



**The Abdus Salam  
International Centre for Theoretical Physics**



**2137-51**

**Joint ICTP-IAEA Advanced Workshop on Multi-Scale Modelling for  
Characterization and Basic Understanding of Radiation Damage  
Mechanisms in Materials**

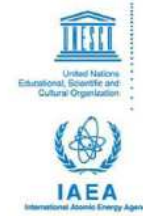
*12 - 23 April 2010*

**Dislocation dynamics simulations of post-irradiation plastic deformation in austenitic  
stainless steels**

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Miramare – Trieste, Italy

Christian Robertson  
CEA-Saclay France  
DEN/DMN/SRMA

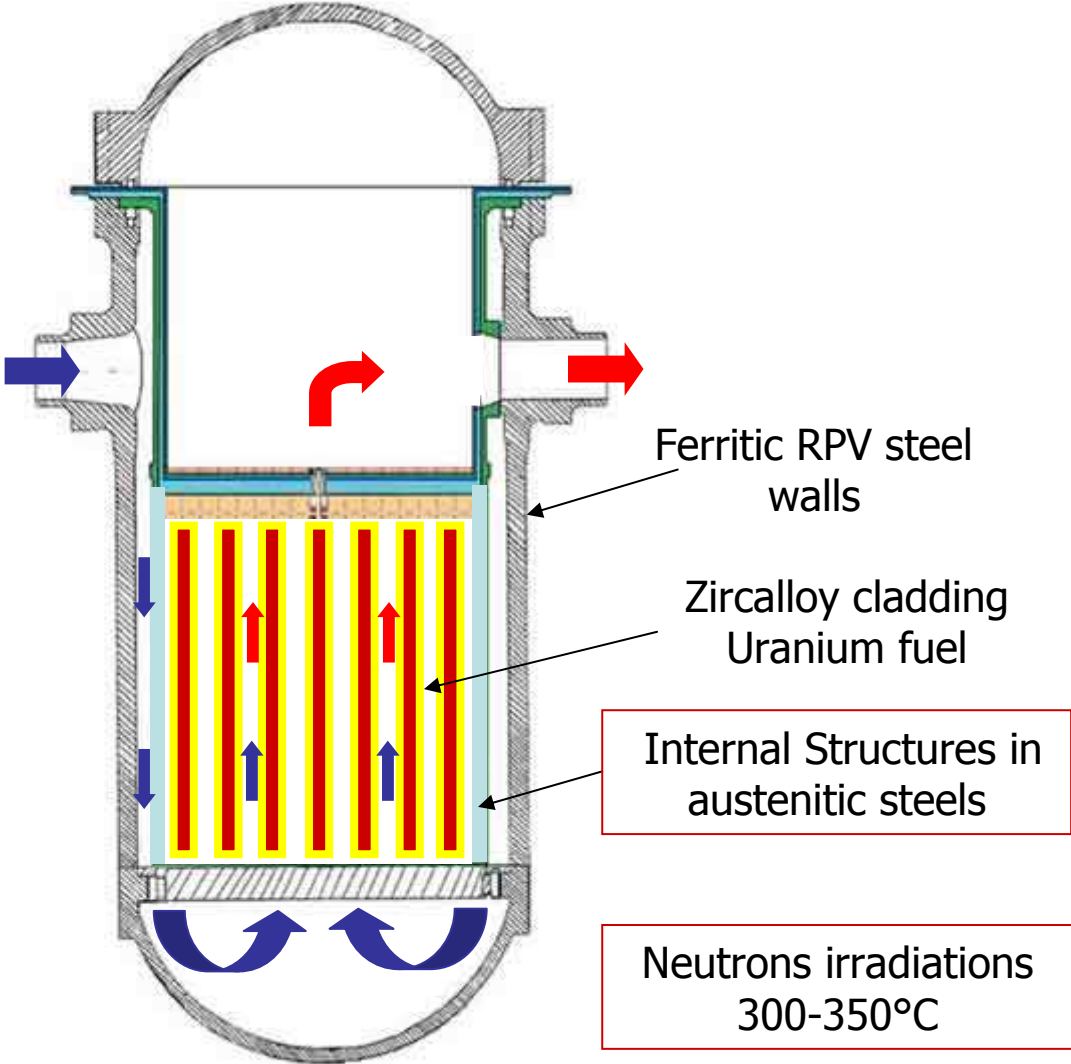
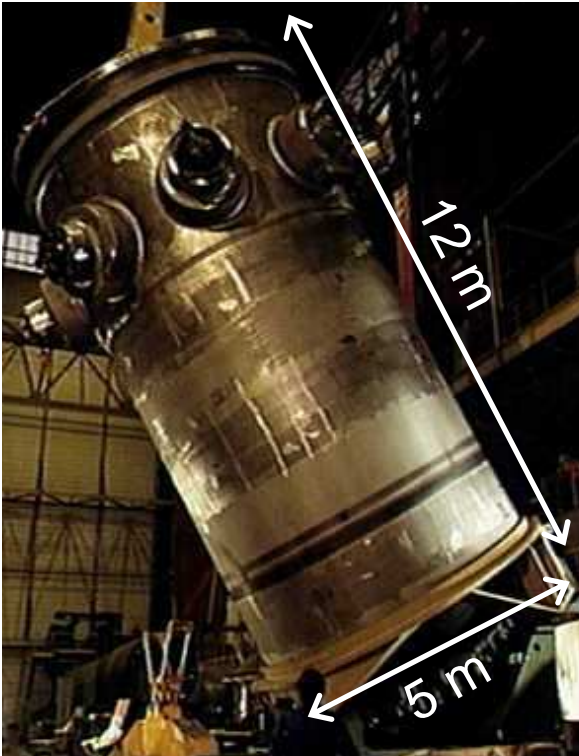
[christian.robertson@cea.fr](mailto:christian.robertson@cea.fr)

DD simulations of post-irradiation plastic deformation in austenitic steels

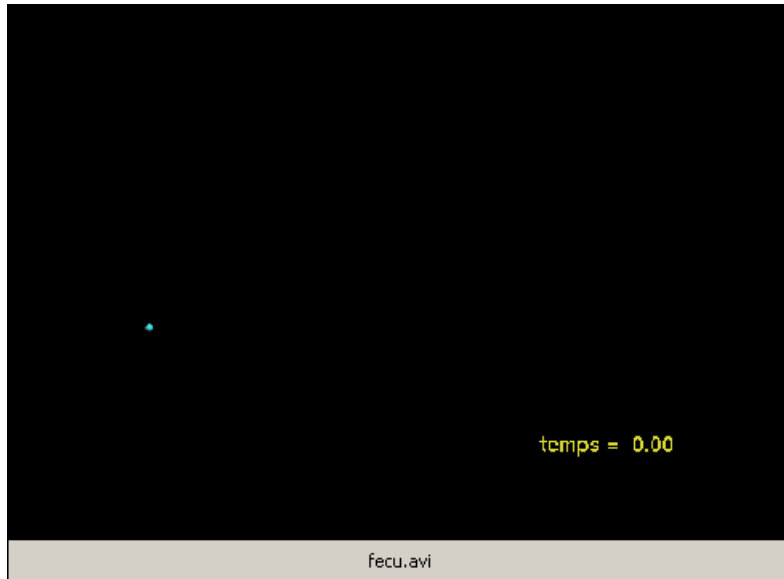


# Irradiation-induced modifications in different metallic alloys

RPV in Pressurized Water Reactors



# What happens, during irradiation of metals?

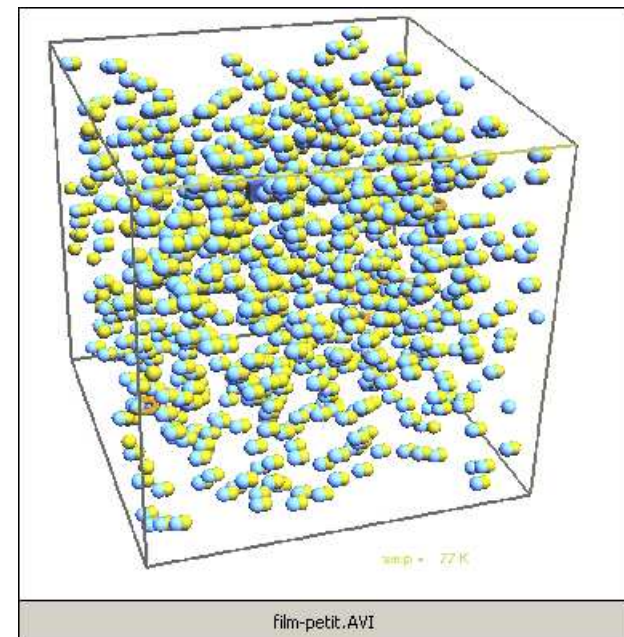
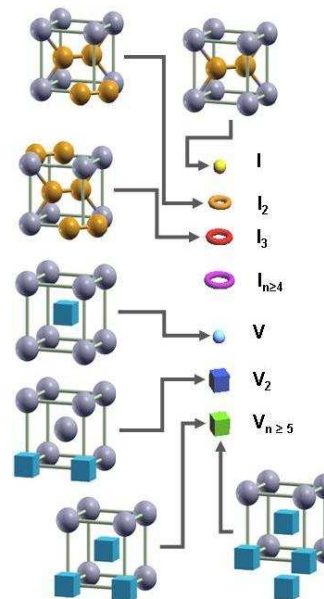


Short term evolutions:  
collision cascades (MD)

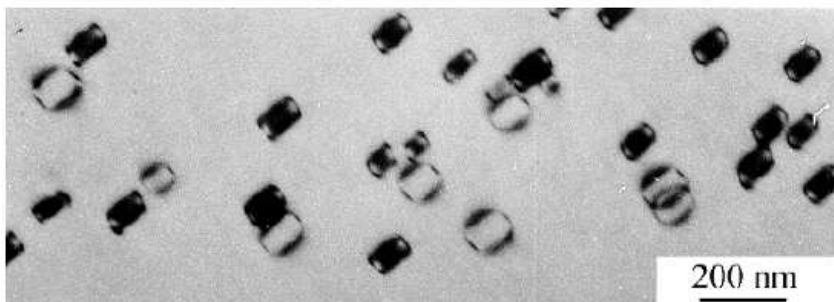
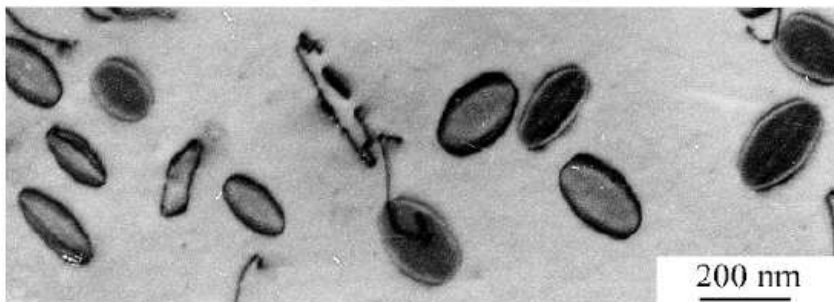
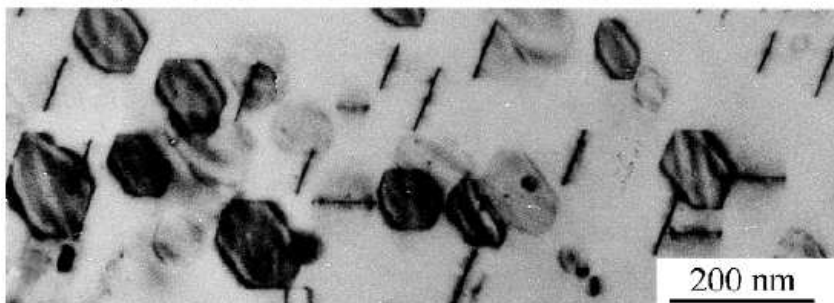
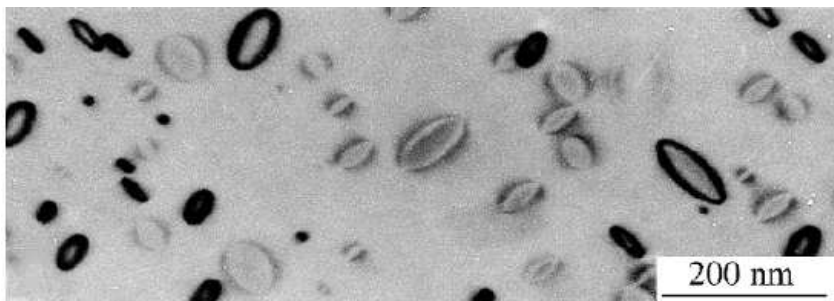
Courtesy CEA/SRMP

Long term evolutions: diffusion,  
germination, growth of point  
defect cluster (KMC)

Courtesy CEA/SRMP

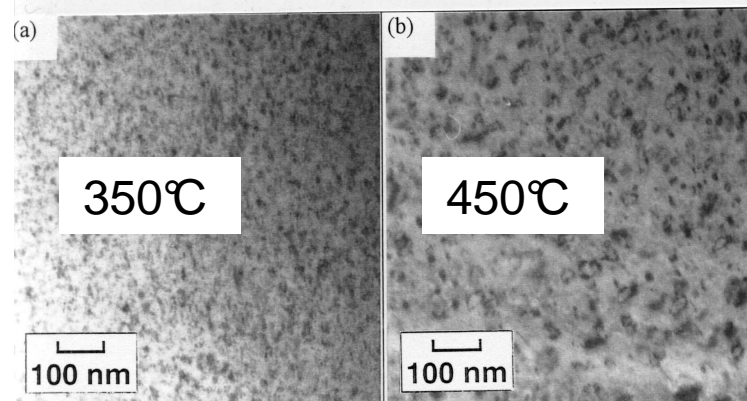


# Irradiation defect loops in various metals

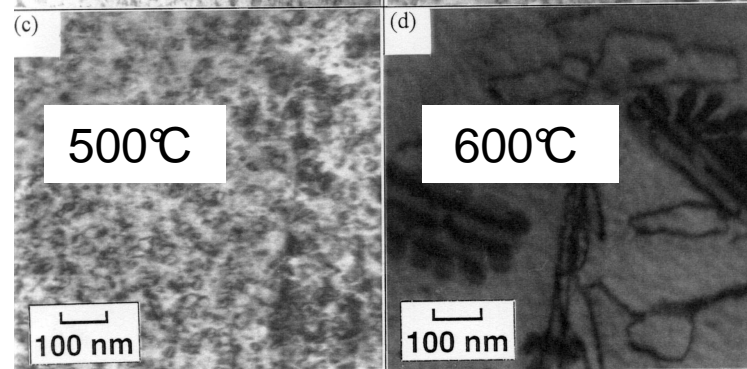


Austenitic steel, 3 dpa

Al



Cu



Ni

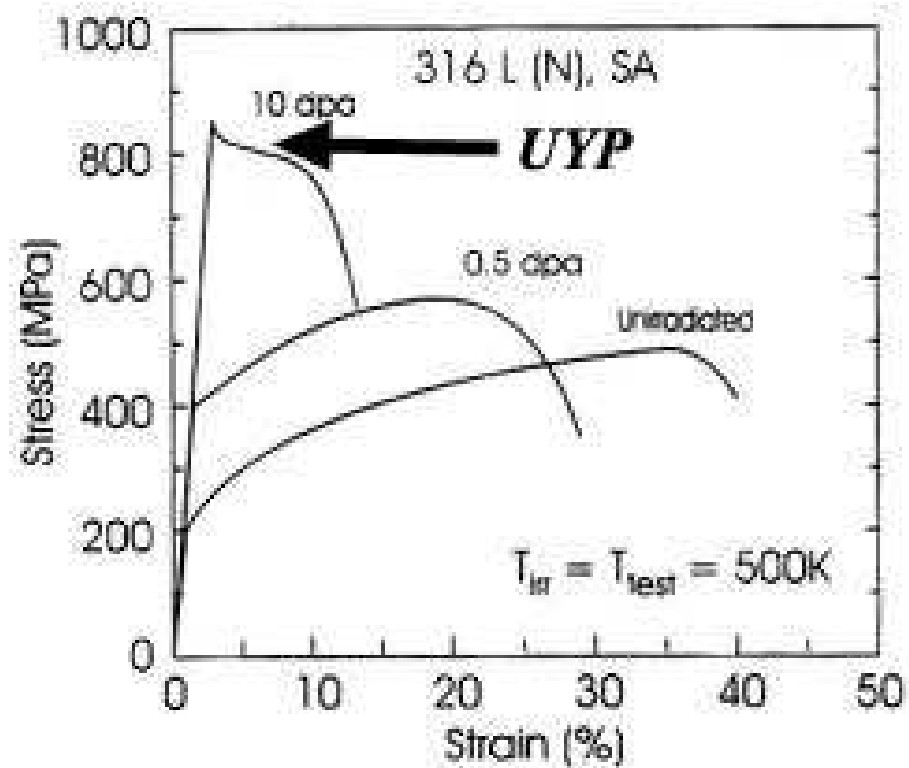
Irradiation 10 dpa  
(Kiritani 1994)

Fe

# Consequence of loops in tensile properties of metals

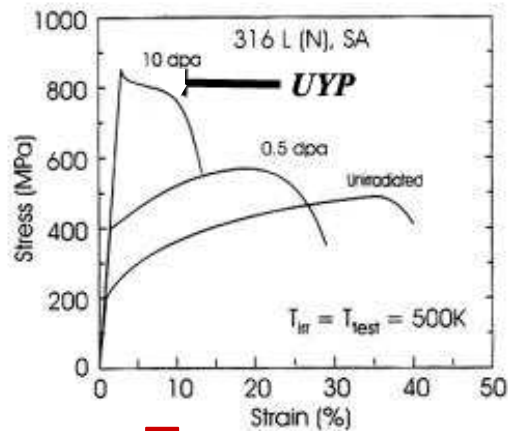
## Irradiated Austenitic Stainless Steels

Dose dependent reduction of ductility  
Hardening (< 3 dpa)  
Hardening/Softening instability(> 3 dpa)

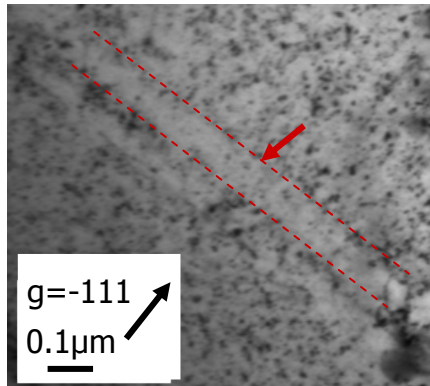


# Post irradiation plasticity: a multi-scale phenomena

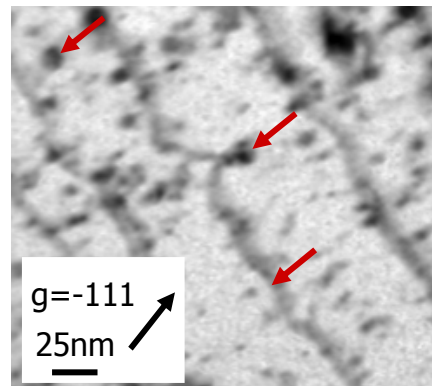
## Irradiated Austenitic Stainless Steels



- Millimeter  
Dose dependent reduction of ductility  
Hardening  
Hardening/Softening ( $> 3$  dpa)



- Micrometer  
Strain localisation into clear bands



- Nanometer  
Interstitial Frank loops  
Dislocation/loop interactions

- Q1

How works the observed strain localisation mechanisms?

- Q2

The reason of the hardening/softening observed at high dose ( $> 3$  dpa) ?

- Approach

Step1: Observation

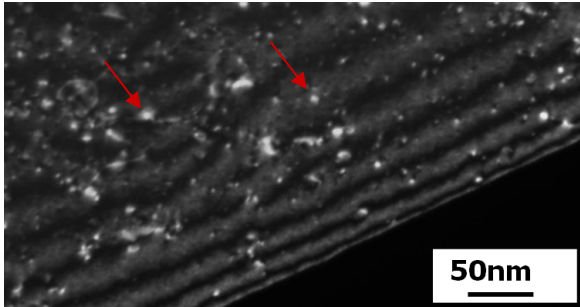
Step2: Analysis with the help of simulations



# Step1- Plastic deformation after irradiation: observations

Goal: Accurate information on dislocation/defect cluster interactions

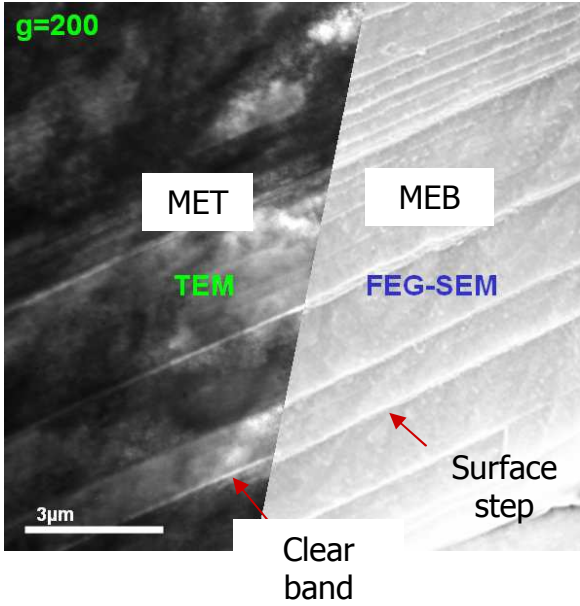
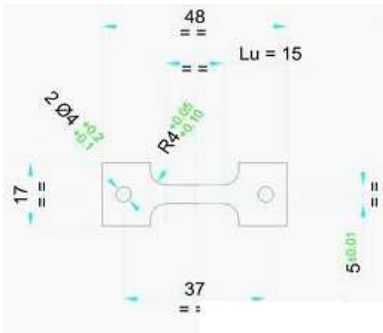
Irradiations:  $T_{irr} = 350^{\circ}\text{C}$   
Ions Xe 95 MeV  
2-5 dpa - 7  $\mu\text{m}$  thick



Frank loops  
Mean diameter  $\sim 6\text{-}10\text{ nm}$   
Density  $\sim 10^{22}\text{ m}^{-3}$   
Representative of saturation density

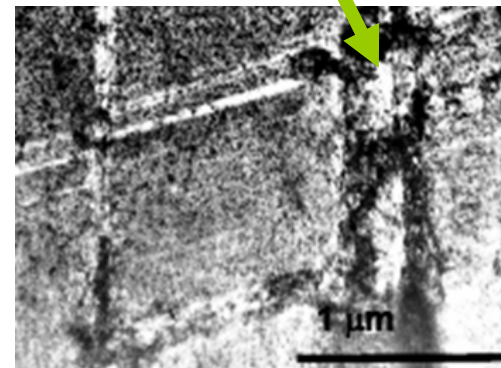
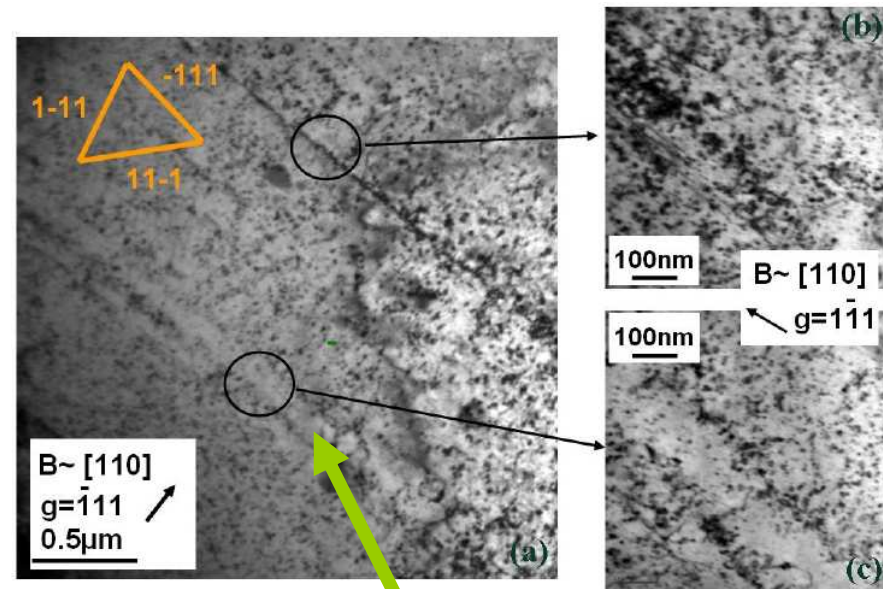
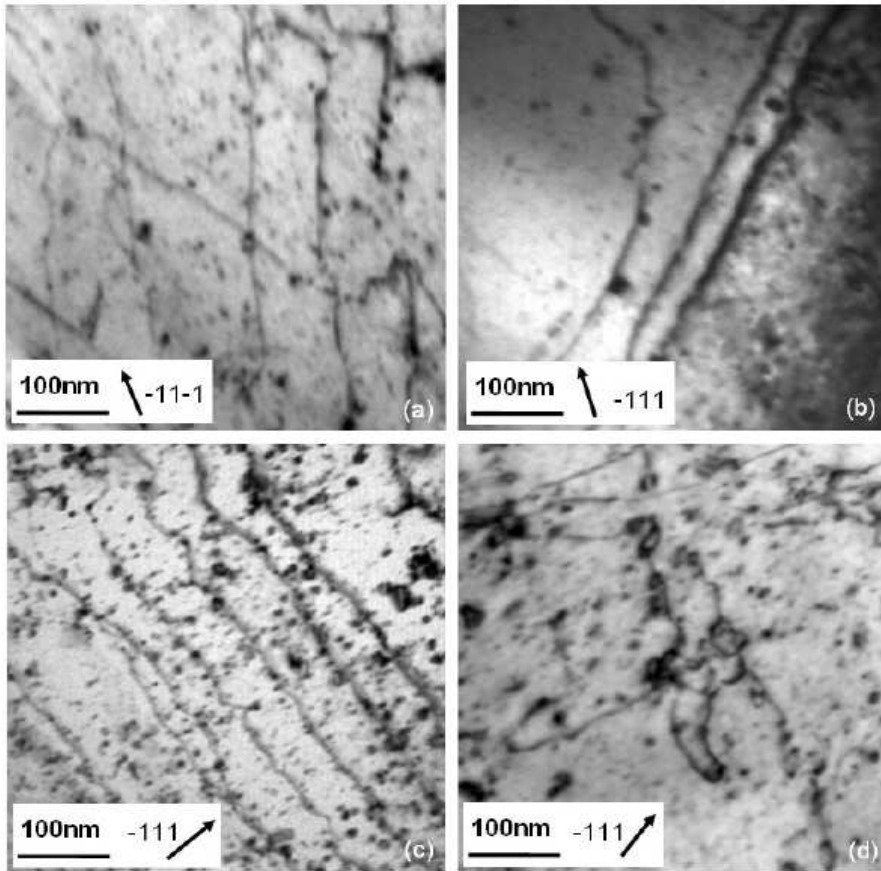
Foil preparation: back-side electrocpolishing

Deformation: Uni-axial Tension  
 $T_{def} = 350^{\circ}\text{C}$   
 $\epsilon = 8\%$   
 $\dot{\epsilon} = 10^{-4}\text{ s}^{-1}$



Strain localisation

# Plastic deformation after irradiation: observations



(K. Farrell ORNL)

Neutrons Irradiations ( $\sim$  cm)

Mostly screw type pile-ups  
Jog size  $\sim$  Irradiation loop size

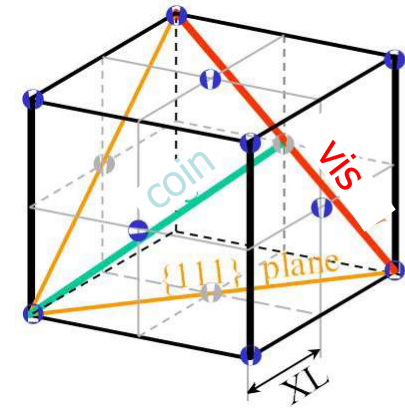
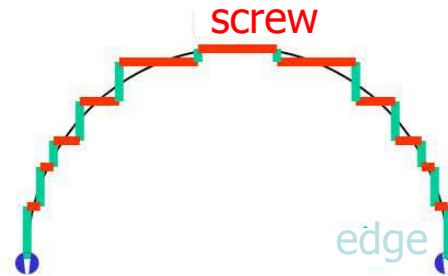
# Dislocation dynamics

Dislocation model

12 slip systems for fcc crystals

Edge and screw segments

Discrete lattice



Lattice spacing and time step

Nanometric loops

DD	« micro »	« nano »
XL	10 b	0.08 b
dt	$5 \cdot 10^{-10}$ s	$5 \cdot 10^{-14}$ s

Interaction rules

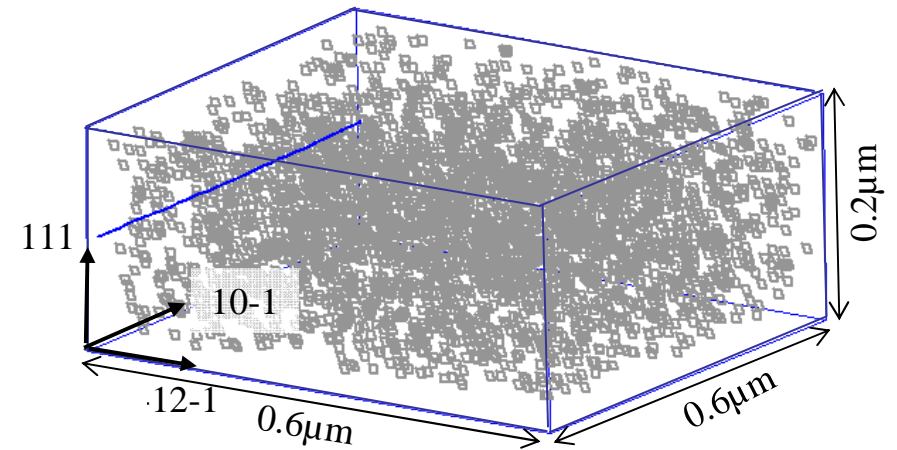
~~Cross-slip~~ - Climb - ~~Obstacle~~ - Recombinaison - Jonction

Model fcc crystal: Cu elastic parameters

## Examined configuration

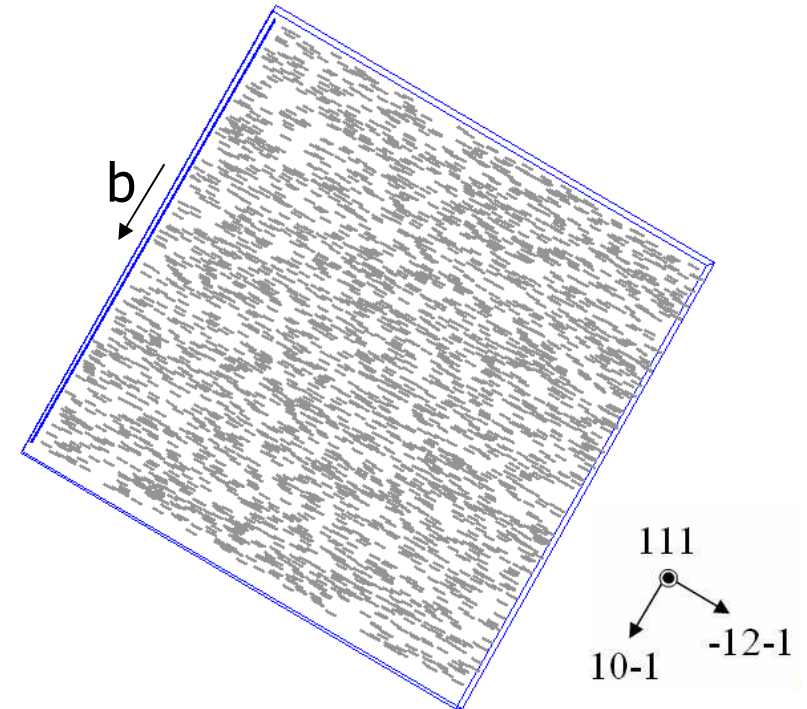
### Frank loop population

- Random position
- $D=10\text{nm}$ ,  $L=50\text{nm}$



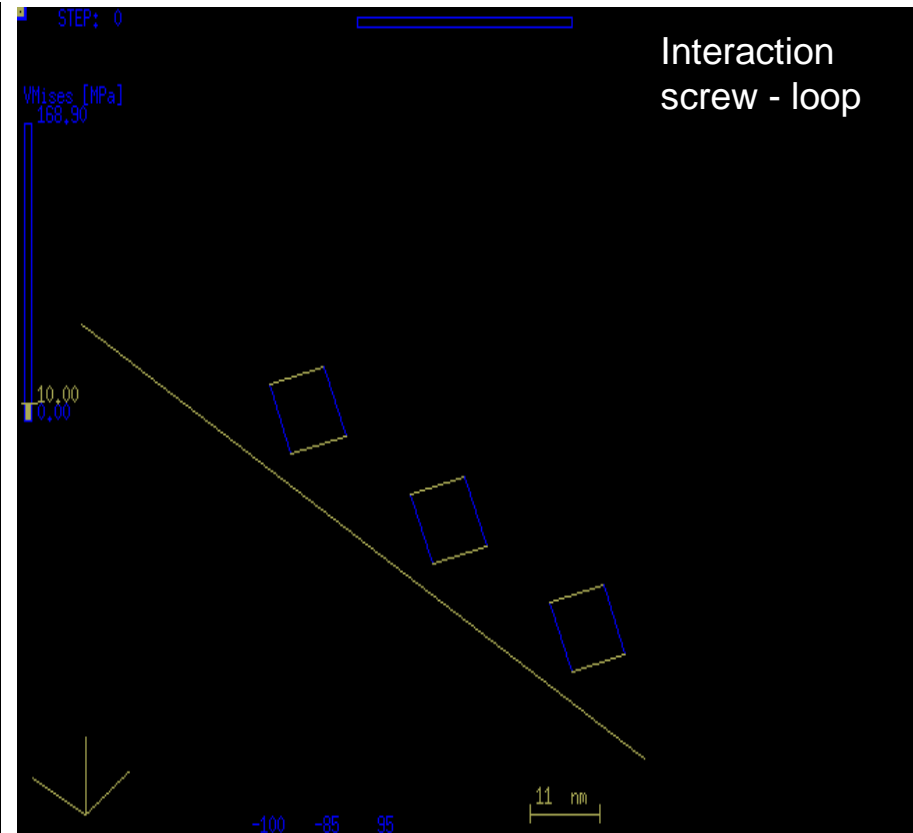
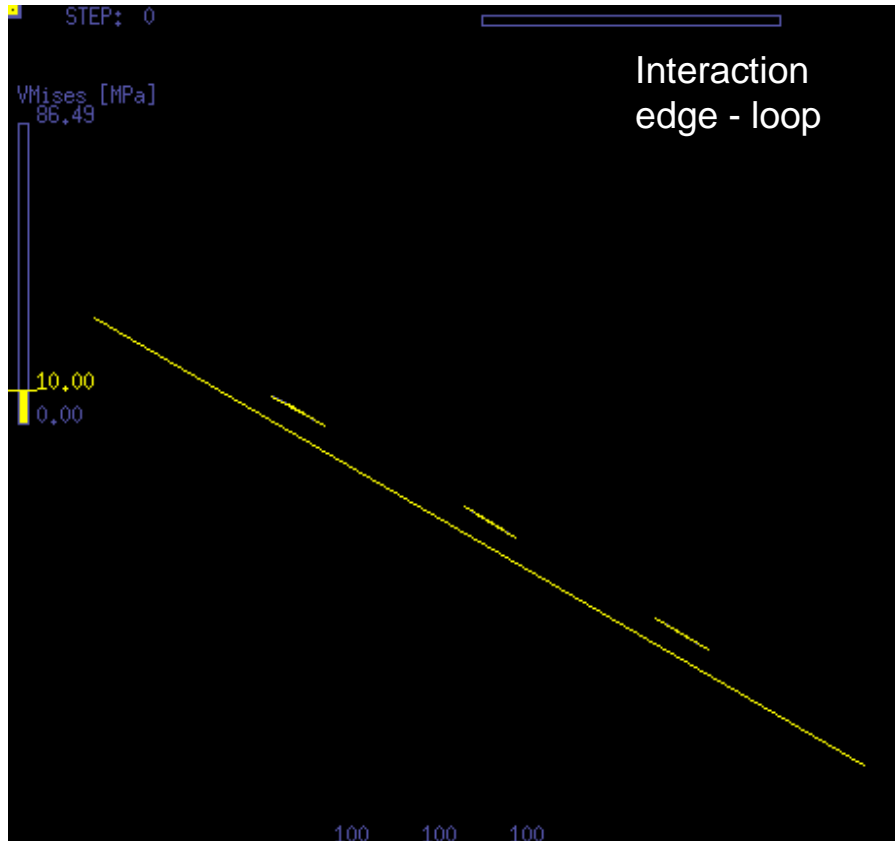
### Loading conditions

- Single slip
- $\dot{\epsilon} = 1.2 \cdot 10^5 \text{s}^{-1}$



# *Dislocation/loop interactions*

Rules come from MD simulation results

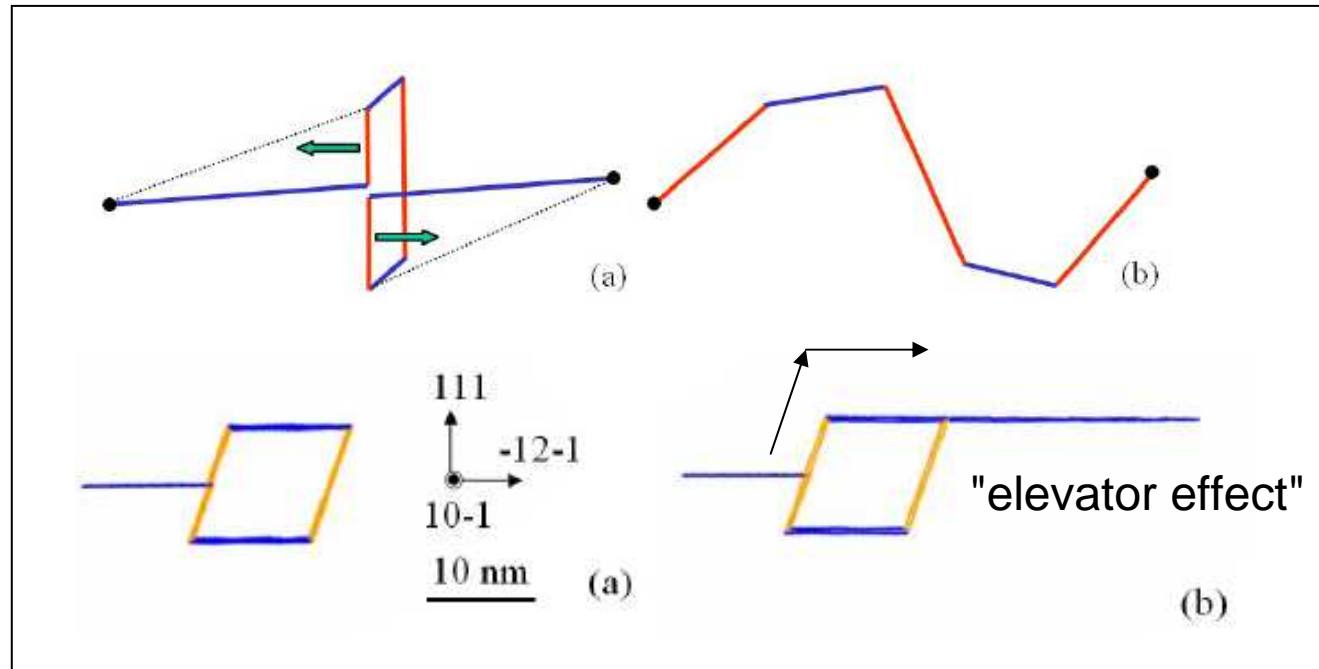


# Elementary dislocation/loop Interactions

Local interaction rules implemented in DD → MD simulations

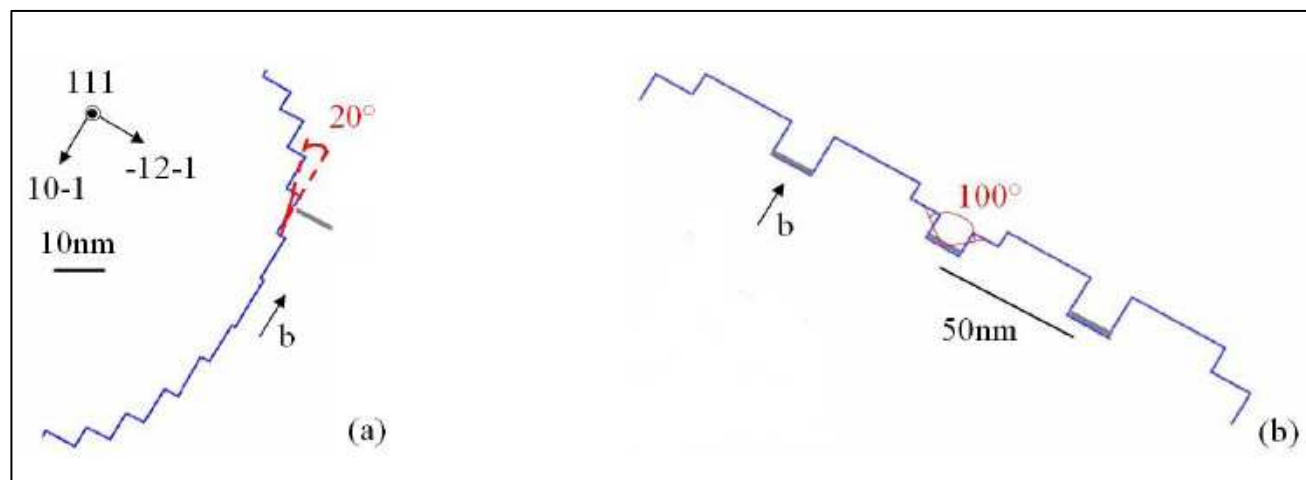
Screw/loop

$\sigma_{crit} = 340 \text{ MPa}$



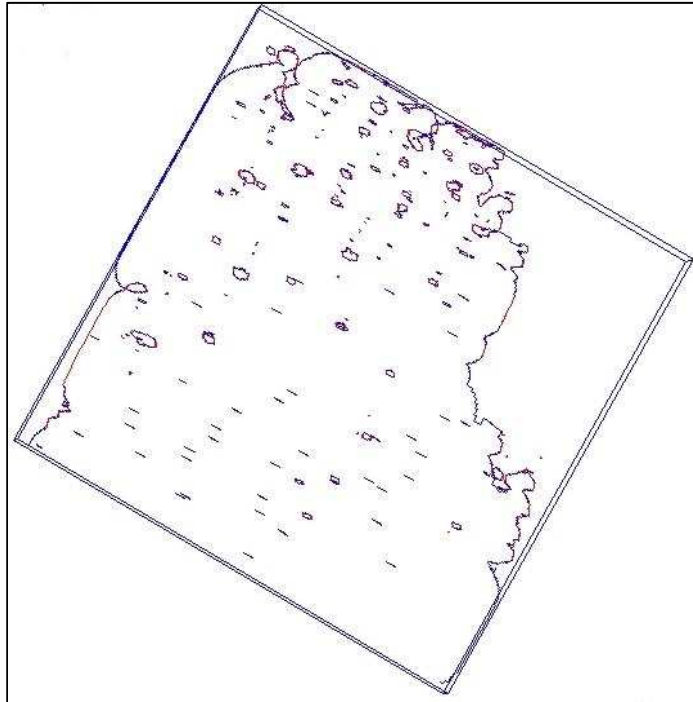
Edge/loop

$\sigma_{crit} = 130 \text{ MPa}$



Case 1: Glide of an isolated dislocation  $\tau_{nuc} \gg \tau_{obst}$

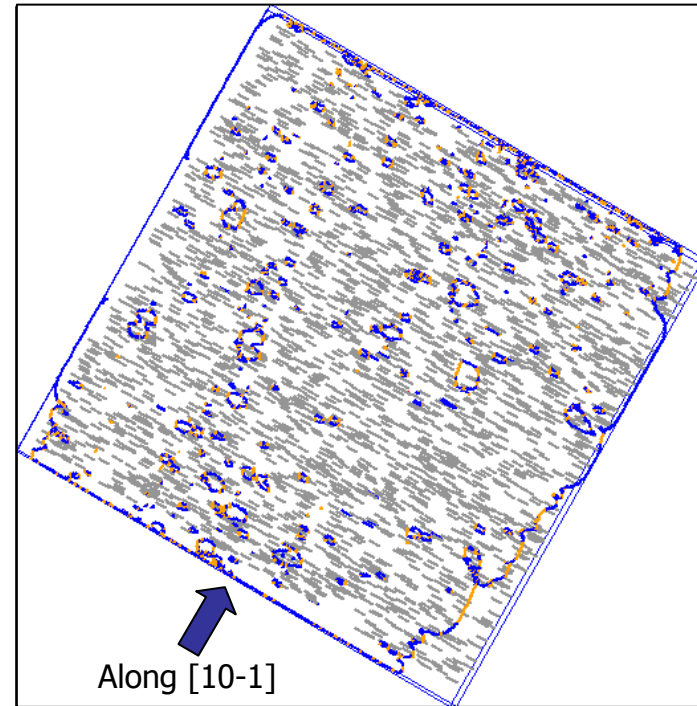
Edge-type source



Max. stress applied ~ 120-160MPa

Easy shear propagation  
Generation of long screw segments  
Planar slip

Screw-type source



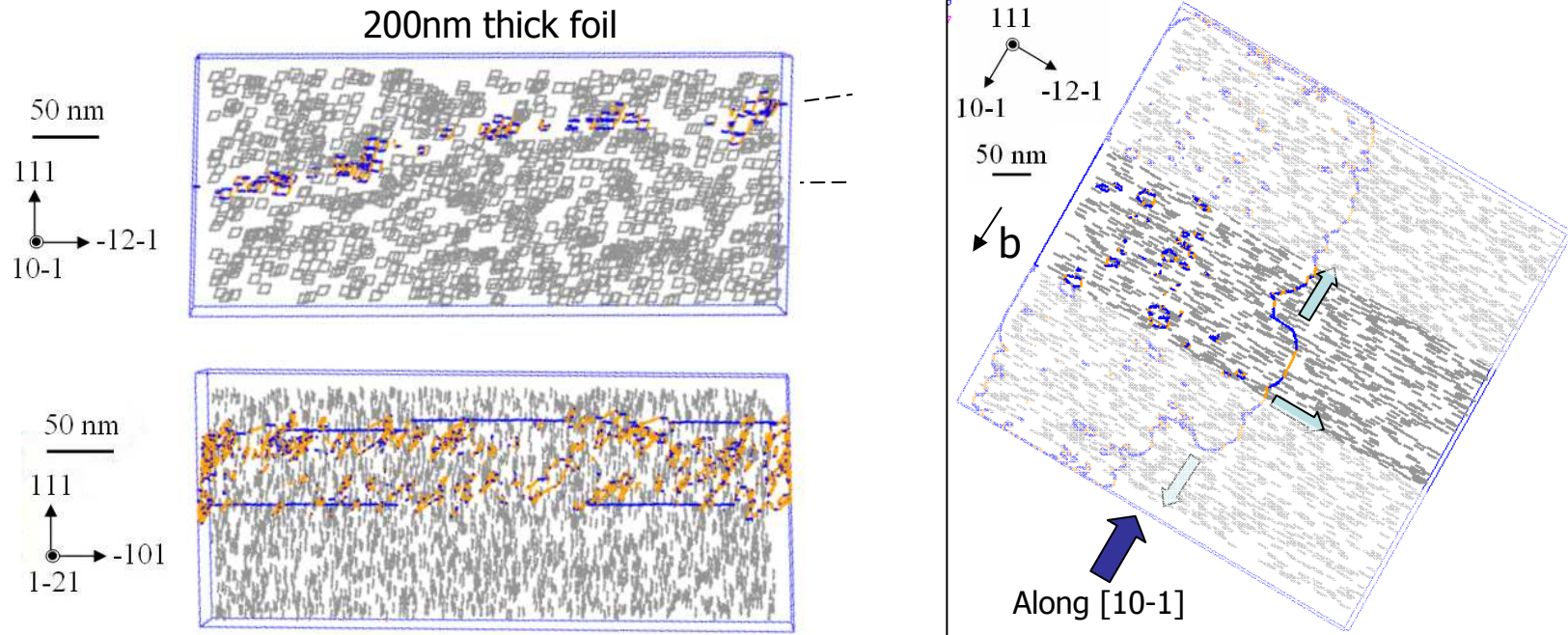
Max. stress applied ~ 200-260MPa

Loop unfaulting  
Helix  
Strong pinning

➡ **Plasticity is controled by screw dislocations**

*Case 1: Glide of an isolated dislocation*  $\tau_{nuc} \gg \tau_{obst}$

Screw-type source



Helix unpinning → Glide plane change

→ **Average glide plane is tilted away from initial plane**

Activation of a screw line → Edge portions → Transport of the helix

→ **Partial loop cleaning**

→ **No clear band of finite thickness**



## Case 2: Collective glide of dislocations

$$\tau_{nuc} < \tau_{obst}$$

### Screw-type source

$$\tau_{nuc} = 120 \text{ MPa}$$

$$\tau_{obst} \begin{cases} \text{edge} = 120-160 \text{ MPa} \\ \text{screw} = 200-260 \text{ MPa} \end{cases}$$

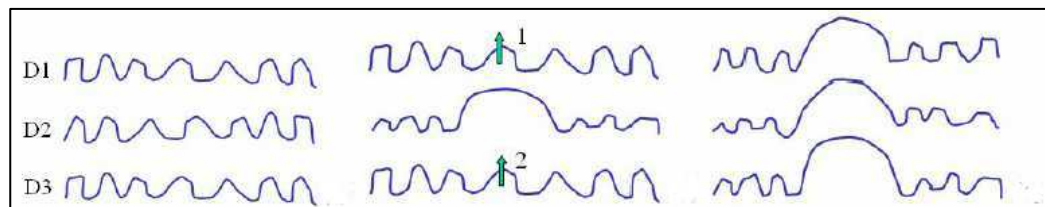
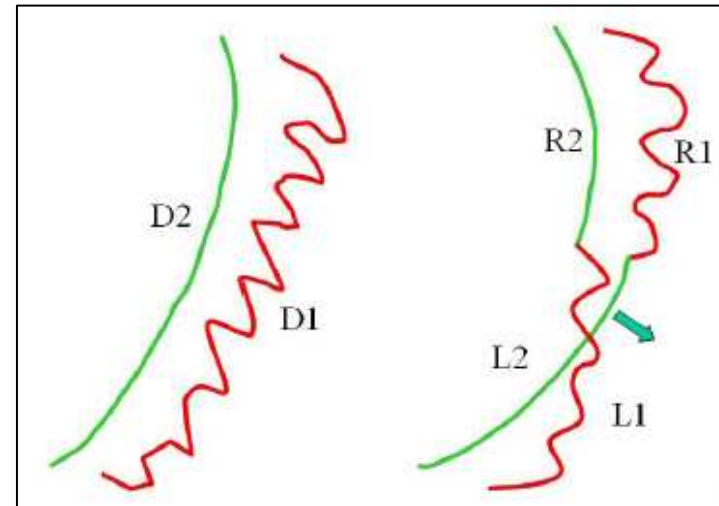
### Dislocation glide with the help of *collective* effects

#### Collective effects

Dislocation pile-ups

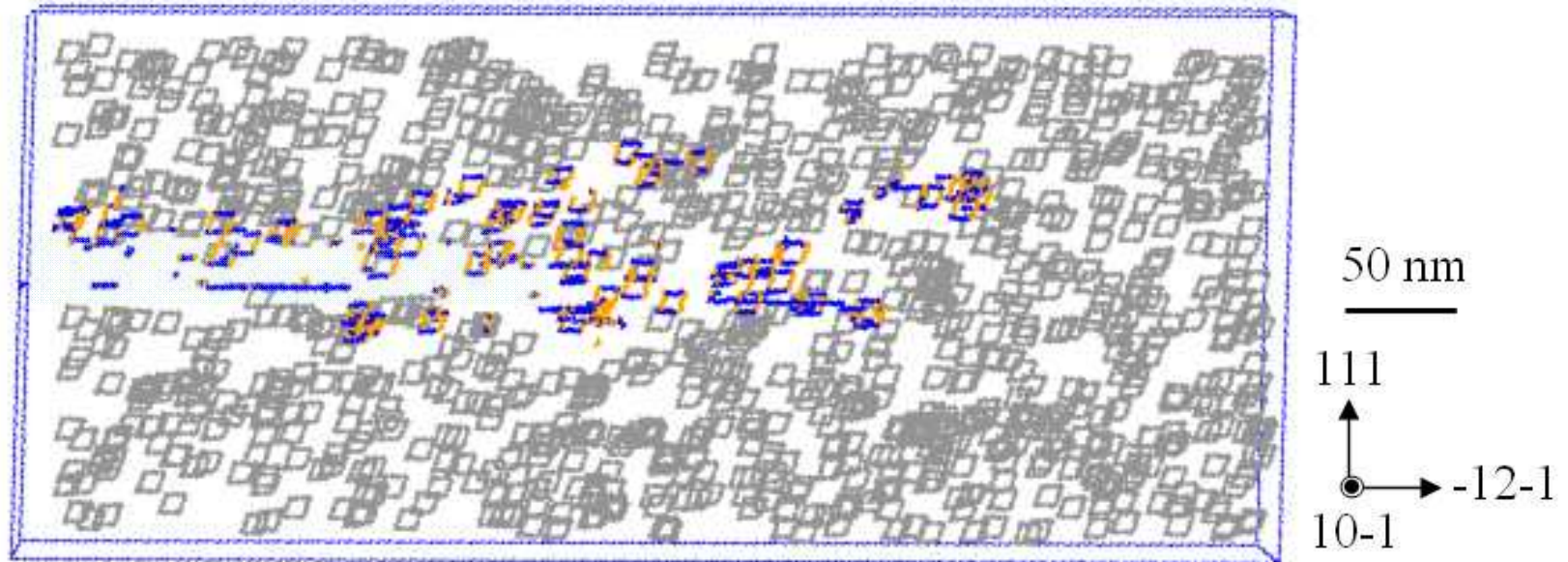
Arm exchanges

Dislocation avalanches



*Case 2: Collective glide of dislocations*

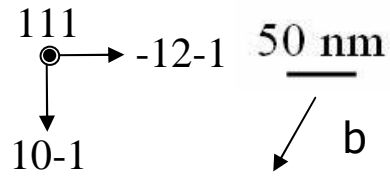
$$\tau_{nuc} < \tau_{obst}$$



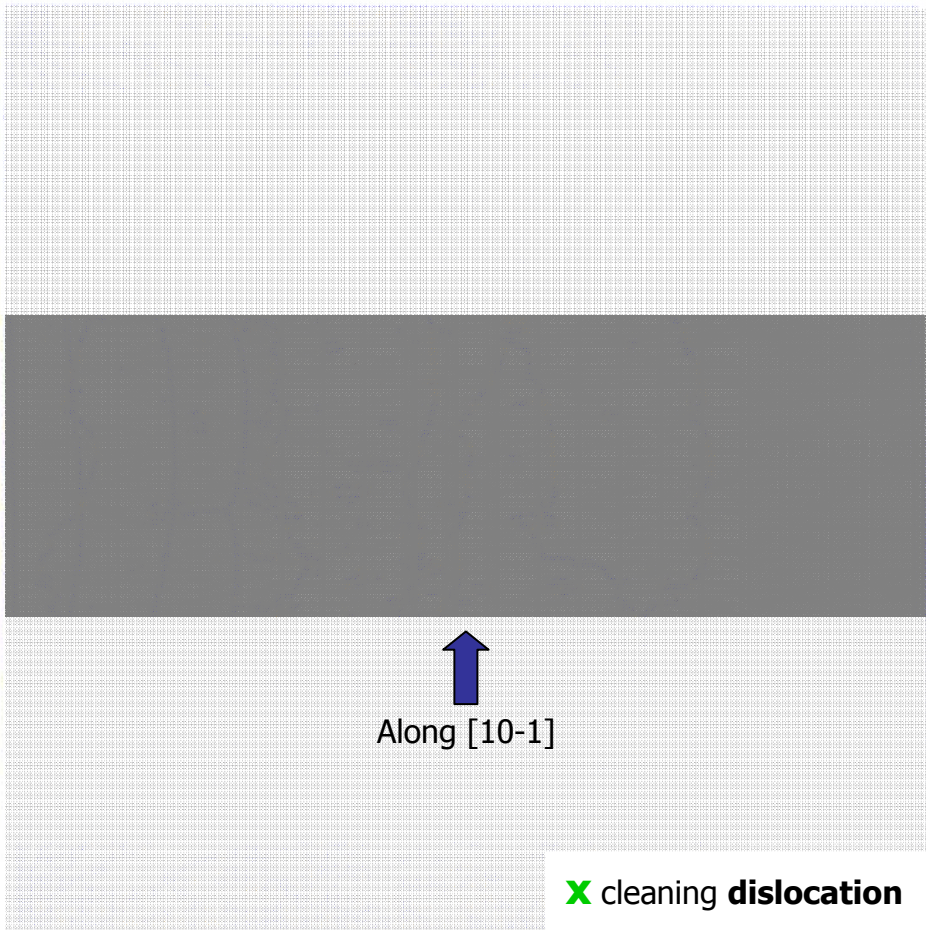
→ **Formation of a clear band // (111) of finite thickness**

# Case 2: Collective glide of dislocations

$$\tau_{nuc} < \tau_{obst}$$



## Microstructure of a clear band

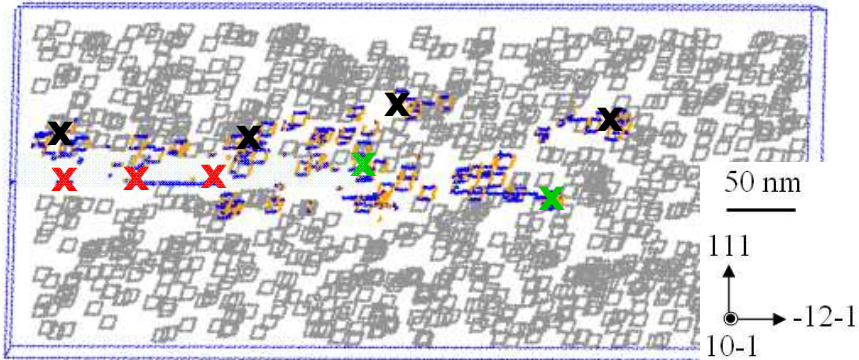


- x cleaning **dislocation**
- x pushing **dislocation**
- x** arrested **dislocation**

Leading dislocations  
 Dislocations with helix jogs  
 Clear and broaden the band

Following dislocations  
 Straight dislocations  
 "Push" the leading dislocations

At band periphery  
 Accumulation of debris



## Case 2: Collective glide of dislocations

$$\tau_{nuc} < \tau_{obst}$$

### Broadening

Helix



Re-emission of screw dislocations in new slip planes

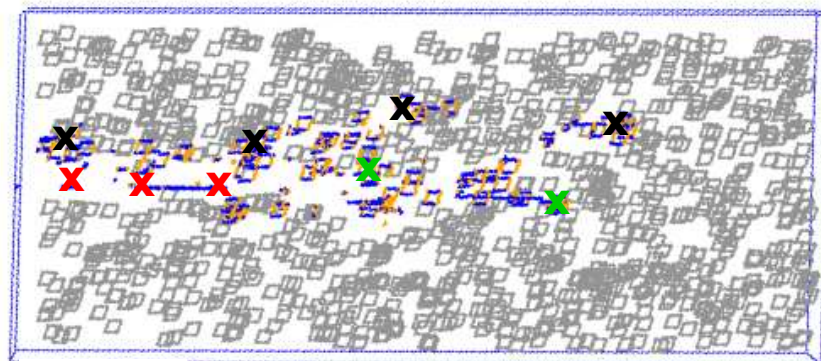


Dislocation pile-ups



Activation of dislocation glide on the new slip planes

$$\tau_{dislocation} = \tau_{applied} + \tau_{pile-up} (\text{number of dislocations}) > \tau_{unpinning} (\text{defect strength, defect density})$$



50 nm

111

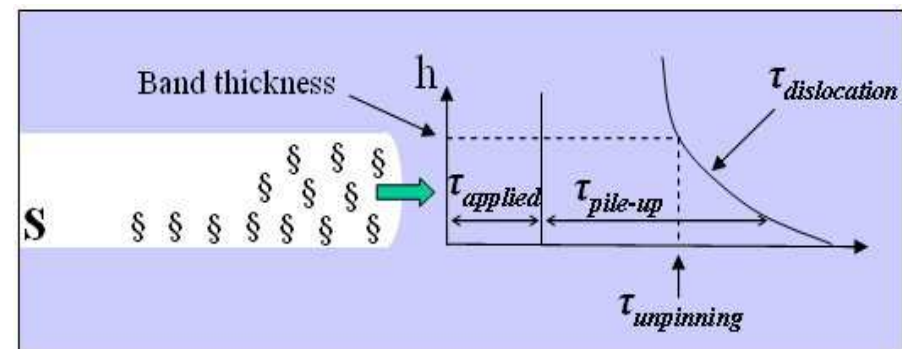
10-1

→ -12-1

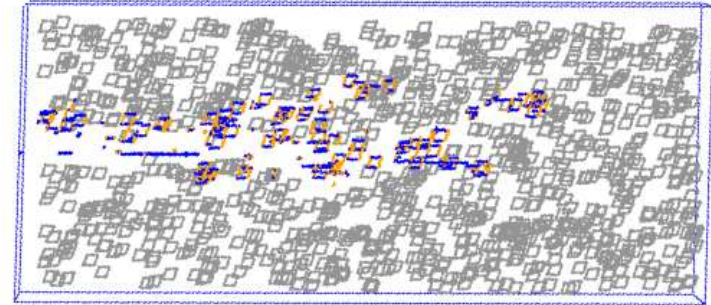
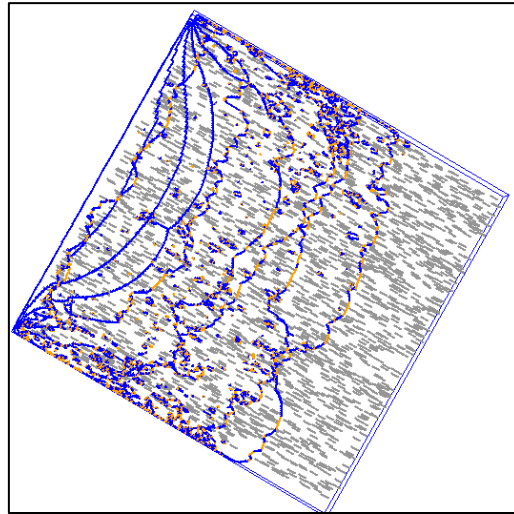
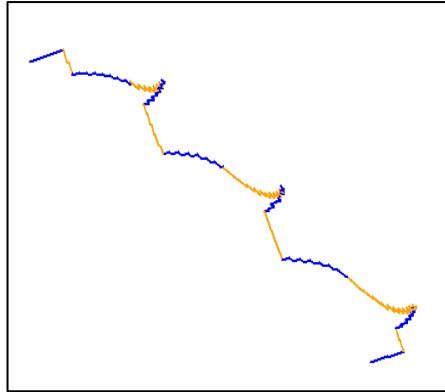
**x** leading dislocations

**x** pushing dislocations

**x** arrested dislocations



## *Conclusion Dislocation Dynamics analysis*



### **Clear band formation scenario**

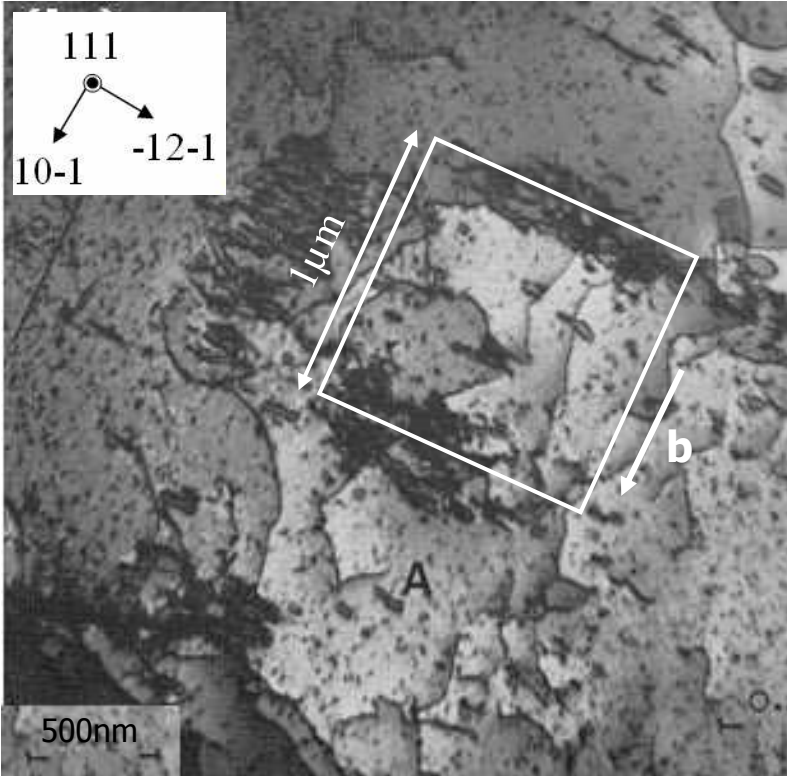
- Helix clear the initial defect loops
- Clear band broaden by slip plane change
- Collective effects necessary



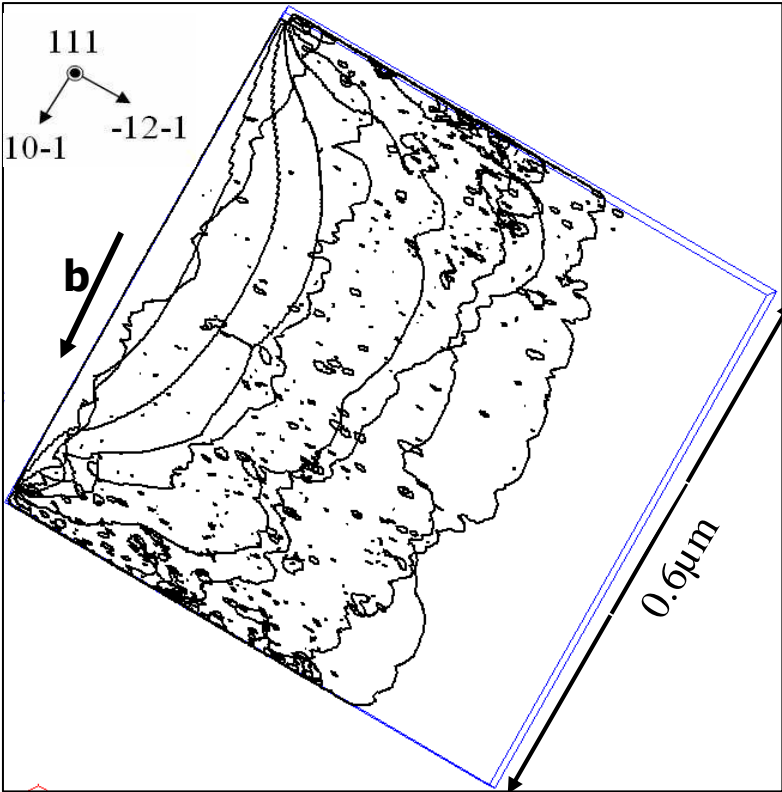
IV Comparaison simulation / experiment

Comparison simulation / experiment

**Microstructure of clear bands**



Cu irradiated and deformed:  
foil parallel to the clear band [Sharp, 67]

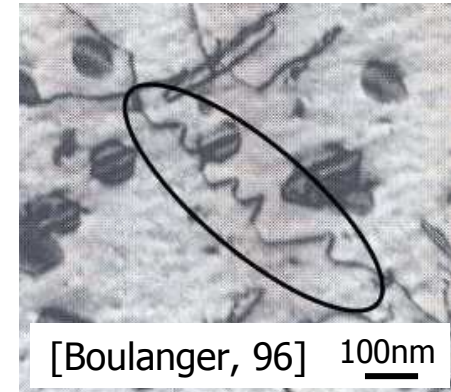
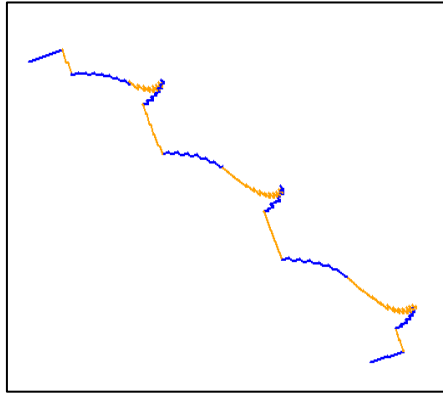
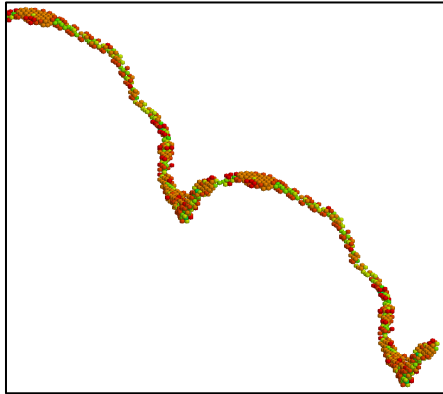


Dislocation Dynamics

- ➡ Longs screw segments
- ➡ Accumulation of jogs and debris in small zones

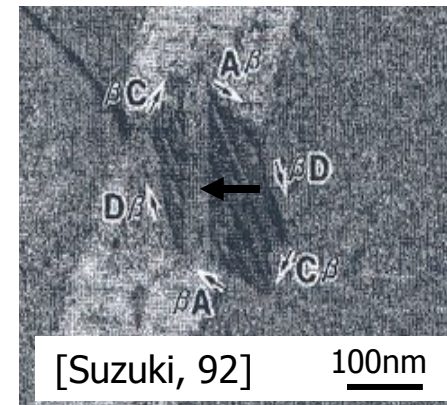
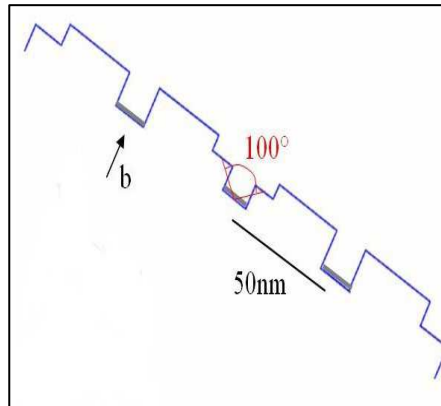
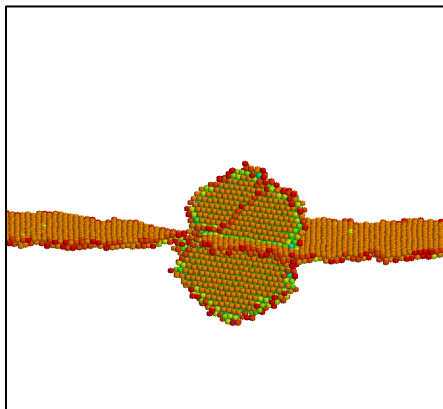
# Comparison simulations / experiments

## Elementary dislocation/loop interactions

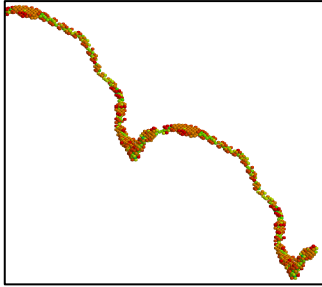


→ Absorption in the form of helix

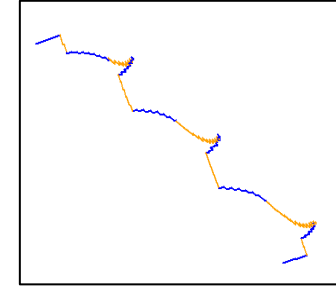
→ Loop shear



## Conclusions



New clear band formation scenario  
Multi-scale modelling



### Molecular Dynamics

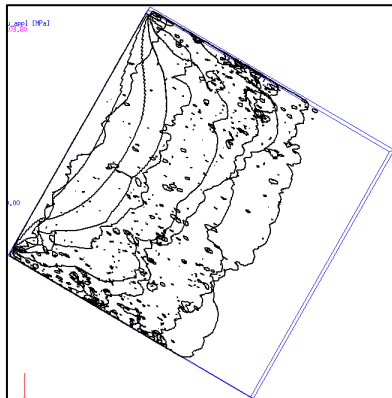
Dislocation/loop interaction mechanisms

- Difference screw / edge
- Screw = Unfaulting - Absorption - Helix – Change of glide plane

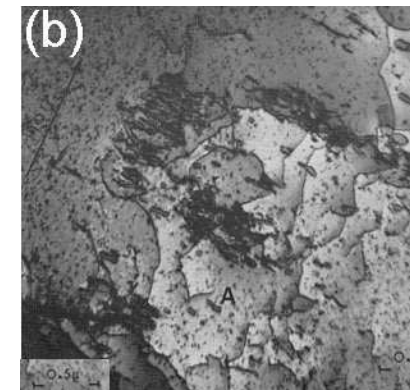
### Dislocation Dynamics

Collective interaction dynamics (leading to finite thickness clear band)

- Realistic description of elementary dislocation/loop interactions
- Cleaning + Broadening = helix + dislocation pile-ups



Experimental validation





**THE END**