



The Abdus Salam
International Centre for Theoretical Physics



2234-1

**Meeting of Modern Science and School Physics: College for School
Teachers of Physics in ICTP**

27 April - 3 May, 2011

Materials engineered by layer-by-layer deposition

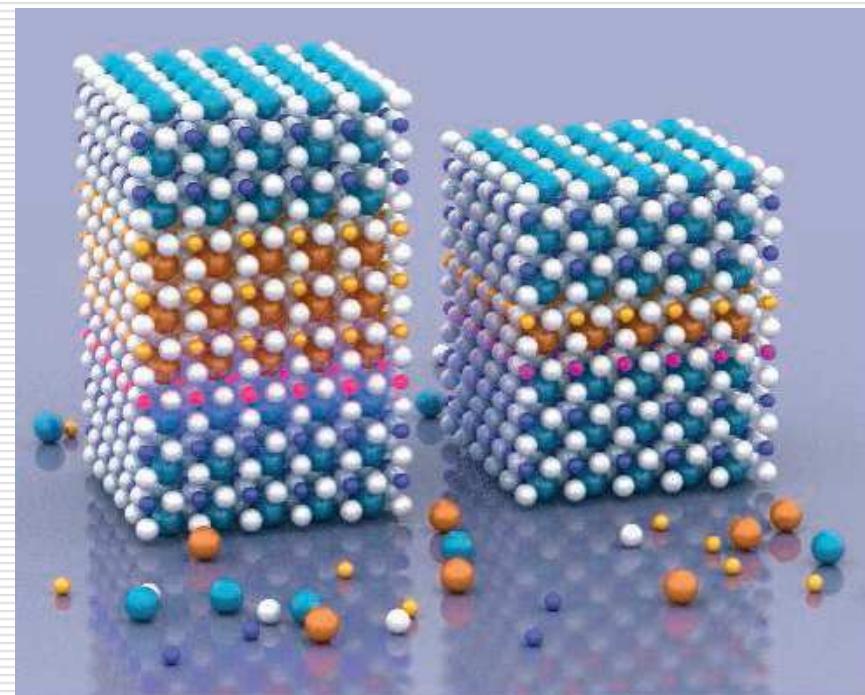
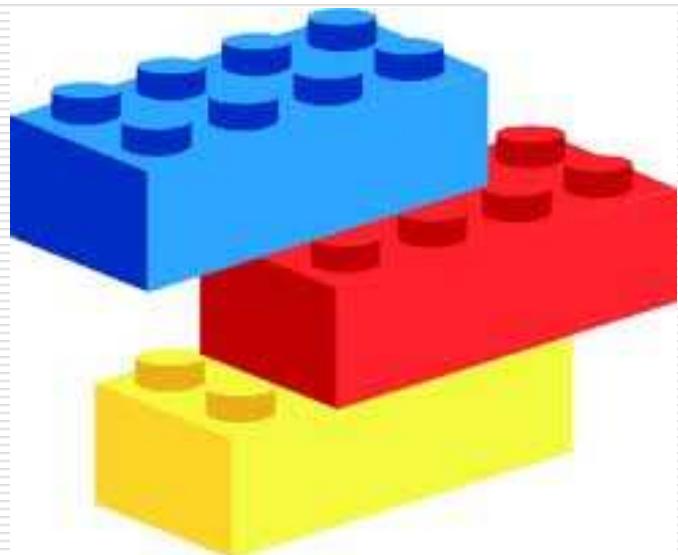
Giuseppe Balestrino
CNR SPIN & University of Rome "Tor Vergata"
Rome
ITALY

Materials engineering by layer-by-layer deposition

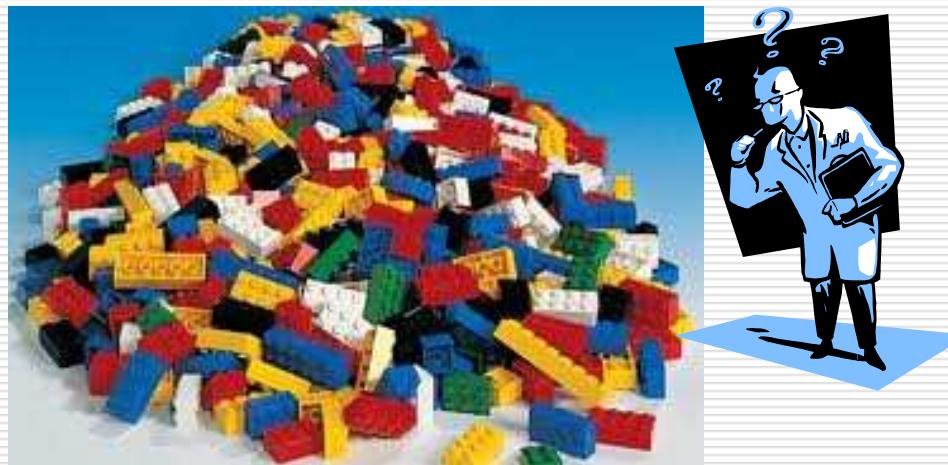
**Thin oxide films with novel or enhanced physical properties
obtained by design**

G. Balestrino

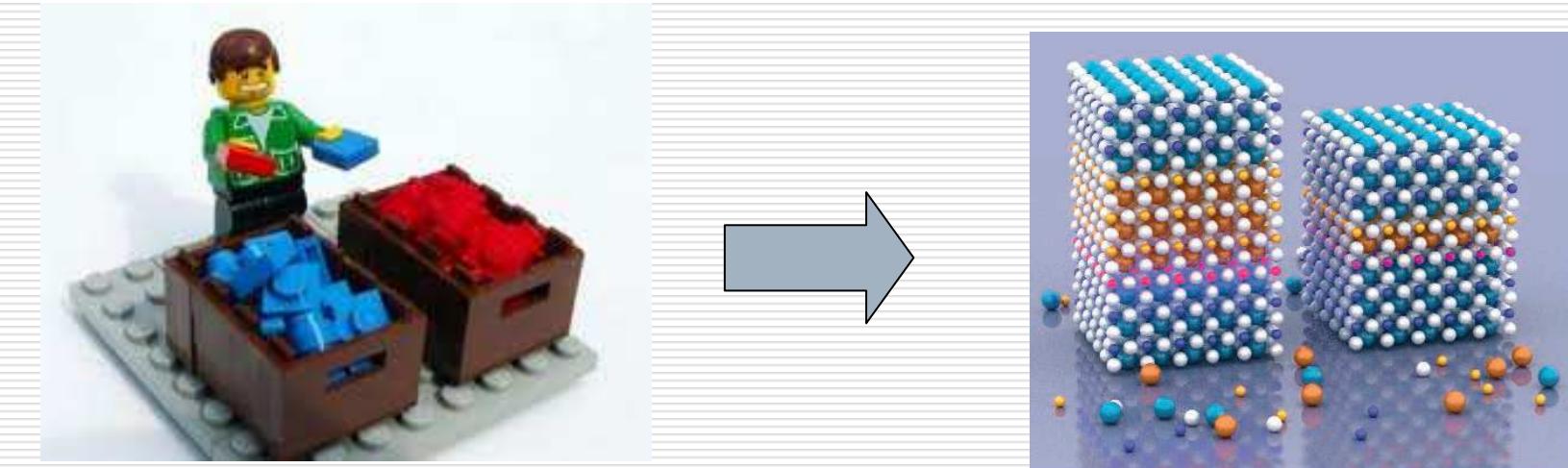
CNR SPIN & University of Roma "Tor Vergata"



Choosing the right bricks

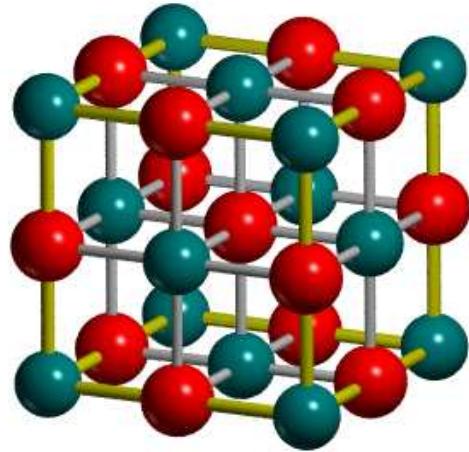


- Chemical compatibility
- Structural compatibility

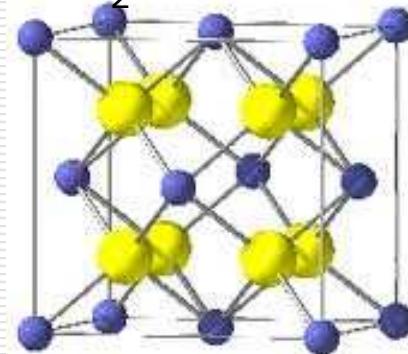


Oxide structures may be quite complex

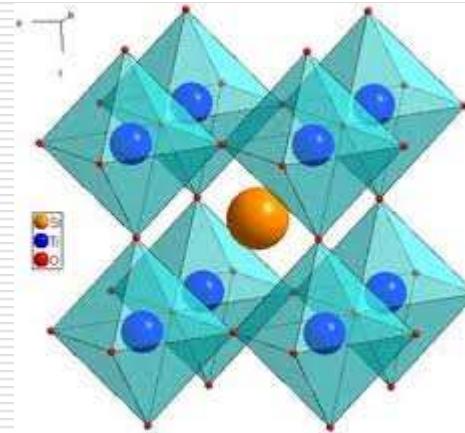
MgO structure



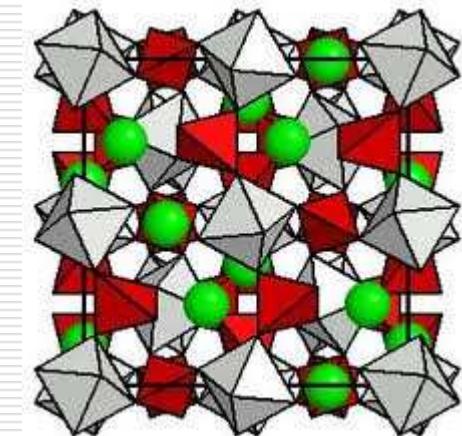
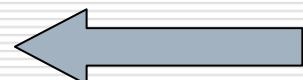
Fluorite structure



Perovskite structure
 SrTiO_3

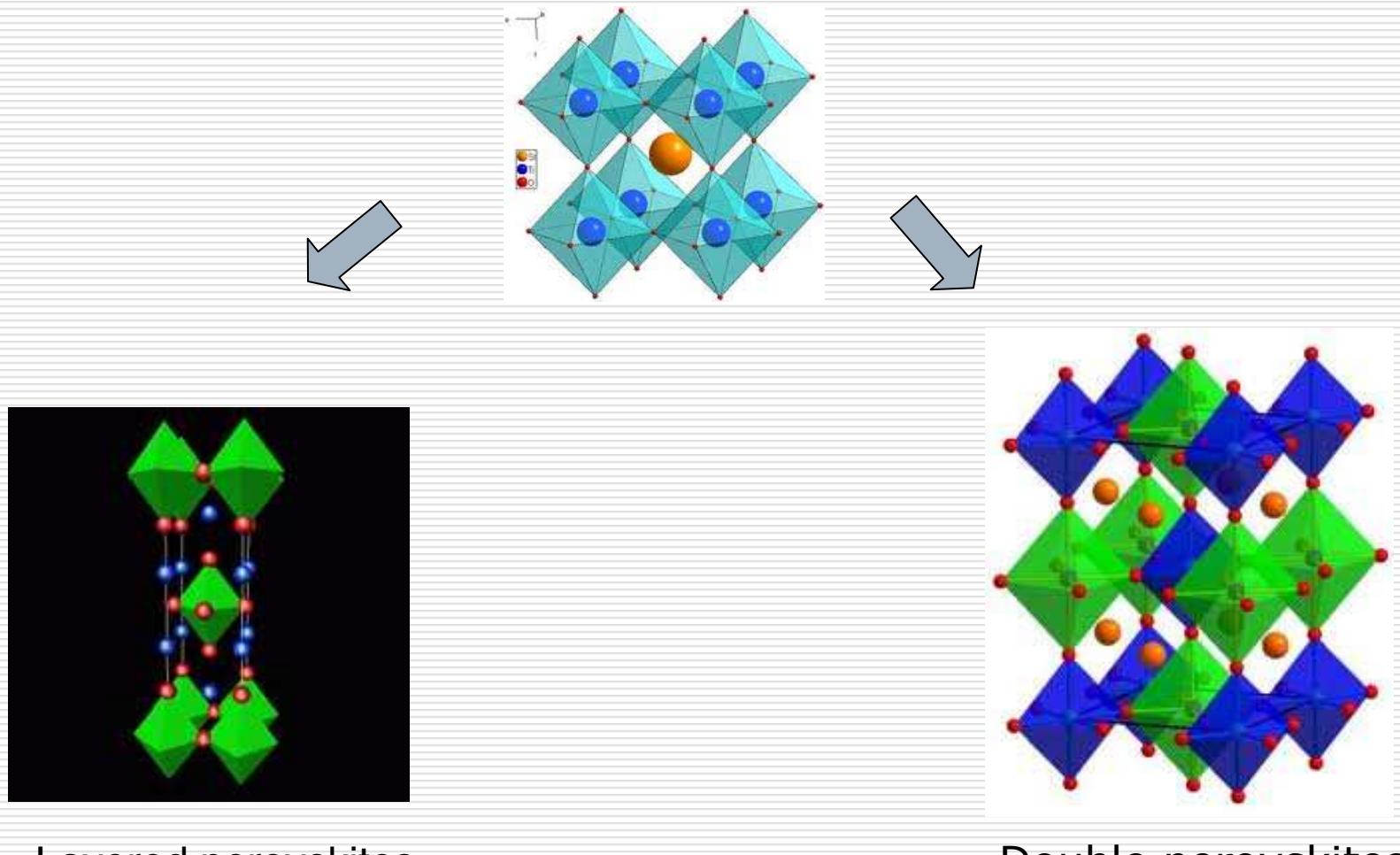


and much more !



Garnet structure

Perovskite-like structures: a good compromise

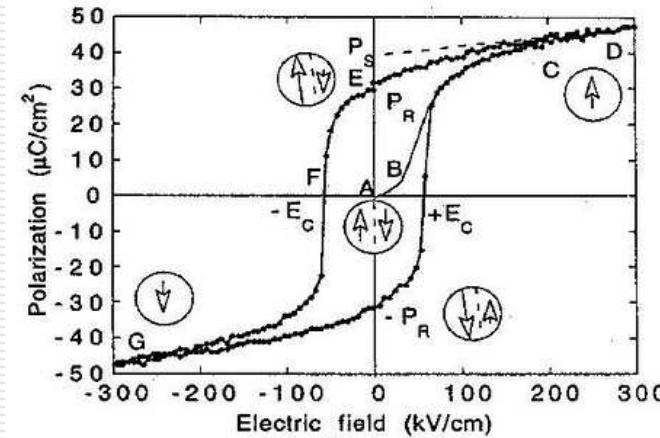
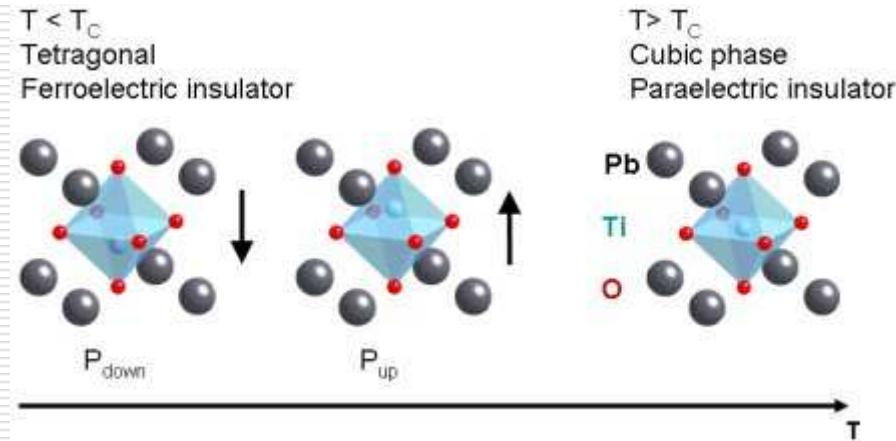


Layered perovskites

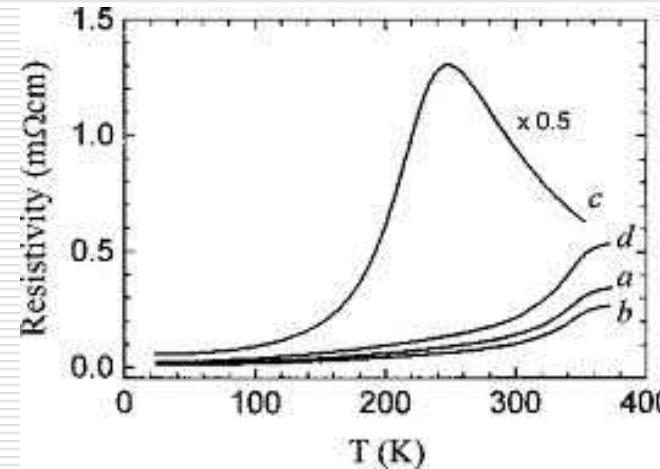
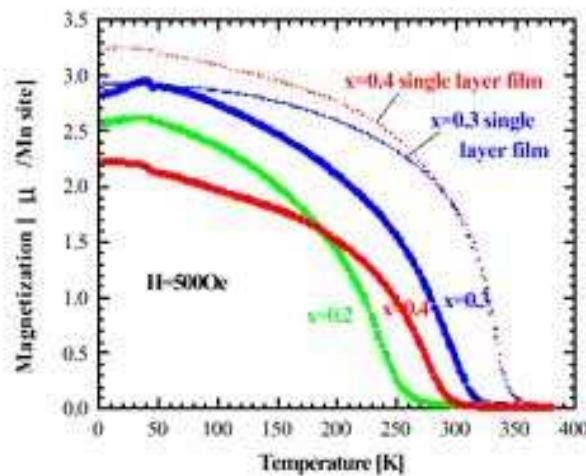
Double perovskites

Functional properties of perovskites

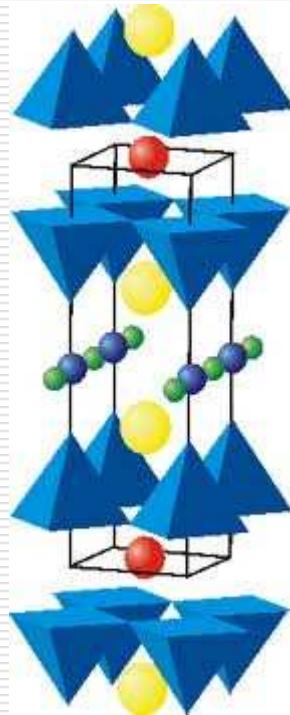
Ferroelectricity



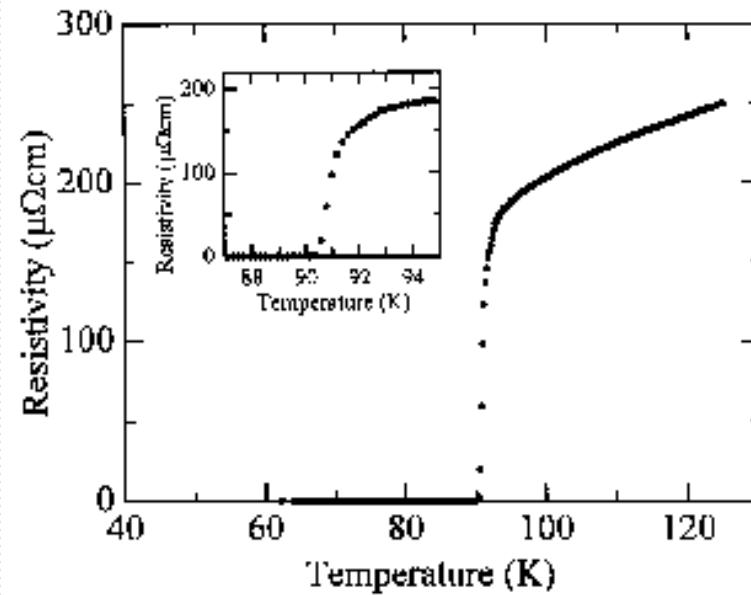
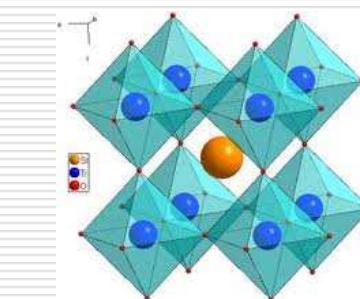
Half metallic, ferromagnetic



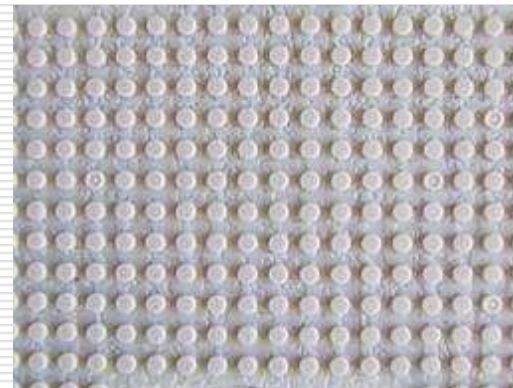
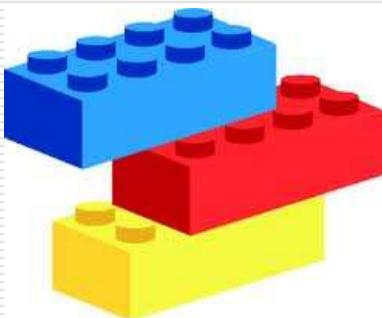
HTS layered superconductors



YBCO



Choice of the right substrate



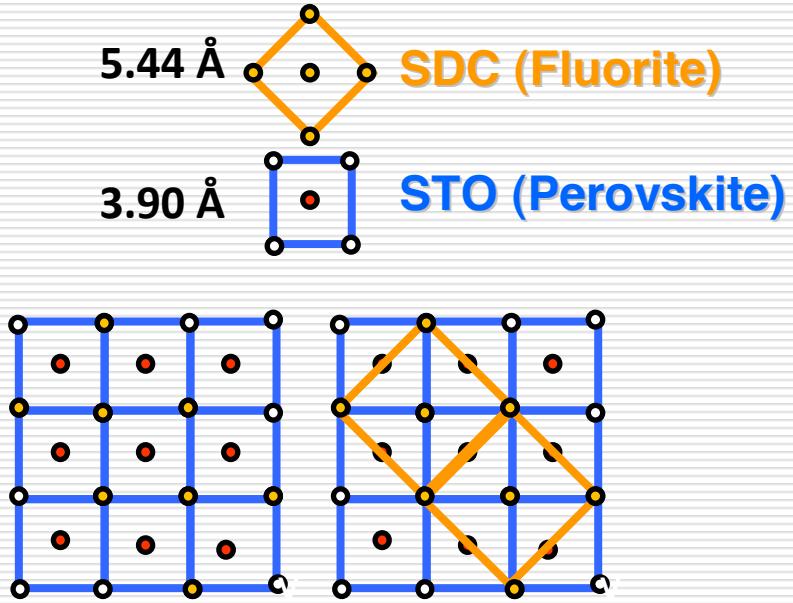
A large variety of commercial substrates available with the perovskite structure

SrTiO_3 cubic $a=3.905 \text{ \AA}$

LaAlO_3 cubic $a=3.76 \text{ \AA}$

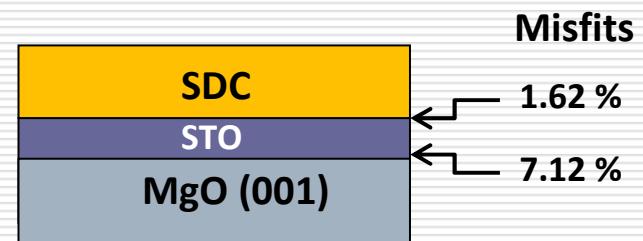
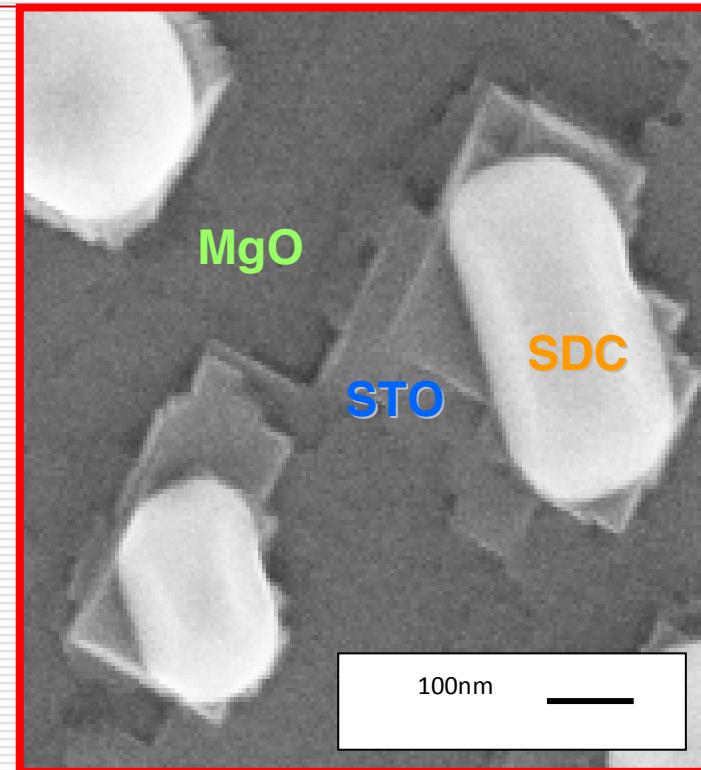
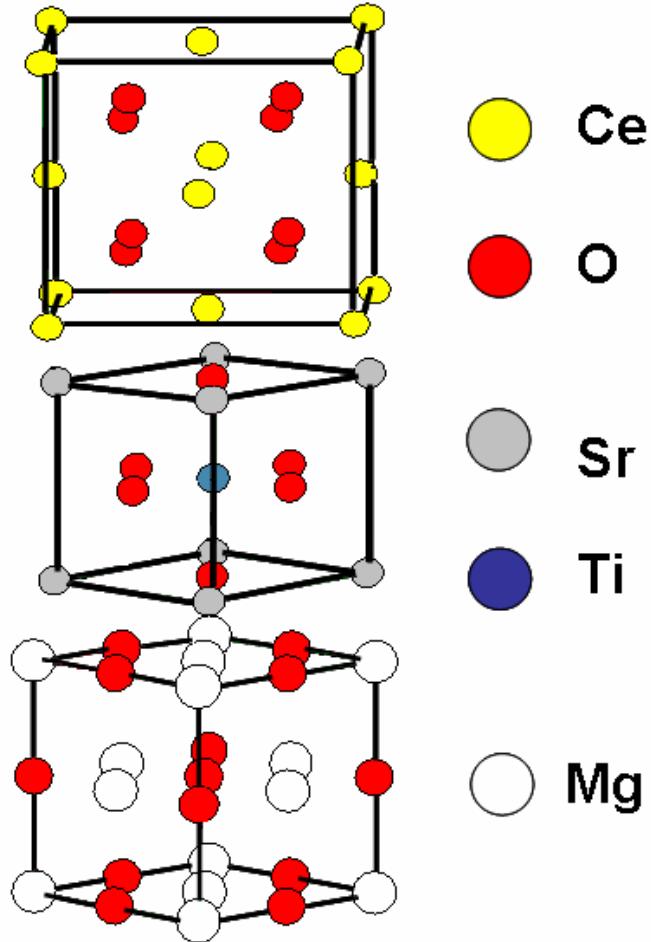
NdGaO_3 pseudocubic $a=3.86 \text{ \AA}$

$Sm_{0.2}Ce_{0.8}O_2$ (SDC) on STO



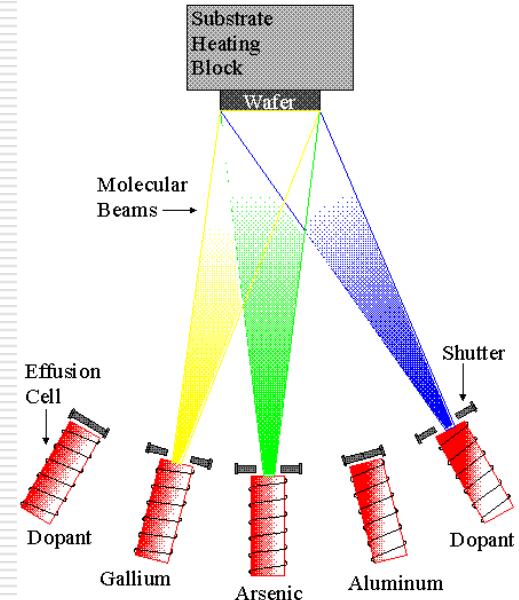
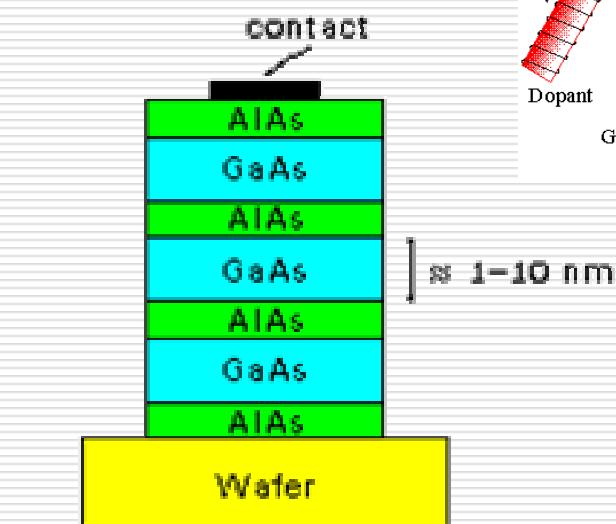
$$d_{110}^{SDC} = 5.44 \text{ \AA} / \sqrt{2} = 3.85 \text{ \AA} \quad \Rightarrow \quad \frac{d_{100}^{STO} - d_{110}^{SDC}}{d_{100}^{STO}} \cong 1.3\%$$

SDC/STO/MgO(001)

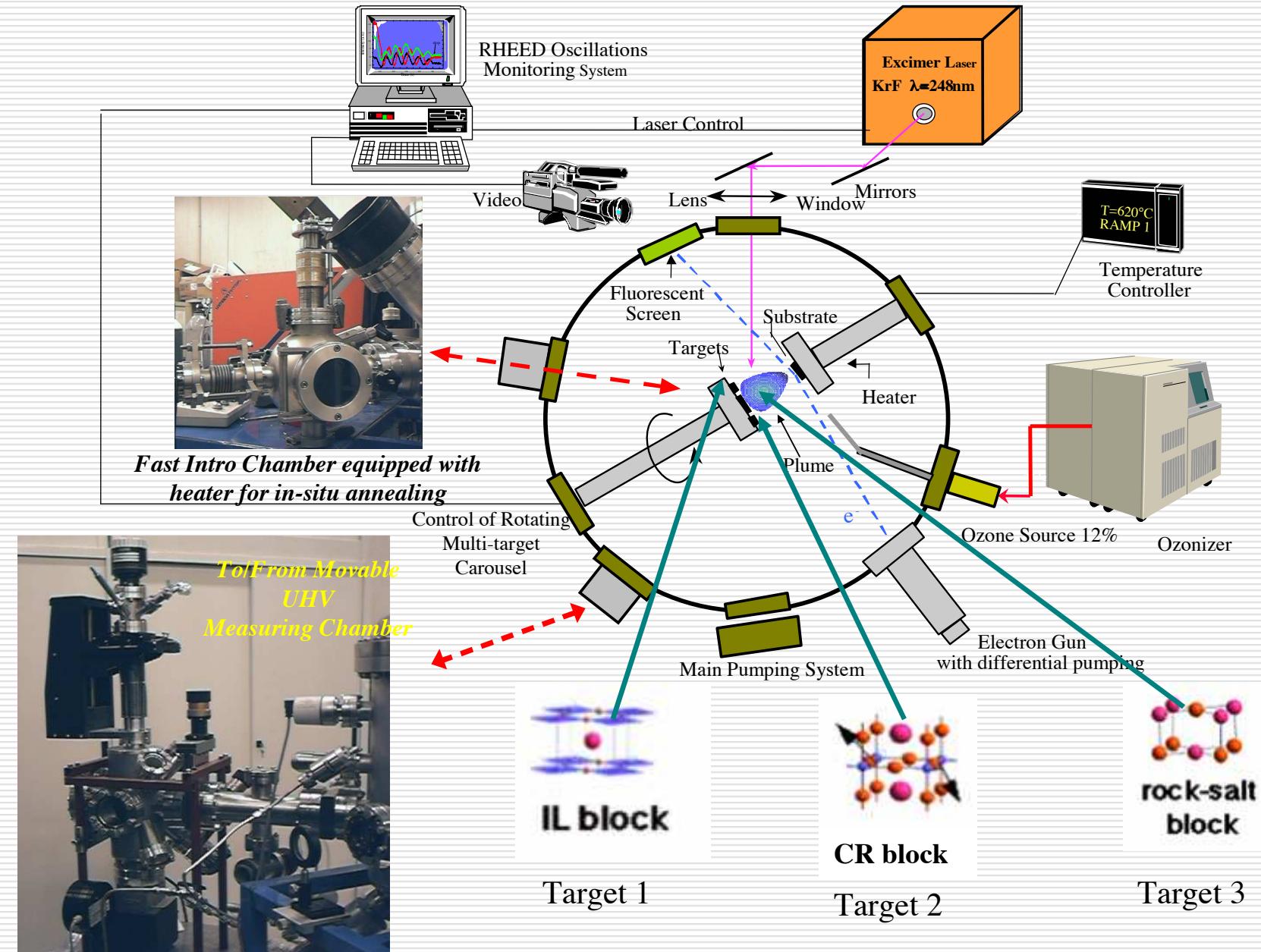


Schematic sketch of the correlation between the cubic cells of SDC, STO, and MgO.

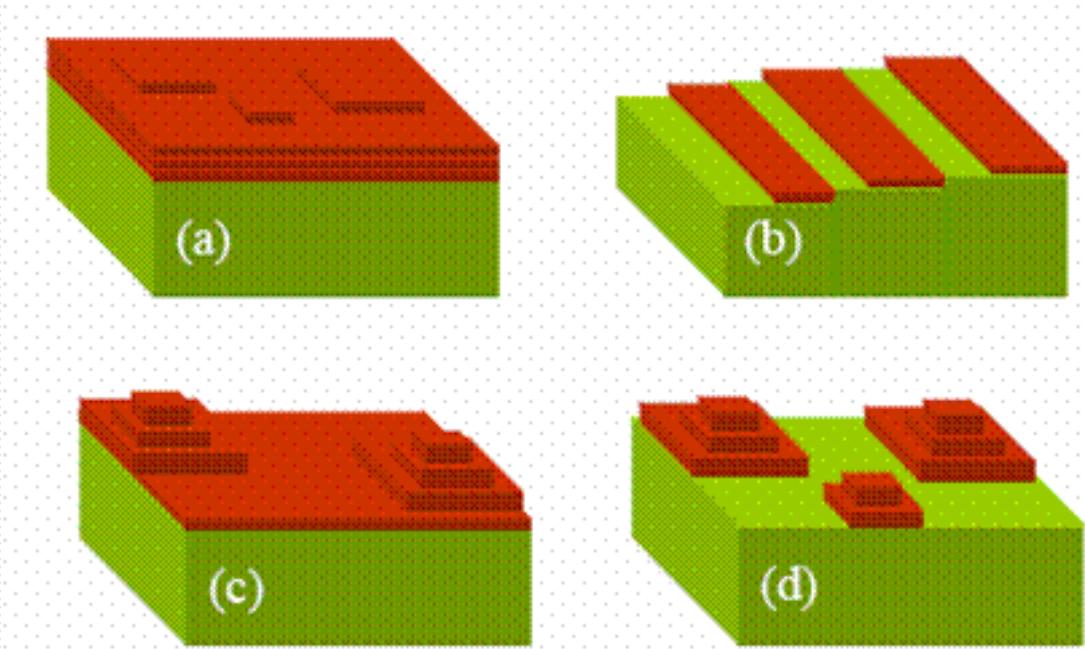
Deposition techniques: MBE



Pulsed Laser Deposition (PLD)

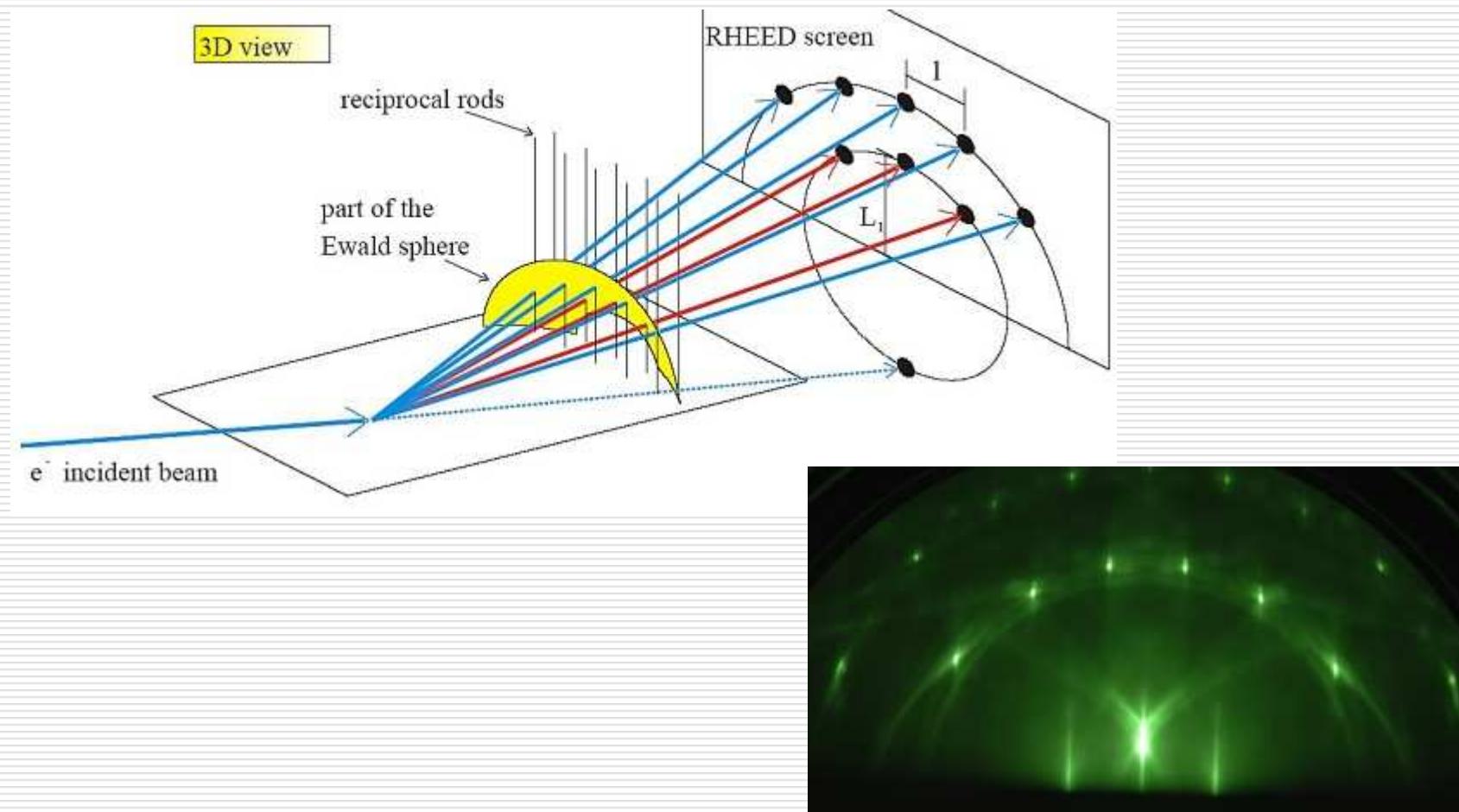


Growth mechanisms

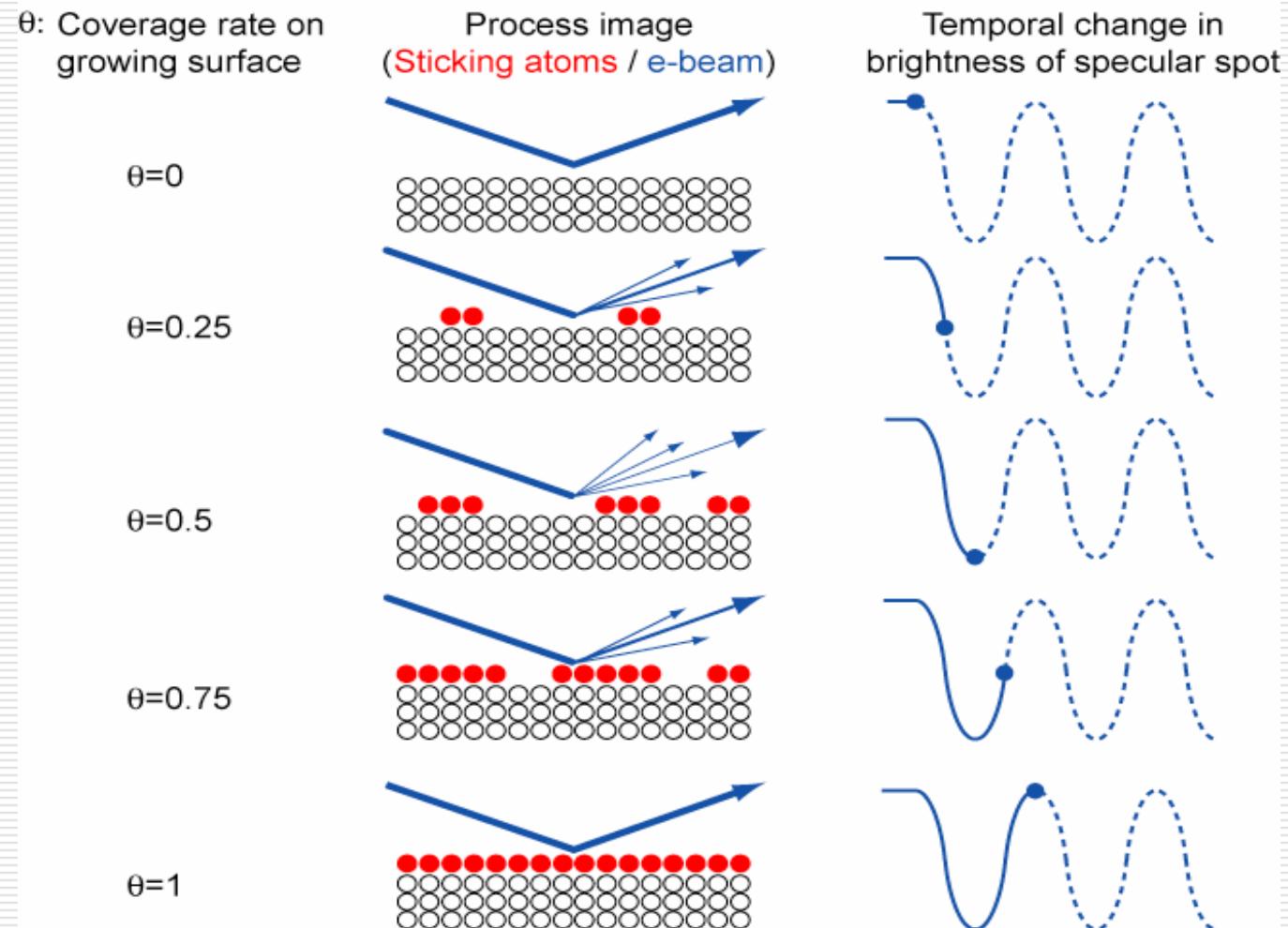


- (a) Frank-Van der Merwe or layer-by-layer growth,
- (b) step-flow growth,
- (c) Stranski-Krastanov growth,
- (d) Volmer-Weber growth.

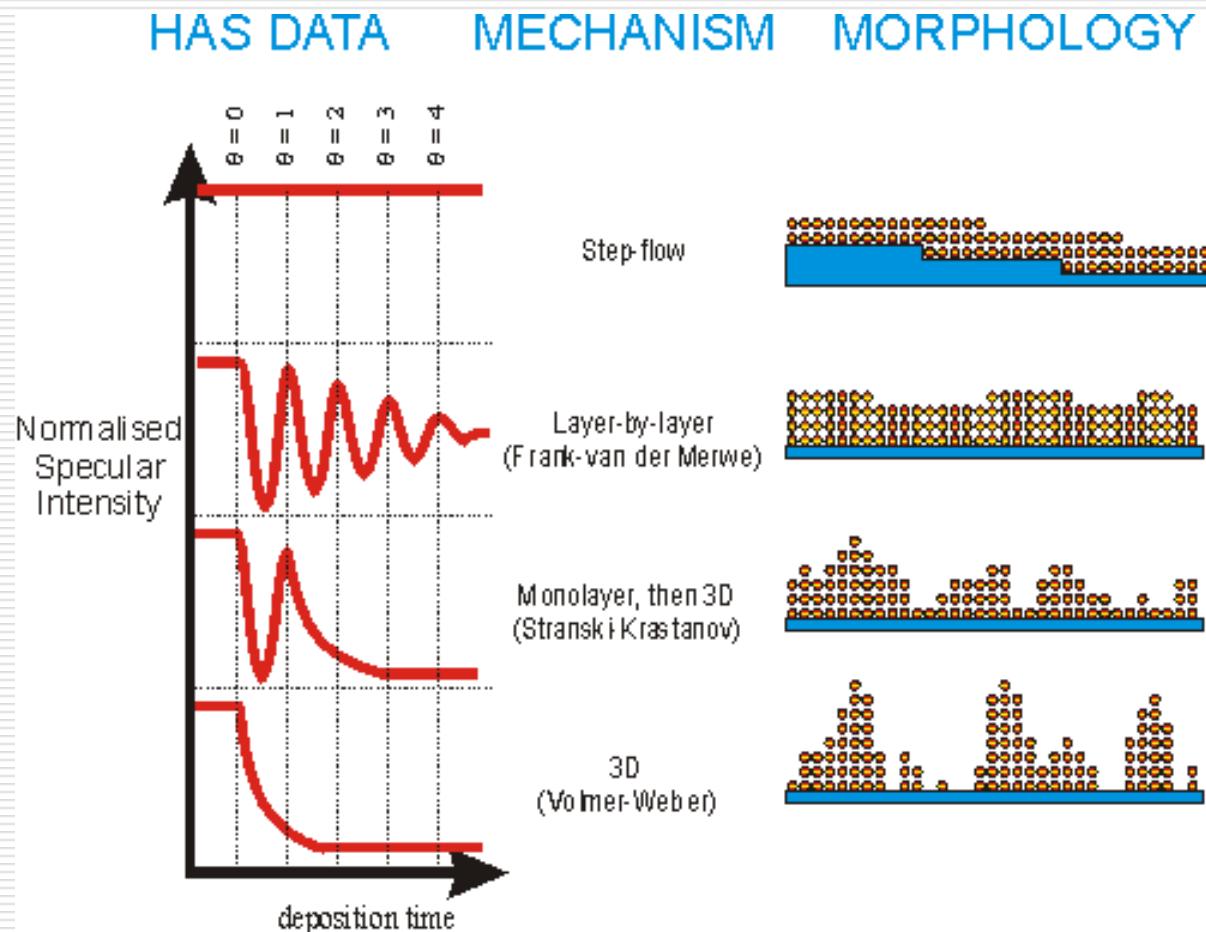
RHEED geometry



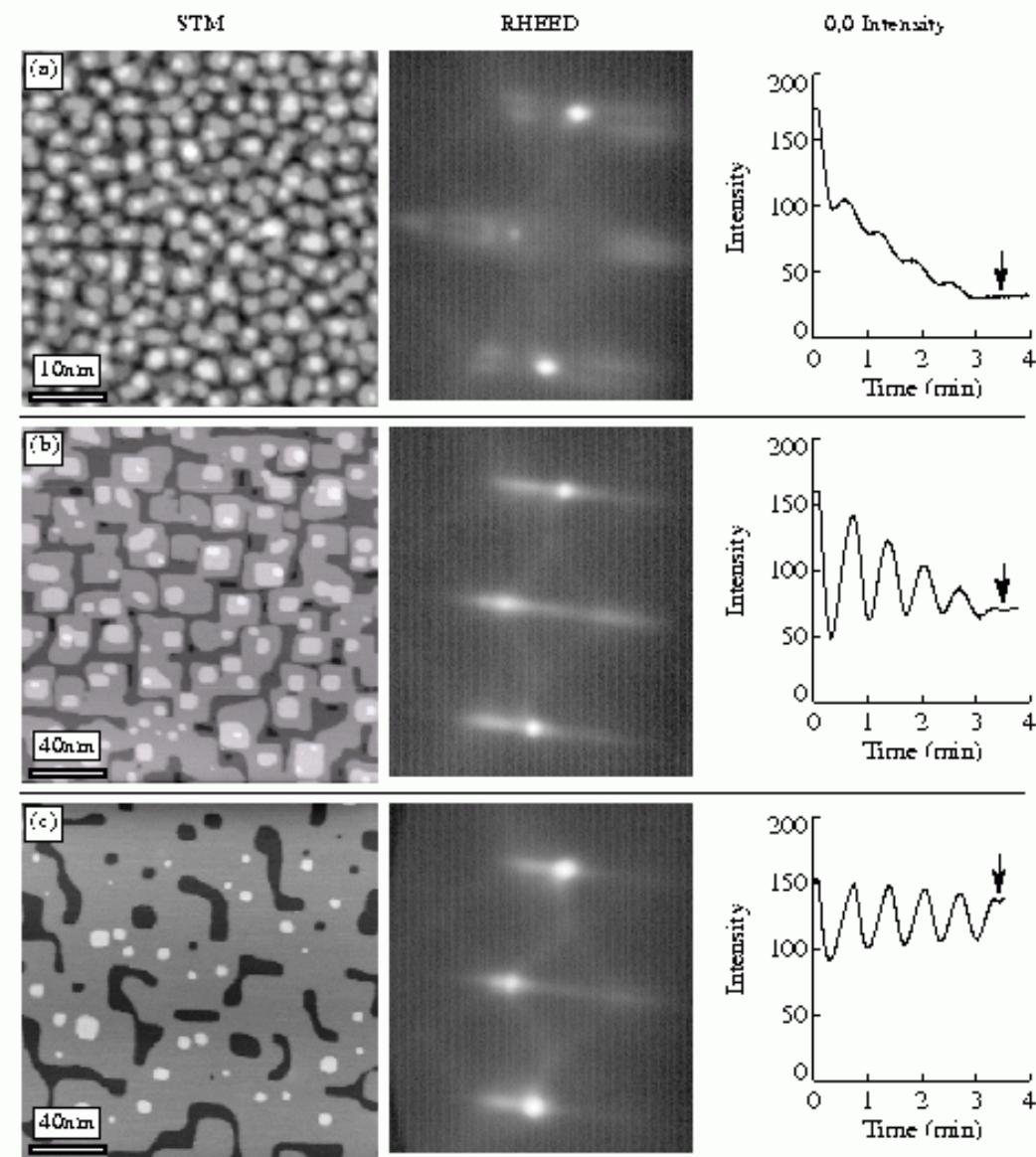
RHEED intensity oscillations



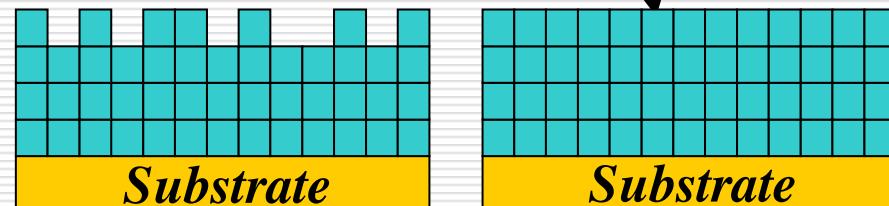
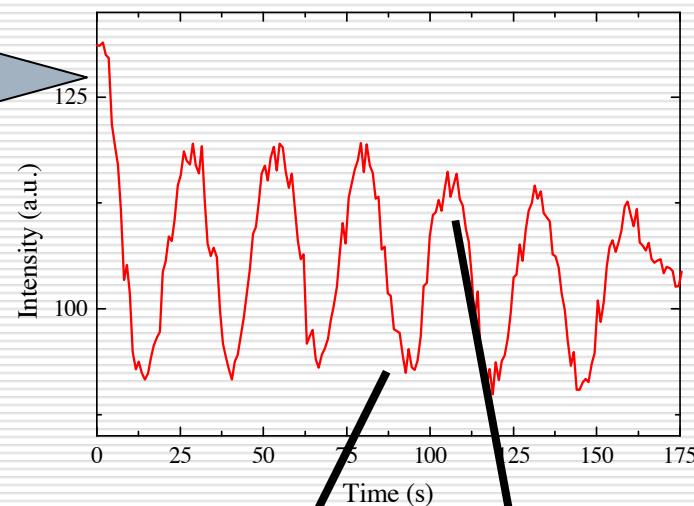
RHEED oscillations vs. surface morphology



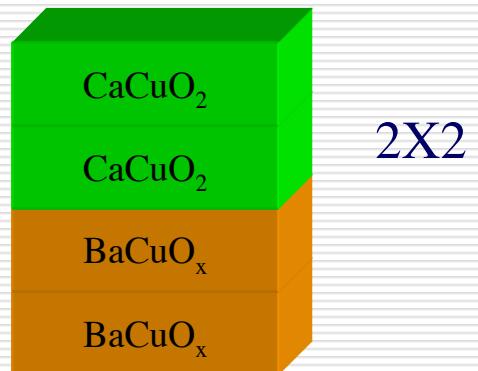
RHEED vs. surface morphology



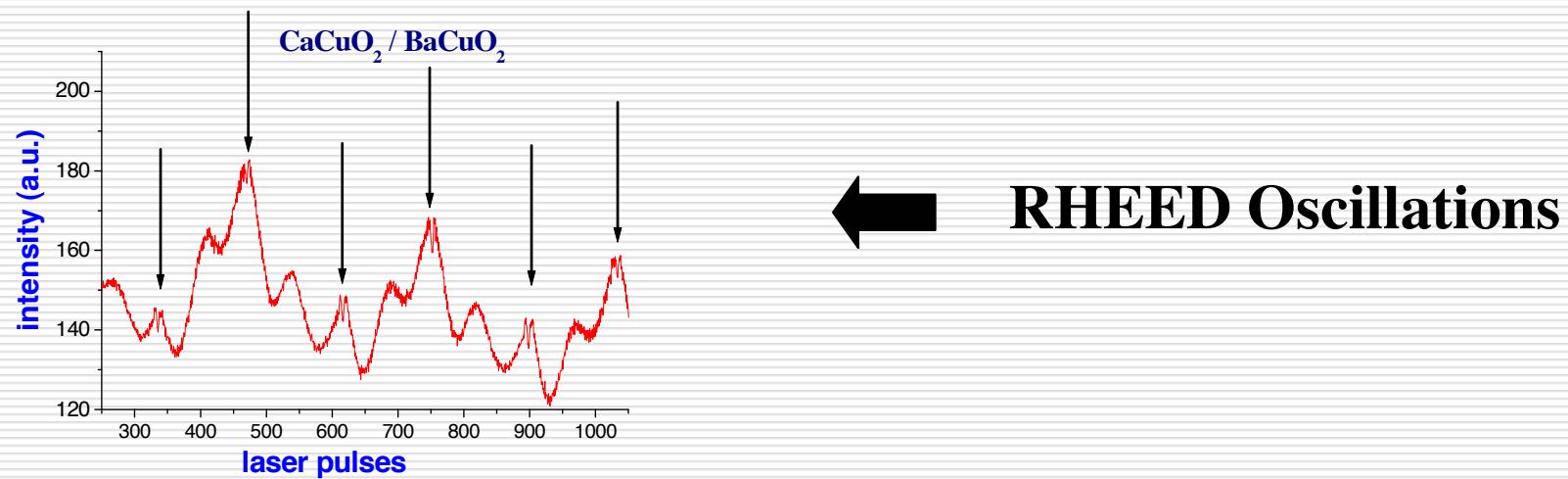
$\text{La}_{0.7} \text{Sr}_{0.3} \text{MnO}_3$ /film on NdGaO_3 : RHEED oscillations



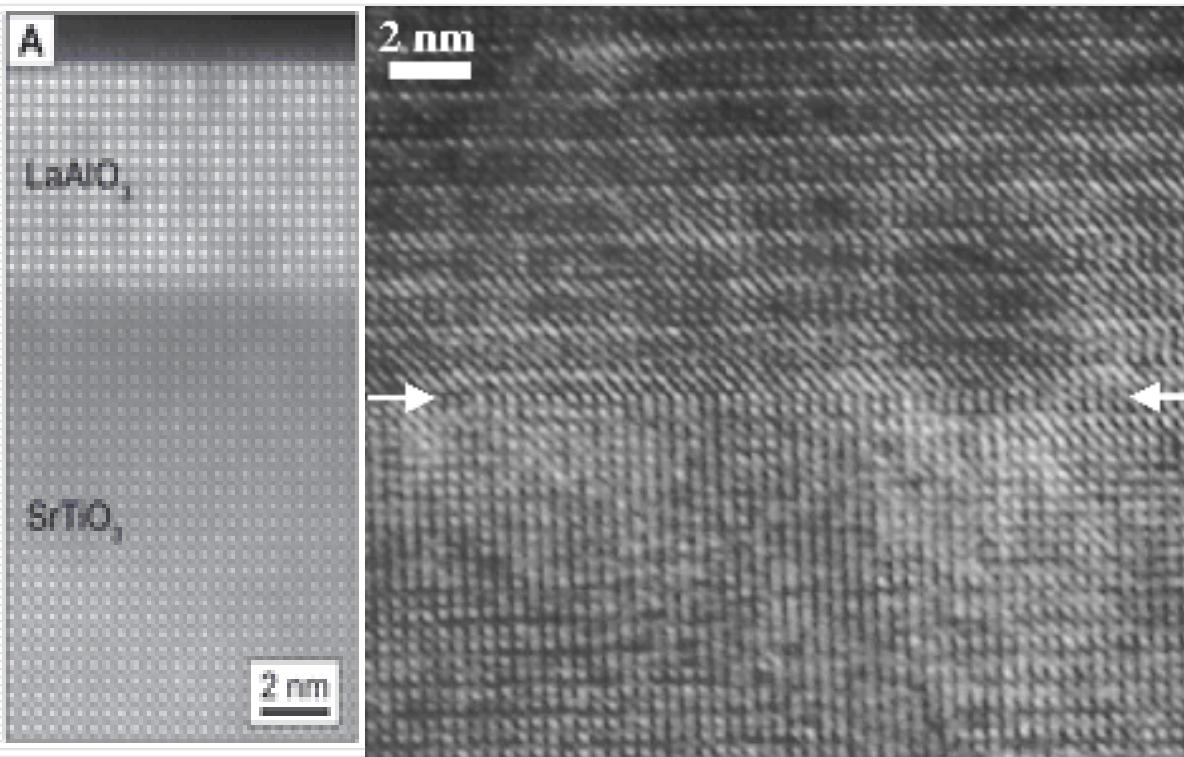
(BaCuO₂)₂/CaCuO₂₂ SL by Laser MBE technique



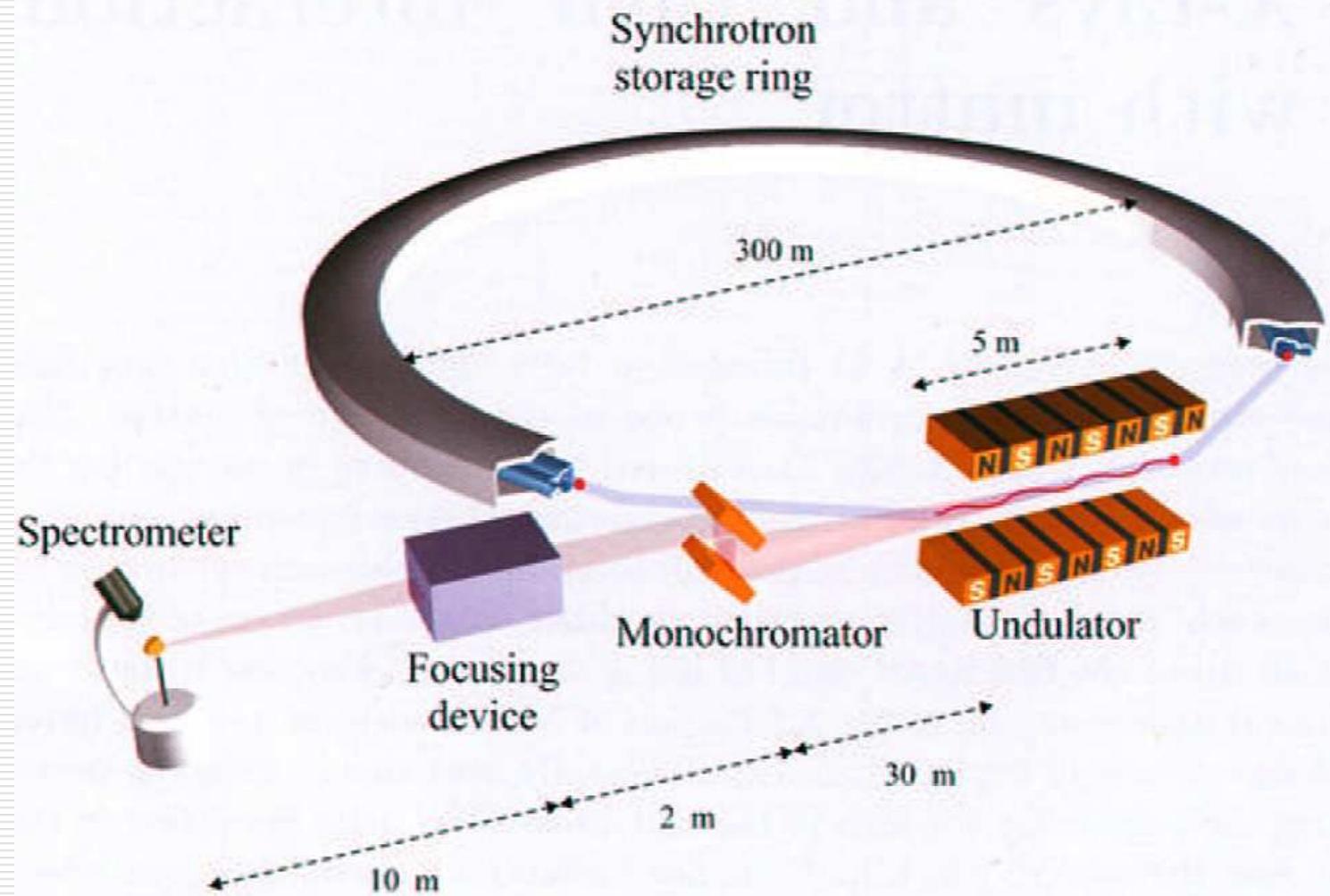
Multitarget Pulsed Laser Deposition



HRTEM of oxide interfaces and superlattices



Synchotron characterization



Characteristics of synchrotron radiation

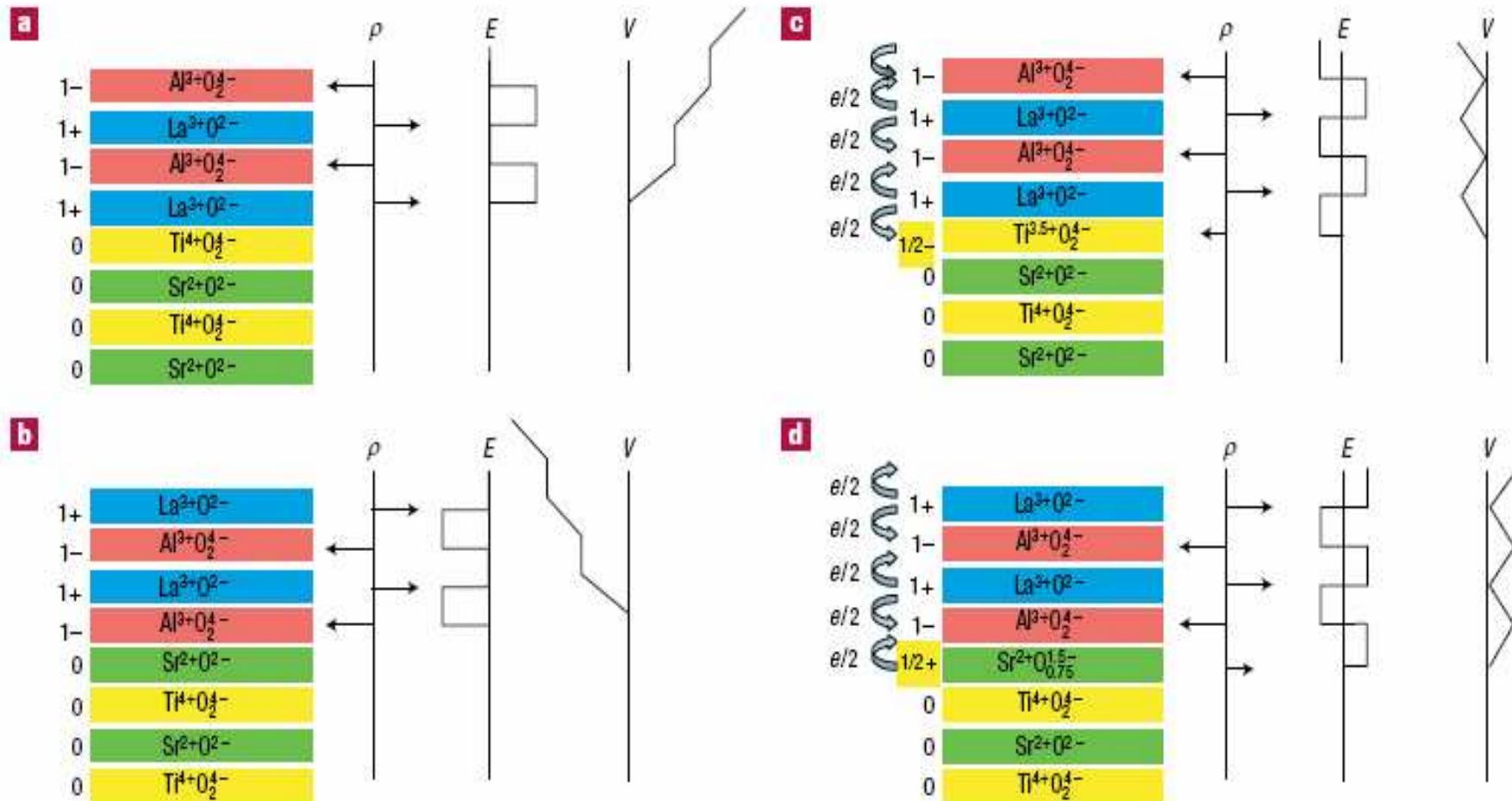
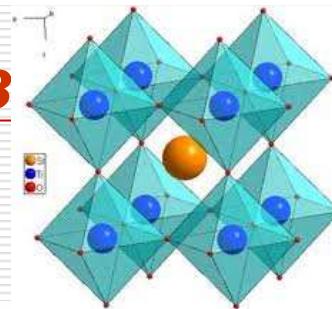
- High brilliance, exceeding other natural and artificial light sources by many orders of magnitude: 3rd generation sources typically have a brilliance larger than 10^{18} photons/s/mm²/mrad²/0.1%BW, where 0.1%BW denotes a bandwidth $10^{-3}\nu$ centered around the frequency ν .
- High collimation, i.e. small angular divergence of the beam
- Widely tunable in energy/wavelength by monochromatization (sub eV up to [the MeV range](#))
- High level of polarization (linear or elliptical)
- Pulsed [light emission](#) (pulse durations at or below one [nanosecond](#)),

An ideal tool for ultrathin layers (few u.c.) and interfaces

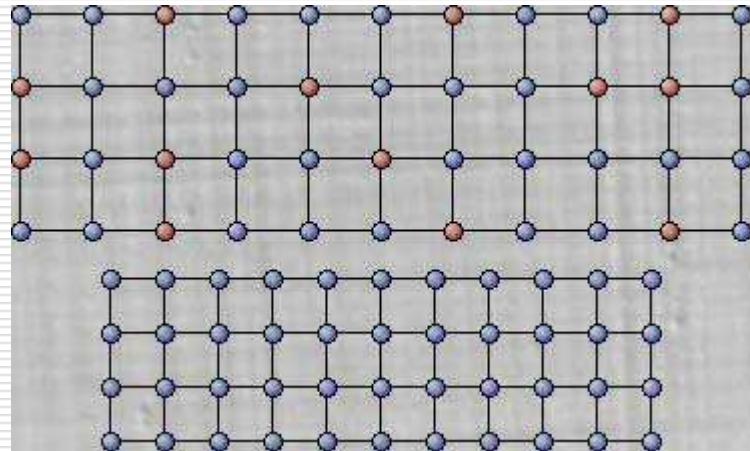
Interface effects

- Epitaxial strain as a consequence of lattice misfit
- Non equilibrium distribution of ions and vacancies (space charge region)
- Polarity discontinuity
- Electrical charge transfer

Interface effects: polar discontinuity $\text{LaAlO}_3/\text{SrTiO}_3$

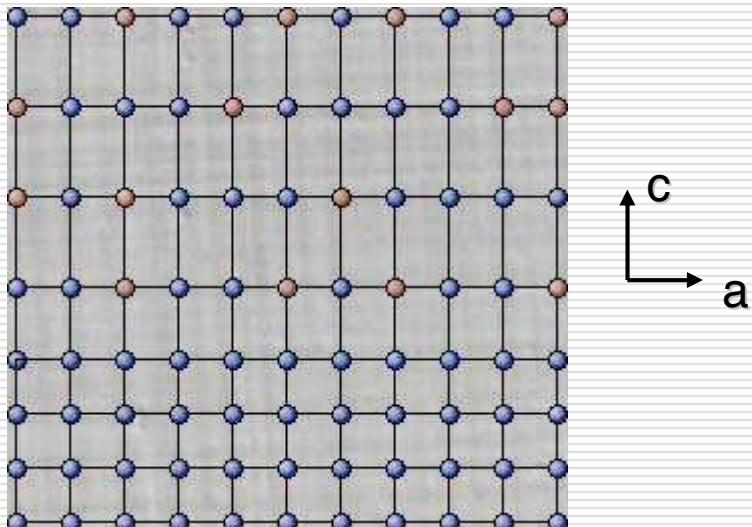
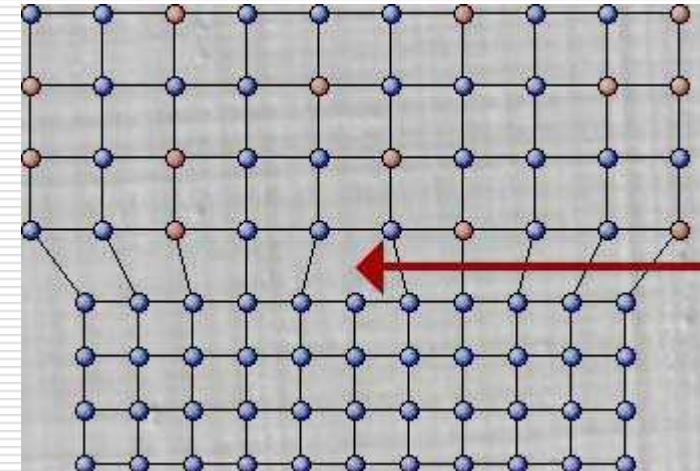


Interface effects: epitaxial strain



$\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$

LaAlO_3



$$\frac{\Delta c}{c} = - \frac{2\nu}{1-\nu} \frac{\Delta a}{a}$$

$$\nu \approx 0.3 - 0.5$$

Three examples

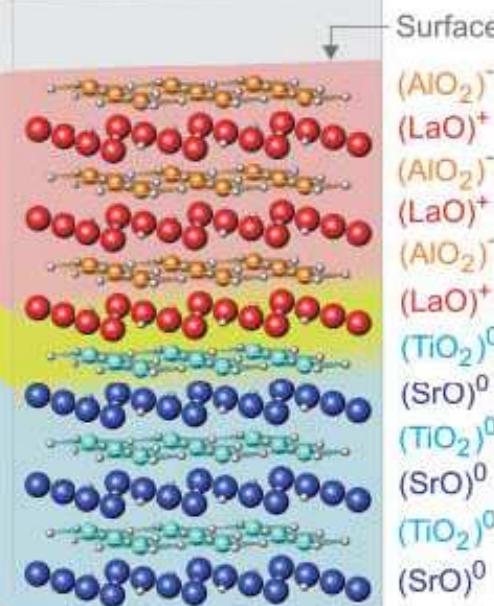
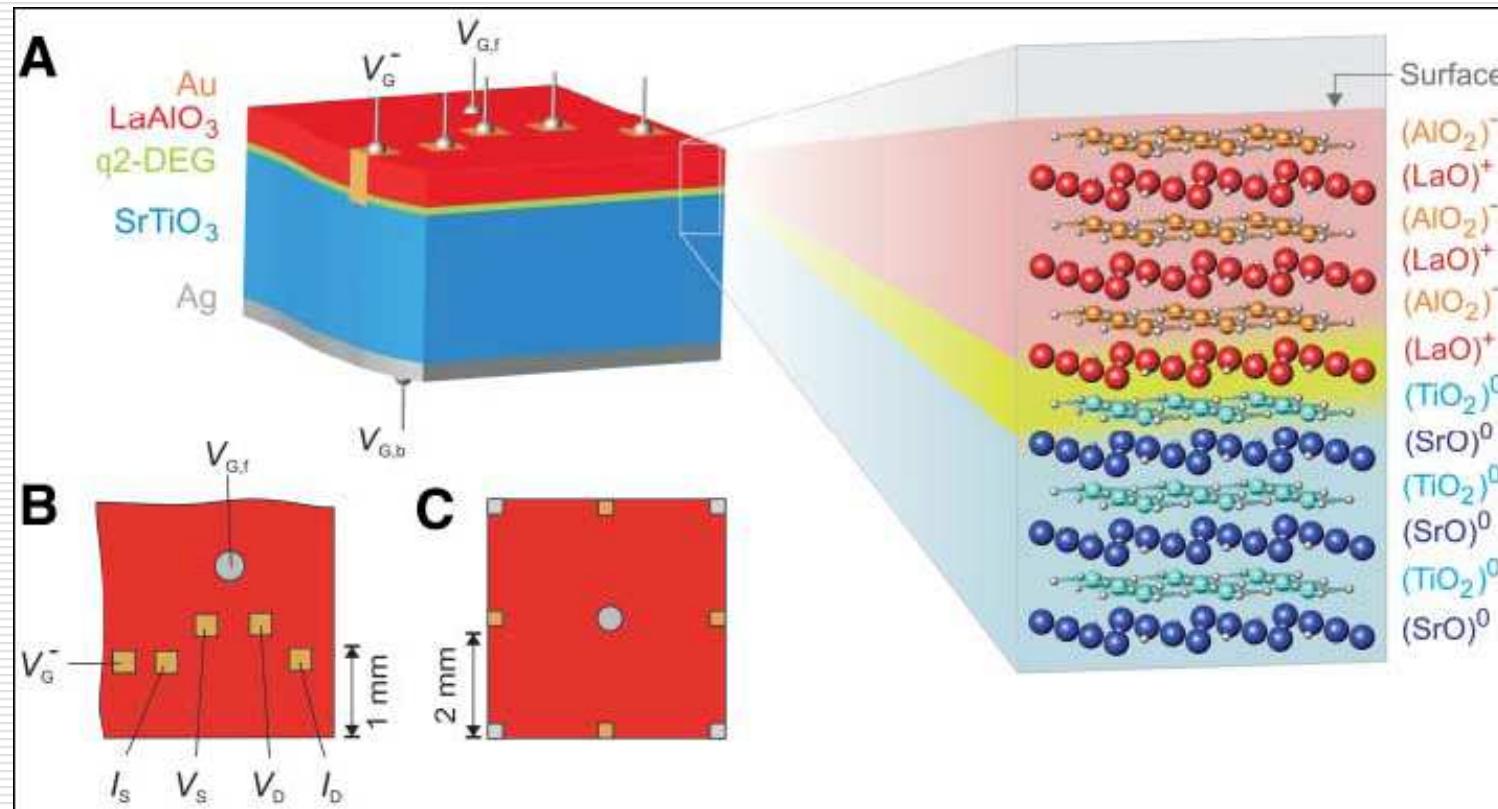
- 2D electron gas in oxide heterostructures
- Induced superconductivity in oxide heterostructures
- Enhanced ionic conductivity in oxide heterostructures

Engineering complex oxide heterostructures

Tunable Quasi-Two-Dimensional Electron Gases in Oxide Heterostructures

S. Thiel,¹ G. Hammerl,¹ A. Schmehl,² C. W. Schneider,¹ J. Mannhart^{1*}

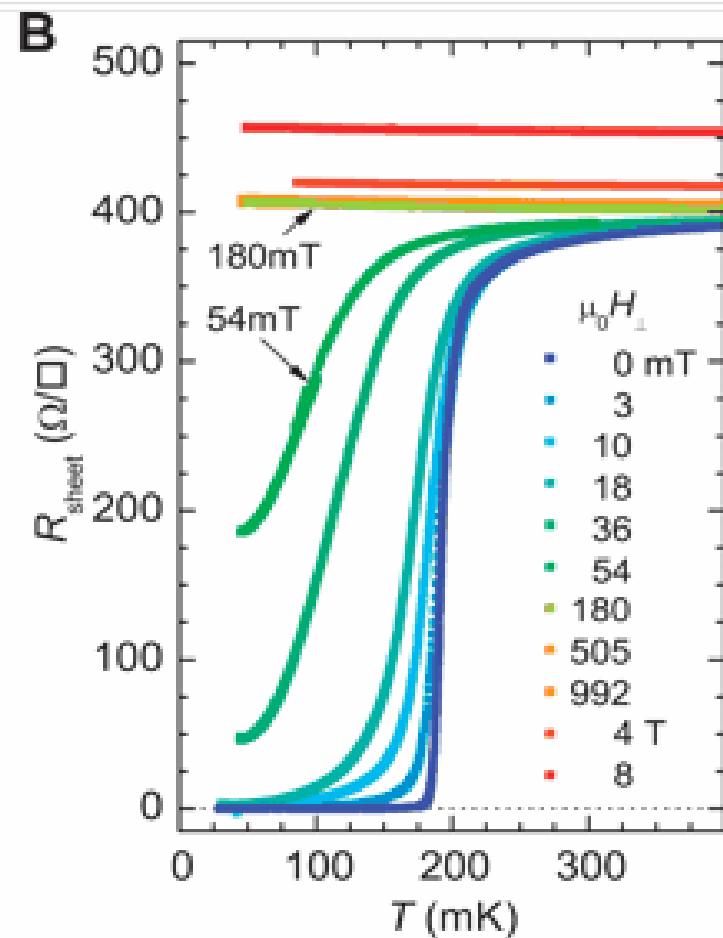
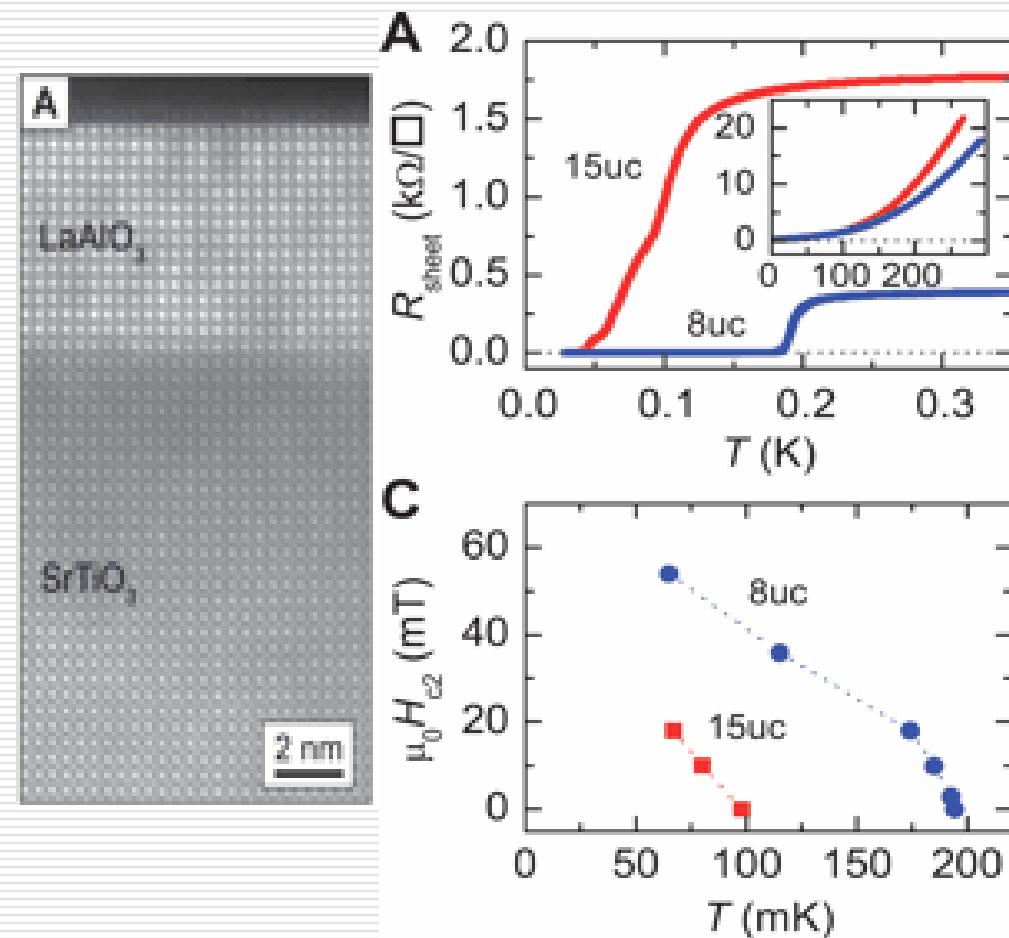
Science 313, 1942 (2006)



Superconductivity at the interface LAO/STO

Superconducting Interfaces Between Insulating Oxides

N. Reyren et al., *Science* 31 August 2007: Vol. 317. no. 5842, pp. 1196 - 1199

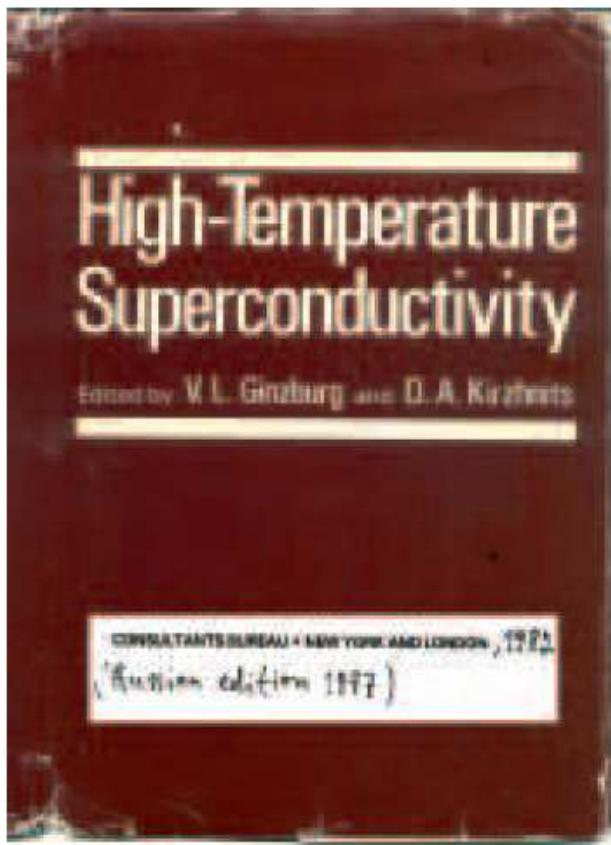


Induced superconductivity in oxide based heterostructures

Nobel Lecture, December 8, 2003

by

Vitaly L. Ginzburg



High-Temperature Superconductivity
(Moscow: Nauka, 1977)

From the Nobel Lecture (citing his book)

"On the basis of general theoretical considerations, I believe at present that the most reasonable estimate is $T_c \approx 300$ K. ...omissis... In this scheme, the most promising materials – from the point of view of the possibility of raising T_c – are, apparently, layered compounds and dielectric–metal–dielectric sandwiches....

And now oxide heterostructures come!

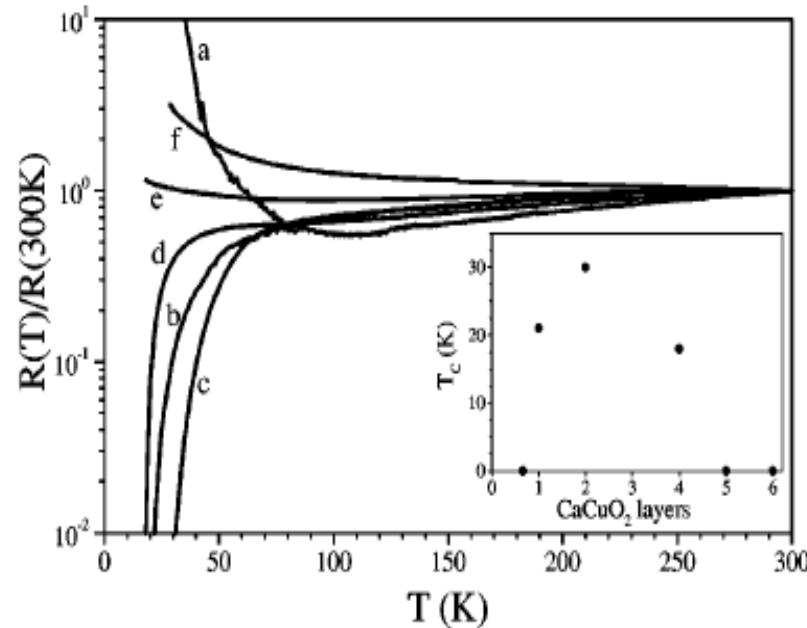
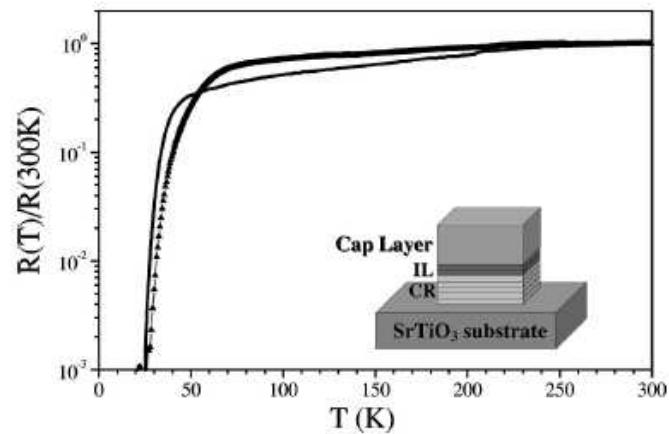
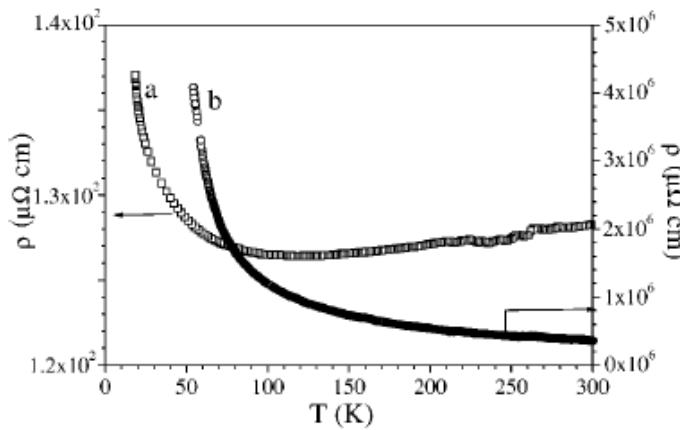
Ginzburg V L Phys. Lett. 13 101 (1964)

Recent experimental evidence

PHYSICAL REVIEW B 66, 094505 (2002)

Superconductivity in the surface layer of $\text{SrTiO}_3/(\text{Ba}_{0.9}\text{Nd}_{0.1})\text{CuO}_{2+x}/\text{CaCuO}_2$ heteroepitaxial structures

G. Balestrino, S. Lavanga, P. G. Medaglia, P. Orgiani, and A. Tebano



Further evidence

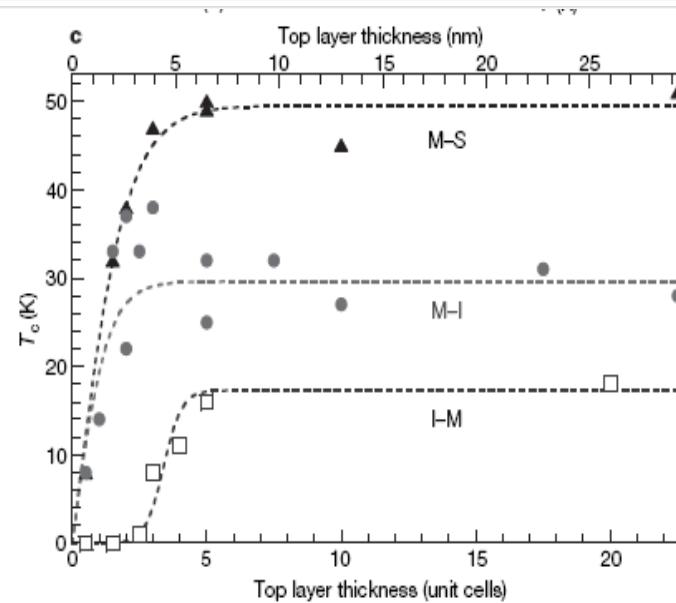
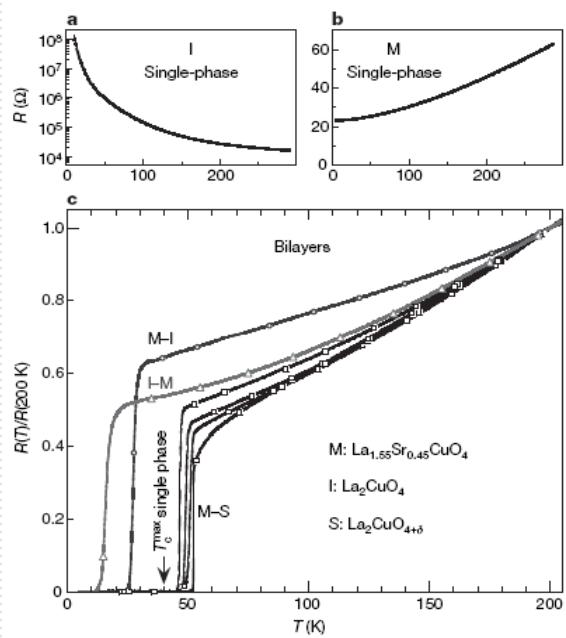
nature

Vol 455 | 9 October 2008 | doi:10.1038/nature07293

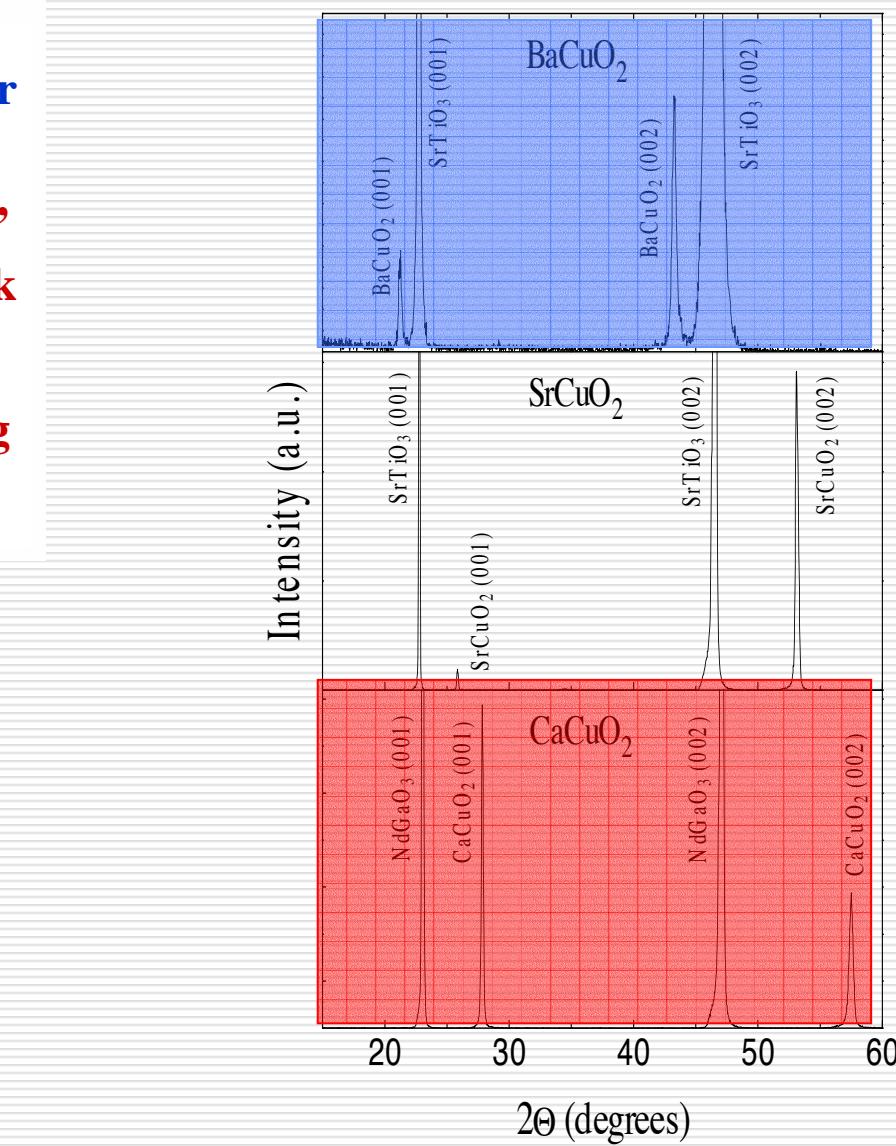
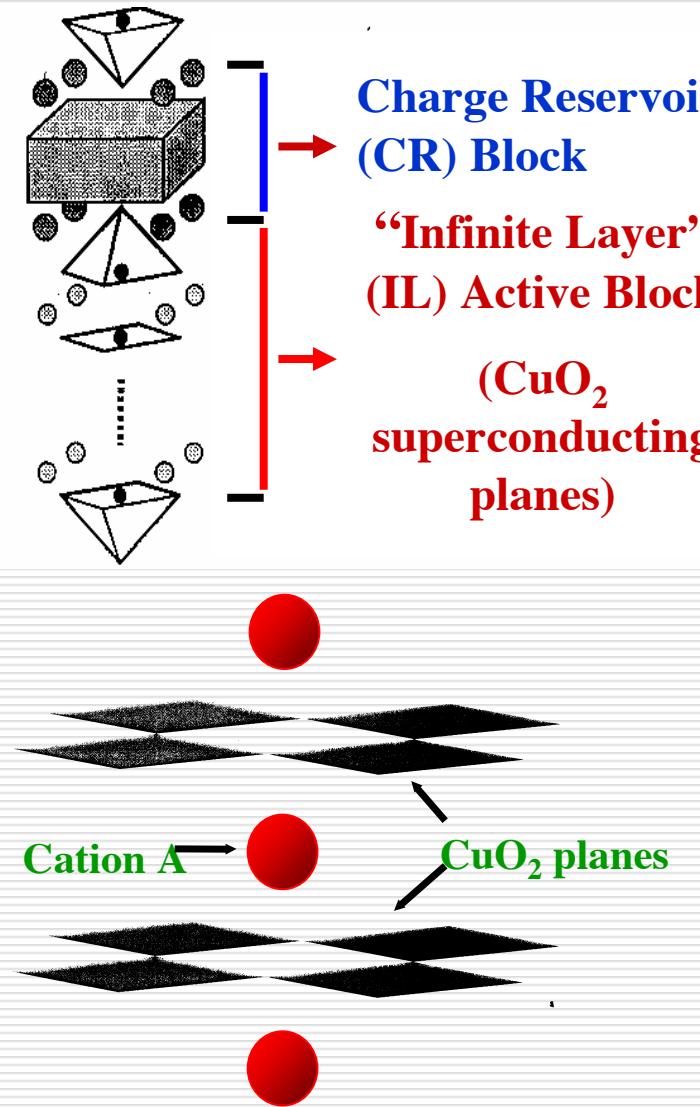
LETTERS

High-temperature interface superconductivity between metallic and insulating copper oxides

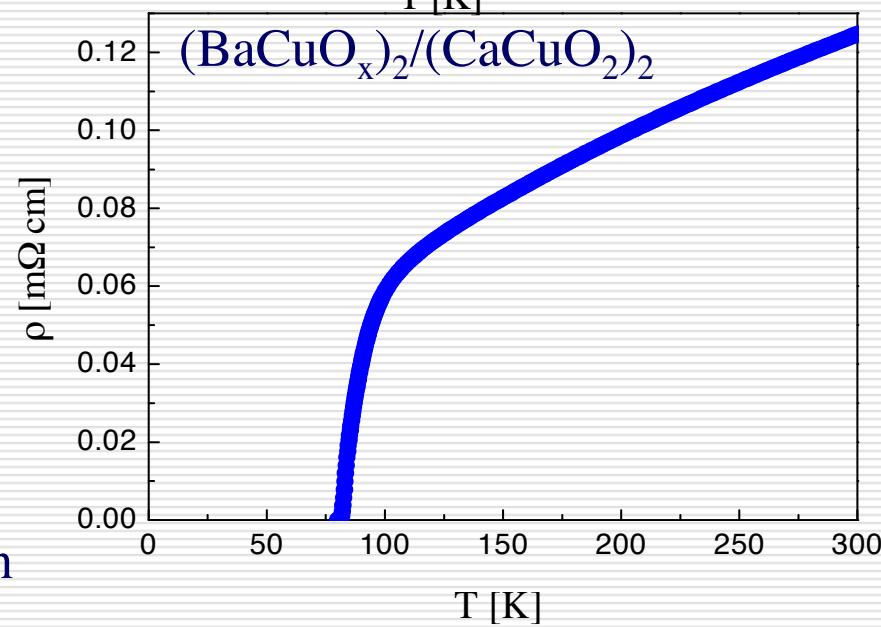
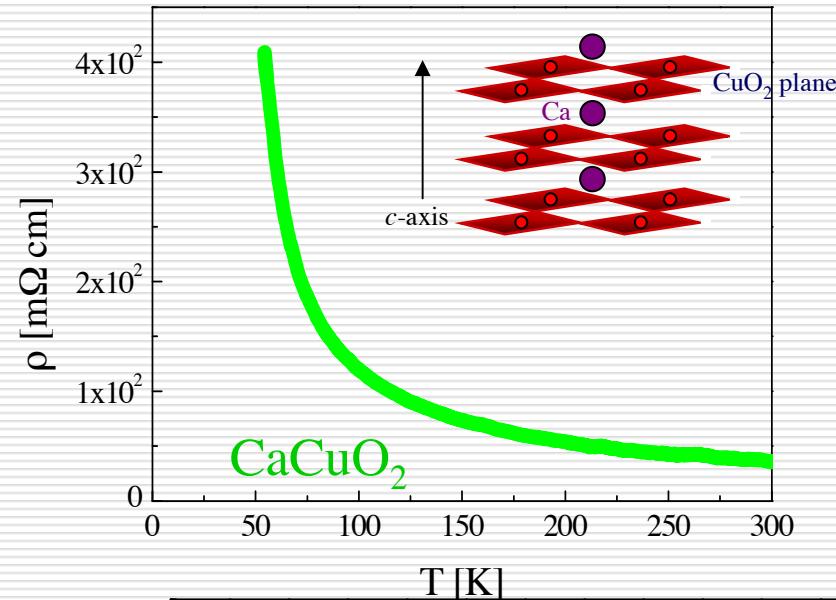
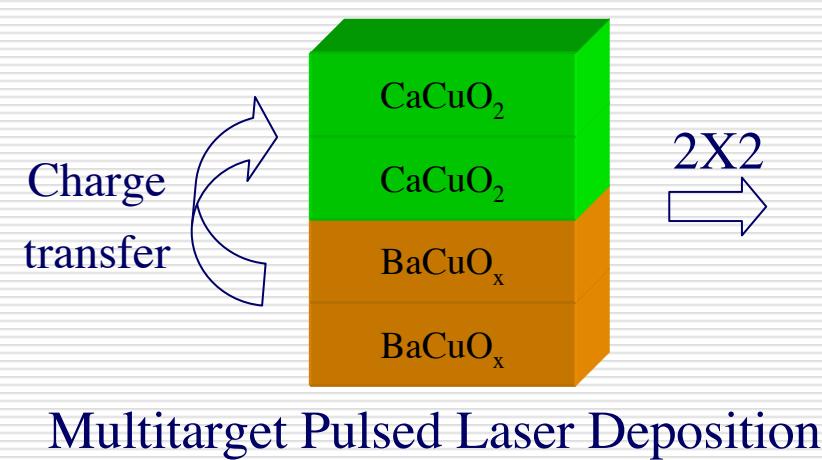
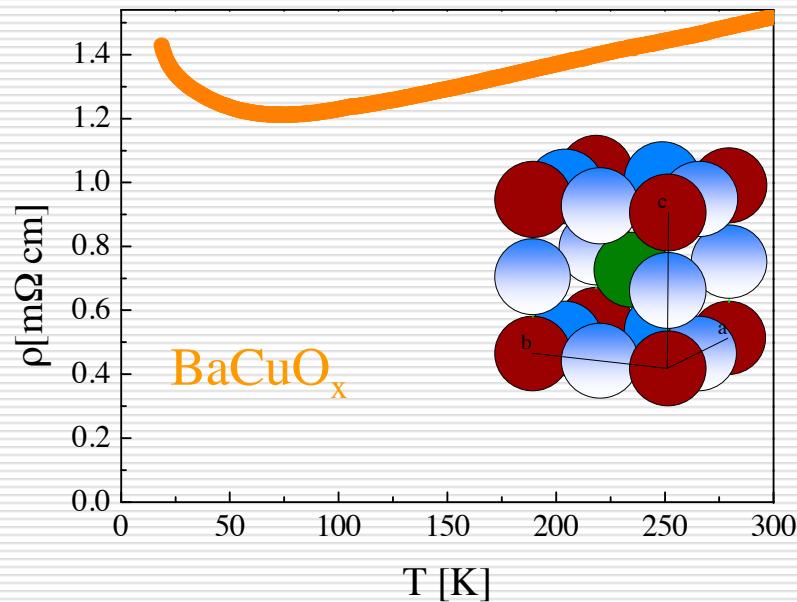
A. Gozar¹, G. Logvenov¹, L. Fitting Kourkoutis², A. T. Bollinger¹, L. A. Giannuzzi³, D. A. Muller² & I. Bozovic¹



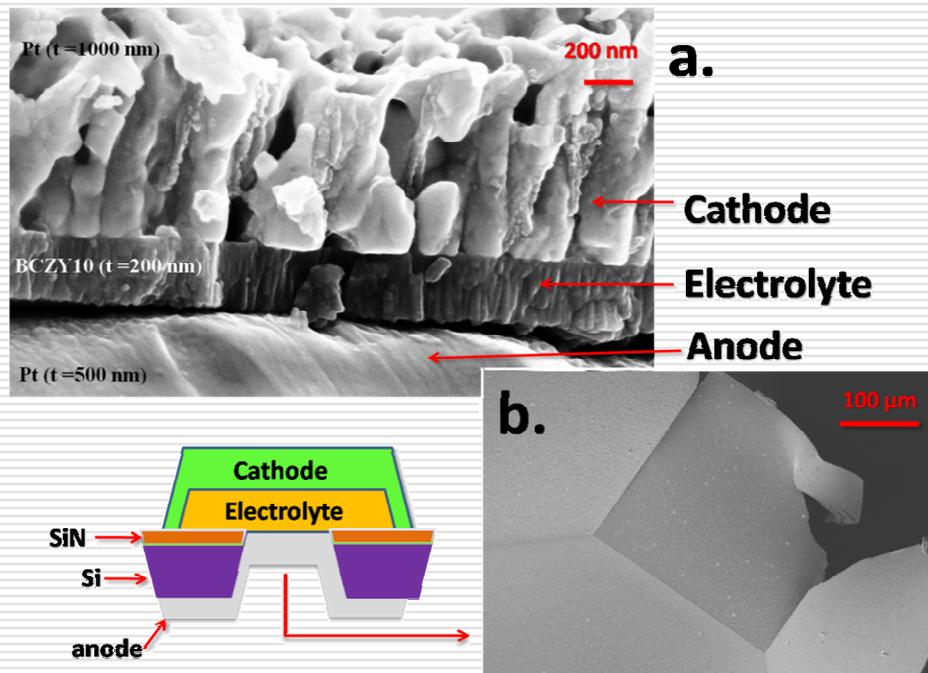
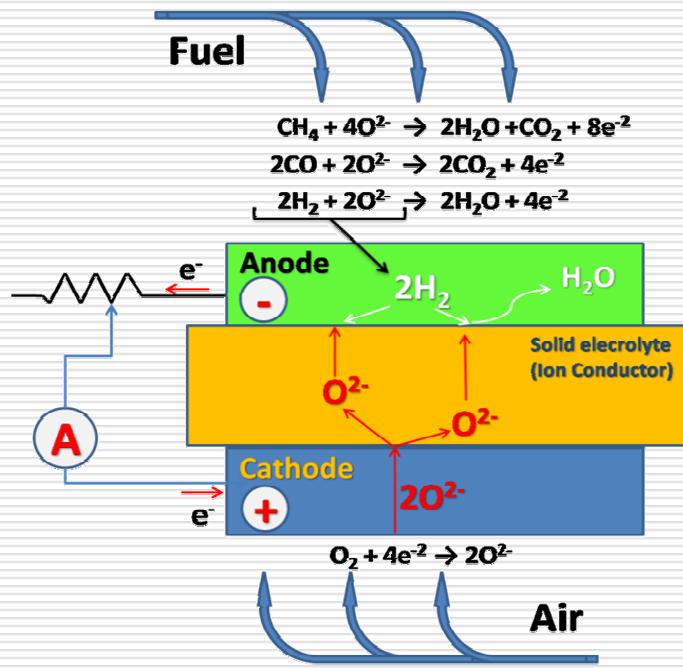
Engineering superconducting superlattices



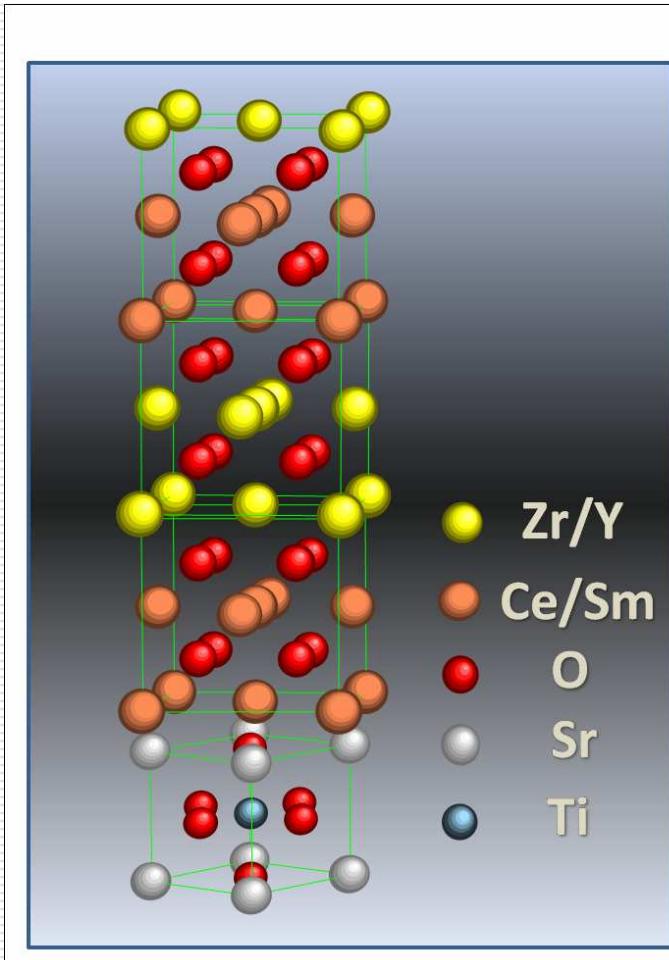
Superconducting superlattices on atomic dimensions



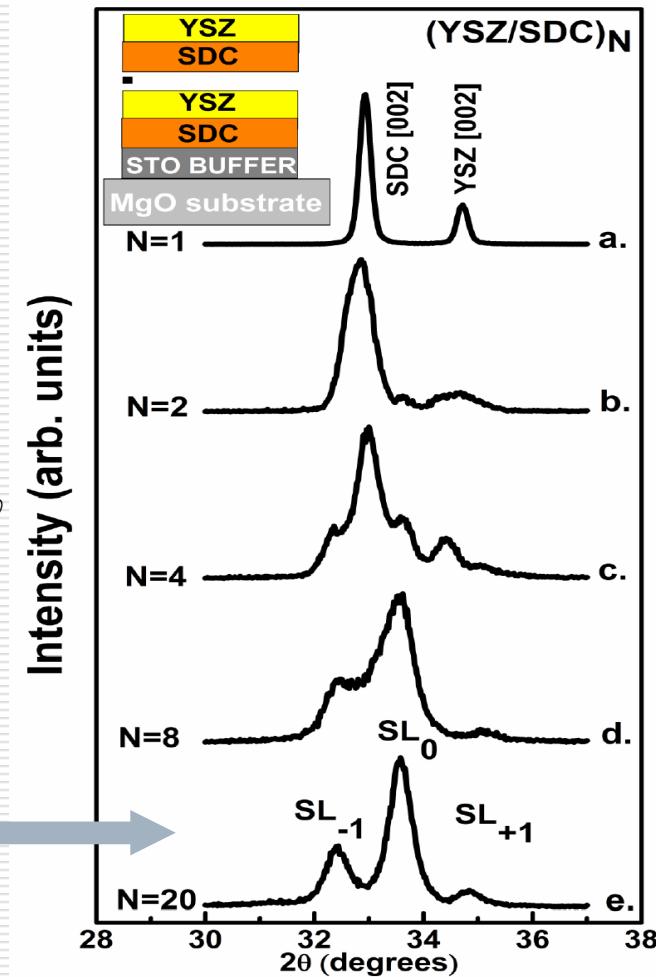
μ SOFCs based on thin film technology



$\text{Y}_{0.08}\text{Zr}_{0.92}\text{O}_2$ (YSZ)/SDC superlattices on STO buffered MgO



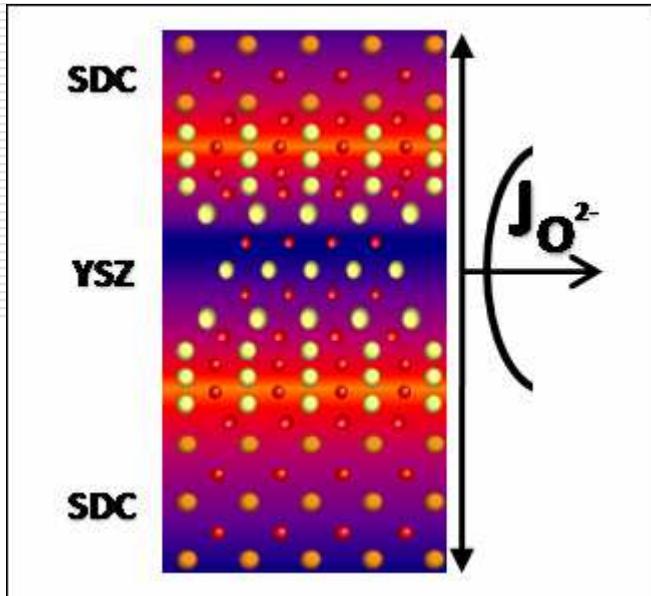
$$\frac{d_{100}^{\text{SDC}} - d_{100}^{\text{YSZ}}}{d_{100}^{\text{SDC}}} \approx 5\%$$



XRD θ - 2θ patterns of several $(\text{YSZ}/\text{SDC})_N/\text{STO}/\text{MgO}$ heterostructures with approximately the same overall thickness but N ranging from 1 to 20 (fig.a. to fig.e.). A sketch of the heterostructure is shown at the top.

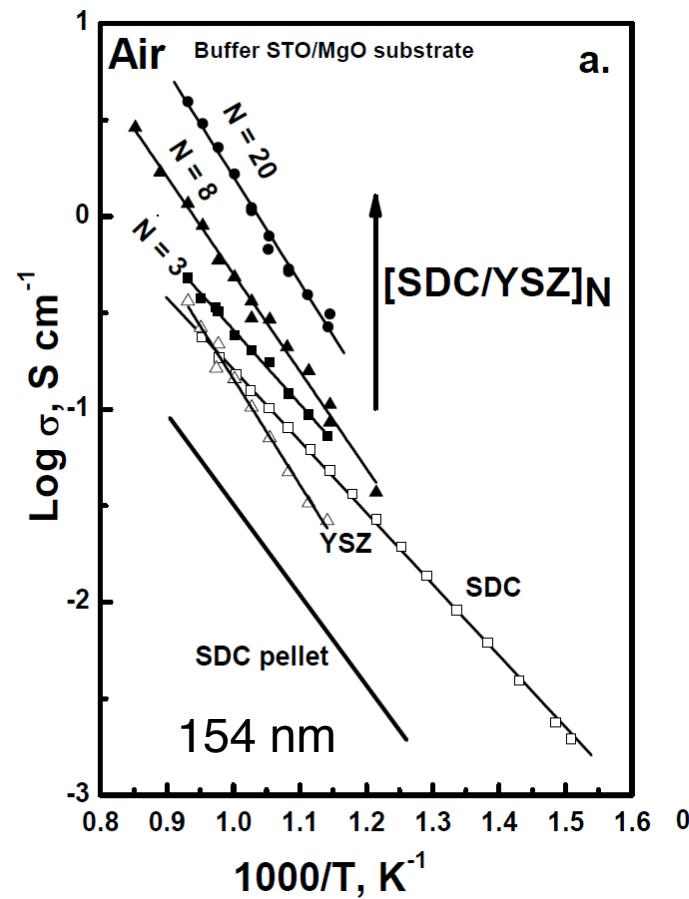
Electrochemical characterization of

$[SDC/YSZ]_N/STO/MgO$



$$\ln\left(\frac{\sigma_{int}}{\sigma_{bulk}}\right) \approx \frac{2}{3} \frac{\Delta V^M}{RT} \frac{Y}{1-\nu} \frac{\Delta a}{a}$$

N. Schichtel et al. Phys. Chem. Chem. Phys., 2009, vol 11, p 3043



$$\ln\left(\frac{\sigma_{int}}{\sigma_{bulk}}\right) \approx 2.8 \quad \ln\left(\frac{\sigma_{N=20}}{\sigma_{YSZ}}\right) \approx 2.4$$

Th

Ex

Conclusions

Recent developments in thin film growth techniques have opened new perspective for the deposition of heterostructures based on complex oxides.

This has made possible to engineer oxide heterostructures with novel and interesting physical properties.

Few examples were given

Chances are good that the field of “oxide electronics”, boosted by the large variety of oxide functional properties, will experience a fast development in the near future.