

2440-10

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Total Energy and Force Methods**

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Simulation of phase change materials for data storage

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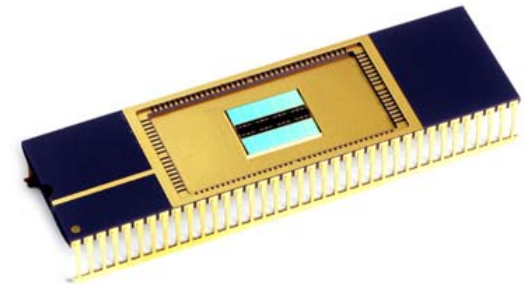
Atomistic simulations of phase change materials for data storage

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Phase Change Materials for electronic and optical data storage

Optical data storage: DVD



Electronic non-volatile memory: Phase Change Memory Cell

Chalcogenides alloys $\text{Ge}_2\text{Sb}_2\text{Te}_5$ (GST) , GeTe

fast and reversible change between crystalline and amorphous phases (50 ns)

Phase change Materials

Two states system → possibility of storing a "0" or "1" bit

Large difference in properties between the two phases
(crystal - metallic amorphous - semiconducting)

Resistivity changes by 3 orders of magnitude → PCM

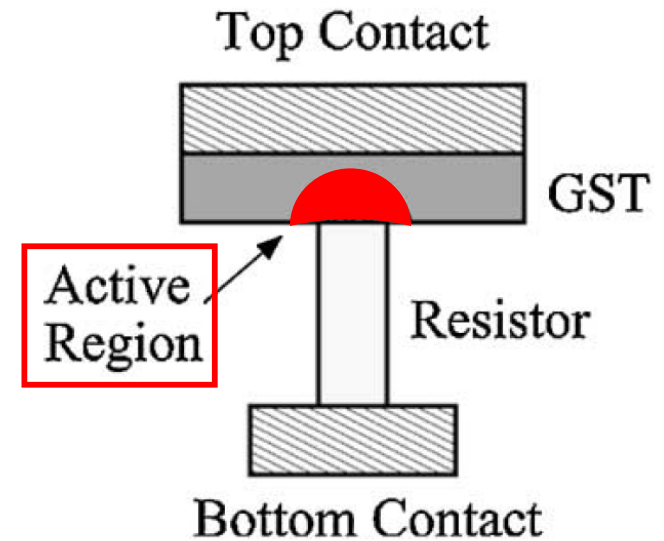
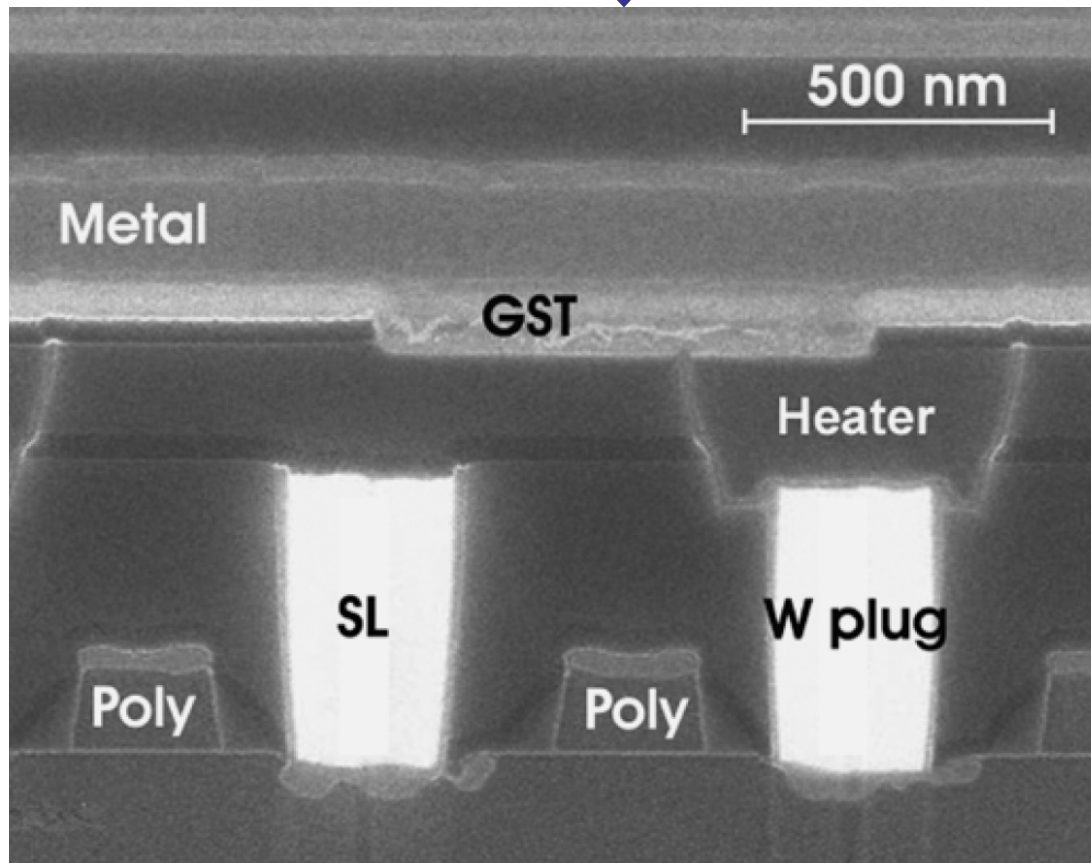
Reflectivity change 30 %: → optical storage (DVD)

Transition induced by heating (Joule or laser irradiation)

The Phase-Change memory cell

Schematic representation

SEM cross section



- Active region: a small drop within GST film undergoes the phase transition
- Phase-change by heating via Joule effect

Concept first proposed by
Ovshinsky in 1968

PCM faster and with better scalability than Flash

April 2010



Numonyx (now bought by Micron) commercialized 90 nm PCM device
Research center based in Agrate Brianza (Milano-Italy)

April 2011

First mobile phone with PCM inside
commercialized by Samsung



July 2012

Micron: 45 nm devices in production

Open issues

- Structure of the amorphous phase
- Origin of optical contrast between amorphous and crystal
- Origin of resistivity contrast
- Mechanism of phase change

Wuttig and Yamada, Nat. Mat. 6, 824 (2007)

Lacaita and Wouters, Physica Status Solidi A 205, 2281 (2008)

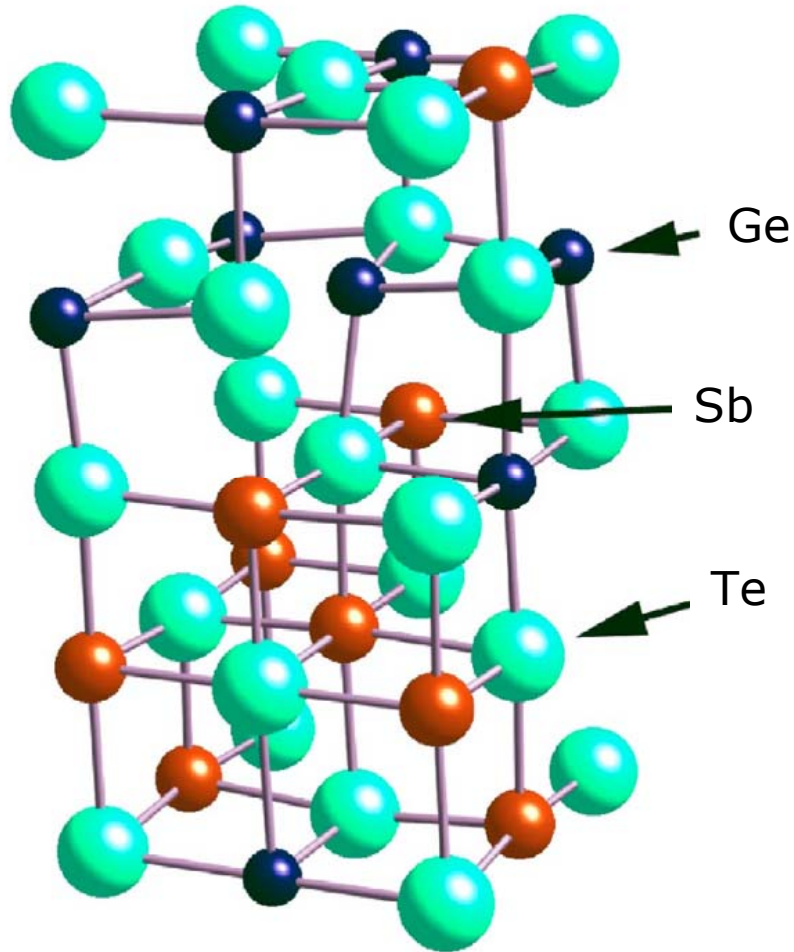
Raoux, Annu. Rev. Mater. Res. 39, 25 (2009)

Raoux , Welnic, Ielmini, Chem. Rev. 110, 240 (2010)

Lencer, Salinga, Wuttig, Adv. Mater. 23, 2030 (2011)

GST crystalline phase: Rocksalt

❖ NaCl-type



- One sublattice fully occupied by Te
- The other randomly occupied by Ge, Sb and 20% vacancies
- Octahedral environment

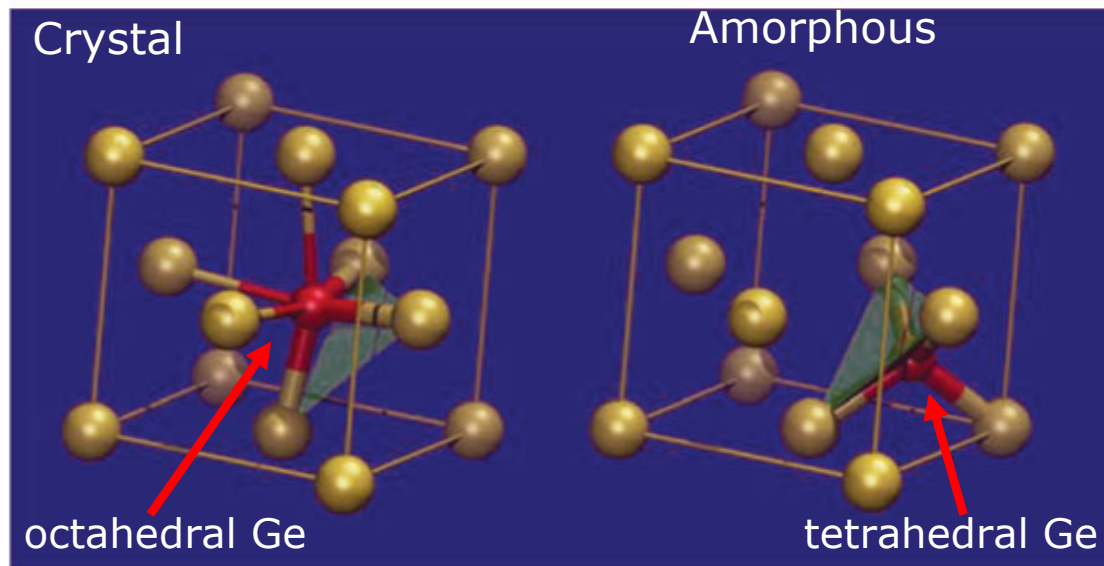
each atom 6-fold coordinated

Metastable phase: transition to hexagonal structure at ~250 °C

Structure of amorphous GST

- EXAFS/XANES data suggest 4-fold Ge coordination
- Ge in tetrahedral environment in amorphous GST
- Phase transition cubic \leftrightarrow amorphous: octahedral \leftrightarrow tetrahedral change of Ge environment

[Kolobov et al., *Nat. Mater.* **3**, 703 (2004)]



Octahedral \leftrightarrow tetrahedral change responsible for electronic and optical contrast

Ab-initio models of amorphous GST

Model of amorphous GST generated by quenching from the melt in ab-initio molecular dynamics

- Caravati, Bernasconi, Kühne, Krack, and Parrinello, Appl. Phys. Lett. 91, 171906 (2007)
- Akola and Jones, Phys. Rev. B 76, 235201 (2007)
- Hegedus and Elliott, Nature Mater. 7, 399 (2008).
- Z. M. Sun et al, Appl. Phys. Lett. 93, 241908 (2008)
- M. Xu et al, Phys. Rev. Lett. 103, 195502 (2009)

Phase Change Materials from First Principles

- First principles molecular dynamics simulations
- Density Functional Theory (GGA-PBE + pseudopotentials)
- Code Quickstep, open source CP2K project www.cp2k.org
Basis sets: Gaussian (wavefunctions) + plane waves (elec. density)

Second generation Car-Parrinello method

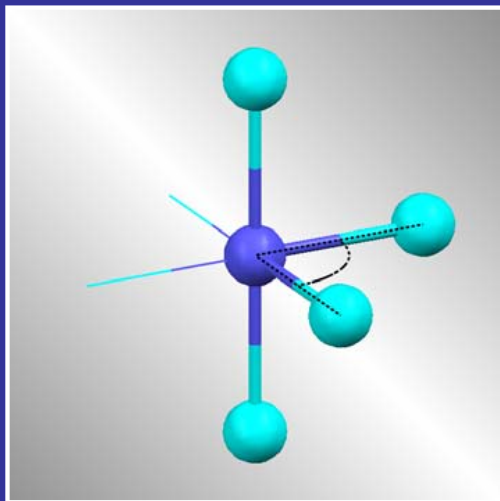
Factor 25 speed up with respect to conventional BO-MD CP2K
(Kühne, Krack, Mohamed, Parrinello, PRL 2007)

270-500 atoms, quenching from the melt in 80-100 ps

Coexistence of octahedra and tetrahedra in amorphous PCM

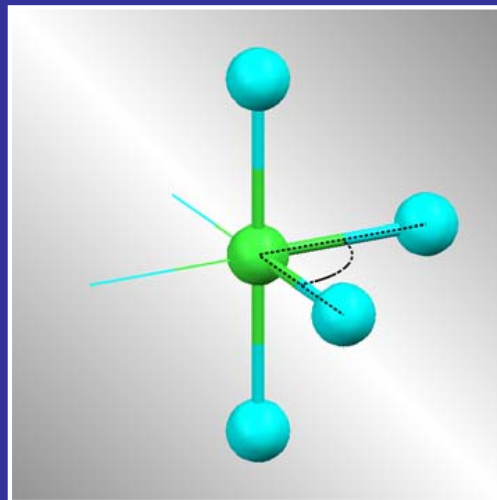
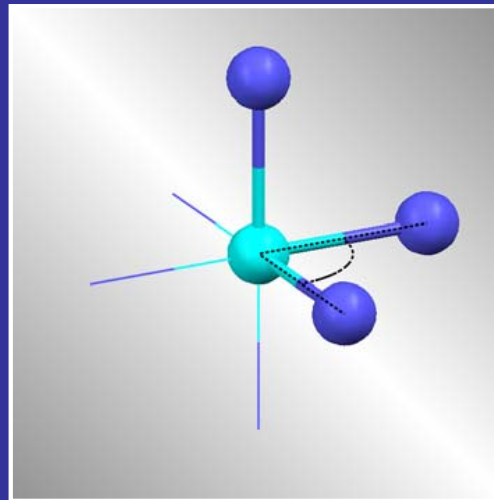
(defective) octahedral
Ge, Sb and Te

4-coor Ge



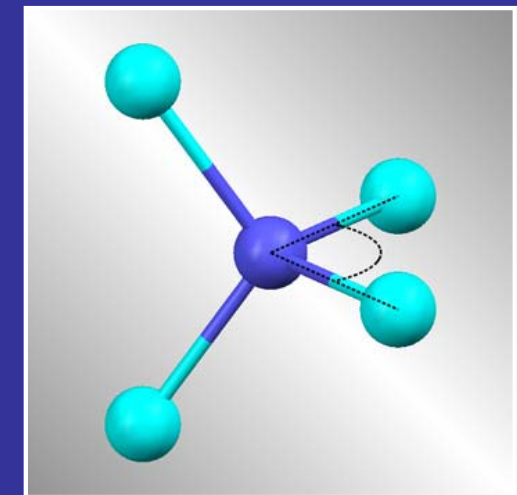
3+1 coordination
→ p-bonding
only p valence states

3-coor Te



4-coor Sb

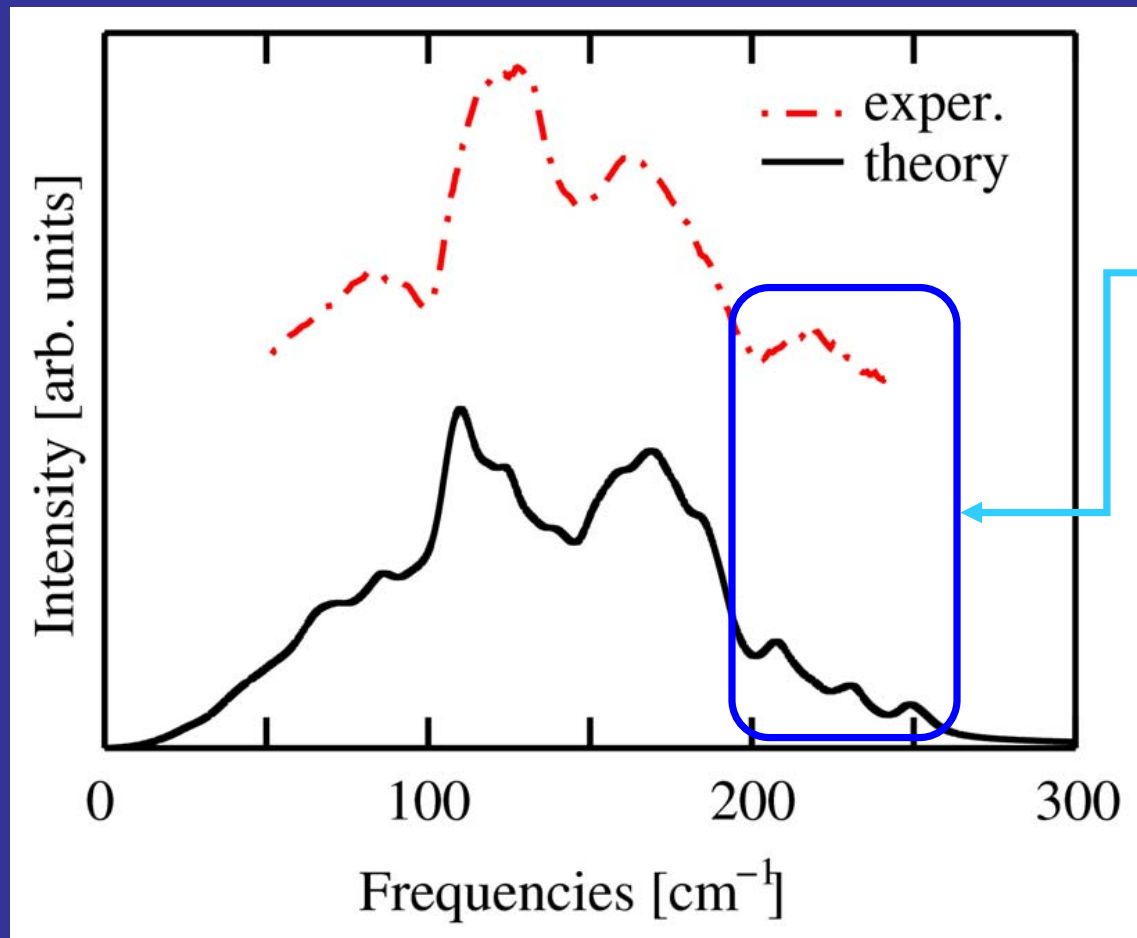
tetrahedral
4-coor Ge, 25 %



sp^3 bonding
s + p valence states

Tetrahedral Ge: vibrational fingerprint

Amorphous GeTe: Raman spectrum (Bond Polarizability Model)



Phonon freq. > 190 cm⁻¹:

- Localized on tetrahedral Ge

⇒ signature of tetrahedral Ge

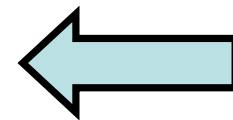
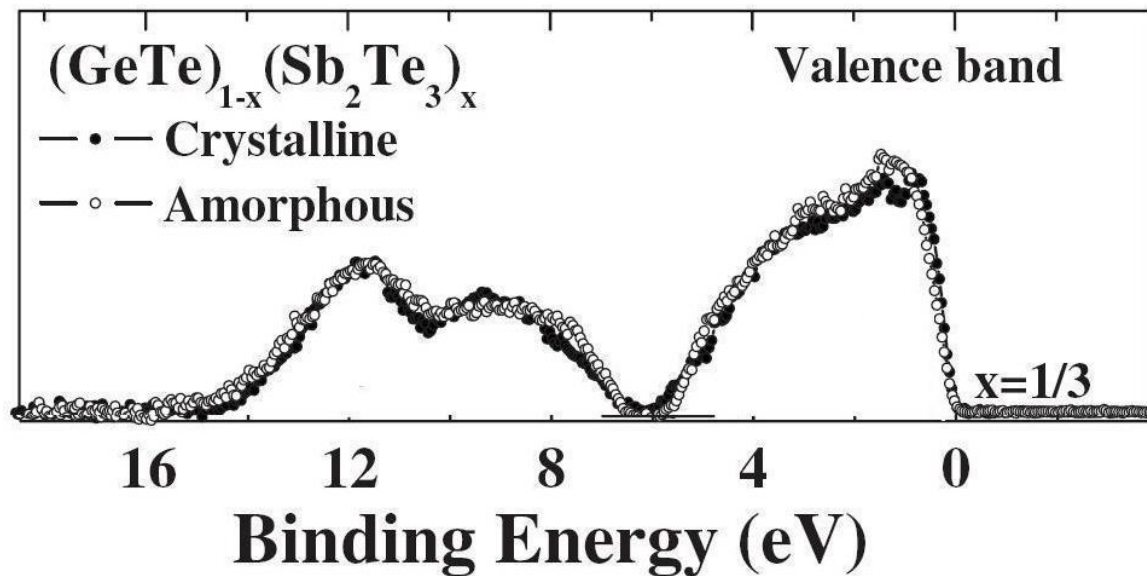
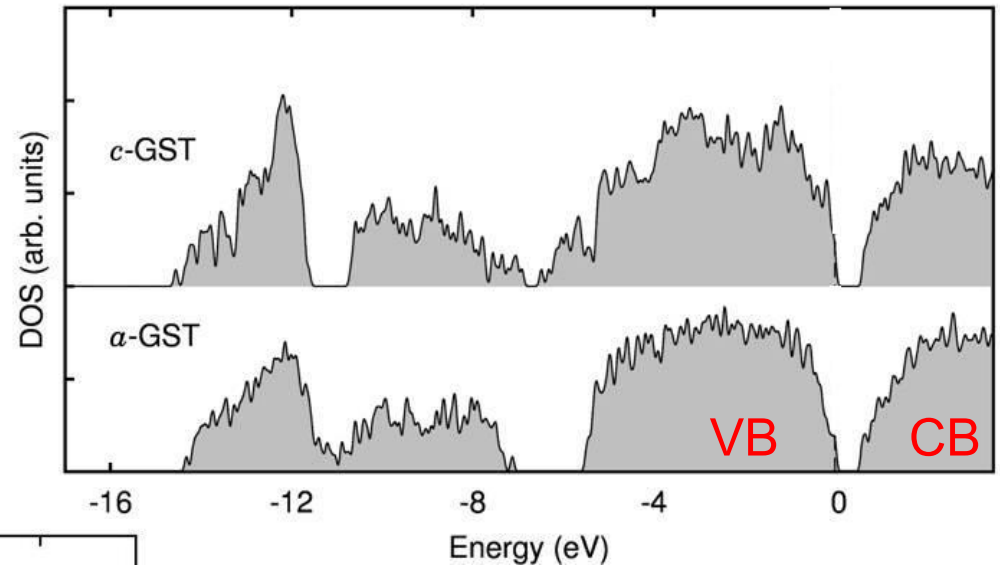
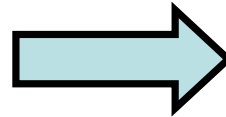
Exp: Andrikopolous et al J. Phys. Chem. Solids (2007); De Bastiani et al PRB (2009).

Mazzarello, Caravati, Angioletti, Bernasconi, Parrinello
Phys. Rev. Lett. 104, 085503 (2010)

Electronic properties

Electronic Density of States very similar in amorphous and crystalline phases

Theory



Exp. Photoemission
J-J. Kim et al, PRB 07

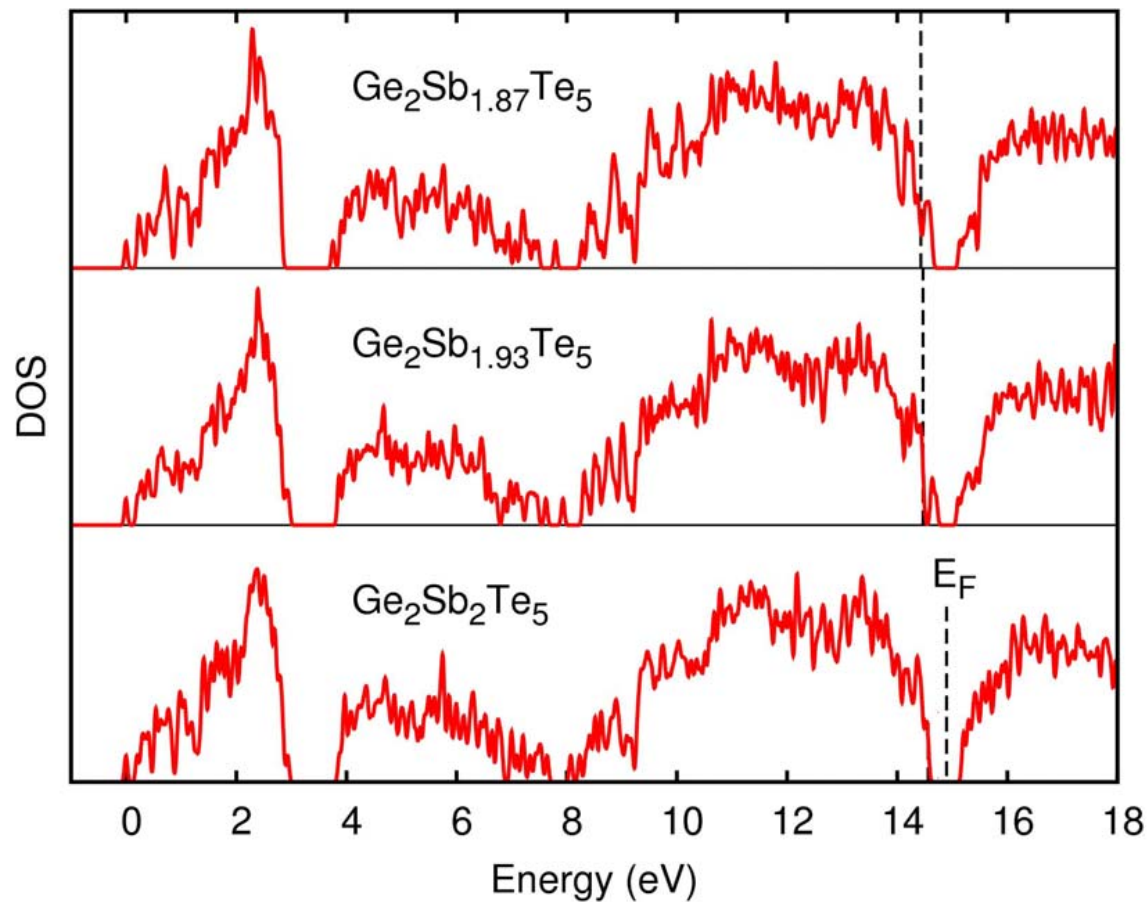
Optical Tauc gap:

c-GST \rightarrow 0.5 eV

a-GST \rightarrow 0.7 eV

Electronic properties: cubic crystal

GST stoichiometric crystal is semiconducting



Ge/Sb deficiency: \rightarrow holes in VB

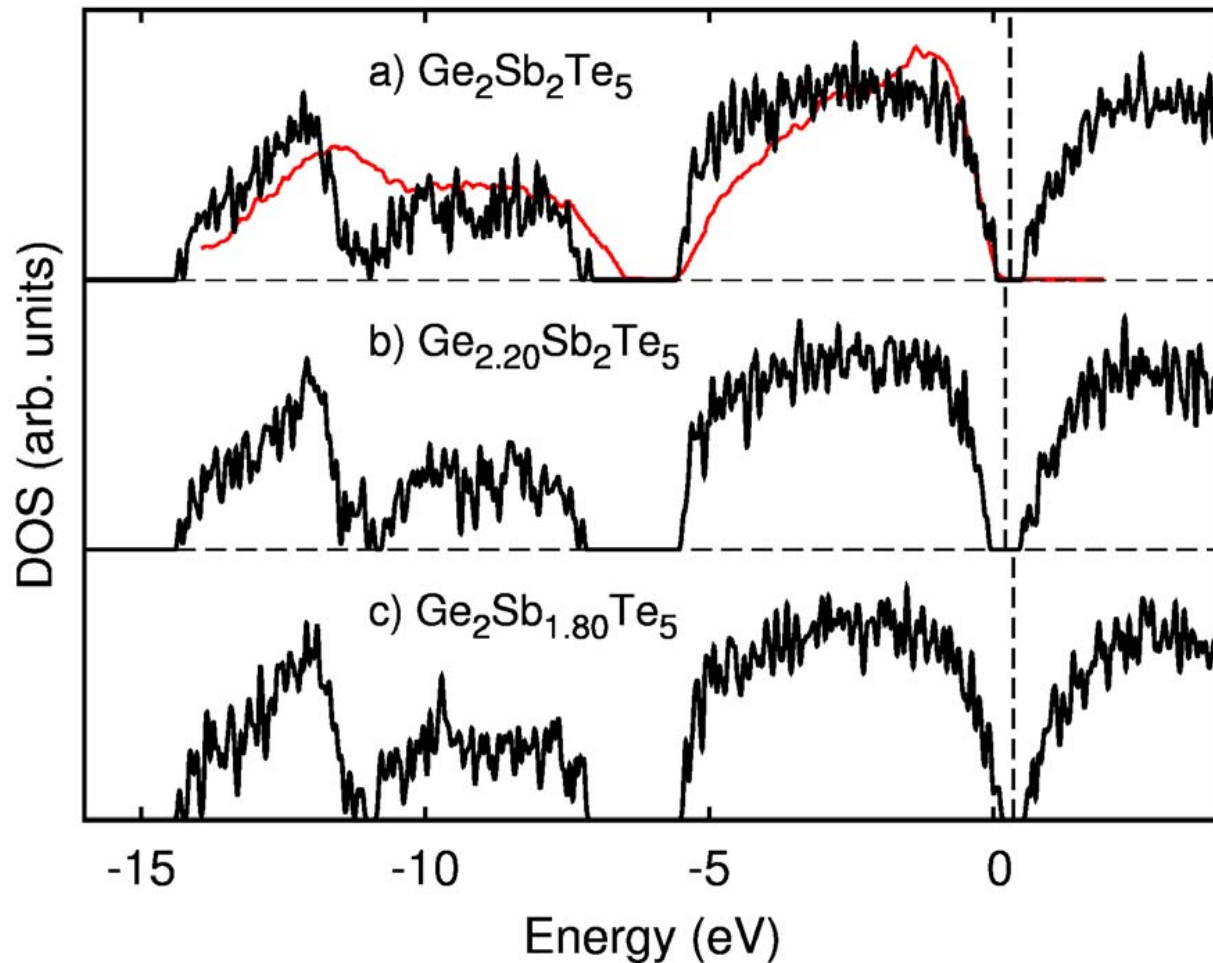
Ge/Sb excess: \rightarrow electrons in CB

HSE hybrid functional

Degenerate p-type semiconductor due to Sb deficiency

Electronic Properties

Non-stoichiometric GST: amorphous phase



Crystal can be doped
Amorphous can not

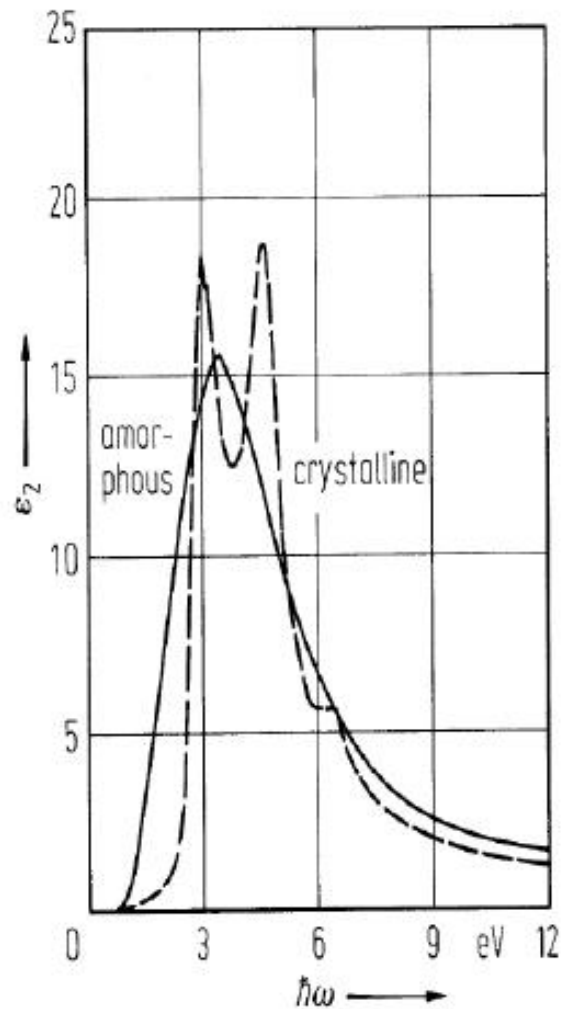


This explains
electronic contrast
(PCM)

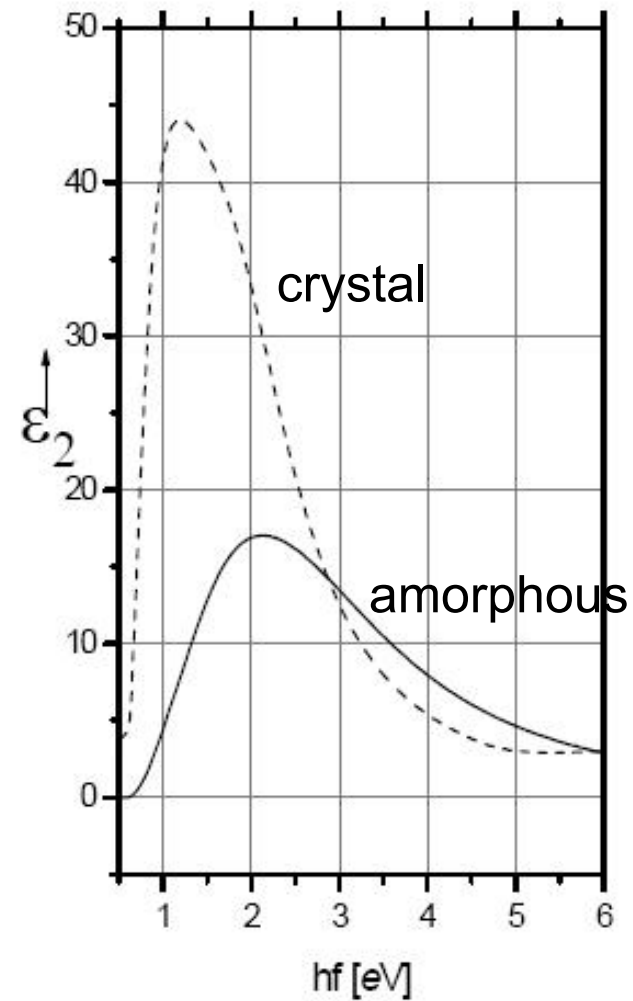
Ge excess, Sb deficiency $\rightarrow E_f$ at midgap

Optical contrast between amorphous and crystal

GaAs



$\text{Ge}_2\text{Sb}_2\text{Te}_5$



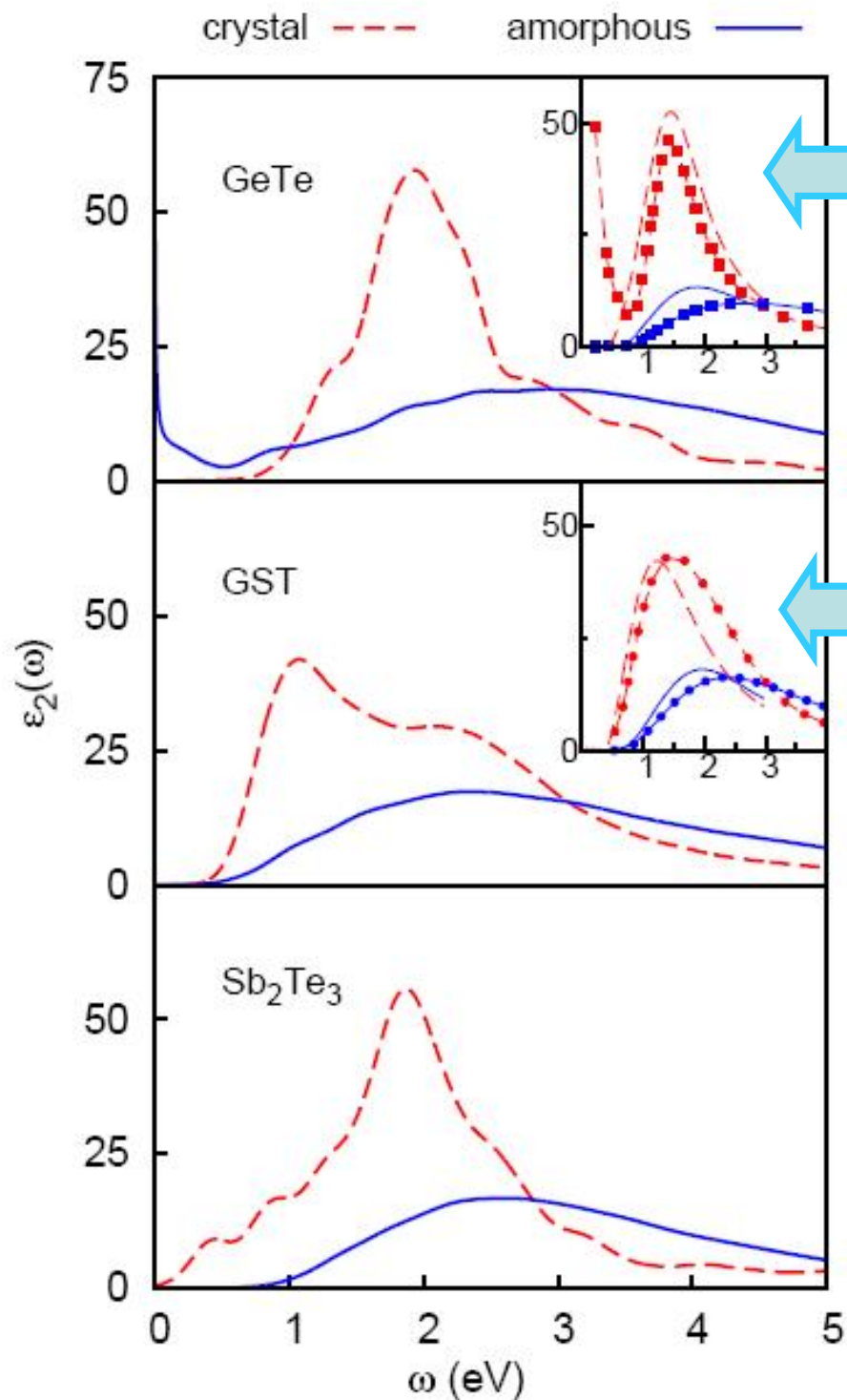
Optical contrast in PCM due to change in optical matrix elements

$$\epsilon_2(\omega) = \frac{8\pi^2}{3V_o N_{\mathbf{k}} \omega^2} \sum_{v,c,\mathbf{k}} |\langle c, \mathbf{k} | \mathbf{p} | v, \mathbf{k} \rangle|^2 \delta(\omega - E_{c,\mathbf{k}} + E_{v,\mathbf{k}})$$

- Presence of Ge tetrahedra
(Welnic, Botti, Reining, Wuttig, PRL 2007)
- Angular disorder in p-bonding in amorphous phase
Resonant valence bonds present in crystal and lost in amorphous

Shportko, et al., Nat. Mat. 2008

Huang and Robertson, PRB 2010



Exp. (Welnic 2007; Shportko 2008)

Optical contrast in PCM

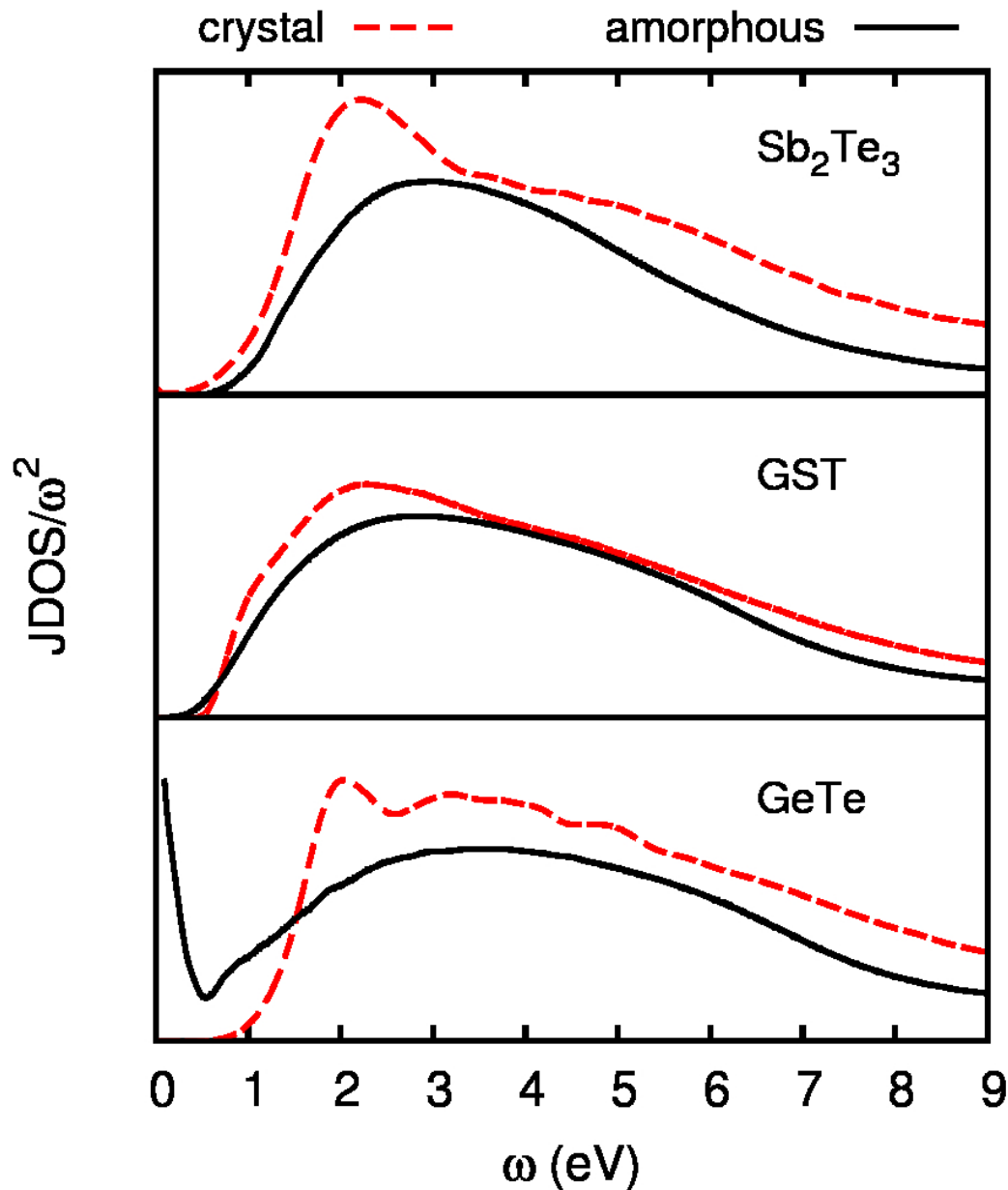
Exp. (Welnic 2007; Orava 2008)

$\epsilon_2(\omega)$ dielectric function
in RPA with hybrid
functional HSE

Caravati, Bernasconi, Parrinello

J. Phys. Cond. Matt. 22, 315801 (2010)

Joint Density of States



(Huang and Robertson PRB 2010)

Angular disorder in p-bonds in amorphous phases



decrease of the optical matrix elements



optical contrast

This explains why DVDs work

Open issues

- Structure of the amorphous phase
- Origin of resistivity contrast
- Origin of optical contrast



Origin of high crystallization speed

large simulations cells and long simulations time needed!

Large scale atomistic simulations of PCM

DFT results as database to develop interatomic potentials based on Neural Network to simulate 10^3 atoms for 10 ns

(Behler and Parrinello PRL 2007)

already applied to elemental carbon, silicon, sodium, ZnO

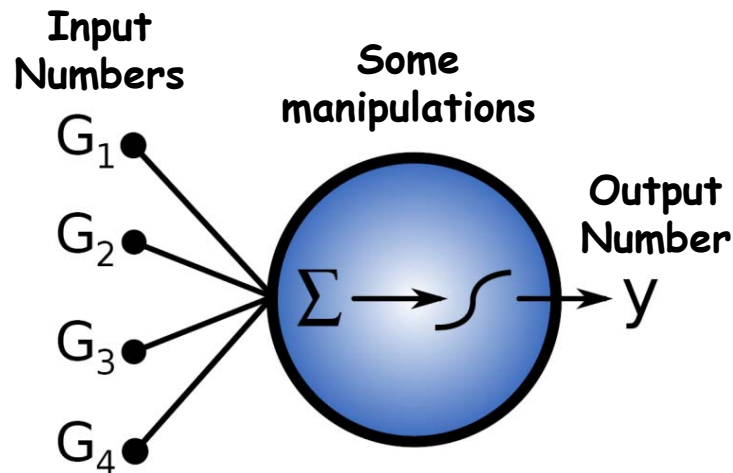
Neural Networks - Symmetry Functions

Total energy as sum of atomic energies $E = \sum_i E_i$



$$E_i = F(\{G(\bar{r})\})$$

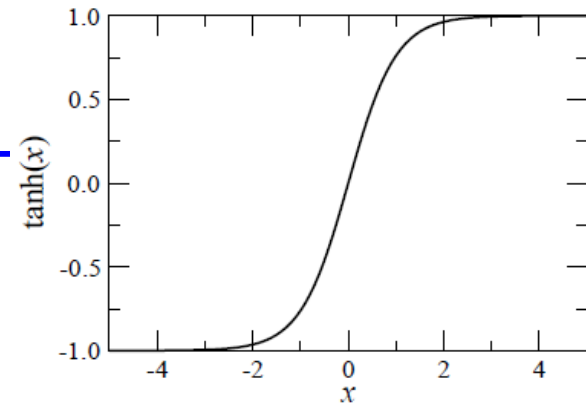
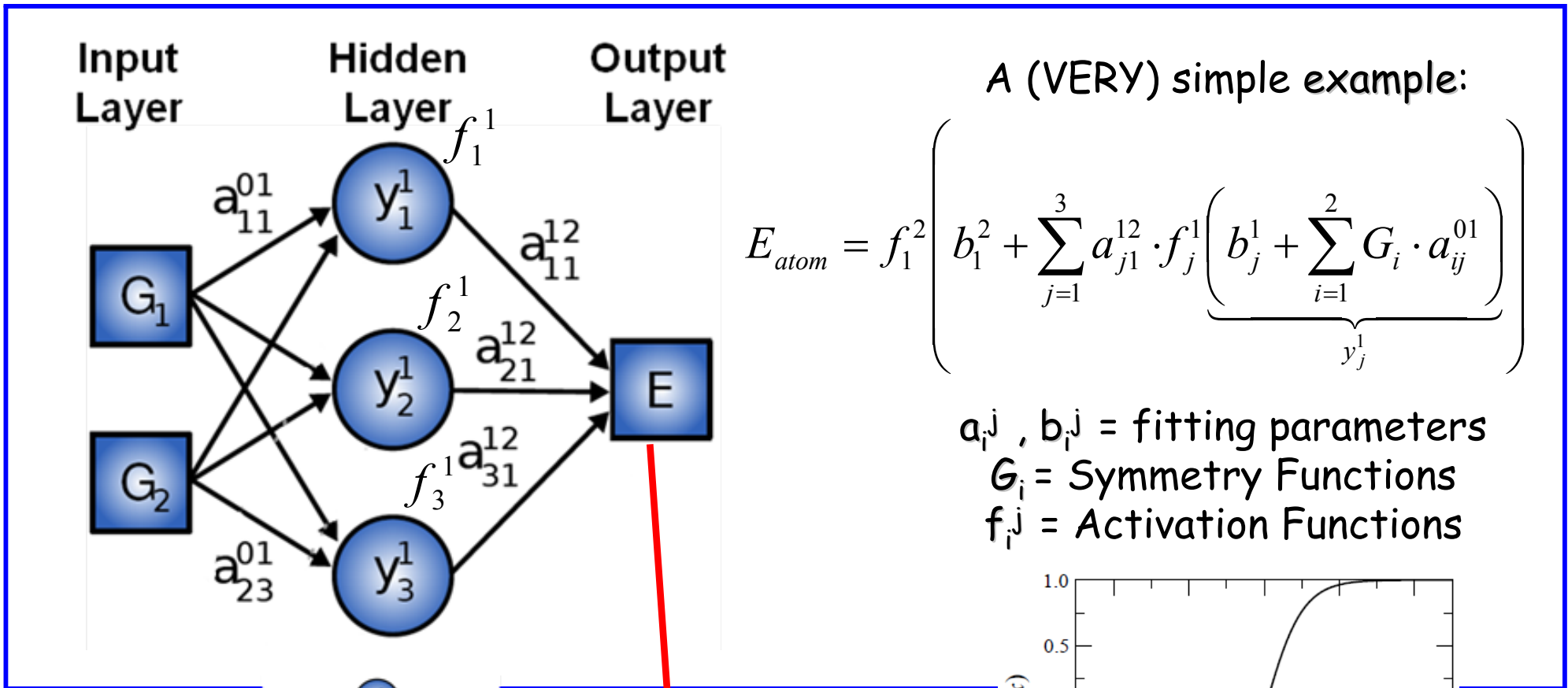
Symmetry functions $\{G\}$:
information on local atomic environment up
to a certain cutoff radius



F is EXTREMELY complicated,
but it IS analytic!

Input numbers: functions of atomic positions (symmetry functions)
Some manipulations: linear and non-linear combinations of input numbers
Output number: energy of the input configuration

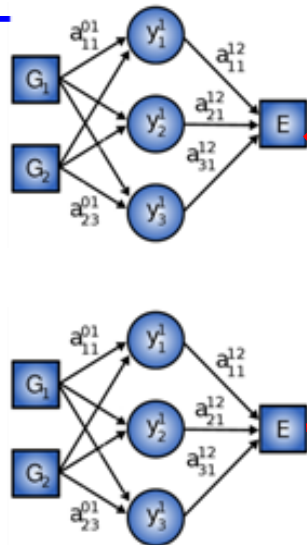
Neural Networks - The algorithm



another atom...

-
-
-
-

N other atoms!



Σ -> Total energy!

Neural Networks - GeTe

Building of the Neural Network Potential

159 symmetry functions for each atom
NN topology: 3 hidden layers of 20 nodes each
About 8000 fitting parameters.
No long range interactions
Code RuNNer (Bochum) + DL_Poly

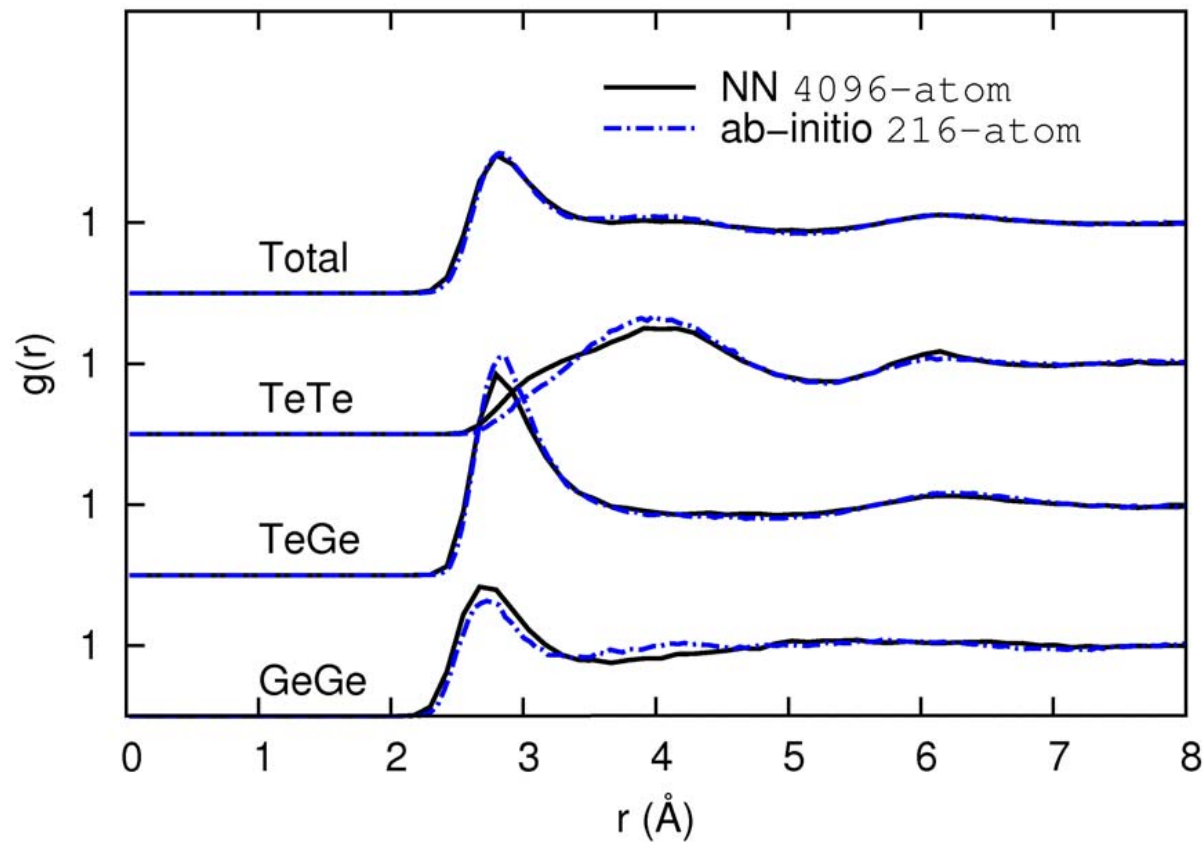
DFT-PBE calculations of selected structures for NN fitting (Quantum-Espresso code)

energy of 30000 structures (64- 128-atom cells, PBC) of crystalline, liquid and amorphous phases of GeTe at normal conditions and high pressure

DFT simulations of liquid and amorphous GeTe 216-atom cells CP2k code

Neural Network: liquid GeTe

pair correlation functions



NN: 4096-atom cell

Sosso, MB, et al. PRB 2012

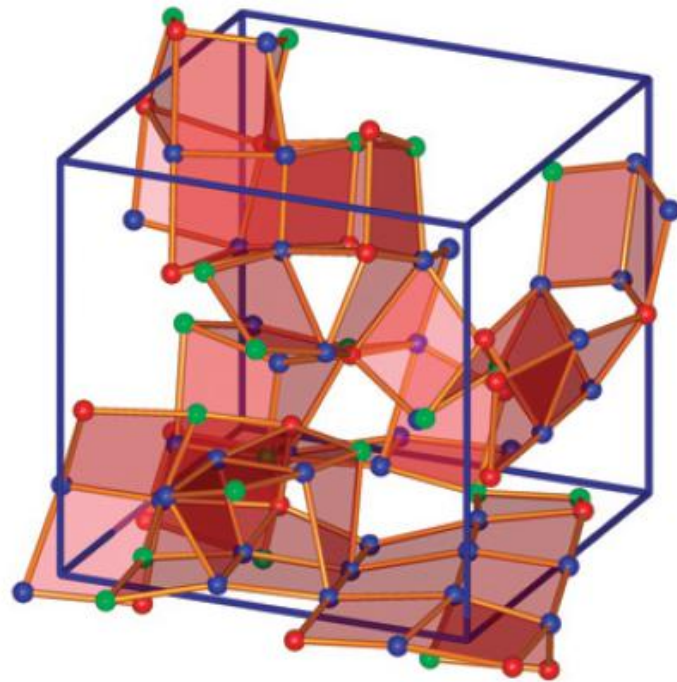
Diffusion coefficient

$D = 4.29 \cdot 10^{-5} \text{ cm}^2/\text{s}$

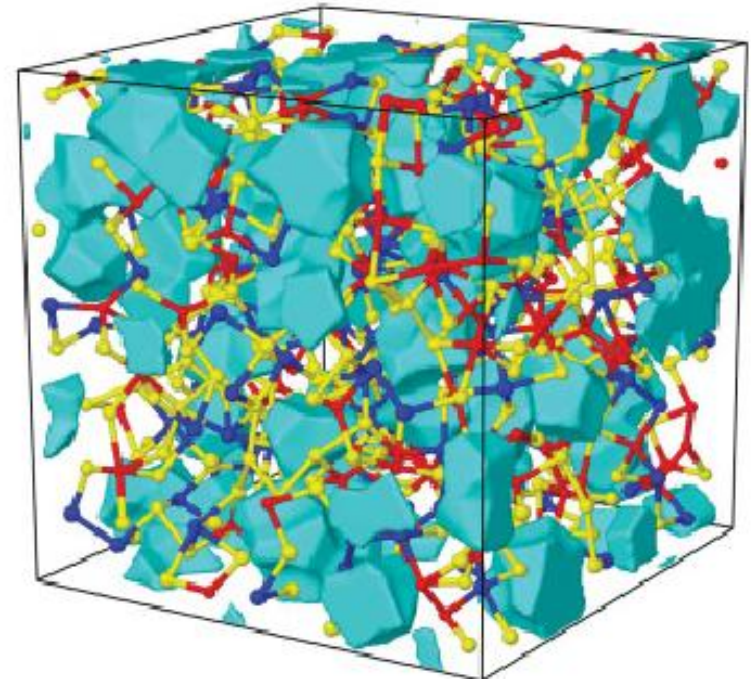
(DFT $4.29 \cdot 10^{-5} \text{ cm}^2/\text{s}$)

NN 10^4 faster than DFT for 4000 atoms

Origin of fast crystallization of phase change materials



α -GST



Rings

(Hegedus, Elliott Nat Mat 2008)

Cavities

(Akola and Jones PRB 2007)

but what's about the supercooled liquid?

Classical nucleation theory

Nucleation rate $I_{ss} \propto D \exp(-G_c/(k_B T))$

D = diffusion coefficient

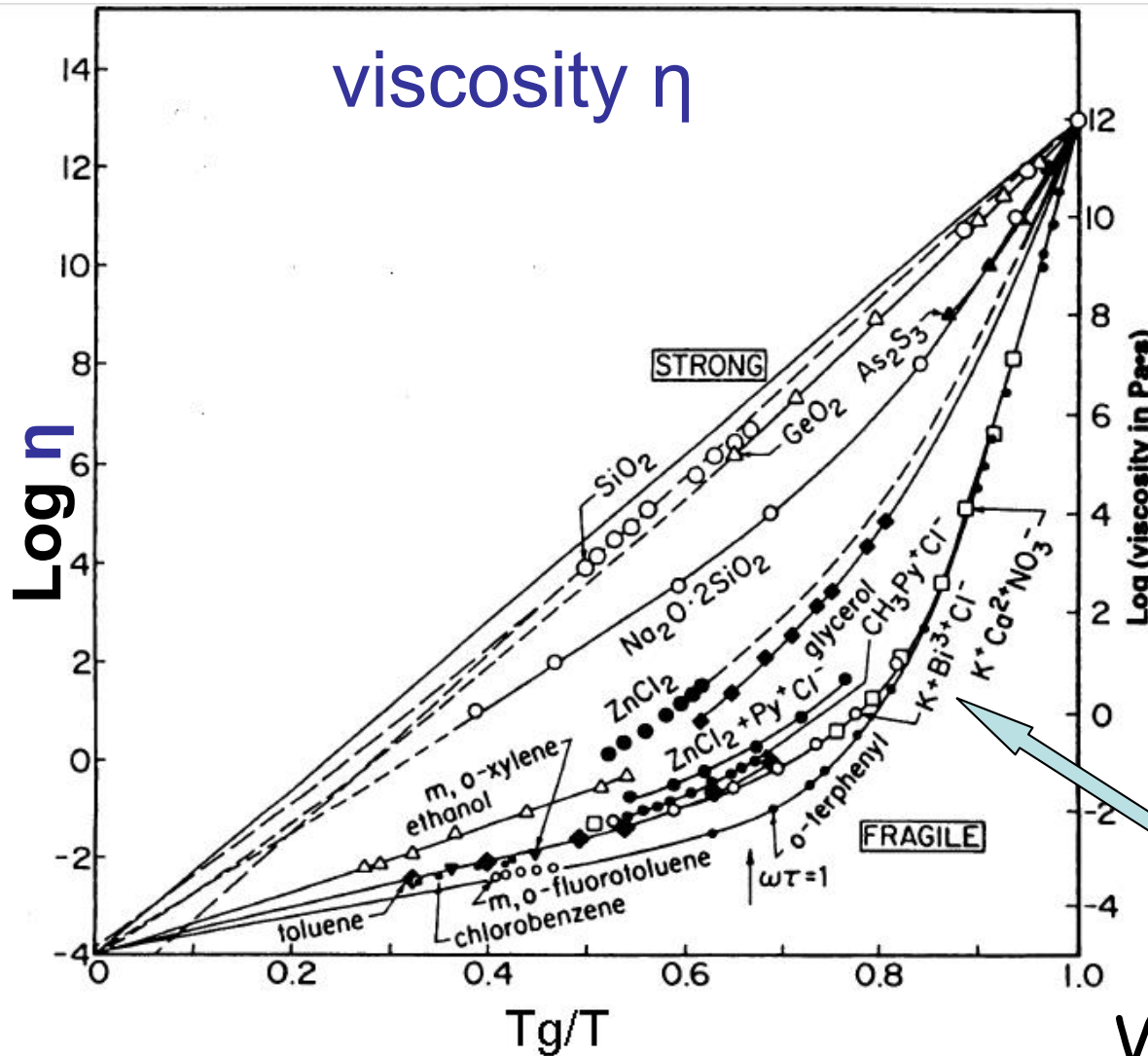
Free energy critical nucleus $G_c = 16\pi\sigma^3/(3\Delta\mu^2)$

$\Delta\mu$ = free energy difference liquid & cryst
driving force for crystallization

Crystal growth speed $u \propto D(1 - e^{-\frac{\Delta\mu}{k_B T}})$

Supercooled liquids: fragile or strong

Angell, Science 267, 1924 (1995)



fragility index

Silica $m = 15$

PVC $m = 195$

$$m = \left. \frac{d(\log_{10}\eta)}{d(T_g/T)} \right|_{T=T_g}$$

$$\eta \sim e^{\frac{1}{A(T-T_k)}}$$

Vogel-Tammann-Fulcher

Are phase change materials fragile liquids?

NN molecular dynamics simulation of
supercooled liquid GeTe

4096-atom cell

Simulation time of 2 ns

Melting temperature

Thermodynamic integration for free energy

$$F_{NN} - F_{ref} = \int_0^1 d\lambda \langle U(\lambda) \rangle \quad U(\lambda) = \lambda U_{NN} + (1 - \lambda) U_{ref}$$

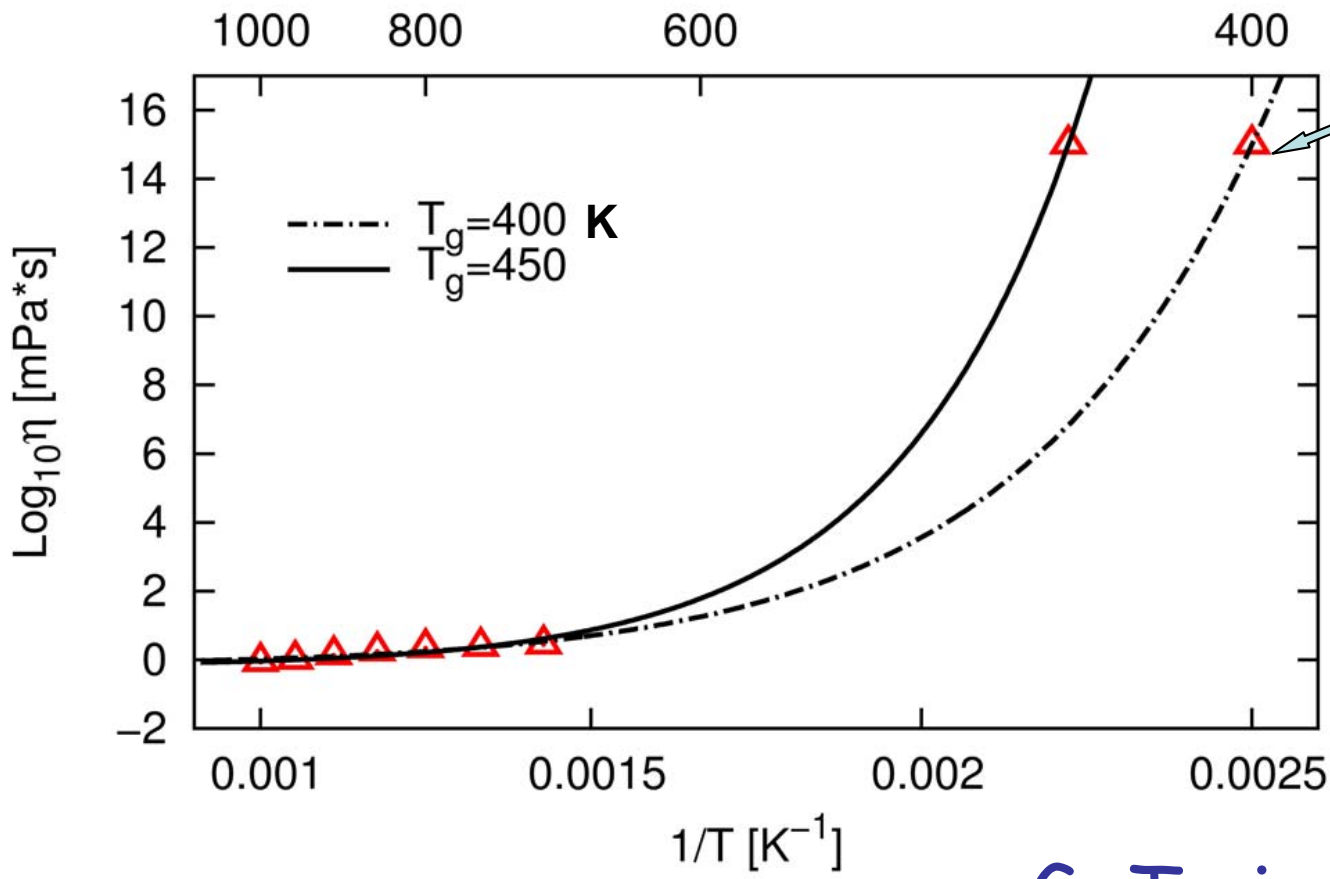
reference = harmonic crystal or LJ fluid

$$\mu_{liquid}(P, T_m) = \mu_{cryst}(P, T_m)$$

$$T_m = 1001 \text{ K} \quad \text{exp. } 998 \text{ K} \quad \text{GeTe}$$

Viscosity of supercooled liquid GeTe

Green-Kubo $\eta = \frac{V}{k_B T} \int_0^\infty \langle \sigma_{xy}(t) \sigma_{xy}(0) \rangle dt$

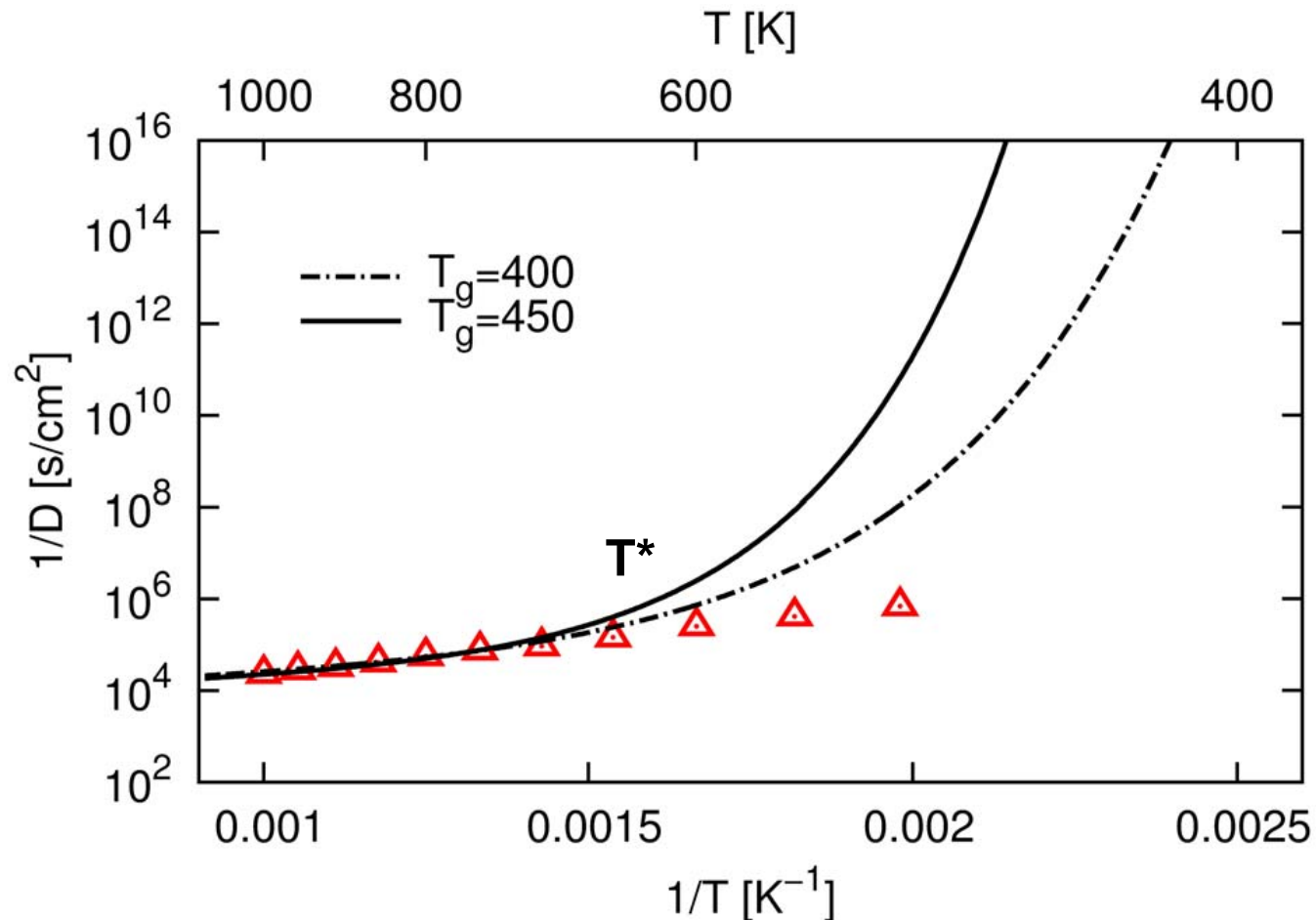


modified VTF
(Mauroa PNAS 2009)

fragility index $m=104$
(Silica =15, PVC =195)

GeTe is a fragile liquid !

Breakdown of Stokes-Einstein relation



$$D \propto \frac{T}{\eta}$$

not valid for

$$T < T^*$$

$$D = 10^{-6} \text{ cm}^2 / \text{s}$$

at 500 K

Self diffusion coefficient D computed from mean square displ. over 50 ps

Decoupling of diffusivity from viscosity in fragile GeTe liquid

Fast crystallization of amorphous just above T_g

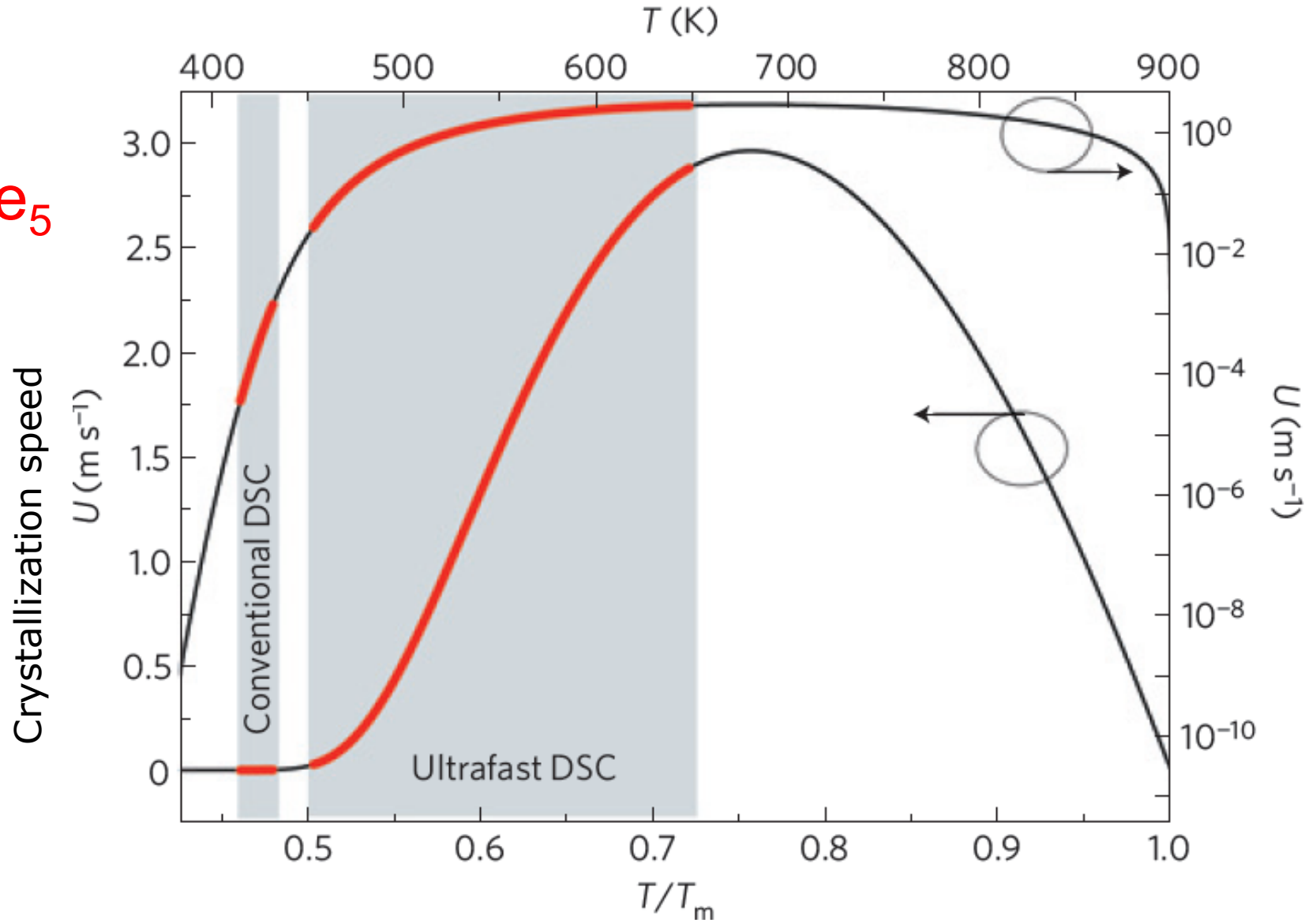
high diffusivity (fragility and breakdown of Stokes-Einstein)
at high supercooling

both large self-diffusion coefficient D and

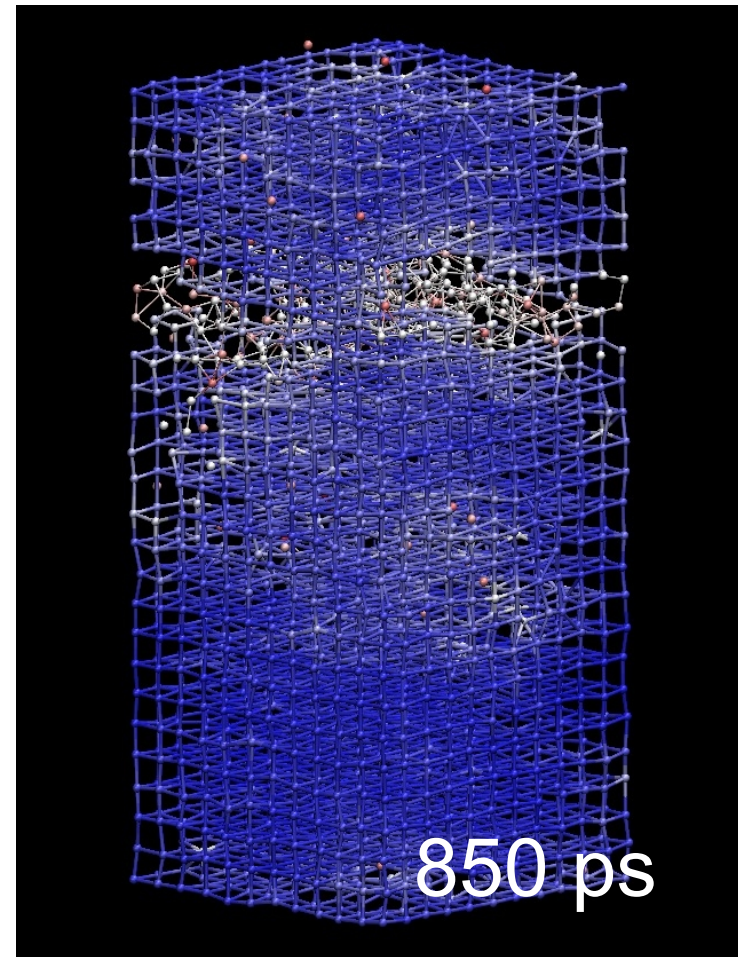
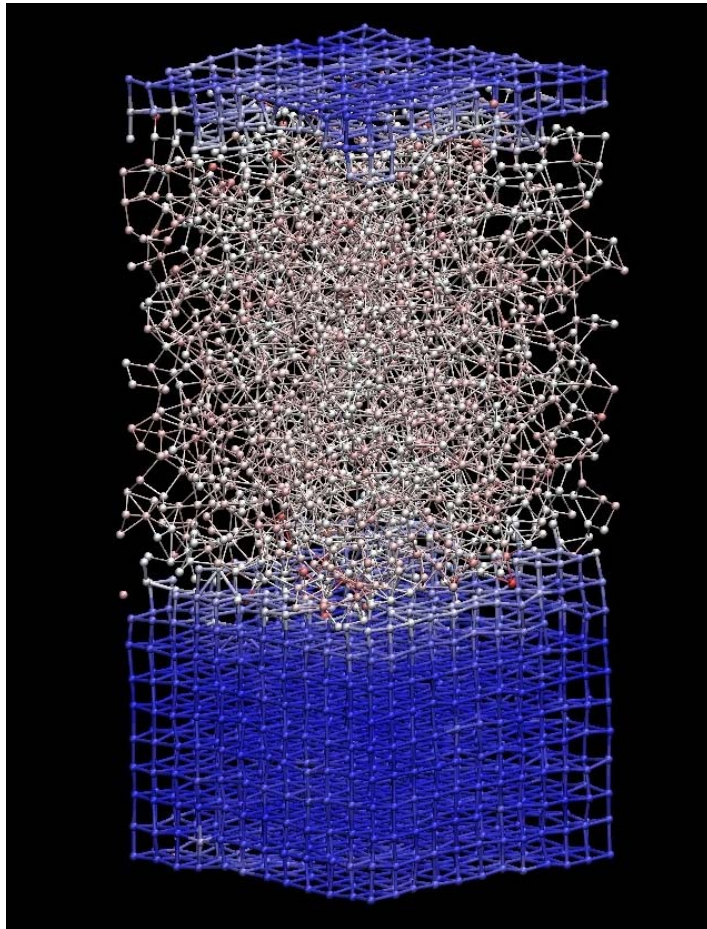
large $\Delta\mu$ driving force for crystallization

Ultrafast DSC (10^4 K/s)

Orava, Greer et al, Nat. Mat. 11, 279 (12)



Crystallization of supercooled GeTe at the liquid-amorphous interface



T= 600 K speed of crystal growth 2 m/s

Coworkers

Milano-Bicocca: G. Sosso, E. Spreafico, D. Colleoni, D. Mandelli
G. Miceli, S. Gabardi, S. Angioletti-Uberti

ETHZ, USI Lugano: S. Caravati, M. Parrinello

PSI-Villigen: M. Krack

Aachen University: R. Mazzarello

Bochum University: J. Behler

Mainz University: T. Kühne

Experimental colleagues at Micron (Agrate-Brianza)

Thank you for attention !

Publications

Caravati et al, Appl. Phys. Lett. 91, 171906 (2007);
Phys. Rev. Lett. 102, 205502 (2009) ; Phys. Rev. B 81, 014201 (2010) ;
J. Phys. Cond. Matt. 21, 255501 (2009); 22, 315801 (2010);
J. Phys. Cond. Matt. 23, 265801 (2011).

Sosso et al, J. Phys. Cond. Matt. 21, 095410 (2009); 21, 245401 (2009);
Phys. Rev. B 83, 134201 (2011); Phys. Rev. B 85, 174103 (2012);
Phys. Rev. B 86, 104301 (2012); Phys. Stat. Sol. B, 249, 1880 (2012).

Spreafico et al, Phys. Rev. B 83, 144205 (2011).

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Gabardi et al, J. Phys. Cond. Matt. 24, 385803 (2012).

Mandelli et al. Phys. Stat. Sol. B, 249, 2140 (2012).