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Use of the IBA DataFurnace

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IBA techniques: Practical Session Use of the IBA DataFurnace

Introduction

The DataFurnace can be used for all particle scattering data (ie. RBS, EBS, ERD, NRA but *not* PIXE at present) to obtain depth profiles, and is a sophisticated code. Some acquaintance with it is also useful to all who may make use of the IBC's depth profiling facilities.

Frequently used buttons 13DF v7.1.3 Notepad Help ... Spectra Setup Run ND 88 📘 🛠 Æ Exit <mark>5 NDF Control</mark> 2 (urnac Directory: 1 Open batch file (batch/open) Filenames: Geometries: Associations Charge (uC 4 Plot spectra (spectra/plot spectra) acall->aas 3.498436 acall bcall hcall->has 3,833 5 Plot fit (viewNDF/data_fit) Associate 6 Plot structure (viewNDF/best_structure) 7 Plot profiles (viewNDF/profiles) 10 Setup NDF (setup/NDF) Link 12 Results (viewNDF/view_results) Sample # 1 💌 🕂 Structure: aunisim mple 13 Run NDF for this sample (runNDF/runSample) 15 Exit & store settings

We start with an introduction to the windows:

The buttons are accelerators. Most of the functions can be accessed through the menu (these are indicated in the box). Note also that keyboard accelerators are indicated on their menu commands.

A "sample" (#20) is defined by one or more spectra (#19). Typically we have two spectra collected simultaneously from two detectors in the chamber (see "Simultaneously collected spectra": this option, #27, can be used if the relative charges are known well enough).

The structure file (#22) restricts the state space which is searched for a solution.

The geometry file (#21) determines the geometry *associated* (#28) with each spectrum.

The batch (#17) is a set of samples. These will often be similar samples with the same geometry and structure files.



You can either open a batch file directly with "Open Batch" (#1) or switch the directory by clicking on the directory window (#18). If there is no batch file in a directory you have to start somewhere. All related spectra are gathered together into one directory. NDF only works inside a directory: there are very many output files and we use the directory structure to organise them together.

Simultaneously collected spectra						
🥨 WINDF v7.1.3	Link Associations					
Batch Spectra Setup Run NDF View NDF Window Notepad Help	Link Association: bcal1->bas [1]					
Image: Second	With Association: Scall>aas Properties C ERD with range foil, one spectrum, charges kept equal Multiple simultaneous spectra, charge ratios kept equal DK Cancel Remove Link					
Filenames: Geometries: Associations: Characteristics: acall->aas beall 28	 24 Changes in the batch file marks it "unsaved" 25 Use the "Link" button to link associations 26 Detectors A & B are linked 27 Use the right option! 28 Associate each spectrum to its geometry file 					
Sample # 1 • + Structure: aunisim Sample: Batch: 25						

Exercise 1: Calibrations

Open /examples/calib1. This is Example 1 on the website (<u>www.ee.surrey.ac.uk/ibc/ndf</u>)

The calibration sample is Au/Ni/SiO₂/Si substrate, where the metal films are very thin (\sim 1.5nm) and the oxide film is \sim 150nm thick. This sample has been discussed extensively in Jeynes *et al*, NIM **B137** (1998) 1229.

We can use the 4 elemental edges to do a linear regression to obtain the gain g (in keV/channel) and offset o (in keV) of the electronics calibration:

E = g.C + o

where E is the energy of the scattered particle and C is the channel number in which the elemental edge appears. In principle, the detector does not register the energy E but a slightly lower energy due to energy loss through the detector dead layer and other effects (this is discussed extensively in the NIM B137 reference) but we usually ignore this "pulse height defect" effect. Actually, the fact that the offset o is non-zero is due to the pulse height defect.



In the example the calibration is already done, but we will show the working.

Look at the spectra (Step 1). The elemental edges are at channel numbers 417, 358, 281, 199 (Au, Ni, Si, O edges respectively) for the spectrum from the A detector (red data), and 394, 343, 278, 204 for the B detector.

We have constructed a spreadsheet (calspc1.xls) to do the linear regression, and the values of gain and offset are entered in the *geometry file* (#21).









Using the simulator we check that the gain and offset, and the solid angle, are reasonable. For the solid angle we check that the Si signal from the SiO2 is the right height using the RoI window.





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The structure file clearly shows that the O signal runs far too deep! What is wrong? There is a perfectly good fit for the high energy part of the spectrum but somehow a stupid structure is being found! Run NDF, and *while it is running*:

🗖 ndf				
R.P. Webb, J.	Phys. D: Appl.	Phys. 36 (2003)	R97-R126 (Topical	Review).
Normal coolin 128 channels FWHM will be Isotopic dist Data will not PROFULE outpu	g convoluted ribution will b be smoothed t will not be n	e used ormalised		
Will fit 6 sa	mples			
Initialising t0 = .636E+05 LMarkov = 1 cooling = 0.5	parameters samp 467 24	le 2: acal1 bcal1		
Start simulat	ed annealing			
T = .636E+05	X2 = .159E+06	Best = .159E+06	var = 1.000000	
T = .333E+05 T = .241F+05	X2 = .344E+04 X2 = .344E+04	Best = .344E+04 Best = .344E+04	var = 0.974401 var = 0.928043	
T = .127E+05	X2 = .340E+04	Best = .340E+04	var = 0.912143	
T = .663E+04 T = .422E+04	X2 = .231E+04 X2 = .230E+04	Best = .231E+04 Best = .230E+04	var = 0.928043 var = 0.907007	
T = .221E+04 T = .116E+04	X2 = .225E+04 X2 = .215E+04	Best = .225E+04 Best = .215E+04	var = 0.912143 var = 0.855391	
T = .607E+03	X2 = .215E+04	Best = .215E+04	var = 0.804129	



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press *refresh* in the yellow screen: this shows the RoIs being used in the geometry files. The data is fitted in the RoI but not outside it (at low energy).



Now inspect the results file. Note that the solid angle ("Omega") is different from Sample 1. Actually we used different geometry files!

Direct inspection of various files						
WARNING! Changing Files this way WILL lead to inconsistancies in WINDF	aas.geo - Notepad	🗐 🖡 cal.spc - Notepad				
Look in: Celb1 Celb1 Celb1	Ele Edt Format Wew Help 1 120 120 1000 12 1000 12 1000 12 1000 12 1000 12 100 12 11 12 12 11 12	File Edit Format Yew Help acall 3.498436 aas unisi bcall 3.833 bas acall 3.498436 aa unisi bcall 3.633 bas acall 3.498436 aa unisi bcall 3.633 ba acall 3.498436 aa unisi bcall 3.633 ba acall 3.498436 aa unisi bcall 3.633 ba acall 3.498436 aal unisi bcall 3.633 ba acall 3.498436 aal unisi bcall 3.833 ba acall 3.498436 aal unisi bcall 3.833 ba bcall 3.833 ba				
Eitting with NDE, 2						



The problem now is that the fitted structure does not match what we know about the sample, that it is a layered structure with pure layers and sharp interfaces. The solution found is a *valid* solution *consistent* with the data, but we wish to know if our expected solution (4 pure layers of Au, Ni, SiO2, Si) is *also* consistent with the data. To do this we *restrict the state space* within which solutions are searched for.



So we can force NDF to find the solution we are expecting. Note that a good fit is obtained, indicating that this is a *valid* solution. IBA data are often *ambiguous*, and in this case the ambiguity is due to the limited energy resolution of the data which limits the film thicknesses that the data can determine. We have effectively told the code that very thin films are present which our systems cannot prove.

Ď cal04.prf - Notepad		×
<u>File E</u> dit F <u>o</u> rmat <u>V</u> iew <u>H</u> elp		
4 29.02 0.0000 0.0000 0.0000 100.0000 27.05 0.0000 0.0000 100.0000 0.0000 1513.41 0.0000 100.0000 0.0000 122914.01 100.0000 0.0000 0.0000 thickness(at/cm2), comp_molecules molecules 4 4.977 si 6.000 silo26 9.125 Ni 5.904 Au The IBA Data Furnace v7.9c 22Jul04 run 03/04/2005 09:28:14 Filename: batch cal.spc str aunisi3.str	cal04.pr [.]	
	>	1.1

You can *look* at the result file using the Notepad editor, and the results are output in a form convenient for putting straight back into the simulator. We can therefore simply store this file as NDF.PRF, and run the simulation.

The point now is that the calibration sample structure is very well known and we want only to verify the calibration parameters. So we don't actually need to do a full fitting, which does not use any knowledge of the sample (except what is in the structure file); a simple local minimisation will do instead. This is much quicker (and needs less attention to the details of the structure file).





Pure simulation can be used to design analyses and to see what is feasible. Play with it!

Exercise 2: Analysing a batch of data

Open /examples/crest2. This is example 2 on the website <u>www.ee.surrey.ac.uk/ibc/ndf</u> and has been discussed extensively in the *Topical Review*: Jeynes *et al* J.Phys.D 36 (2003) R97

It was the result of a Masterclass involving eighteen year old school leavers, where they analysed an.d wrote up a set of RBS data, which was subsequently published:

A.Belson, D.Brasted, C.Dawes, S.Mashford, H.Sunnucks, J.S.Sharpe, C.N.McKinty, M.Kerford, M.A.Lourenço, A. Kewell, T.Butler, K.P.Homewood, C.Jeynes, R.P.Webb, K.J.Reeson Kirkby, *Ion Beam Synthesised FeSi*₂ – *development of band gap and structure during annealing*, CREST Masterclass project: Presented at ESPRIT Advanced Research Initiative in Microelectronics (MEL-ARI) Athens, October 1999



(This is copied into the document using file/copy on the main menu.)

It shows two spectra collected simultaneously as before, of annealed iron implants in silicon, in a batch of 12 samples with various anneals and implant treatments.



The partial spectra are shown (datafit window menu: file/open-separated-spectra): Fe-green, Si-blue, O-orange. Note the slight channelling below ch.150.

In this case the Fe and Si signals overlap dramatically, making the depth profiles quite hard to extract. We will demonstrate how DataFurnace solves these spectra very easily as a batch.

With an appropriate (not very restrictive) structure file and the correct calibration, the whole batch can be run (runNDF/run_batch).

The calibration (gain & offset) is taken initially from the previous exercise (see figure). But they are changed to allow NDF to fit the O signal accurately.

In this case the only well known energy in the spectra is the Si edge (since the Fe is buried and the O signal is very weak). Therefore, NDF does not have sufficient information to adjust the gain & offset, and we hold the gain constant. Because the fits are very sensitive to elemental edge (sharply rising) signals we allow the offset to vary. This will be determined by the Si edge signal.



The solid angles are also taken from the previous spectra, but in general (and in these in particular) the solid-angle.charge product is determined by the spectra, and so we allow the charge to vary. It turns out that the charge collection for this set of spectra is sometimes not very good (compare the collected charges in batch file calib.spc with the fitted charges in c2.spc, most obviously sample 8 in c2.spc).



