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#### **Hands-on Tutorial on Electronic Structure Computations**

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Ground State Computations with Quantum Espresso: Self-consistent field (SCF) calculations with pwscf

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# **Ground-State Computations with Quantum ESPRESSO**

# Self-consistent field (SCF)

calculations

with pwscf

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# **Content**

### Brief introduction on

- how the scf calculations are performed in pwscf
- what are the main related input parameters

# **Note for experienced users**

For those of you who are already experienced users of pwscf

- please be patent (the fun part will come later)
- it would be nice if you could help for the hands-on part of the session (after the introduction)

# From a practical point-of-view

Complete information on the input variables for pwscf is available in the files:

PW/Doc/INPUT\_PW.txt and PW/Doc/INPUT\_PW.html

These files are the reference for input data and describe all variables

### **Self-consistent DFT calculations**

• The goal is to solve the Kohn-Sham equations:







L. J. Sham

$$\left\{ -\frac{1}{2m_e} \nabla^2 + V_{nuc/ion}(\mathbf{r}) + V_H[\rho(\mathbf{r})] + V_{XC}[\rho(\mathbf{r})] \right\} \psi_i(\mathbf{r}) = \varepsilon_i \psi_i(\mathbf{r})$$

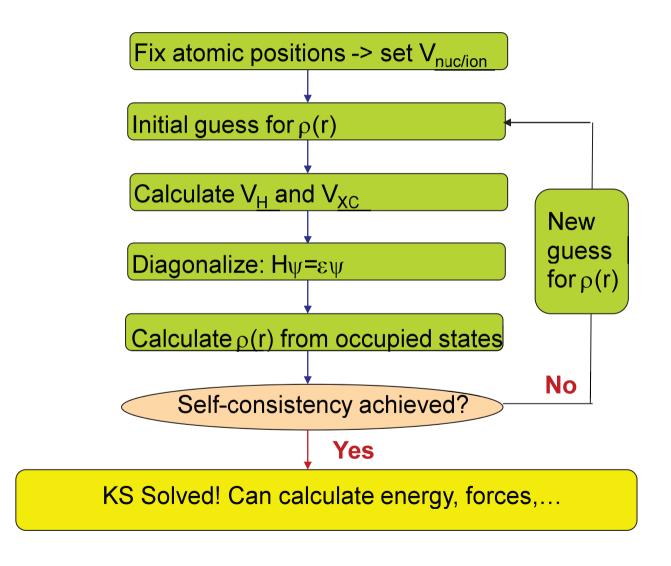
• The solution has to be self-consistent:  $H_{KS}$  depends on  $\rho$  and hence on the solution  $\{\psi_i\}$ 

$$\rho_{in}$$

$$\rho(\mathbf{r}) = \sum_{\varepsilon_i \le E_F} |\psi_i(\mathbf{r})|^2$$

$$\{\psi_i\}_{out}$$

# How to solve KS equations: general flowchart



# Approach/ingredients for efficient computations used in pwscf

- 1. Plane-wave basis set and periodic systems
- 2. Pseudopotentials
- 3. Discrete k-points for numerical integrations over the Brillouin zone

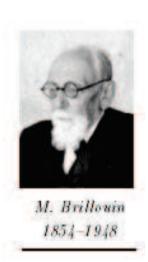
# 1. Plane-wave basis and periodic systems

For periodic systems:

$$\psi_{\mathbf{k}}(\mathbf{r}) = \sum_{\mathbf{G}} c_{\mathbf{k},\mathbf{G}} e^{i(\mathbf{k}+\mathbf{G}) \cdot \mathbf{r}}$$

**G**: reciprocal lattice vectors

**K**: in the Brillouin Zone (BZ)



• In the calculations, the plane-wave expansion is truncated, i.e., the basis includes plane waves up to a given kinetic energy cutoff (E<sub>cut</sub>):

$$\frac{\hbar^2 |\mathbf{k} + \mathbf{G}|^2}{2m_e} \le E_{cut}$$

Input parameter:

ecutwfc

# 1. Plane-wave basis and periodic systems

• In practice, the contribution of high Fourier components (large |k+G|) is small when the electronic orbitals are *not* sharply peaked or have lots of wiggles near the nucleus



In general, increased wiggles/localization

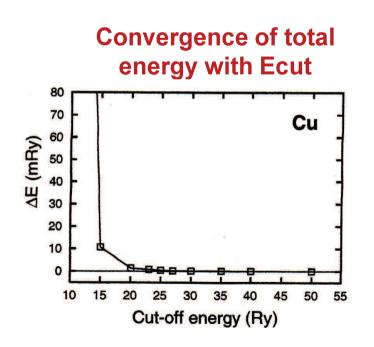


increased Ecut

# First rule:

 One should always check the convergence with Ecut:

variational parameter



F. Favot, EPFL Thesis

# 1. Defining the periodic system

(<u>in input file</u>, under namelist: **sysтем**)

Need to specify the Bravais lattice / unit cell ( pw basis: G vectors, BZ) & the atomic basis

A. Bravais 1811–1863

#### <u>Input parameters</u>:

ibrav sets type of Bravais lattice (sc,bcc,hex,...)

celldm(i) set lengths (and relative angles, if required)

of lattice vectors **a**<sub>1</sub>, **a**<sub>2</sub>, **a**<sub>3</sub>

natnumber of atoms in the unit cellntypnumber of types of atoms

Under field: ATOMIC\_POSITIONS

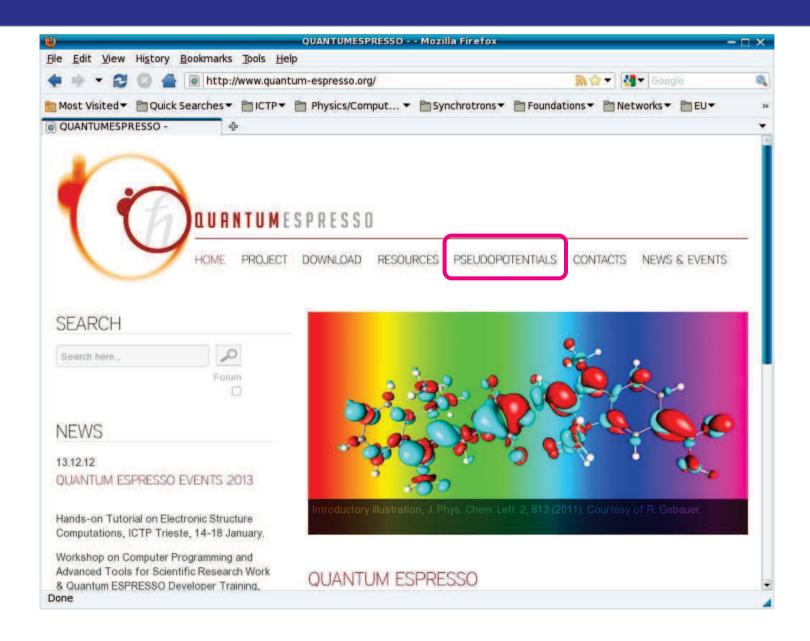
positions of the nat atoms; can be chosen to be given in units of lattice vectors ("crystal") or cartesian coordinates ("bohr", "angstrom" or "alat" units)

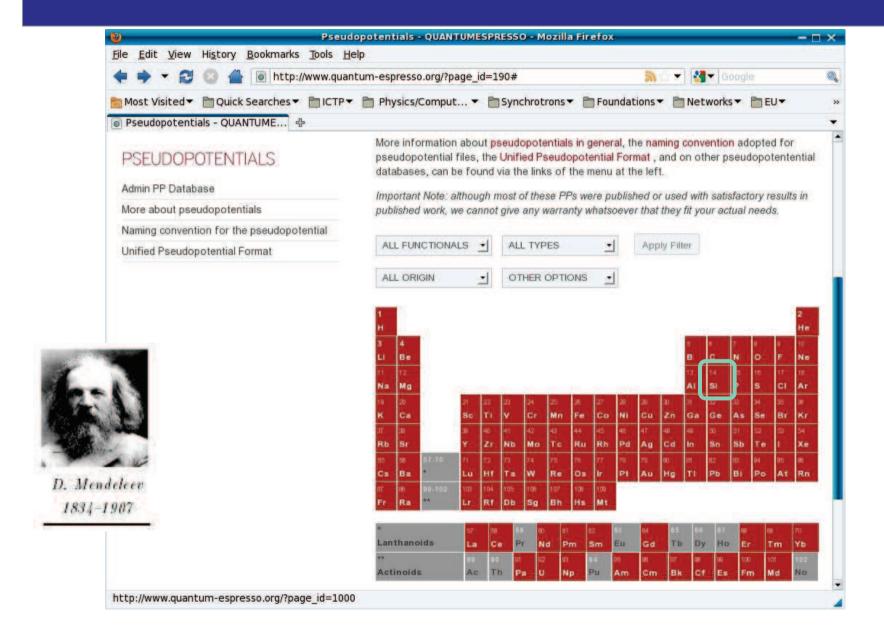
# 2. Pseudopotentials

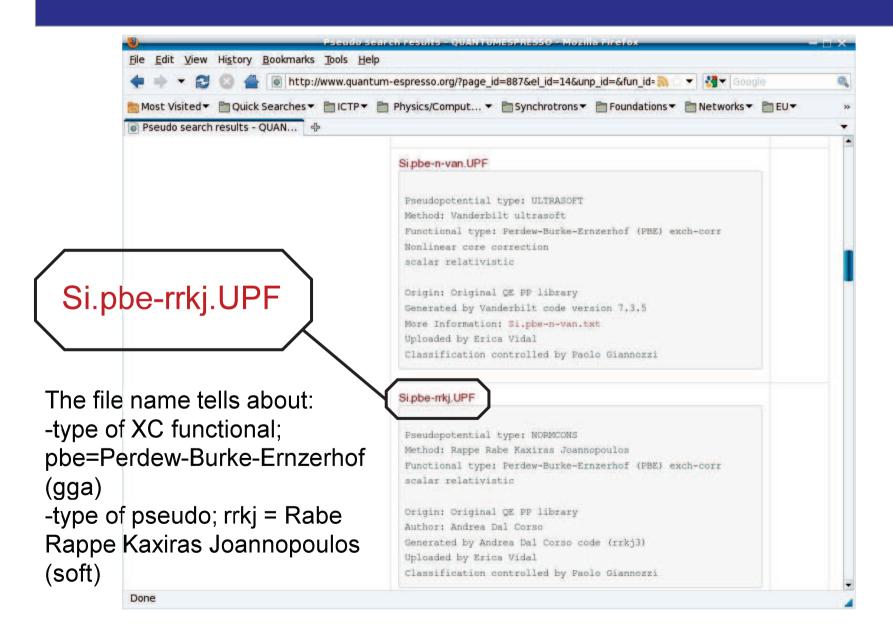
Nuclear potential replaced by pseudopotential:

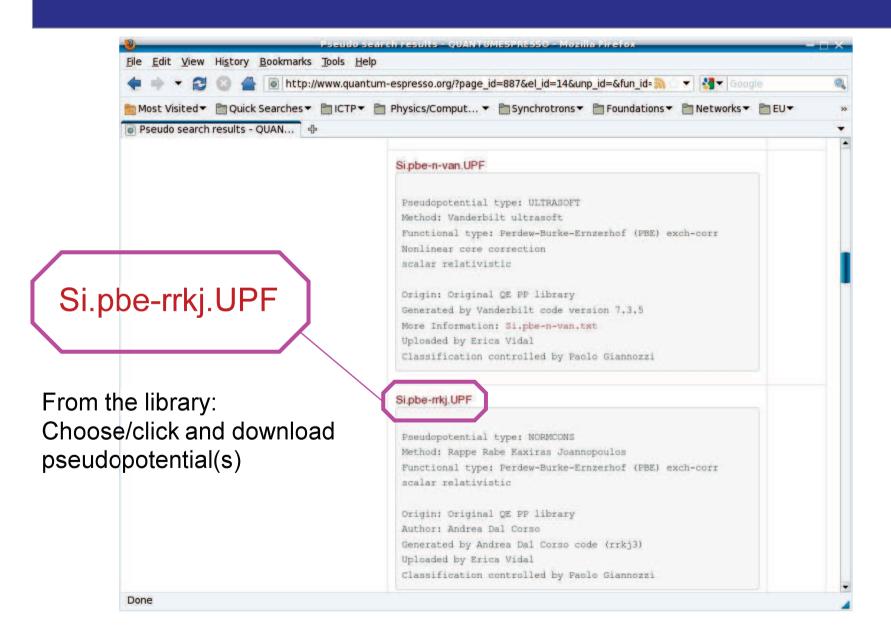
effective potential that reproduces valence-electron wave-functions outside core region (accounts for the effects of nuleus + core electrons)

- > need to solve KS only for valence electrons
- removes wiggles/sharpness of valence wavefunctions lowers E<sub>cut</sub>
- Different types of pseudopotentials: Norm-conserving, ultrasoft,.. (which type to use depends on element)









# 2. Pseudopotentials & input file

• <u>In input file</u>, pseudopotential name under field:

For Si, e.g.,

```
ATOMIC_SPECIES
Si 28.086 Si.pbe-rrkj.UPF
```

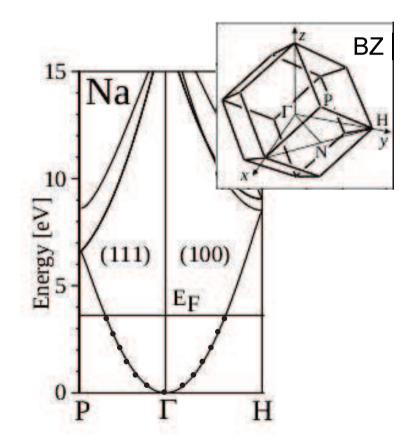
#### Notes:

- all species in input should have pseudopotentials with the same XC functional
- if ultrasoft pseudopotentials are used, the cutoff for (augmentation) charge: ecutrho should be set to 8-12 x ecutwfc (default 4 x ecutwfc)

# 3. Discrete k-points for numerial integrations over the Brillouin-Zone (BZ)

- Many needed quantities (in particular:  $\rho$ , Etot,...) require an integration over the BZ
- Done with sum over a finite nb of k-points in BZ with appropriate weights:

$$\langle Q \rangle = \sum_{\mathbf{k} \in BZ} Q(\mathbf{k}) w_{\mathbf{k}}$$



# 3. Discrete k-points for numerial integrations over the BZ

### Type of BZ sampling points

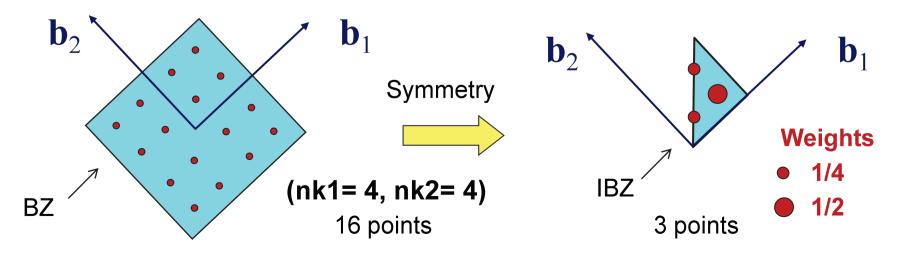
[in chronological order]:

• Special k-point(s): Baldereschi, Chadi-Cohen

Points derived to optimize integration with few k-points for specific classes of crystal structures

### Monkhorst-Pack (MP) grid:

Grid of equally spaced points (nk1, nk2, nk3) along each of the reciprocal-lattice unit-cell vectors; may be centered at the origin or shifted



# 3. Brillouin-zone-sampling points

In input file, under field: K\_POINTS

If "automatic" is specified, uses MP grid.

### Ex.:

```
K_POINTS (automatic)
4 4 4 1 1 1
```

nk1 nk2 nk3 Shift in units  $[\mathbf{b}_1/(2 \cdot nk1), \mathbf{b}_2/(2 \cdot nk2), \mathbf{b}_3/(2 \cdot nk3)]$ 

# 3. For metals: one should use electronic level broadening/smearing

- Metals require lots of k-points for convergence (abrupt cut of band structure at E<sub>F</sub>)
- Introduce smearing/broadening of the electronic levels to accelerate convergence of integrations with k-points

In input file, set

```
occupation = 'smearing'
```

and, e.g.,

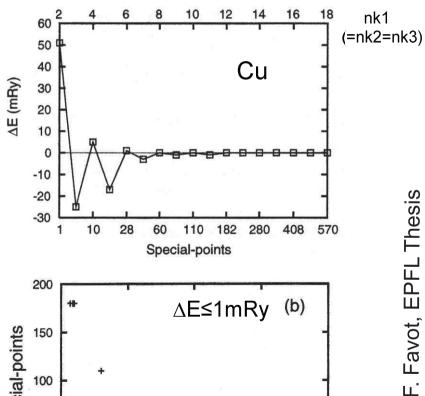
```
smearing ='methfessel-paxton'
degauss = 0.02
type of smearing
smearing width
```

• The smearing should not be too large (the larger the smearing, the faster the convergence, but also the larger the error introduced)

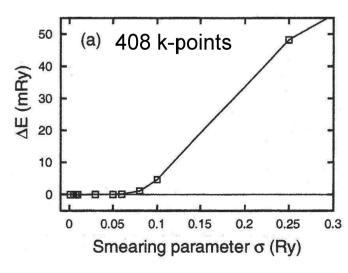
# 3. Checking convergence with k-points

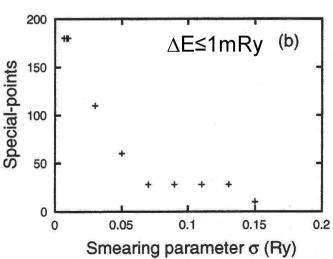
**Second rule:** Check convergence with k points

**Convergence of total** energy with nb of k points

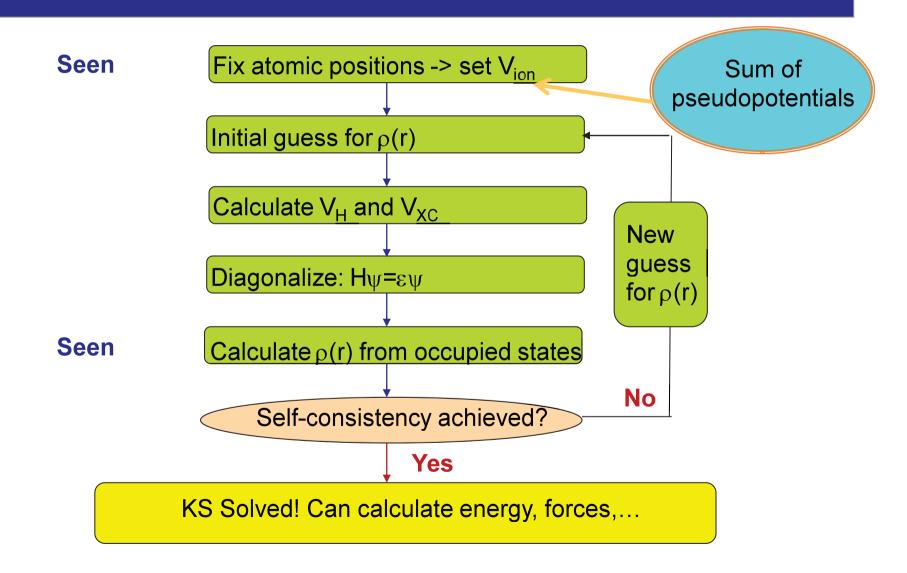


With smearing (Methfessel Paxton):

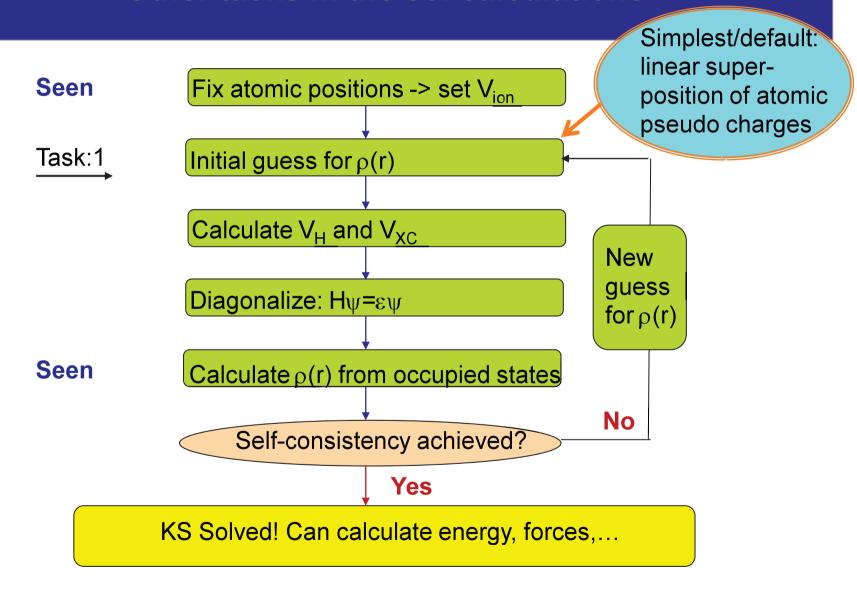




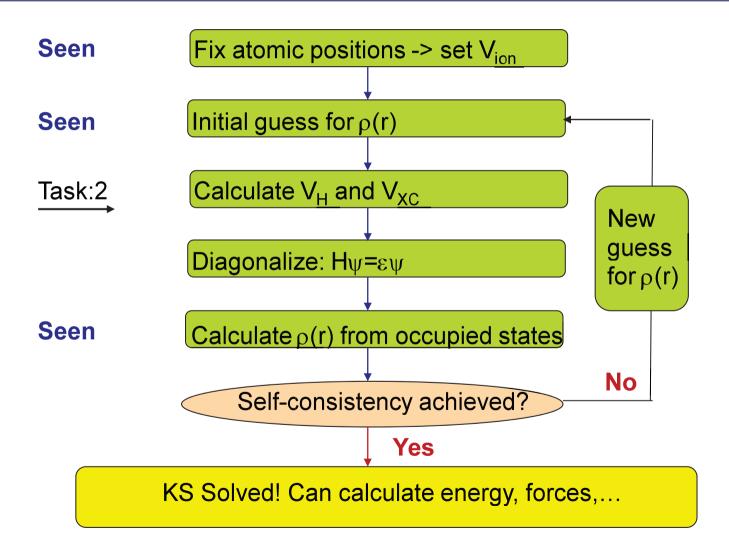
### Flowchart of the scf calculations



### Other tasks in the scf calculations



### Other tasks in the scf calculations



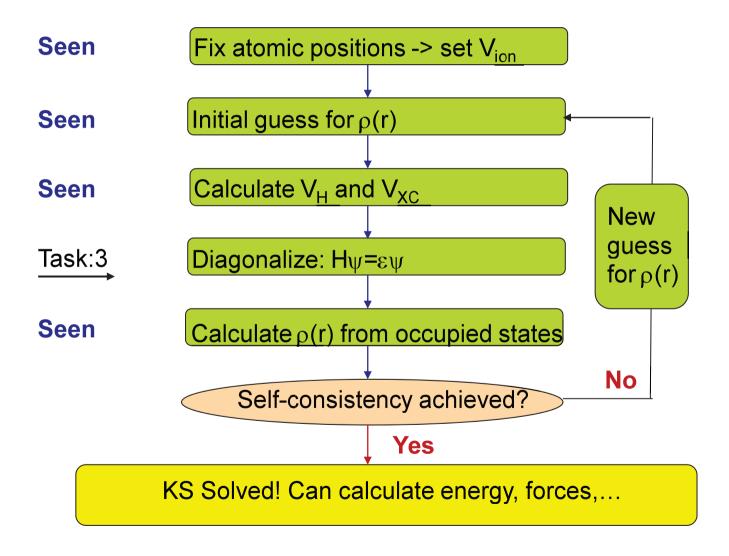
# **V<sub>H</sub> & V<sub>XC</sub>** [ρ]

- $V_H$  conveniently obtained in reciprocal space:  $V_H(\mathbf{G}) = 4\pi \ e^2 \ \rho(\mathbf{G})/G^2$  and efficient use of FFT
- $V_{XC}$  conveniently set directly in real space:  $V_{XC}[\rho(\mathbf{r}), \nabla \rho(\mathbf{r})]$

(the type of XC functional is chosen when specifying the pseudopotential)

Note: one needs then to iteratively apply  $H_{KS}$  on  $\psi_{ki}^{(trial)}$ :  $H_{KS}e^{ikr}u_{ki}$ , and the product  $[V_H(r)+V_{XC}(r)] \cdot u_{ki}(r)$  is conveniently computed in real space

### Other tasks in the scf calculations



# Iterative diagonalization

- In principle, in the plane-wave basis, the hamiltonian matrix size is  $N_{\text{PW}}xN_{\text{PW}}$  and  $N_{\text{PW}}$  is huge in general
- prohibitive for exact diagonalization
- however only a small number of bands/eigenvalues are needed in general compared to N<sub>PW</sub>
  - -> the approach is to use iterative diagonalizations (need mainly H  $\psi_i^{\text{(trial)}}$ )

In input, parameters:

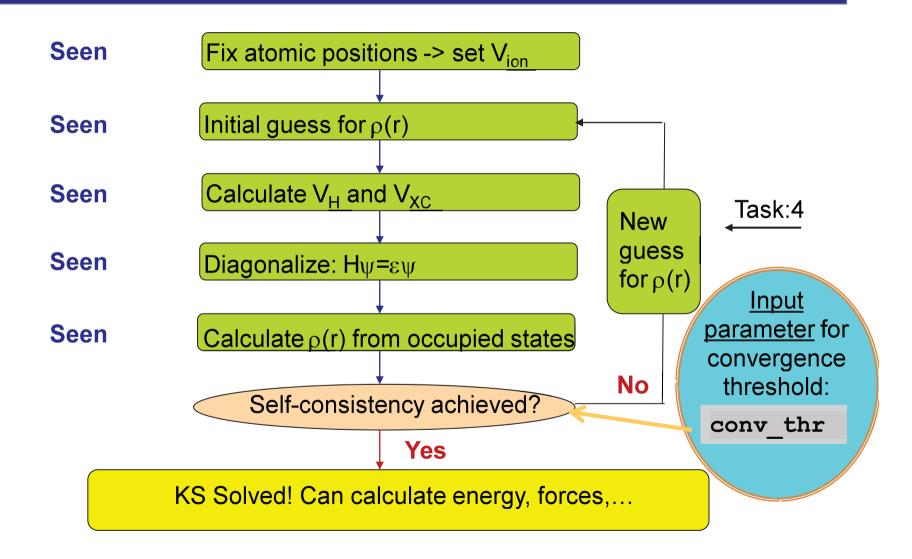
diagonalization

iterative-diagonalization algorithm: default is 'david' (Davidson), but can use also 'cg' (conjugate-gradient type)

nbnd

number of eigenvalues

### Other main tasks in the scf calculations



# **Mixing**

- It may take many, many iterations to reach self-consistency ( $\rho_{out} = \rho_{in}$ )
- Using just  $\rho_{out}(it=i)$  as input charge for  $\rho_{in}(it=i+1)$  usually does not work
- It is necessary to mix, i.e., use some combinations of input and output charges (using also information from the previous iterations) to get/accelerate convergence

#### <u>Input parameters</u>:

mixing mode

mixing scheme, default is 'plain' (Broyden), can be also 'TF' (plain with Thomas-Fermi-like screening)

mixing beta

how much new charge is used at the 1st mixing step (typical values are 0.1 to 0.7)

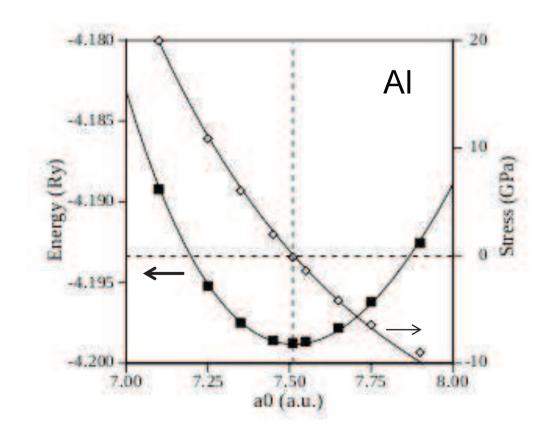
# Output values: total energy, forces, stresses

Can be used to optimize the atomic structure

e.g., lattice parameter of cubic crystal

In input, to print forces and calculate stresses:

tprnfor = .true.
tstress = .true.



C. Fall EPFL Thesis

### **Atomic relaxation calculations**

