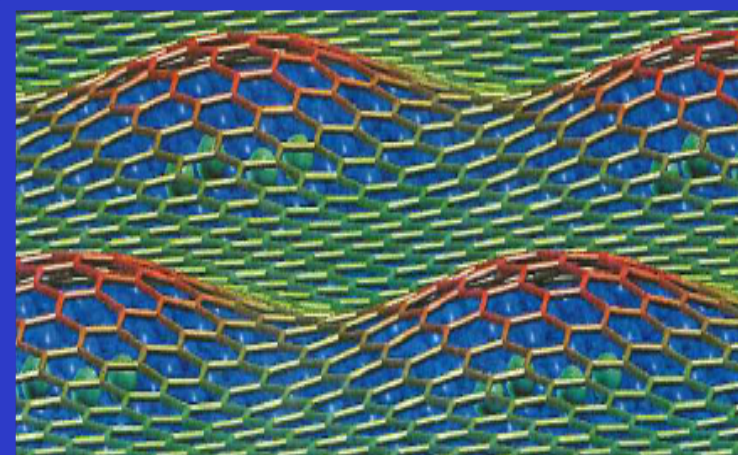
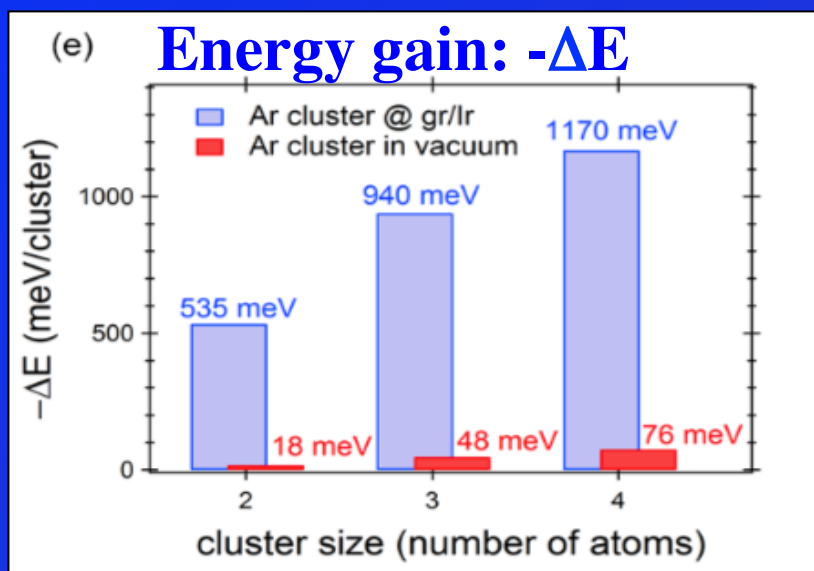
# Graphene nanobubbles on Ir(001)

Ar cluster formation energy:

$$\Delta E = E^{\text{NAr}} - N E^{\text{1Ar}}$$

G. Zamborlini, M. Imam\* et al.  
*Nano Lett.* **15**, 6162 (2015)

\* Computations by Mighfar Imam



- Drastic increase in the energy gain for Ar cluster formation at the Gr/Ir interface (compared to Ar cluster in vacuum), related to the distortion of the physisorbed graphene
- Major gain in energy when two or more bubbles merge into a single one
- The bubble formation is driven by minimization of the energy cost of membrane distortion and loss of adhesion