



2359-28

Joint ICTP-IAEA Workshop on Physics of Radiation Effect and its Simulation for Non-Metallic Condensed Matter

13 - 24 August 2012

Effects of neutron and gamma irradiation on degradation of nonmetallic materials for high temperature applications

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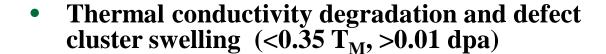
IAEA/ICTP School on Physics of Radiation Effects and its Simulation for Non-metallic Condensed Matter

Trieste, Italy August 13-24, 2012



Radiation Damage can Produce Large Changes in Ceramic Materials

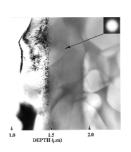
• Amorphization and disordering ($<0.2~T_M,>0.1~dpa$)

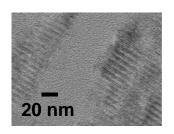


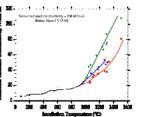
• Phase instabilities from radiation-induced precipitation (0.3-0.6 T_M , >10 dpa)

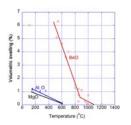
• Irradiation creep ($<0.45 T_M$, >10 dpa)

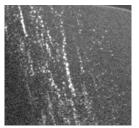
• Volumetric swelling from void formation (0.3-0.6 T_M , >10 dpa)

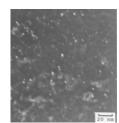


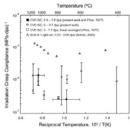


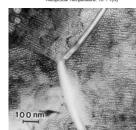












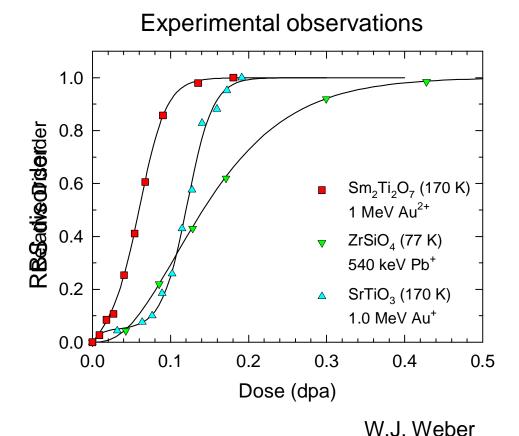


² Managed by UT-Battelle after S.J. Zinkle, Chpt. 3 in Comprehensive Nuclear Materials (Elsevier, 2012)

Overview of dose dependence for different amorphization mechanisms

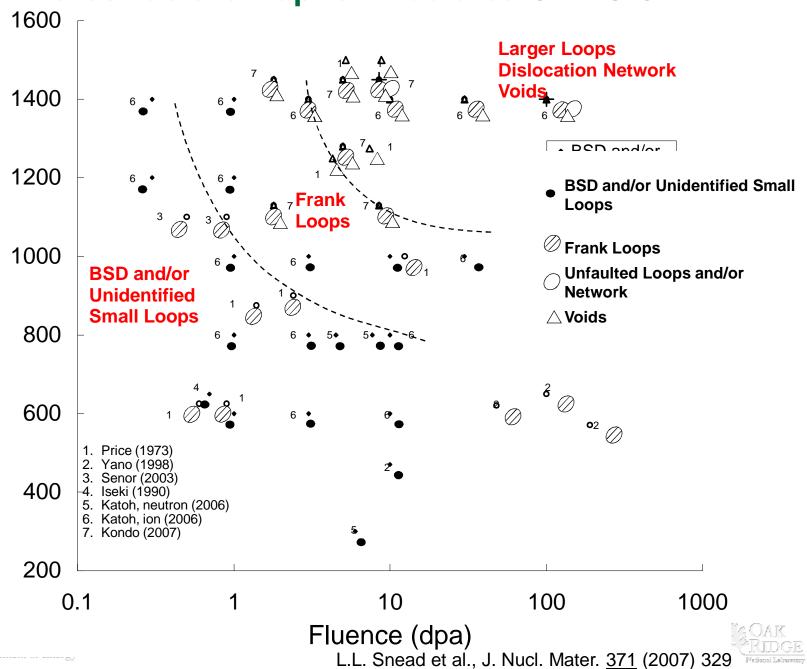
Model predictions 1.0 Direct Impact 0.8 Double Cascade Overlap 0.4 Cefect Accumulation 0.0 0.2 0.4 0.6 0.8 1.0 Normalized Dose

W.J. Weber, Nucl. Instr. Meth. B <u>166</u> (2000) 98



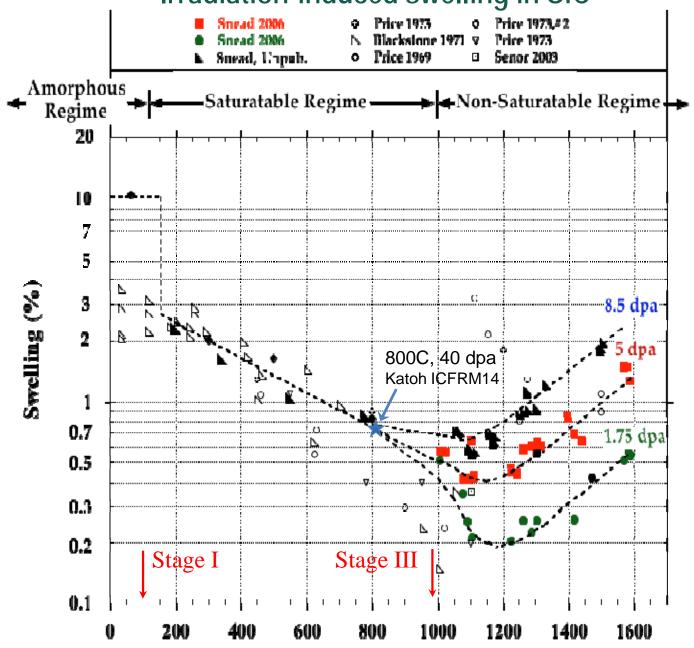


Microstructure Map for Irradiated CVD SiC



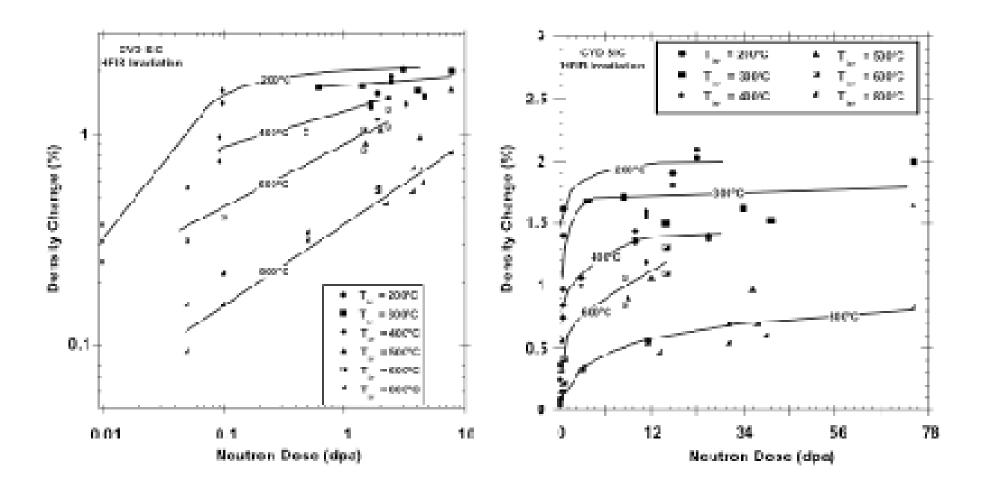
Managed

Irradiation-induced swelling in SiC



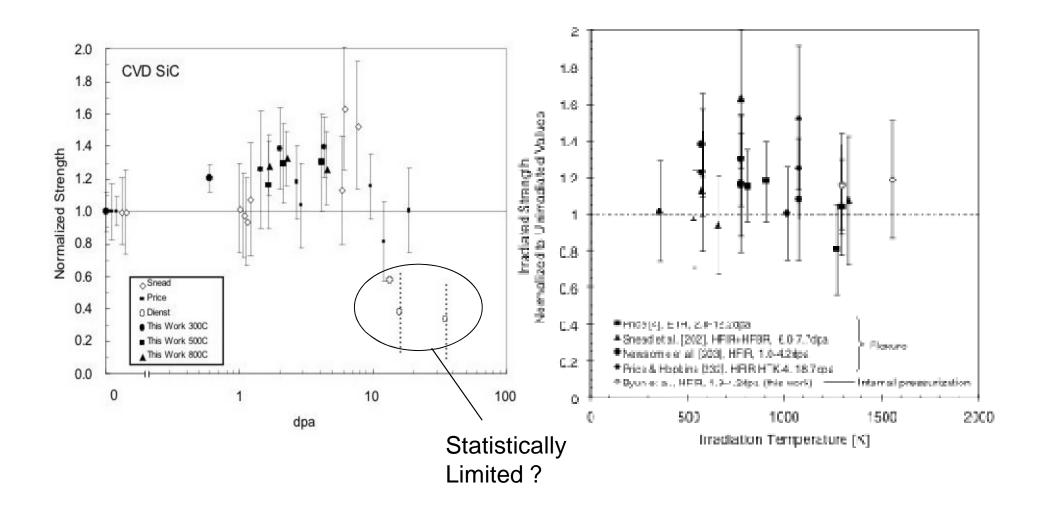


Effect of neutron irradiation on volumetric swelling of SiC



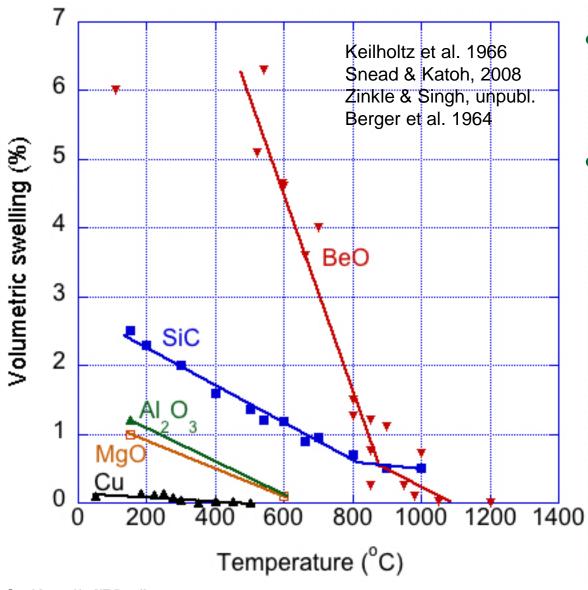


Effect of neutron irradiation on room temperature flexural strength of SiC





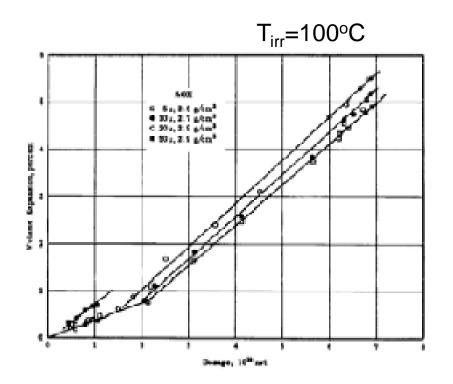
Comparison of point defect swelling behavior in irradiated metals and ceramics

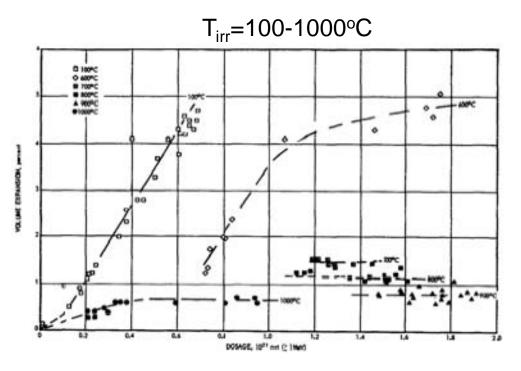


- Large variation in magnitude of point defect swelling in materials
- Metals typically have point defect swelling values near 0.1% (AI, Cu, Ag, Au), compared to 1-5% for ceramics
 - Implies shorter point defect recombination radii or reduced barriers for recombination in metals versus ceramics



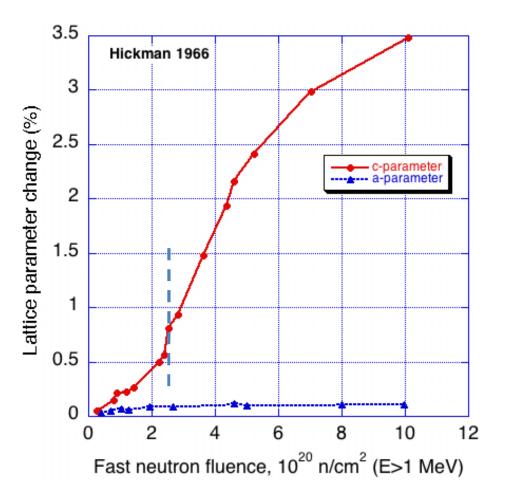
Effect of low temperature neutron irradiation on volumetric swelling in BeO







Effect of fission neutron irradiation near 75°C on the anisotropic swelling and mechanical strength of BeO



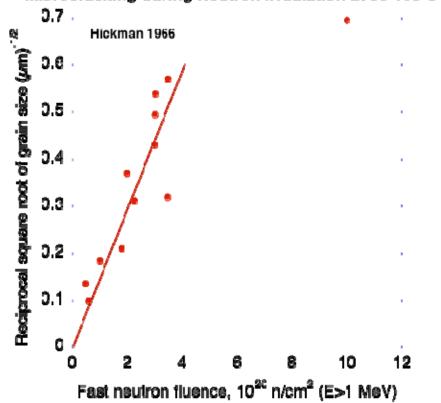
1.2 Hickman 1966 inad Aminad, modalus oʻngalaro ralik a.c 3.6 3.4 0.2 0 10¹⁸ Fast neutron fluence, n/cm2 (E>1 MeV)

B.S. Hickman, in Studies in Radiation Effects, G.J. Dienes, Ed. (Gordon & Breach, 1966) vol. 1 p. 72

L.L. Snead & S.J. Zinkle, STAIF 2005, ed. M.S. El-Genk, AIP conf. proc. 746 (2005) p. 768

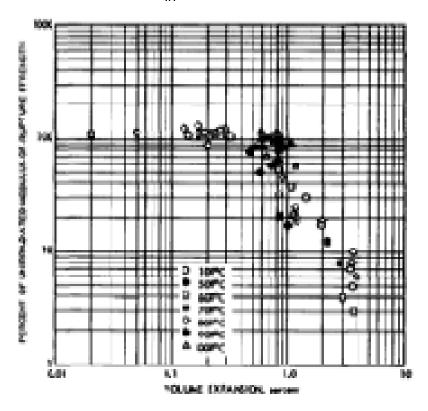
Effect of grain size and volumetric swelling on threshold microcracking in polycrystalline BeO

Effect of BeO Grain Size on the Neutron Fluence to Produce Microcracking during Neutron Irradiation at 50-100°C



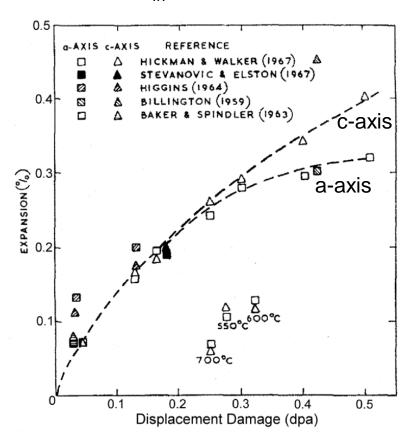
after L.L. Snead & S.J. Zinkle, STAIF 2005, ed. M.S. El-Genk, AIP conf. proc. 746 (2005) p. 768

Flexural strength of BeO vs. volumetric expansion (20 μ m grain size, 1.5x10²¹/m²) T_{irr} =100-1000°C



C.G. Collins, J. Nucl. Mater. <u>14</u> (1964) 69

Anisotropic lattice expansion and grain boundary cracking in neutron-irradiated Al₂O₃



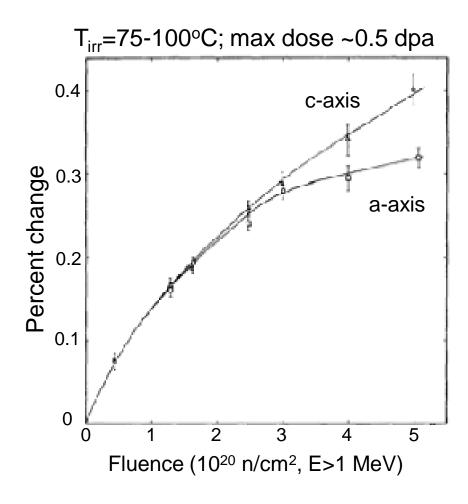
after B.S. Hickman & D.G. Walker, J. Nucl. Mater. 18 (1966) 197

G.W. Keilholtz et al., Nucl. Tech. <u>17</u> (1973) 234

National Laboratory

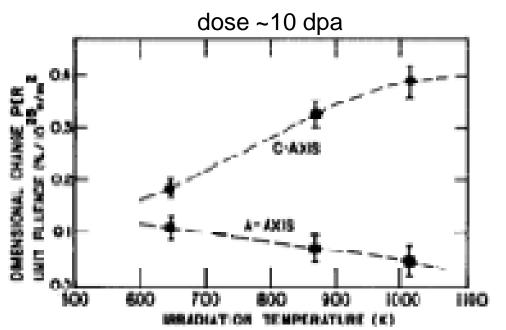
Microcracking and strength degradation observed for volumetric swelling >3-4%

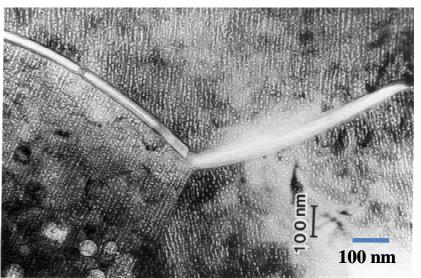
Neutron radiation-induced lattice expansion in Al₂O₃



B.S. Hickman & D.G. Walker, J. Nucl. Mater. <u>18</u> (1966) 197

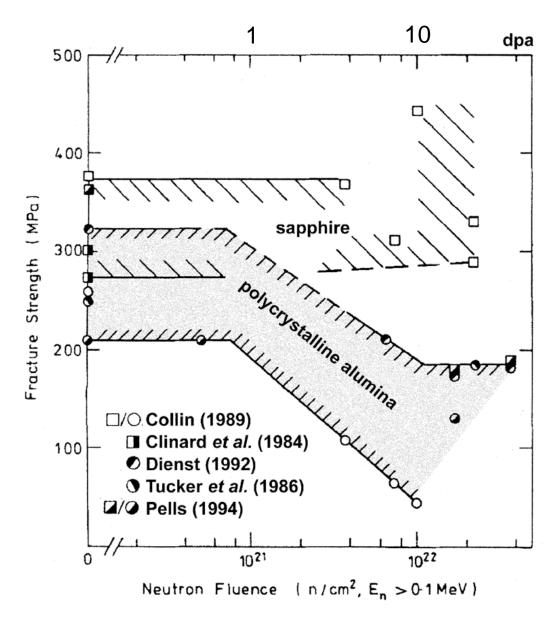
Grain boundary microcracking occurs in Al_2O_3 at high temperatures and doses Managed by UT-Battelle





F.W. Clinard, Jr. et al., J. Nucl. Mater. <u>108&109</u> (1982) 655

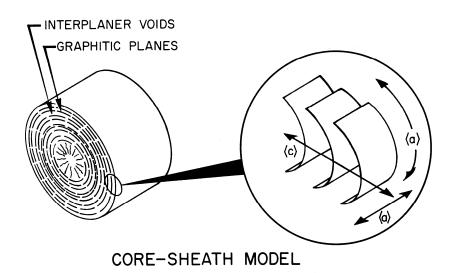
Effect of neutron irradiation on flexural strength of Al₂O₃

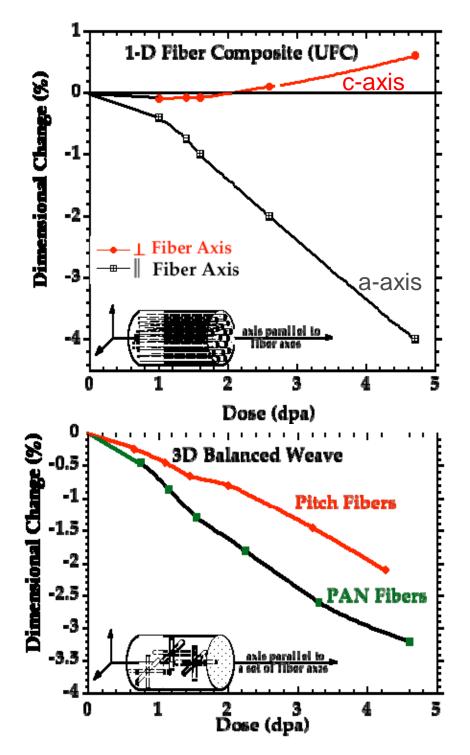




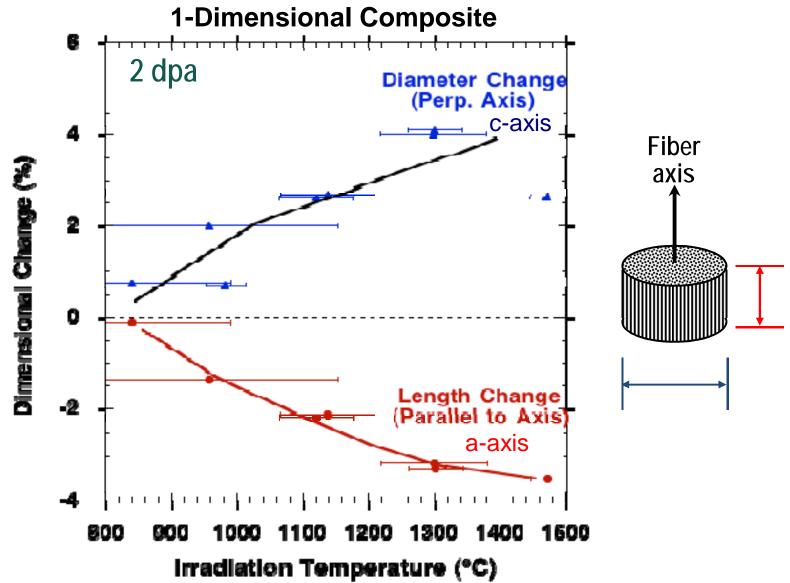
CFC's Under Irradiation

(HFIR, 600°C)



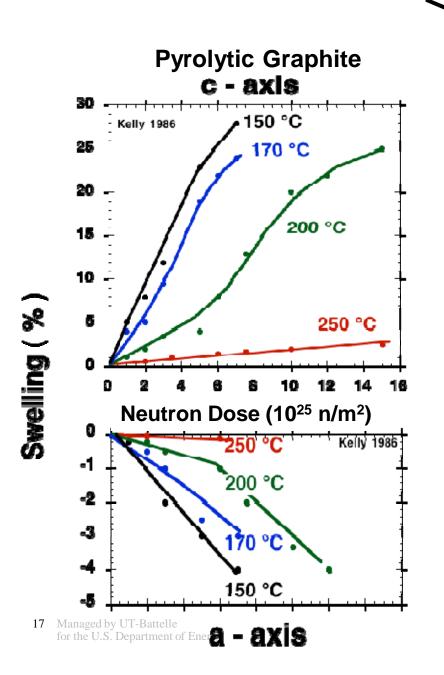


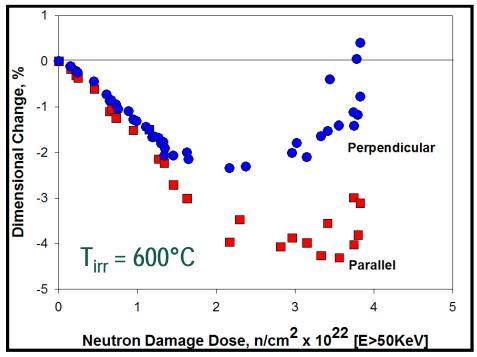
Dimensional Change in 1-D Graphite Composite

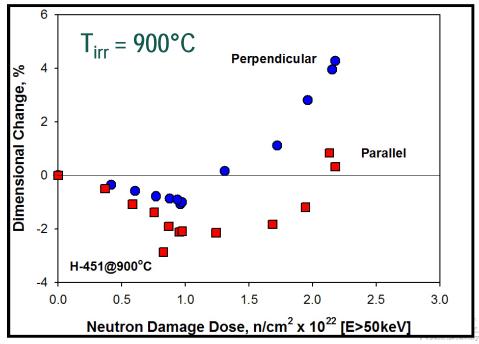




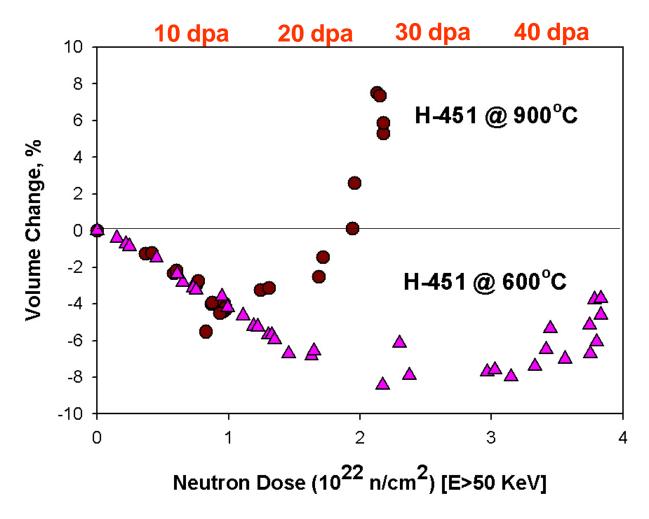
H-451 Nuclear Graphite







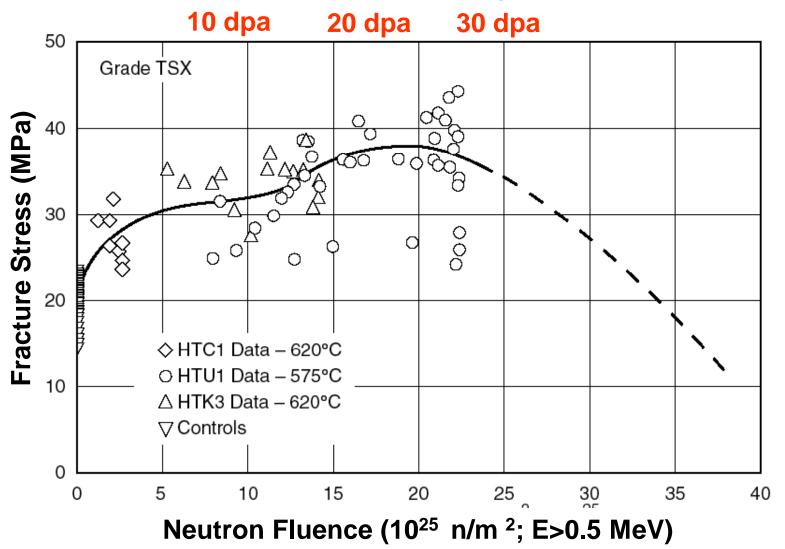
Effect of Temperature and Swelling of Nuclear Graphite



- initial <c> swelling accommodated by closure of intrinsic porosity.
- once porosity filled swelling can begin.
- less initial porosity for higher initial temperature (closure of intrinsic porosity.)

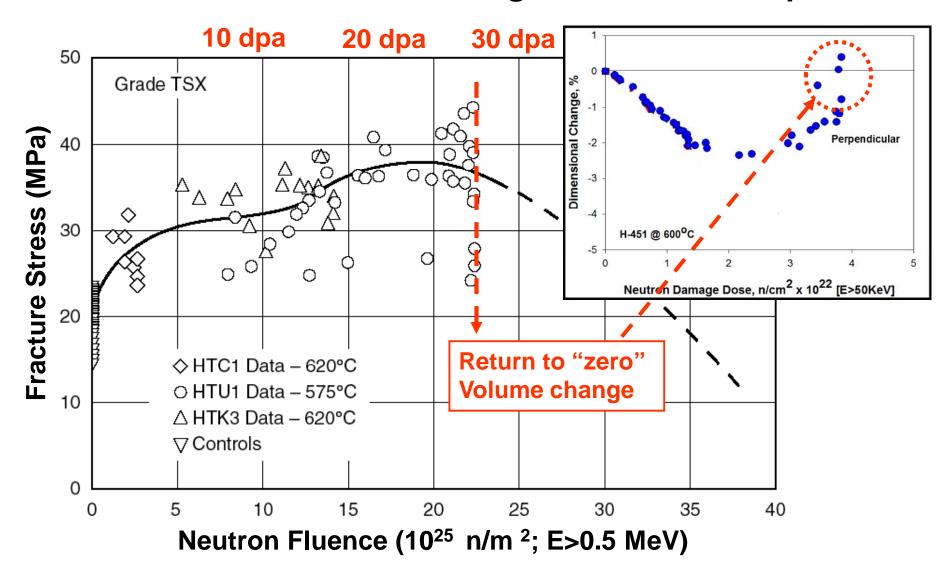


Effect of Irradiation on Strength of Nuclear Graphite



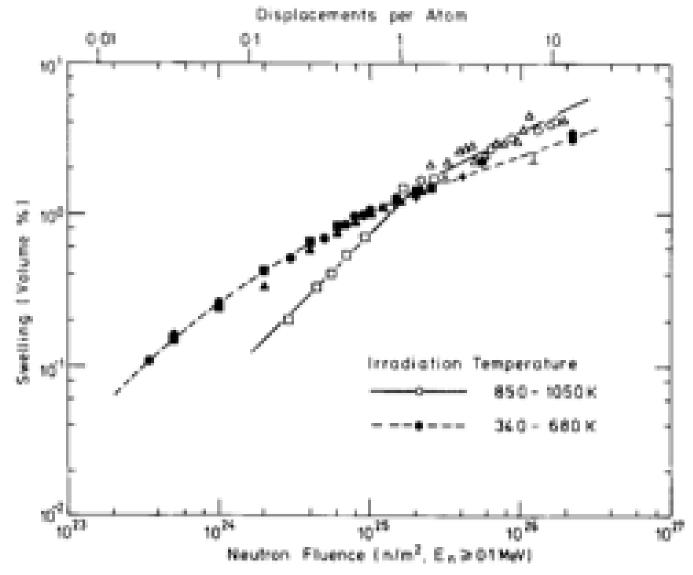


Effect of Irradiation on Strength of Nuclear Graphite





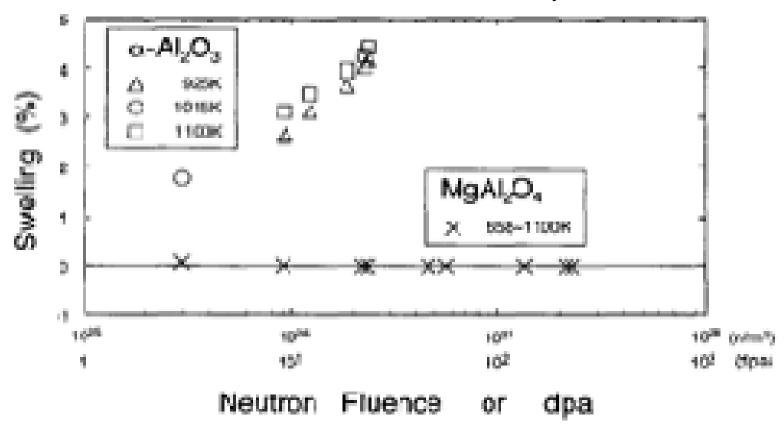
Effect of neutron irradiation on volumetric swelling of Al₂O₃





Effect of neutron irradiation on volumetric swelling of single crystal Al₂O₃ and MgAl₂O₄

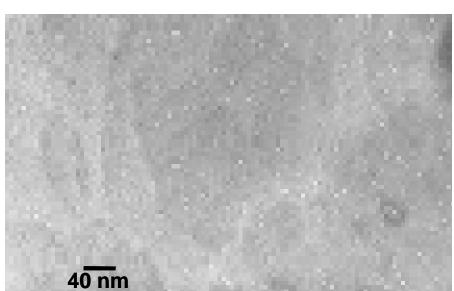
Based on data by Clinard et al. and Garner et al.





Cavity microstructure of 1 MeV He ion irradiated MgAl₂O₄ at 650°C

1x10²¹ He/m²



Cavity formation observed in irradiated midrange region >1µm from the peak implanted zone

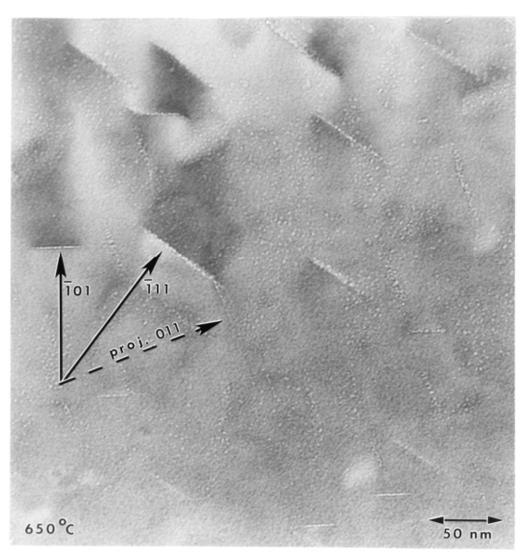
100 nm 1x1022 He/m2 21 dpa, 50 at%He at peak 100 nm

2.1 dpa, 5 at%He

at peak

S.J. Zinkle, Nucl. Instrum. for the U.S. Department of Energy Meth. B (2012) in press

Preferential cavity formation on dislocation loops in dual ion beam irradiated MgAl₂O₄ at 650°C

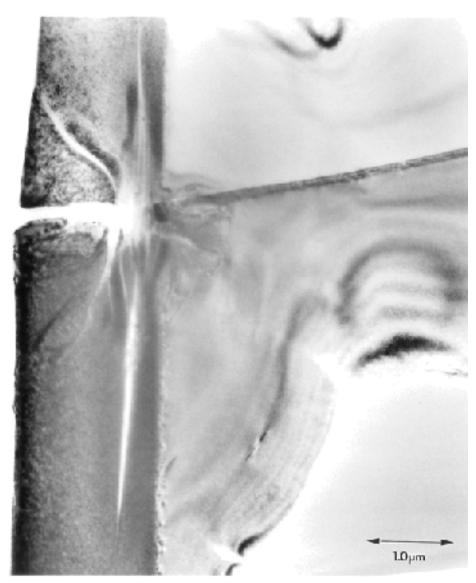


2 MeV AI + 200-400 keV He

26 dpa, 3900 appm He



Grain boundary cavitation in dual ion beam irradiated MgAl₂O₄ at 650°C



2 MeV AI + 200-400 keV He

26 dpa, 3900 appm He at ~0.8 μm depth



Grain boundary cavitation in dual ion beam irradiated MgAl₂O₄ at 650°C

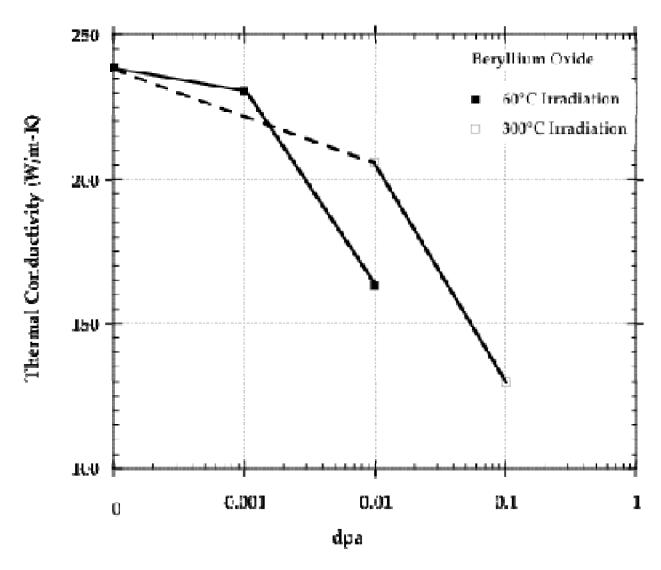


2 MeV AI + 200-400 keV He

26 dpa, 3900 appm He at ~0.8 μm depth

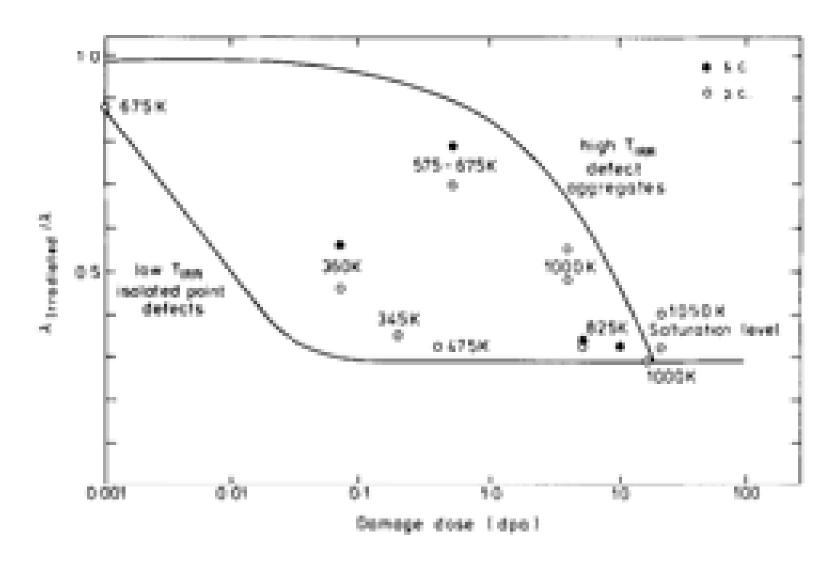


Effect of low temperature neutron irradiation on thermal conductivity of BeO



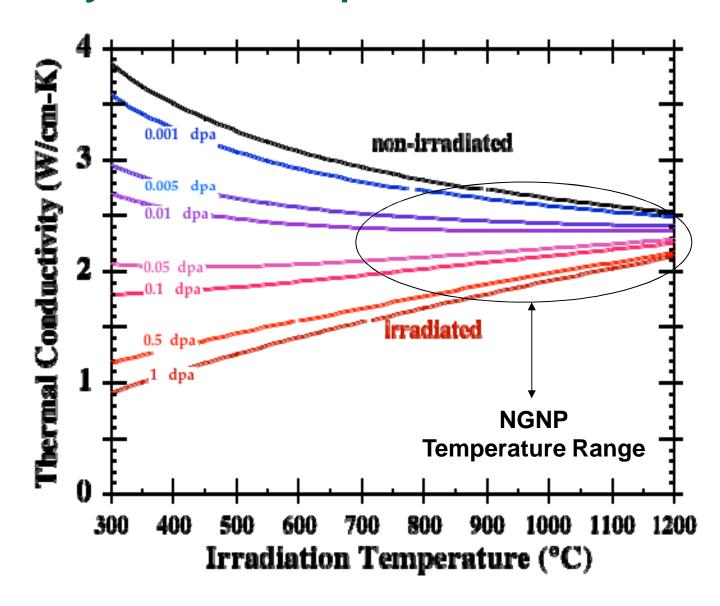


Effect of neutron irradiation on thermal conductivity of Al_2O_3



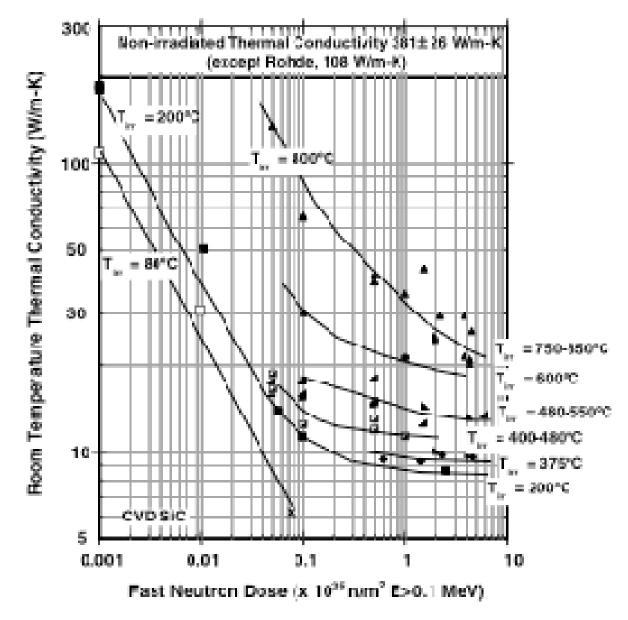


Thermal Conductivity Degradation in Highconductivity Carbon Composite



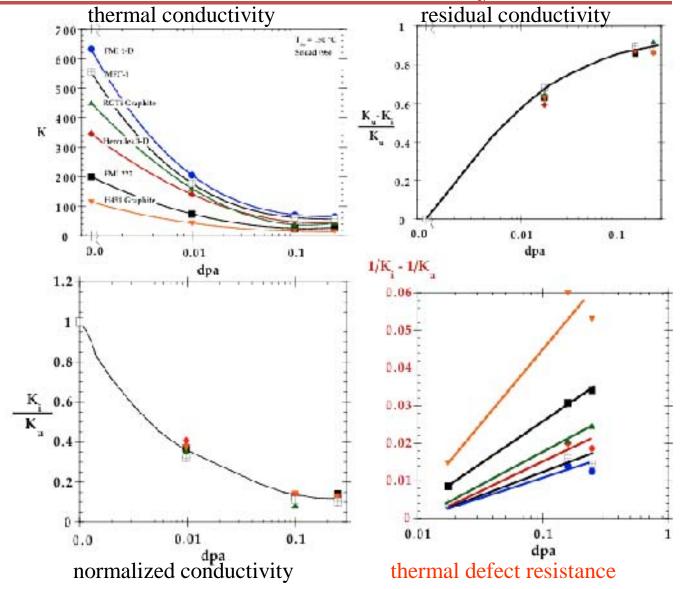
Effect of neutron irradiation on thermal conductivity of

SiC



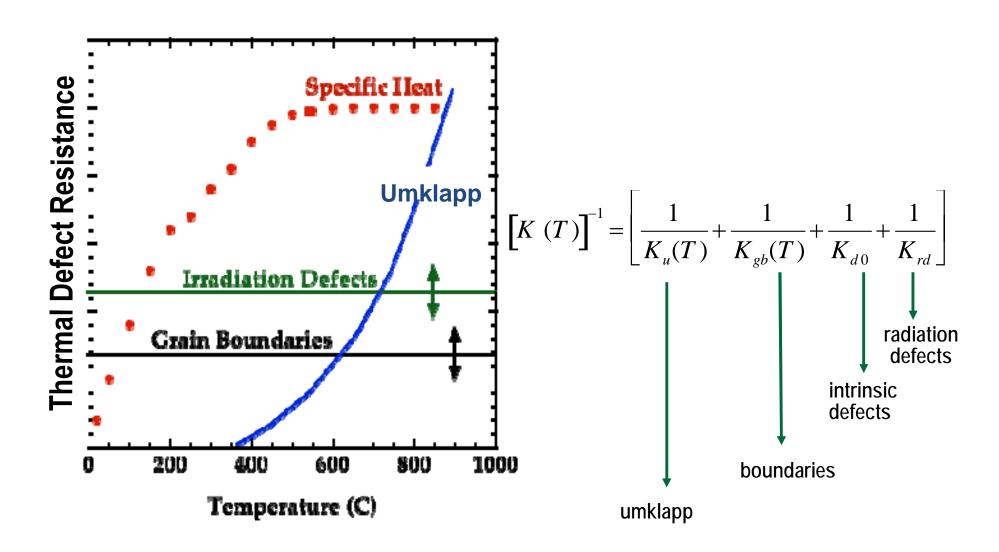


What is the best parameter to characterize radiation-induced degradation of the thermal conductivity of ceramics?





Thermal Conductivity of Ceramics





Adopting Thermal Defect Resistance

Thermal defect resistance

$$\frac{1}{K_{rd}} = \frac{1}{K_{irr}} - \frac{1}{K_{unirr}}$$

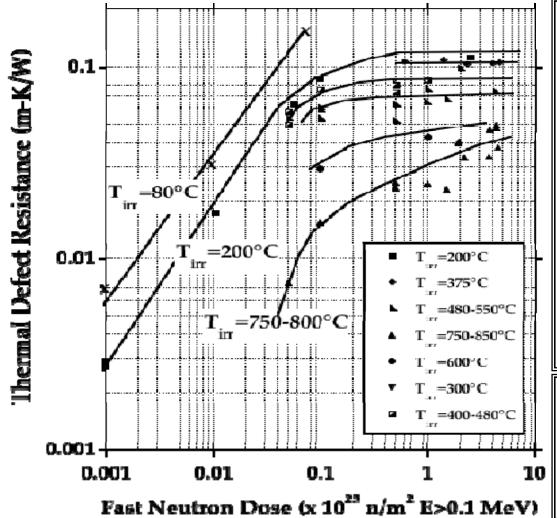
Thermal defect resistance due to low vacancy production

$$\frac{1}{K_{rd}} = \left(\frac{3\pi}{2k_B}\right) \left(\frac{\Omega\omega_D}{v^2}\right) C_v$$

The motivation for using thermal defect resistance is that radiation-induced defects, such as vacancies and clusters, have resistances proportional (or square root dependent) to their concentration and are additive. This gives an easy way to compare stability of ceramics under irradiation.

$$\frac{1}{K_{tdr}} = \frac{1}{K_{irradiated}} - \frac{1}{K_{non-irradiated}}$$

TDR in Point Defect Swelling Regime



Vacancies

For Weak Scattering

$$\Delta \left(\frac{1}{K}\right)_{vac} = \frac{1}{3K_i} \left(\frac{\omega_D}{\omega_p}\right)^2 = \frac{3\pi\Omega\omega_D}{2k_B v^2} C_v$$

For Strong Scattering

$$\Delta \left(\frac{1}{K}\right)_{vac} = \frac{2}{\pi K_i} \left(\frac{\omega_D}{\omega_p} - \frac{\pi}{2}\right)$$

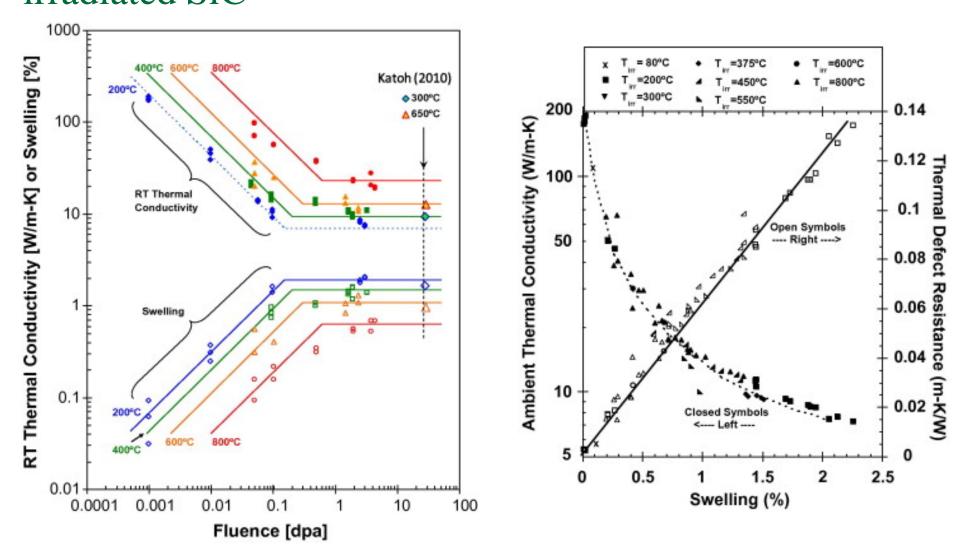
$$= \frac{6\pi^{\frac{1}{2}}}{k_{B}} \frac{1}{\omega_{D}} \left(\frac{\Omega}{aT_{m}}\right)^{\frac{1}{2}} \left(C_{v}T\right)^{\frac{1}{2}} - \frac{2\pi^{2}}{k_{B}} \frac{v^{2}}{aT_{m}\omega_{D}^{3}} T$$

Small Loops

$$\Delta \left(\frac{1}{K}\right)_{loop} = K_i \frac{C}{B} = \frac{24 \pi h^2 R^2}{v} n_{loop}$$



Evolution of swelling and thermal conductivity in irradiated SiC

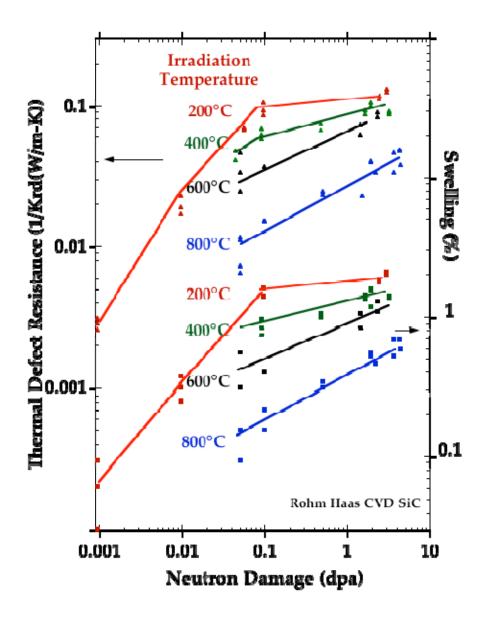


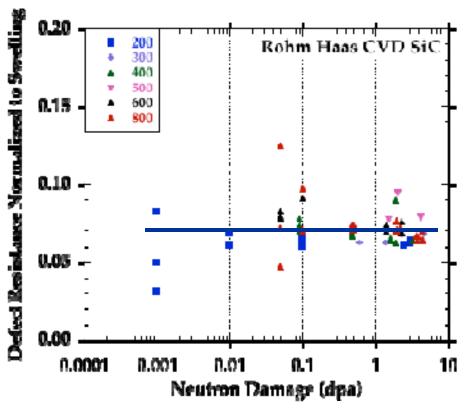
L.L. Snead et al., J. Nucl. Mater. 411 (2011) 330

L.L. Snead et al., J. Nucl. Mater. <u>371</u> (2007) 329



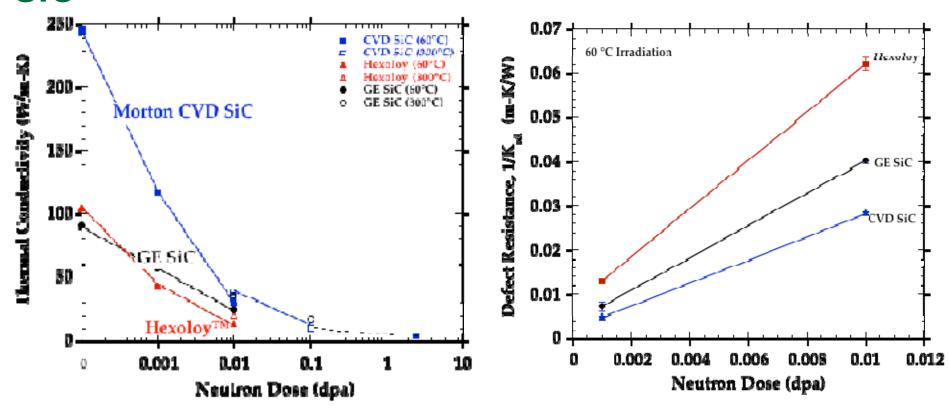
Irradiation-induced Thermal Defect Resistance in SiC





Thermal defect resistance normalized to irradiation-induced swelling is constant independent of irradiation temperature (200-800°C) suggesting single defect type controlling phonon scattering

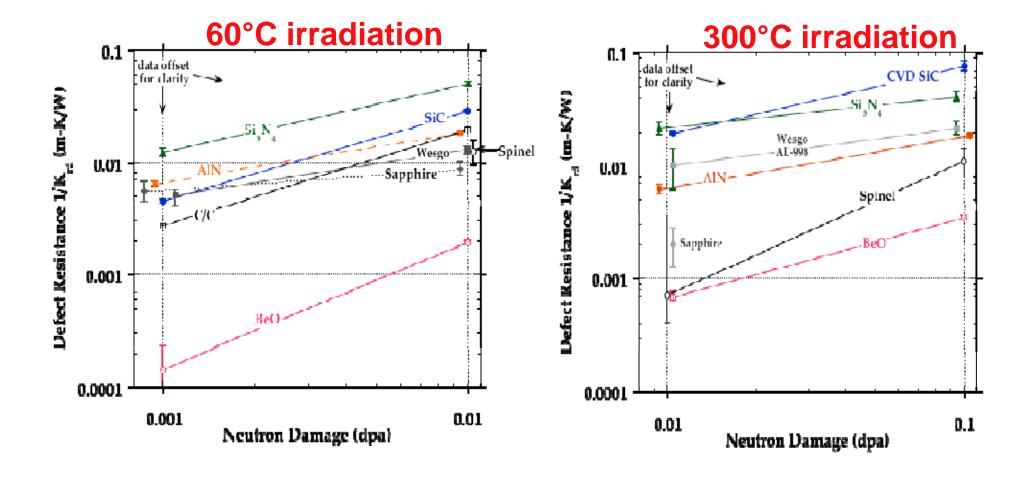
Irradiation-Induced Thermal Defect Resistance of SiC



	CVD SiC	GE SiC
K _{unirr} -K _{irr}	2.35	0.7
% K _{unirr}	6%	22%
Defect Resistance	2.8	2.9

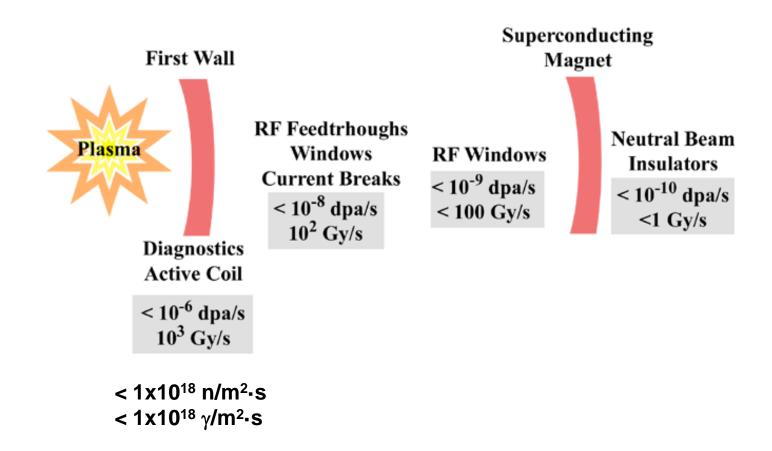
Hexoloy contains boron, which disrupts grain boundary under irradiation. GE SiC and CVD SiC are both high-purity SiC.







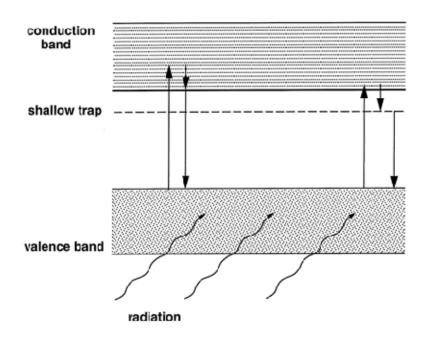
Radiation Environments Anticipated for Ceramic Components in the ITER





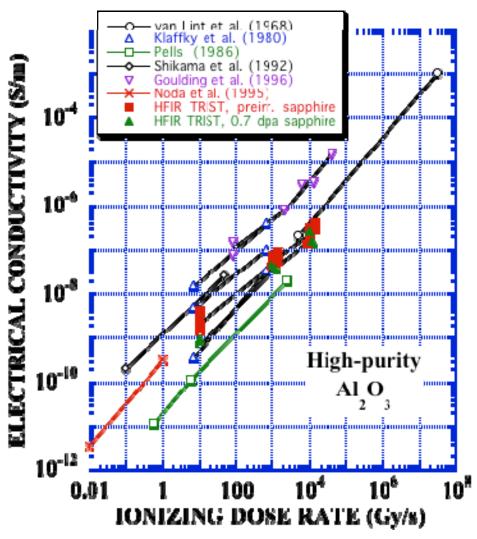
Effect of ionizing irradiation on the electric conductivity of ceramic insulators

Radiation Induced Conductivity in Insulators



$$\sigma_{\text{RIC}} = \sigma_0 + KR^{\delta}$$

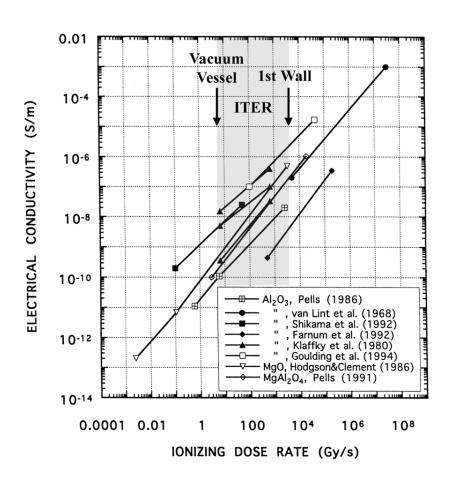
 $\delta \sim \begin{cases} 1 \text{ for insulators} \\ <<1 \text{ for semiconductors and metals} \end{cases}$



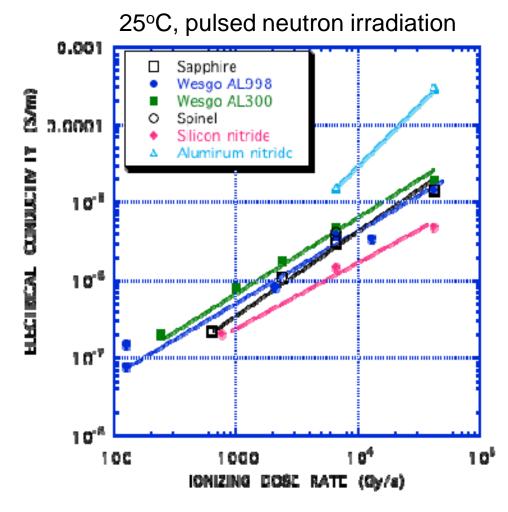
S.J.. Zinkle et al., DOE/ER-0313/22 (1997) p.188



Effect of ionizing irradiation on the electric conductivity of ceramic insulators



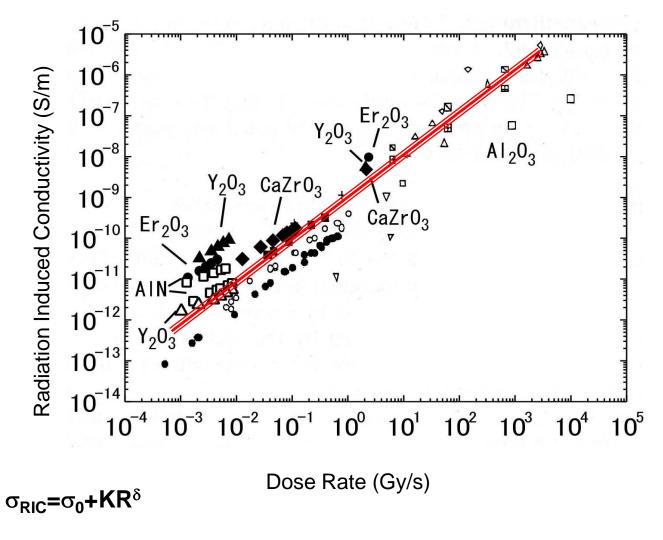
L.W. Hobbs et al., J. Nucl. Mater. 216 (1994) 291



R.H. Goulding et al., J. Appl. Phys. 79 (1996) 2920



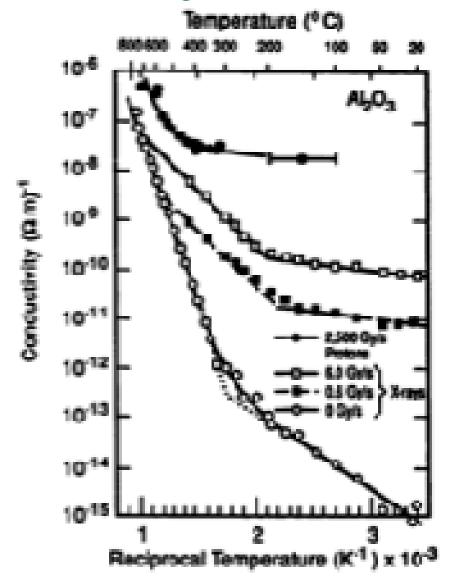
Effect of ionizing irradiation on the electric conductivity of ceramic insulators





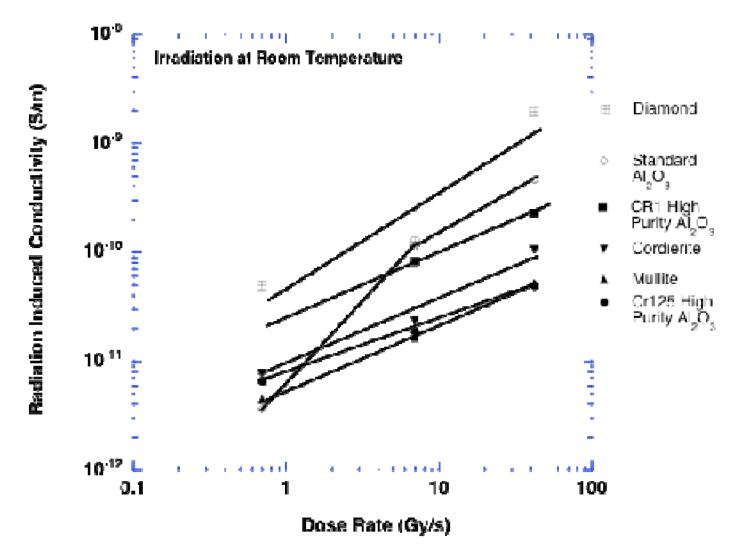


Temperature-dependent electrical conductivity of Al₂O₃ with and without ionizing radiation





Radiation Induced Conductivity in Selected Powder Ceramic Insulated Cables



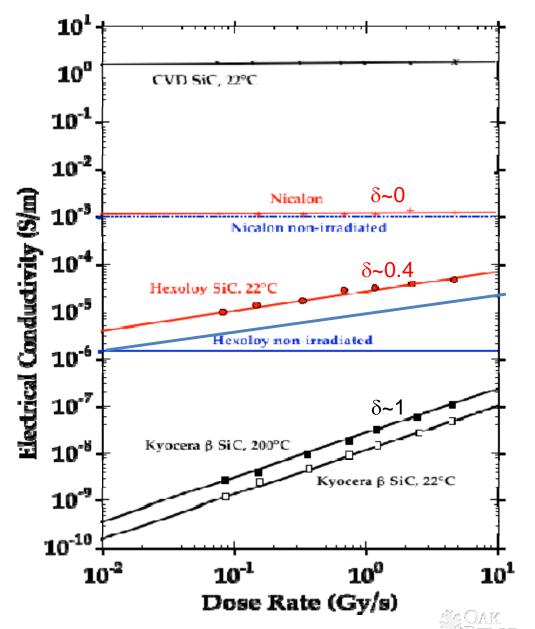


Radiation Induced Conductivity (In-situ) for different grades of SiC

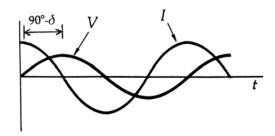
 lonizing radiation excites electrons into the conduction band enhancing conductivity.

$$\sigma = \sigma_o + KR_{\blacktriangledown}^{\delta}$$
 Dose rate

• If base conductivity is high (>10⁻⁴ S/m), RIC is insignificant

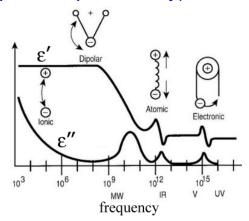


Dielectric loss tangent of ceramic insulators

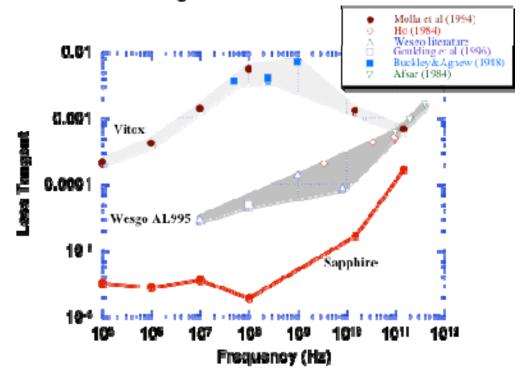


Plots of current *I* and voltage *V* across the capacitor versus time. The voltage lags the current by 90°.

 $tan\delta = \epsilon''/\epsilon'$ (ratio of imaginary to real parts of permittivity)



Measured Loss Tangents for Different Grades of Alumina



need low electrical conductivity (including during irradiation

$$\tan \delta = \frac{\sigma_{DC}}{\omega \varepsilon'} + \frac{\chi''}{\varepsilon'/\varepsilon_0}$$
 Polarization losses can be increased by irradiation-induced defects

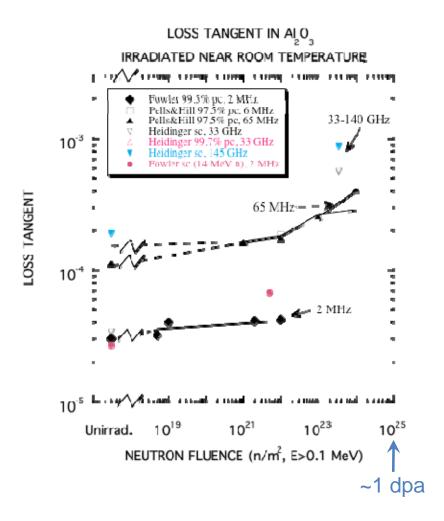
Netional Lebourton

Absorbed power: $P=\omega\epsilon' \tan\delta E^2$

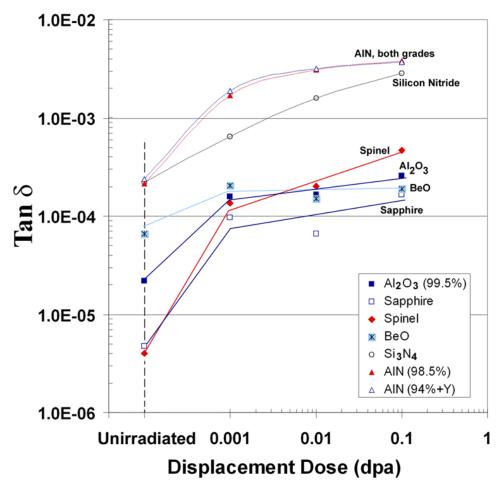
To avoid overheating, need $\tan \delta < 10^{-3}$ (ICH, 100 MHz) or $< 10^{-6}$ ECH, 100 GHz)

Effect of irradiation on the dielectric properties of ceramic insulators

100 MHz loss tangent after 70°C neutron irradiation



L.W. Hobbs et al., J. Nucl. Mater. 216 (1994) 291

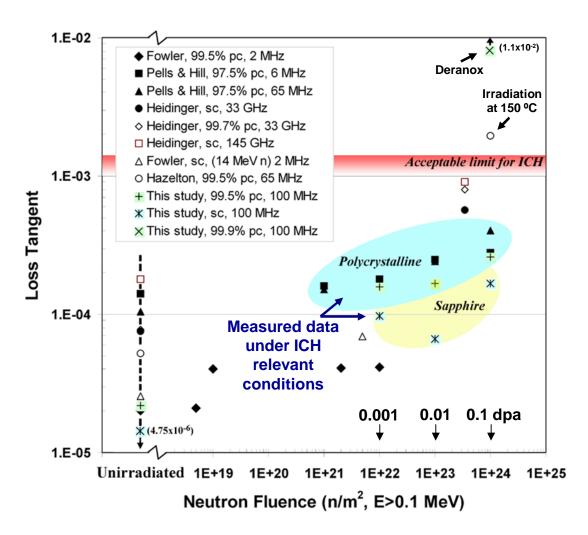


after R.H. Goulding & K.J. Leonard, ITER report G 51 TD 26 FU (2007)

and C. Hazeltine et al., J. Nucl. Mater. 253 (1998) 190



Loss Tangent Database of Irradiated Al₂O₃



- Neutron irradiation near room temperature for polycrystalline (pc) and single crystal (scsapphire)
- Acceptable limit for ion cyclotron heating (ICH), Tan $\delta \cong 0.001$
- Data from this study for Wesgo Al-995 and single crystal sapphire plotted
- Certain grades of polycrystalline Al₂O₃ suitable for use up to 0.1 dpa
- Deranox grade Al₂O₃ exceeds Tan δ limit for ICH application



Conclusions

- Neutron and gamma radiation can produce significant property changes (degradation) in ceramics
- Strength degradation typically occurs for anisotropic (e.g., HCP) crystal structures
 - Graphite is more resilient than BeO, Al₂O₃
- Most pronounced changes typically occur in physical properties (e.g., thermal conductivity)
- Electrical conductivity of insulators exhibits marked prompt increases during exposure to ionizing radiation
 - Effect is small for semiconductors

