



*The Abdus Salam  
International Centre for Theoretical Physics*



**2227-7**

**Joint ICTP-IAEA Workshop on Radiation Resistant Polymers**

*14 - 18 March 2011*

**SELF-HEALING IN POLYMERS**

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# SELF-HEALING IN POLYMERS

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Lecture given at the ICTP/IAEA Workshop on “Radiation Resistant Polymers”, 14-18 March 2011, Trieste, Italy

- SH comes from the nature, ability to heal is a characteristic of living organisms. A damage triggers an automatic healing response.
- Materials that can recover mechanical properties following failure offer “increased safety and service life”.
- Relevant to materials that are used without or with only limited access by men, e.g. Medical applications, civil, aerospace, automotive and power engineering.

## Self-healing Coatings in Practice - Examples

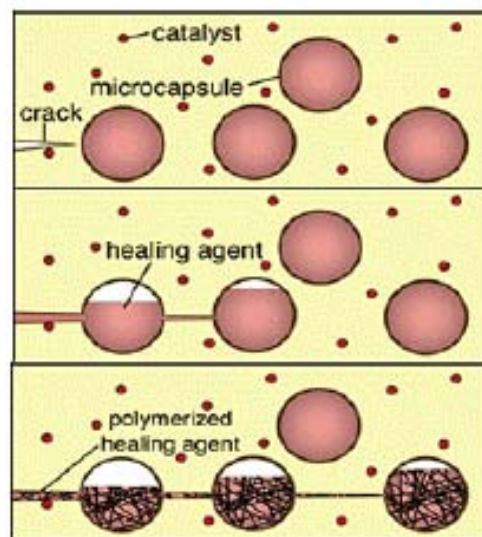
- Nissan X-Trail (2005; „Scratch Guard“)
- Nissan Infiniti (2008; „Scratch Shield“)
- Toyota Lexus (2010)
- Iveco/PPG (driving cab of an innovative concept truck 2010)
- Fiat/PPG (practice test with different test cars 2010/2011)

# Healing of Scratches?

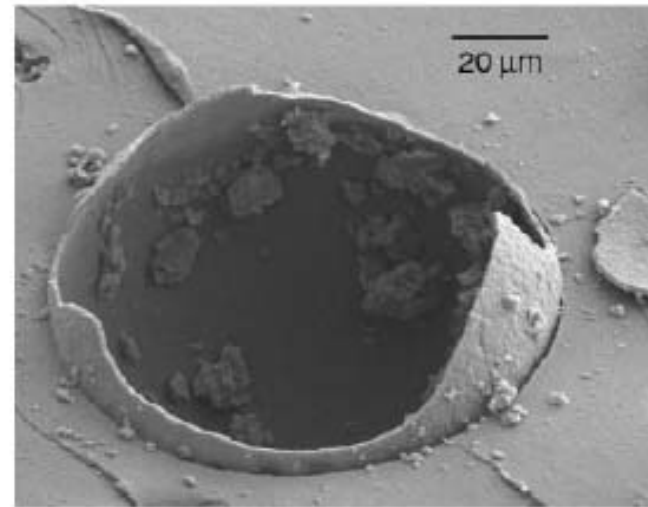


- Polymer networks are made to self-heal by either adding microcapsules filled with uncured resin or by introducing reversible bonds.
- The resin held within the matrix is released upon crack formation and hardens to heal the crack.
- The other mechanism of healing relies on the reversibility of bonds found designed into polymer networks.
- SH polymeric materials are therefore multifunctional composite systems.

# SH by MICROCAPSULE APPROACH



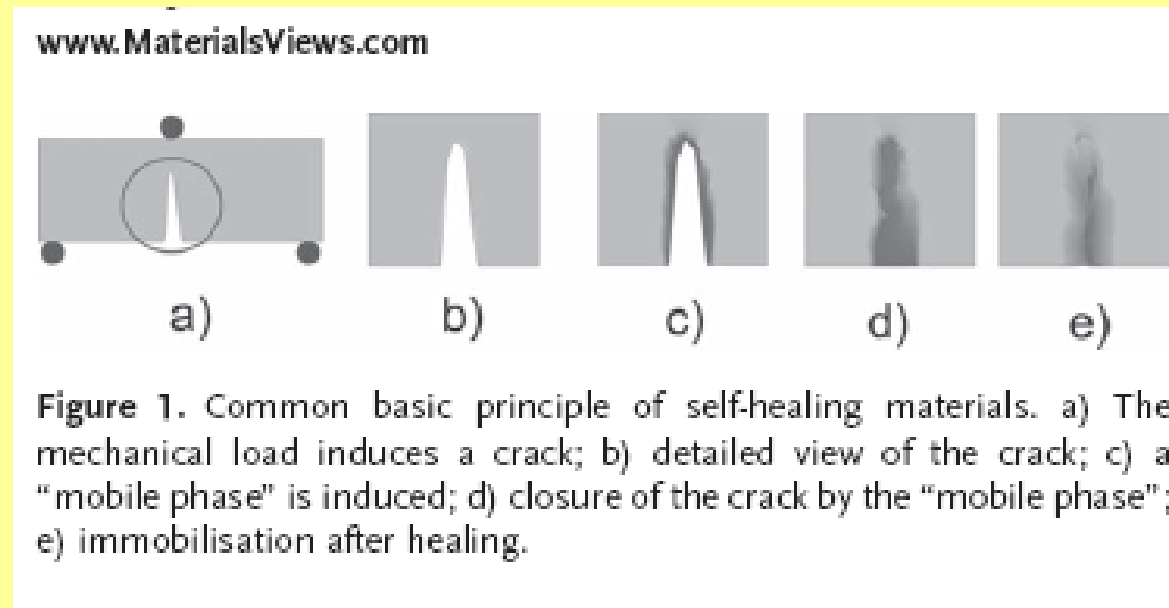
(a)



(b)

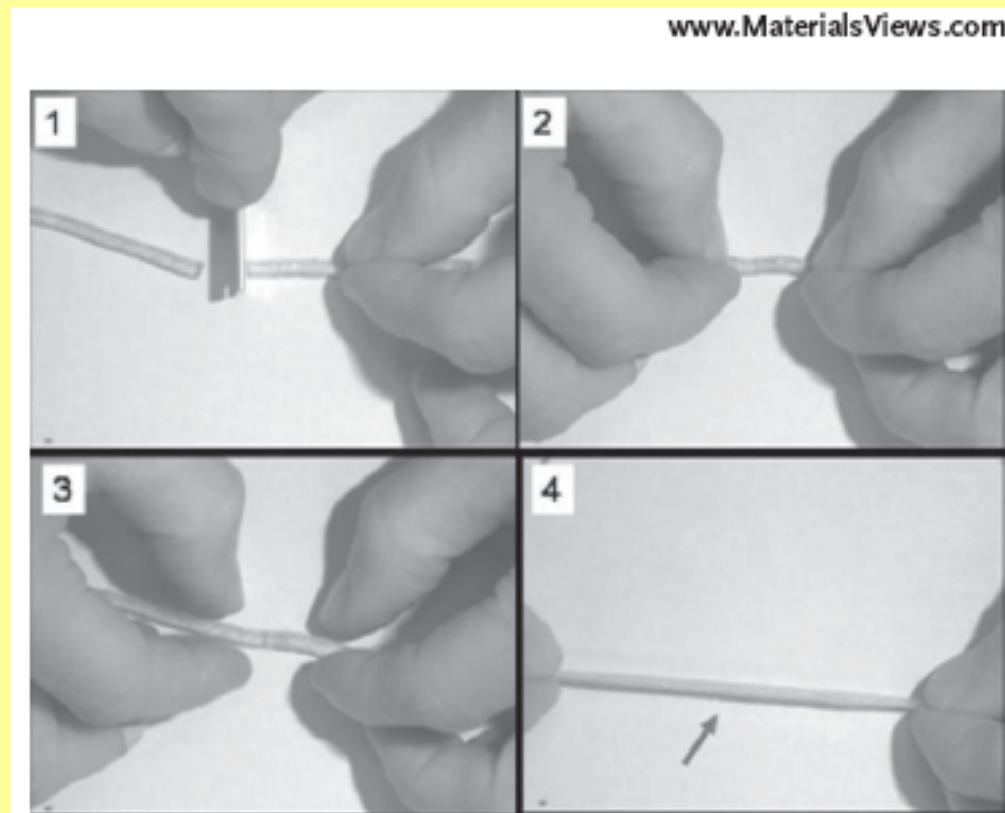
Figure 1: (a) Basic method of the microcapsule approach, (b) ESEM image showing ruptured microcapsule [White et al, 2001]

Prerequisite for a self-healing of a (mechanical damage) is the generation of a mobile phase which can close this crack

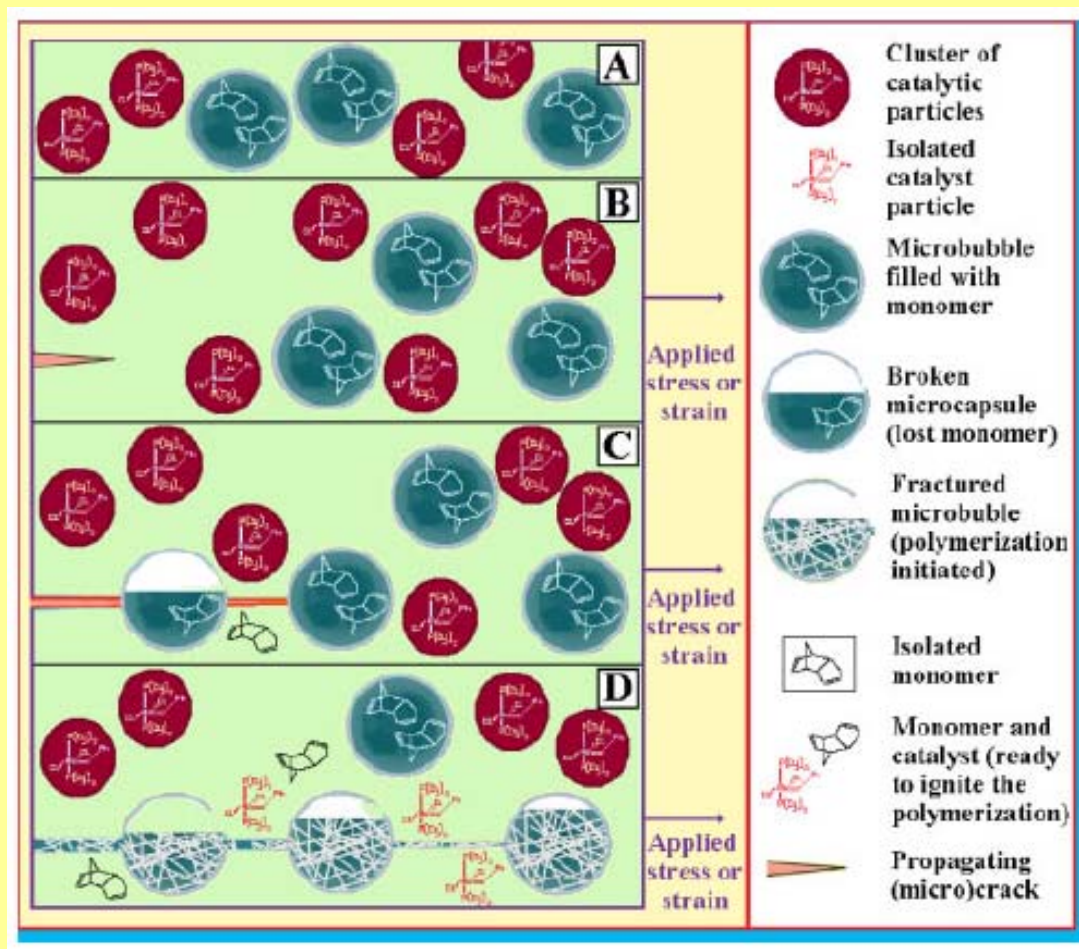


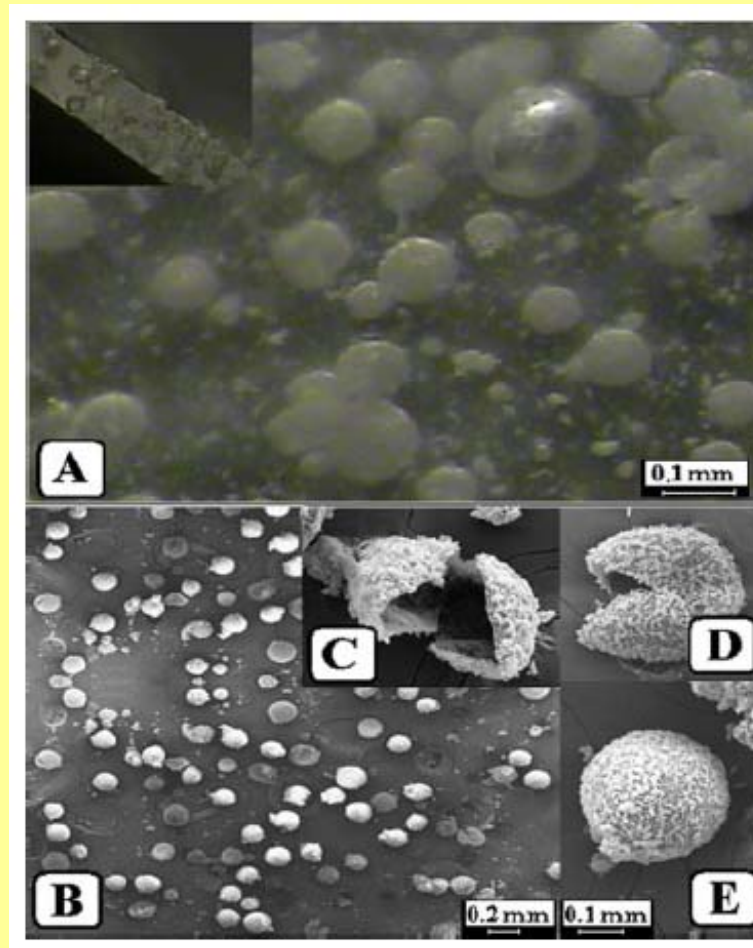


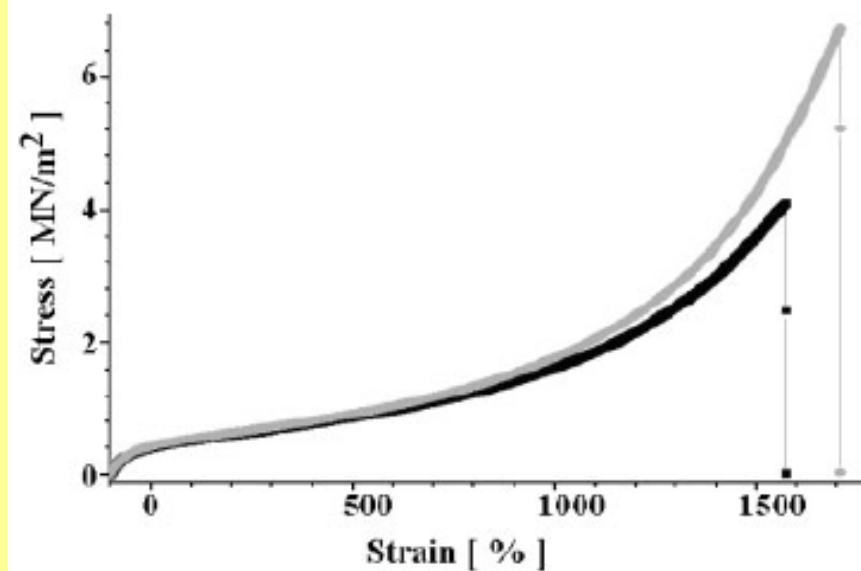
# SH by REVERSIBILITY OF BONDS



**Figure 3.** Self-healing properties of a supramolecular polymer by Leibler and coworkers. Reproduced with permission.<sup>[6]</sup> Copyright 2008, Ludwik Leibler, CNRS.

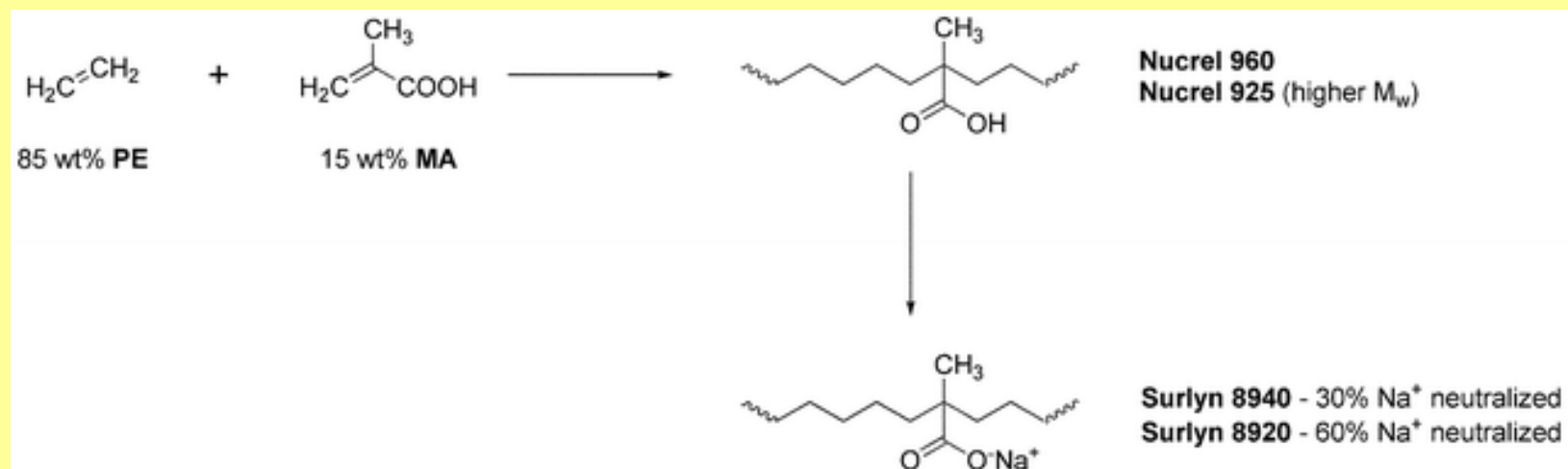




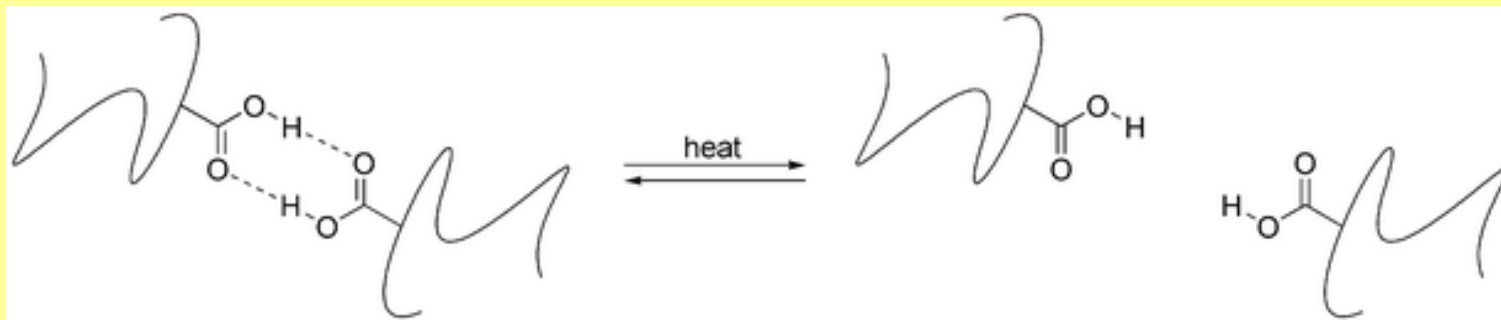


**Figure 5.** Stress-strain dependences for polystyrene-*block*-polybutadiene *block*-polystyrene filled with 5% microbubbles containing DCPD (black line; self-healing features not activated) and polystyrene-*block*-polybutadiene *block*-polystyrene filled with 5% microbubbles containing DCPD and loaded with 1% Grubbs catalyst.

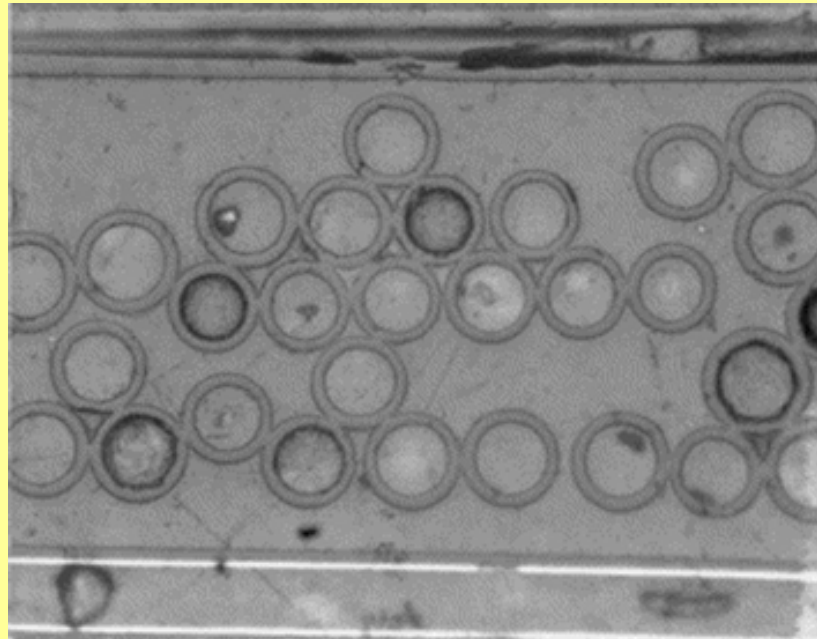
# IONOMERS USED IN SH

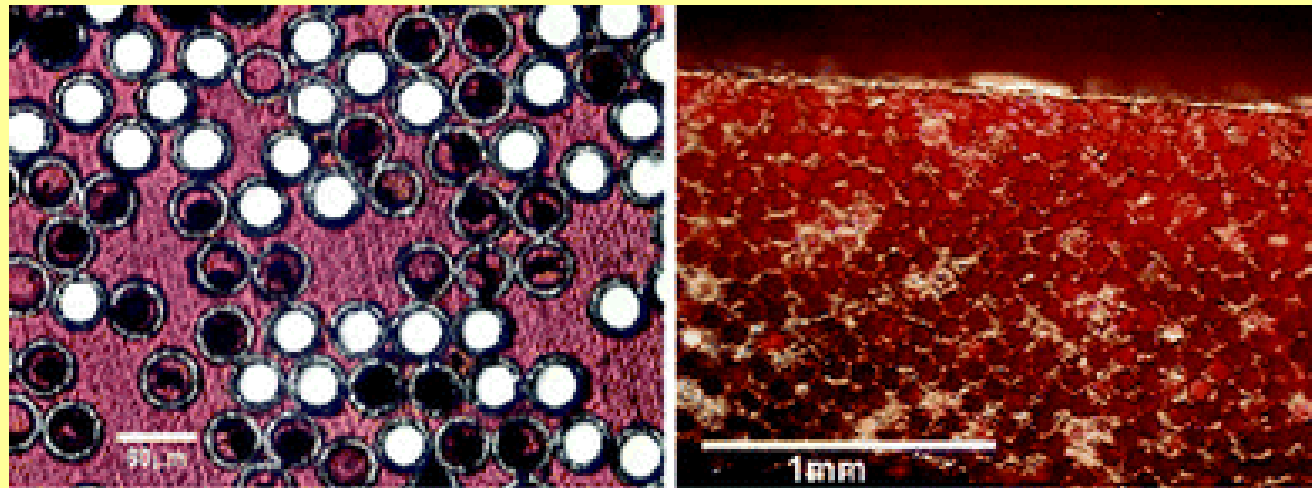


# THERMALLY CONTROLLED HYDROGEN-BOND XL

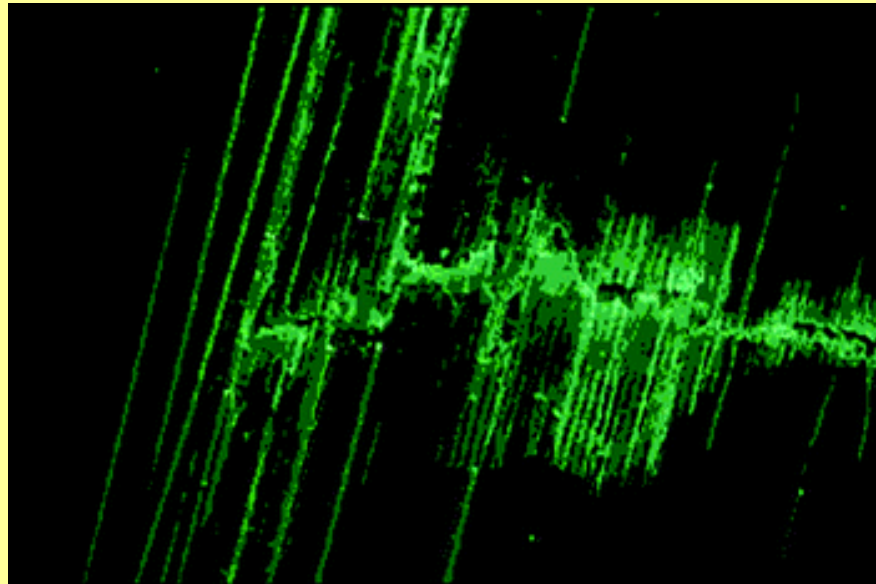


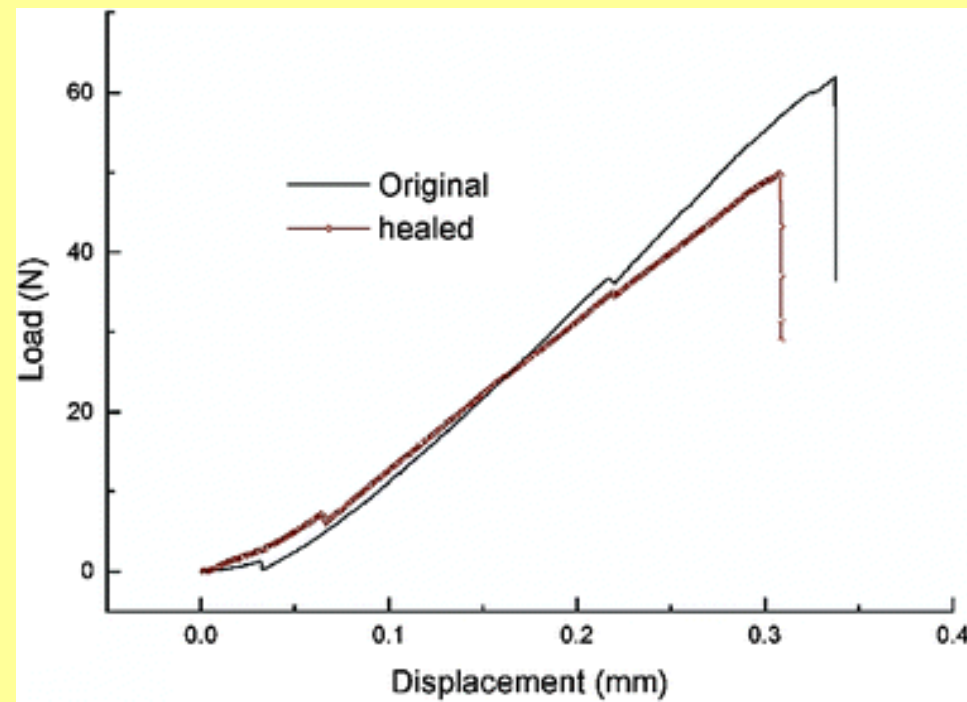
# HOLLOW GLASS FIBRES









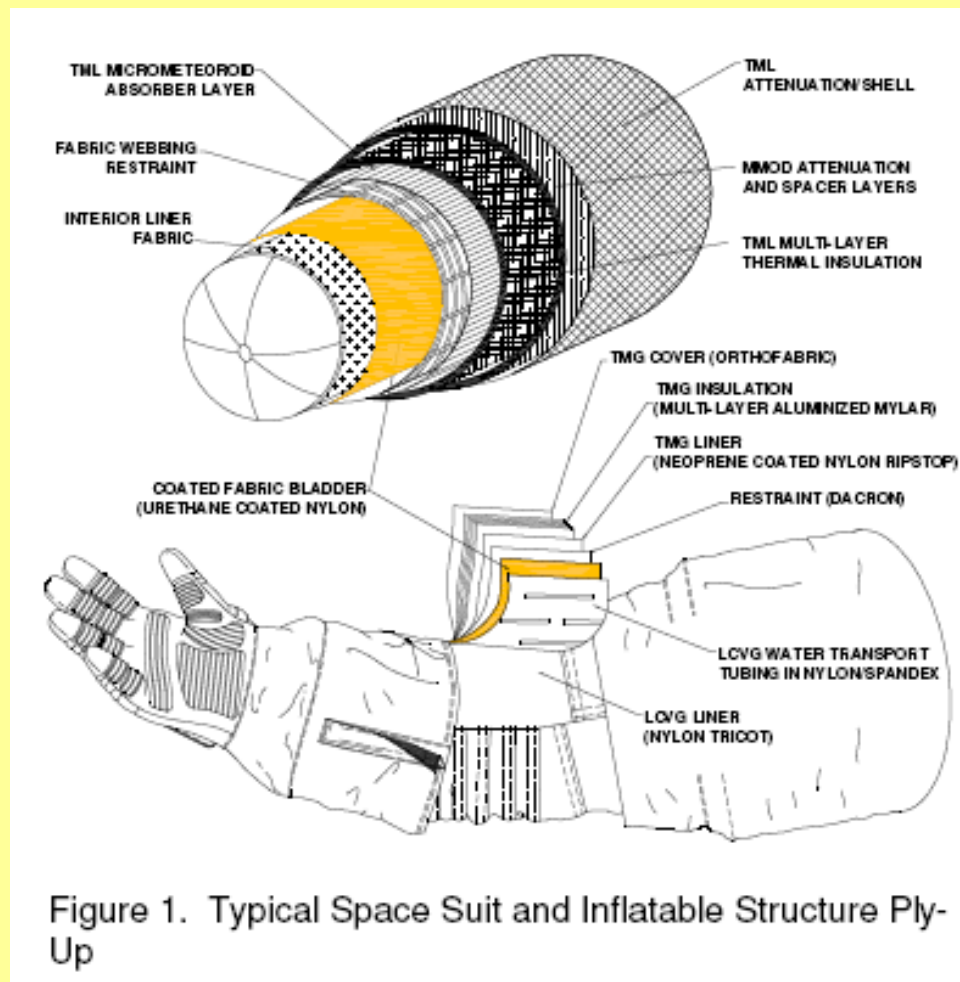


2007-01-3211

# **Self-Healing Technology for Gas Retention Structures and Space Suit Systems**

**J. Ferl, J. Ware, D. Cadogan and J. Yavorsky**  
ILC Dover, LP

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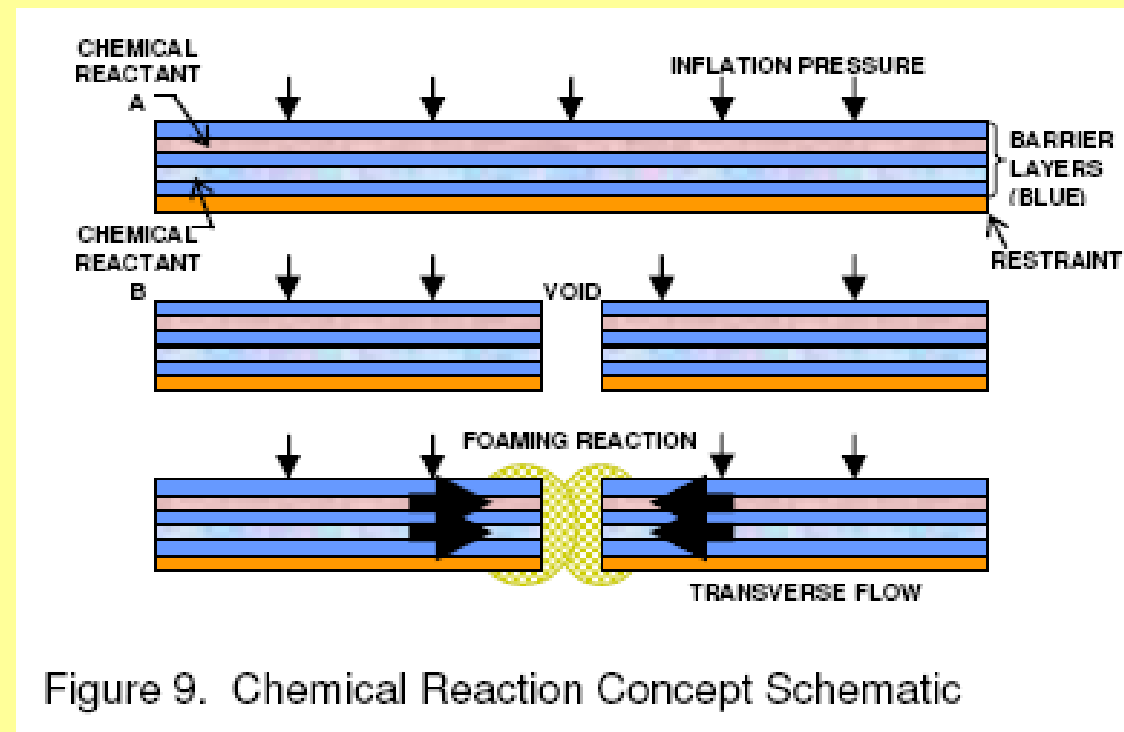


Figure 9. Chemical Reaction Concept Schematic

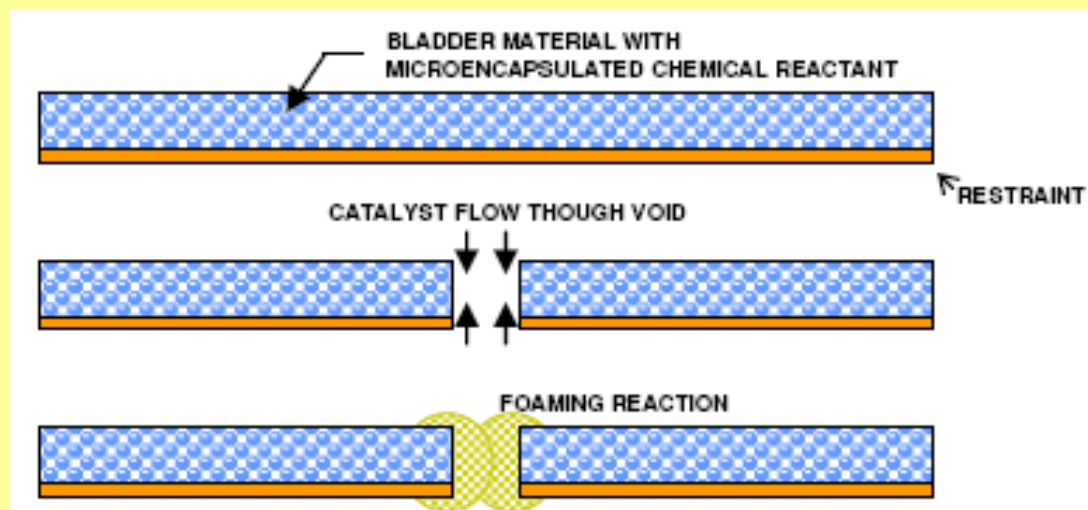


Figure 8. Environmental Reaction Concept Schematic



Figure 17. Self-healing Viscoelastic Gel Localized on Each Pattern Piece of SSA Lower Arm Bladder

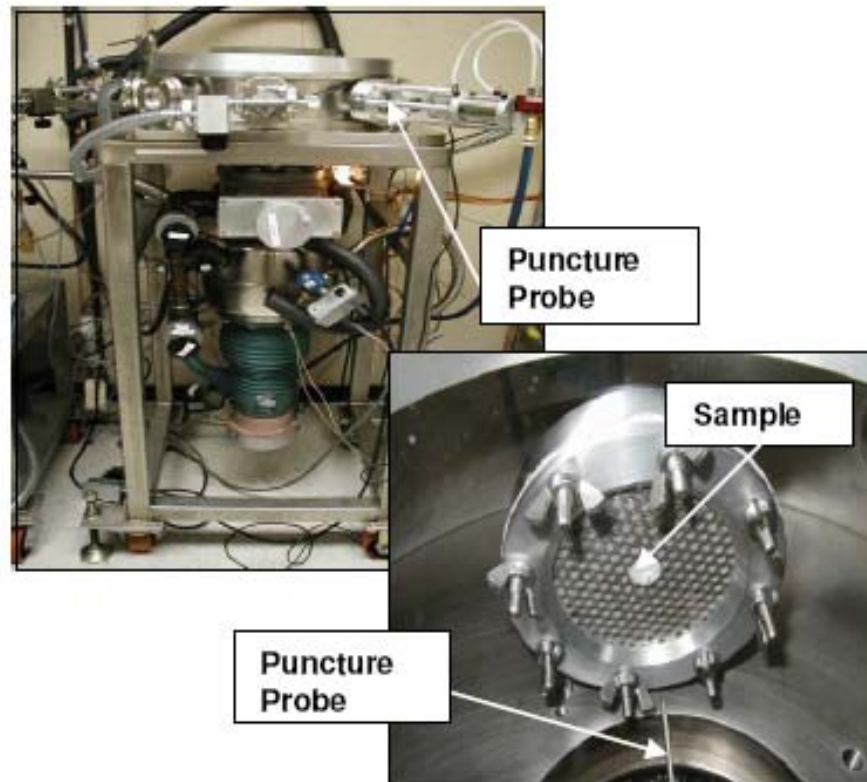


Figure 19. Self-Healing Sample Test Fixture in Vacuum Chamber



# PUNCTURE TESTS IN VACUUM CHAMBER

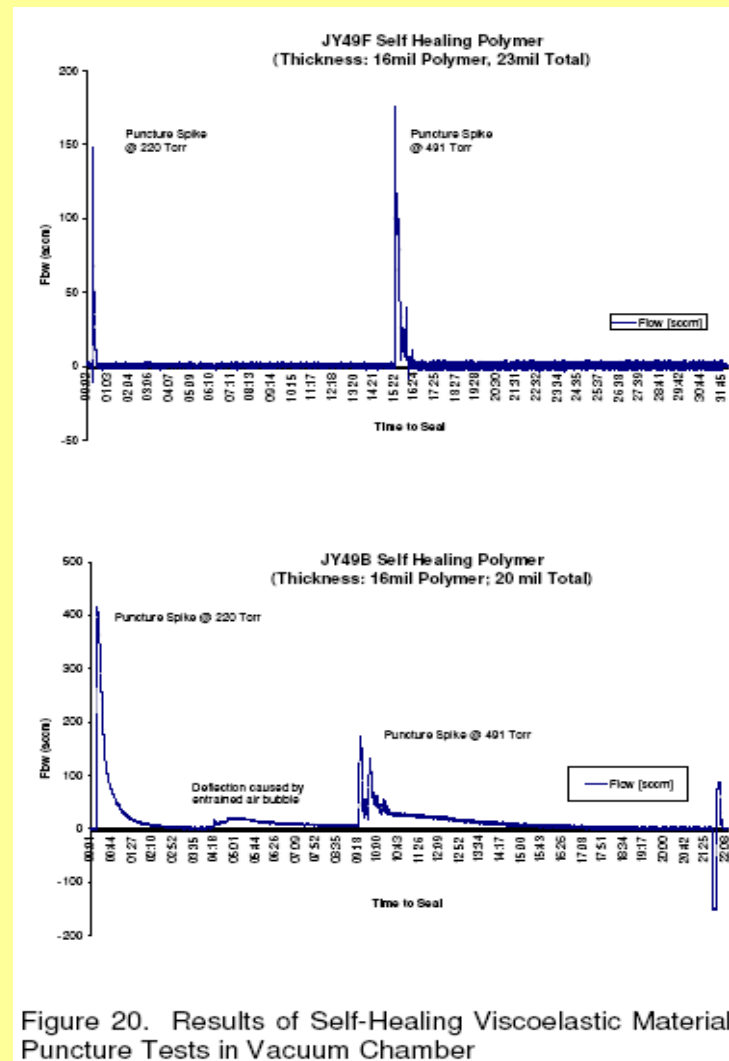


Figure 20. Results of Self-Healing Viscoelastic Material Puncture Tests in Vacuum Chamber

# INVERSE TEMPERATURE-ANNEALING PHENOMENA

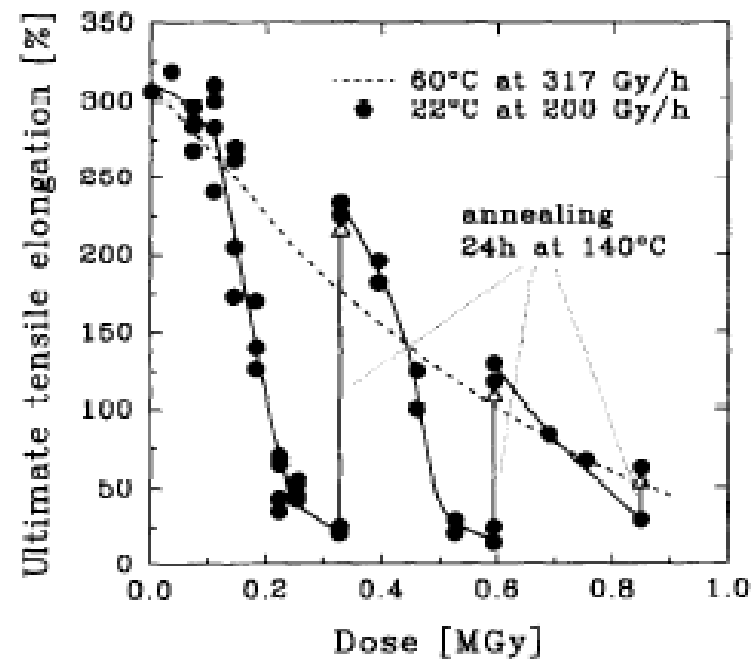


Fig. 1. Decrease in ultimate elongation for material A during degradation at 22°C and 200 Gy h<sup>-1</sup> and the recovery of mechanical properties upon annealing at 140°C for 24 h after 328, 594 and 849 kGy. Note the slower degradation for the material when aged at 60°C.

lina et al.

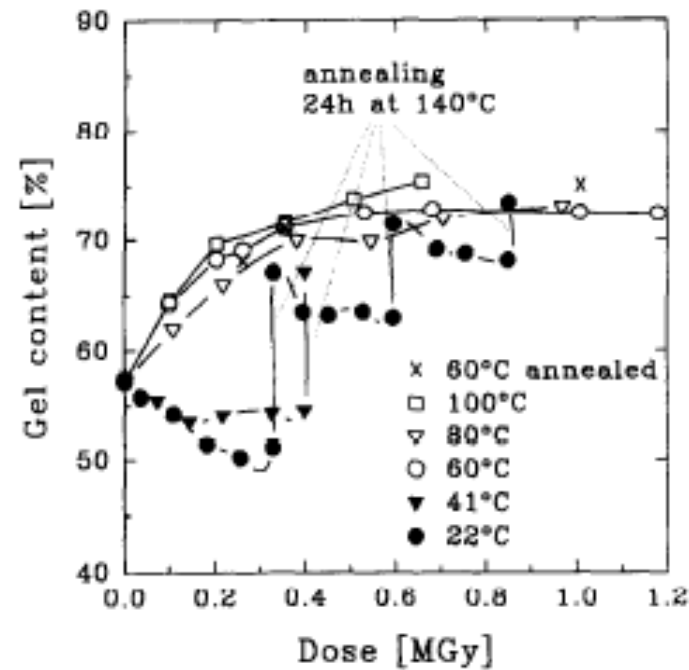
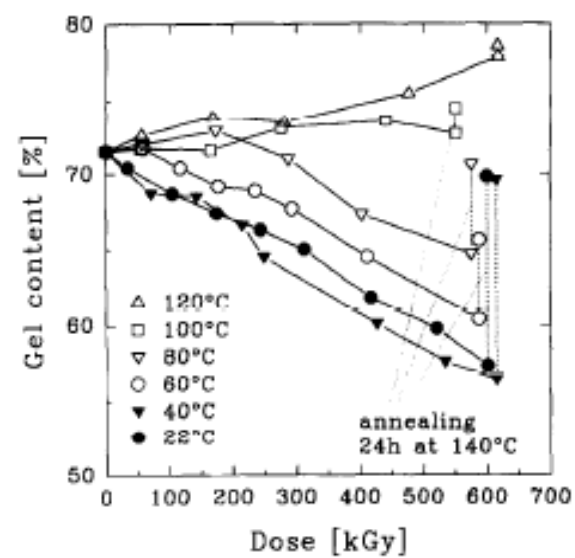


Fig. 2. Changes in the gel content of material A during combined radiation-temperature exposures as a function of temperature showing predominantly scission at lower temperatures and crosslinking at higher temperatures as well as a marked increase in the gel content during annealing of samples aged at 22 and 41°C.



**Fig. 5.** Changes in the gel content of material B during aging at different temperatures and dose rates ranging from 0.21 to 0.35 kGy h<sup>-1</sup>.

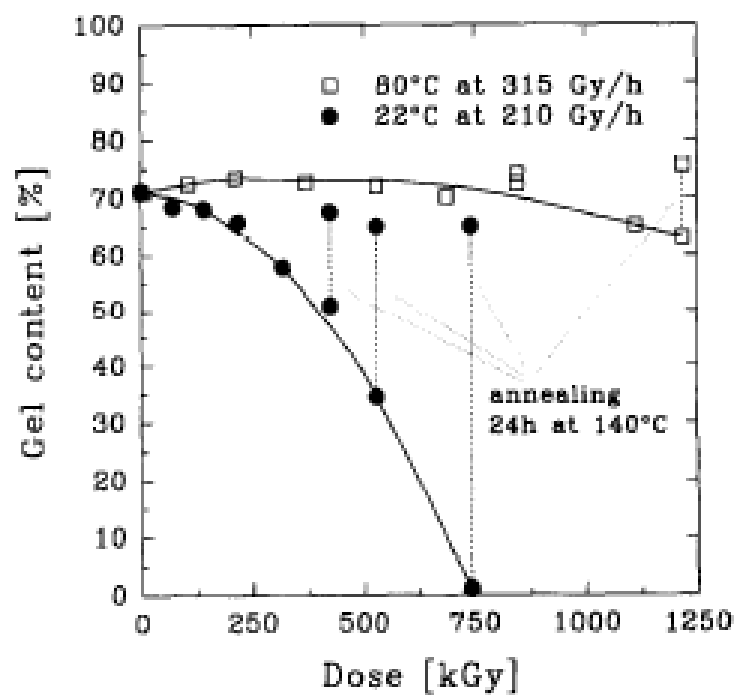


Fig. 6. Changes in the gel content of material C during aging at 22°C (0.21 kGy h<sup>-1</sup>) and 80°C (0.315 kGy h<sup>-1</sup>).

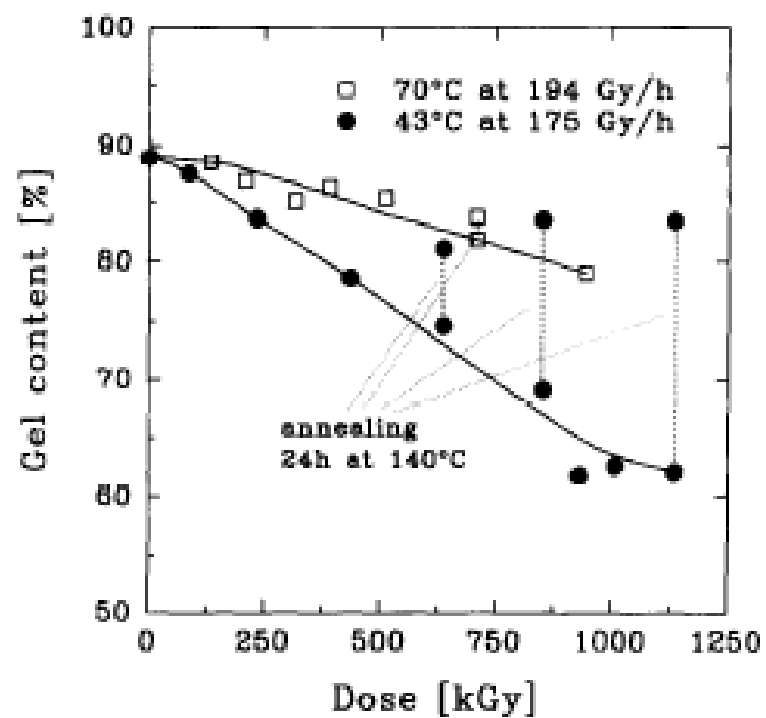


Fig. 7. Changes in the gel content of material D during aging at 43°C ( $0.175 \text{ kGy h}^{-1}$ ) and 70°C ( $0.194 \text{ kGy h}^{-1}$ ).

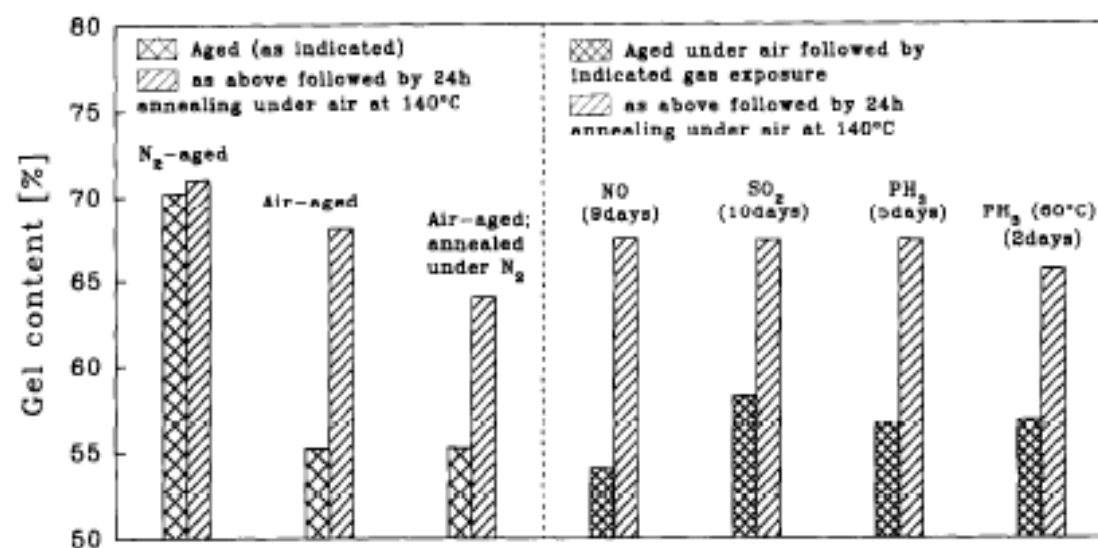


Fig. 11. Gas treatment of material A, aged at 22°C to 400 kGy at 485 Gy h<sup>-1</sup>, and its influence on subsequent crosslinking during annealing at 140°C (deactivation of any hydroperoxides).

**THANK YOU  
FOR YOUR KIND ATTENTION**