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#### Joint ICTP-IAEA Advanced Workshop on Multi-Scale Modelling for Characterization and Basic Understanding of Radiation Damage Mechanisms in Materials

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Relating modeled positron response to experiment

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### **Relating modeled positron response to experiment**

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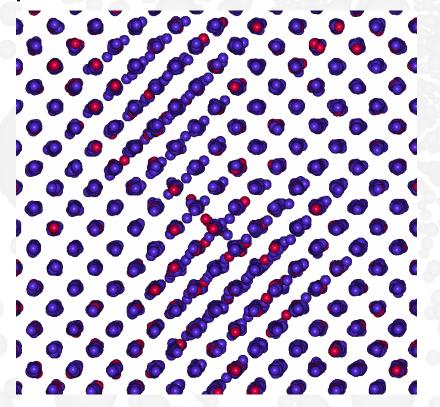
### 2 Outline

Cascades in Fe-Cr system

Fe-Cu system

Conclusions

2, 10, 15 and 20 keV cascades studied (produced in SCK•CEN, Mol) using an EAM potential.

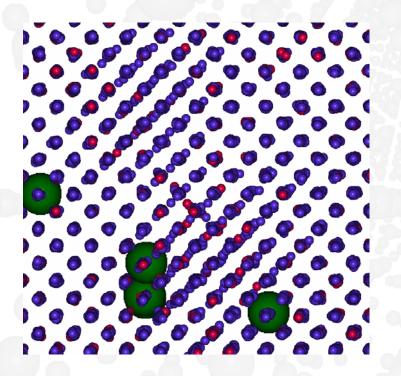


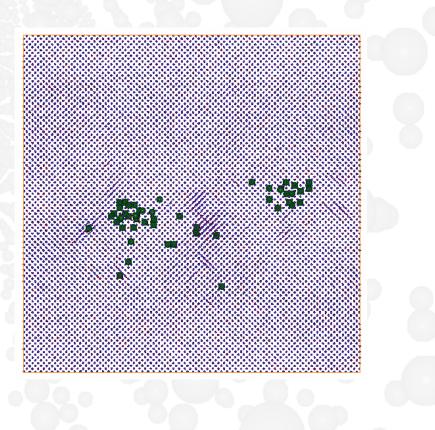
20 keV cascade

defect structure described in D.A. Terentyev et al., JNM **349** (2006) 119

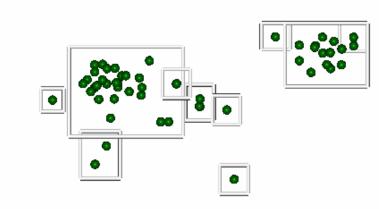
- Free volumes in simulation boxes are found using an automated procedure:
  - a regular 3D mesh (spacing 0.3 Å) is created inside the box
  - all mesh points are scanned and those having distance larger than 1.7 Å to all atoms are marked
  - neighboring marked mesh points are joined and create an open volume defect in the lattice
  - a cut around each open volume defect is taken and parameters for positron calculations are generated
  - cuts corresponding to neighboring open volume defects are joined

Results of free volume analysis (20 keV cascade):



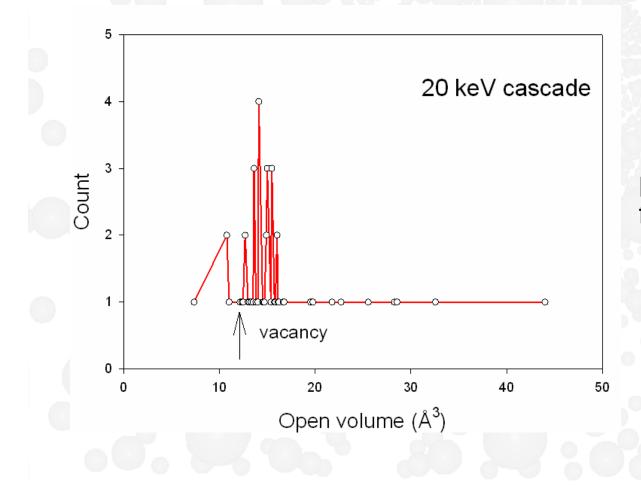


Cuts around free volumes:





Open volume distribution

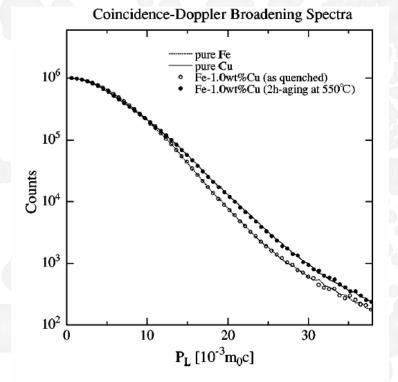


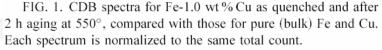
positron lifetimes range from 150 to 220 ps

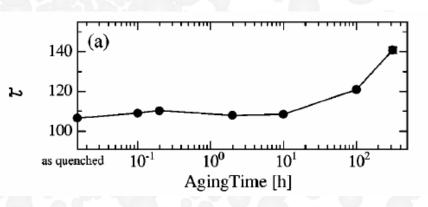
The open volume analysis shows vacancies and small clusters.

But under normal conditions vacancies are very mobile and are not stable.

- Cu clusters in Fe represent a nice and simple system to study positron trapping to precipitates.
  - PA experimental study by Y. Nagai et al., PRB 61 (2000) 6574.



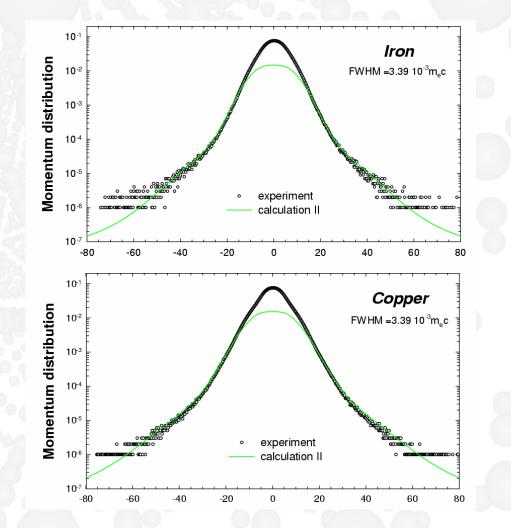




Aging dependence of positron lifetime.

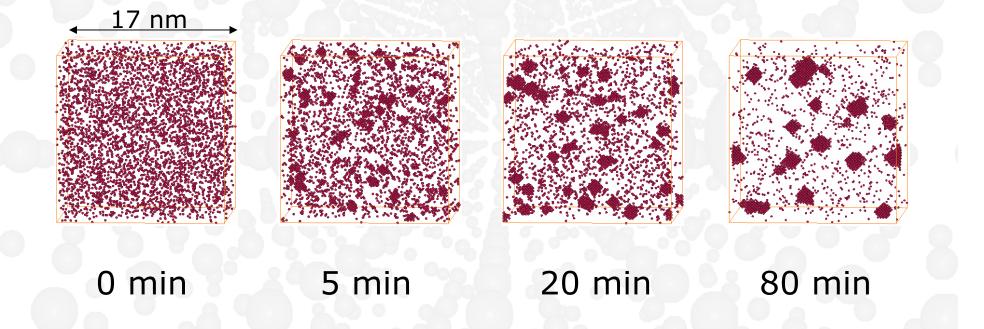
PA CDB experiment:

> pure Fe, Cu – parameter `tunning'

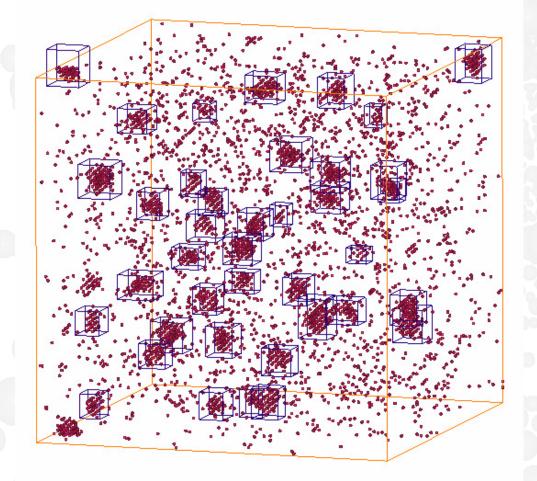


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AKMC simulations (SCK•CEN, Mol)
(1 at%Cu alloy, 60 × 60 × 60 bcc cells, 432000 atoms, 500 °C, Ludwig-Farkas potential, up to 6 × 10<sup>9</sup> steps, rigid lattice):



Simulations boxes too big for ATSUP:

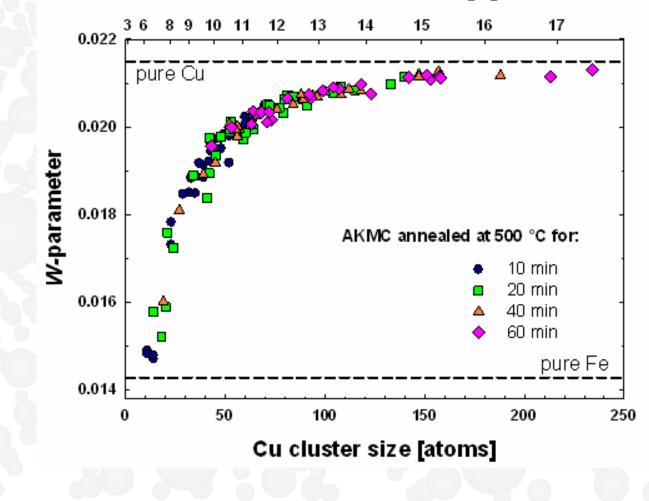


Automatic routine how to select cuts around clusters.

- find all Cu atoms
- determine clusters
  - by 1nn relationship
- make rectangular cuts around clusters
- cuts are `aligned' with the lattice

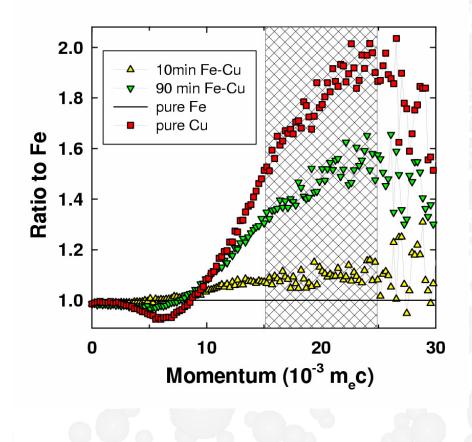
#### Calculation of W-parameter for all snapshopts

Cu cluster diameter [Å]



 $W = \int_{p_2}^{p_2} \rho(p) dp$ p<sub>1</sub> = 15 × 10<sup>-3</sup> m<sub>e</sub>c p<sub>2</sub> = 25 × 10<sup>-3</sup> m<sub>e</sub>c

Simulation vs experiment: specific trapping rate estimate



10 min annealed sample  $W_{exp} = 0.0153$ 

$$W_{\text{exp}} = (1 - \eta) W_{\text{Fe}} + \eta W_{n-\text{Cu}}$$

Simulations:  $n = 45; c = 9 \times 10^{18} \text{ cm}^{-3}$  $W_{45-\text{Cu}} \approx 0.0195$ 

*η* ≈ 21 %

- Simulation vs experiment: specific trapping rate estimate
  - Trapping rate:

 $\eta = \kappa / (\kappa + \lambda_{Fe}) \rightarrow \kappa \approx 2.5 \text{ ns}^{-1}$ 

Specific trapping rate:  $\kappa = \nu c \rightarrow \nu \approx 2.8 \times 10^{-10} \text{ cm}^3/\text{s}$ 

Specific trapping rate for single vacancies in Fe:  $v_{\text{Fe-V}} = 1.3 \times 10^{-8} \text{ cm}^3/\text{s}$ 

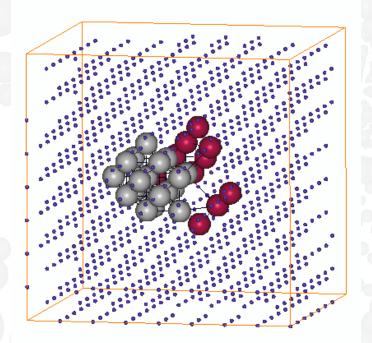
 $v << v_{Fe-V}$  (transition limited trapping regime?)

Supposing the process is 'diffusion limited'  $\kappa = 4\pi RcD_+$  ( $v = 4\pi RD_+$ ) using  $D_+ = 1.0$  cm<sup>2</sup>/s and R(45) = 5.0 Å we get

 $\kappa = 5.6 \times 10^3 \text{ ns}^{-1}$   $v \approx 6.3 \times 10^{-7} \text{ cm}^3/\text{s}$ , which strongly contradicts previous values and the process of trapping to Cu precipitates should be 'transition limited'

In this way, the whole dependence v(annealing time) = v(cluster size) could be evaluated.

Medium size Cu-V clusters have been also studied (simulations performed in SCK•CEN, Mol).

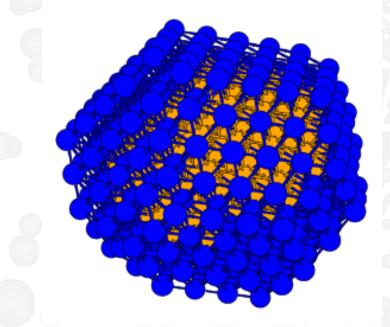


N(Cu) = 10, N(V) = 20

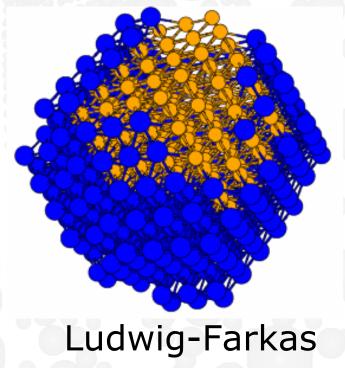
N(Cu)=40, N(V)=10

Complete vs incomplete 'coverage'?

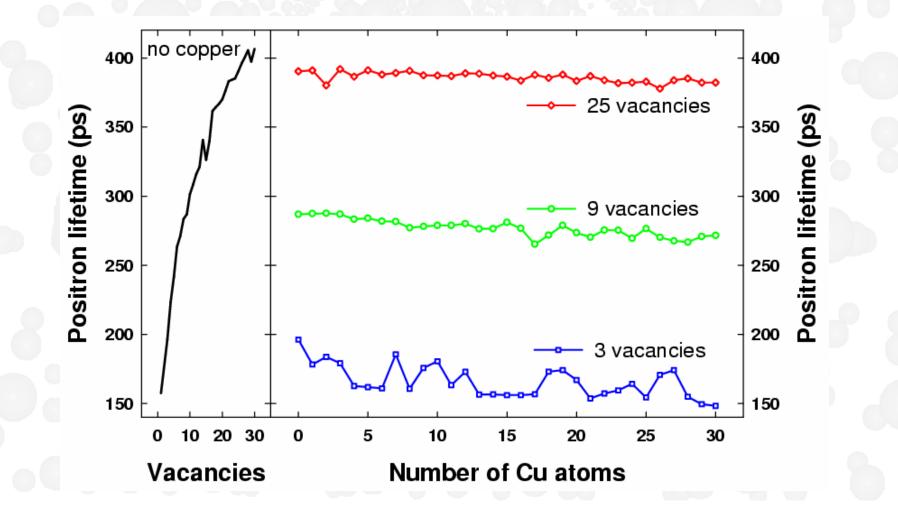
N(Cu) = 300, N(V) = 100



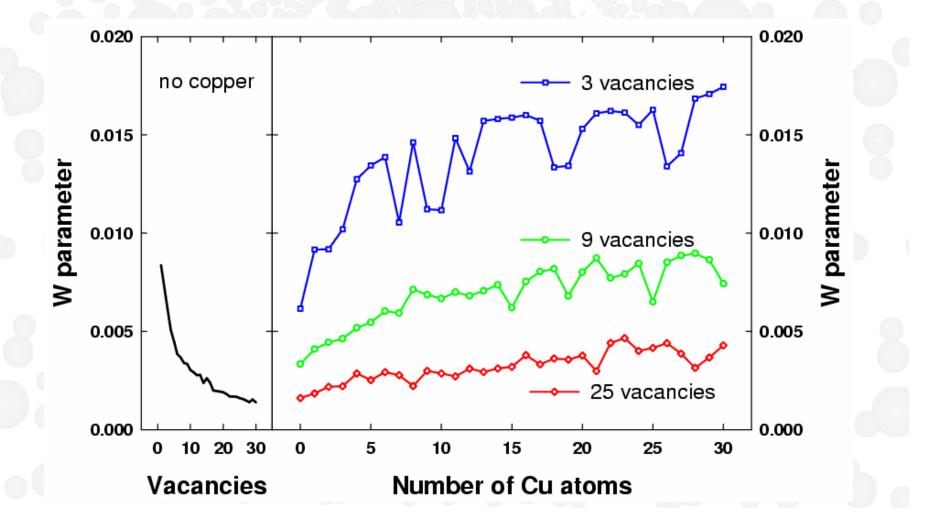
Ackland-Bacon



Calculated dependencies of lifetime on N(Cu) for given N(V).



Calculated dependencies of W-parameter on N(Cu) for given N(V).



Problems with simulations: in some cases there are two Cu-VC clusters in the simulation box.

To our knowledge, there is no experimental study addressing the problem of Cu atoms in the vicinity of vacancy clusters.

Further work is in progress.

### Conclusions

 Irradiation does really produces vacancies and their clusters in Fe-Cr system.

- Calculated lifetimes agree with those seen in experiment.
- Cascades should be simulated considering realistic microstructure (dislocations, GBs, clusters, etc.).

### Conclusions

- Cu precipitations in Fe can be effectively studied with positrons.
- The question of positron trapping into Cu clusters requires further study.
- By combining structure simulations, positron simulations with experiment, average cluster size can be determined.
- The problem of vacancy cluster decoration with Cu atoms can be also addressed with the help of simulations.

