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**EXPERIMENTAL WORKSHOP ON
HIGH TEMPERATURE SUPERCONDUCTORS
(11 - 22 April 1988)**

**THE CRYSTAL CHEMISTRY & STRUCTURES OF
HIGH T_c SUPERCONDUCTING OXIDES**

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① The crystal chemistry & structures of High T_c superconducting oxides
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What is the High T_c superconductor?

discovery of 35K $La_{2-x}Ba_xCuO_{4-y}$
(Bednorz & Müller)

finding of 90K $YBa_2Cu_3O_{7-x}$
(Chu, Zhao, Tarnaka)

finding of 85-110K $Pt-Cu-Ca-O$ oxides
(Maeda) based on $Pt-Cu-O$ oxide
(Michel & Raveau)

finding of 100-125K $Tl-Ba-Ca-Cu-O$ oxides
(Hermann & Sheng)

All of High T_c superconductors:
layer distorted - perovskite structure
or layer Bismuth structure
 $\left\{ \begin{array}{l} Bi_2O_2 \text{ double layers +} \\ \text{distorted perovskite.} \end{array} \right.$

What is the perovskite structure? (fig 1)

② stabilization of cubic perovskite:
radii of A and B cations satisfy
 $R_A + R_O = t * 2(R_B + R_O)$
 $.8 \leq t \leq .9$

$$R_A > R_B$$

For the distorted perovskite, t takes large range of value.

Basic informations: Cubic, $Pm\bar{3}m$
 $a \approx 3.9 \text{ \AA}$

second represent of perovskite structure.
(see fig. 2)

The features of ideal perovskite:

- Not contain close packed oxygen ions
- $B-O$ chain $\rightarrow$ $B-O$ sheet $\rightarrow$ $B-O$ 3 dimensional framework.

Ideal ABO_3 type perovskite:

examples: $CaTiO_3$, $SrTiO_3$ (fig.1)

distorted order,

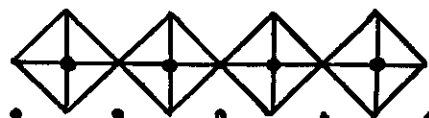
Cubic $\rightarrow$ Tetragonal $\rightarrow$ Orthorhombic

high temperature

low temp. phase

Perovskite-related structure

K_2NiF_4 type: La_2CuO_4 & $La_{2-x}Sr_xCuO_4$

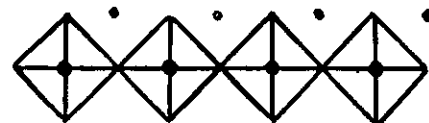


double layer
La-O (La_2O_3)



Cu
O

single layer Cu-O
octahedra



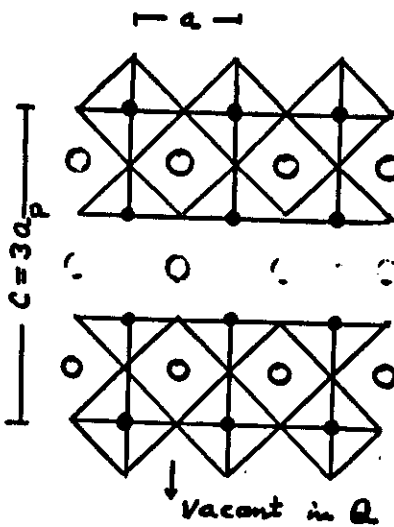
ABO_3 type & derivation

$ABO_3 \rightarrow A_m B_m O_{3m-1}$ ($m=2, 3, 4, 5$)

$m=3$

$A = Y$ and Ba , $B = Cu$

(3)



- CuO square plane layer

- Ba

- Cu-O pyramid layer

- Y (a oxygen layer was taken out)

$YBa_2Cu_3O_7 \rightarrow YBa_2Cu_3O_{7-x}$

$YBa_2Cu_3O_{7-x}$

Vacant in O axes

How to determine the structure:

techniques: x-ray powder diffraction (profile refinement) - Rietveld Method.

.. x-ray single crystal diffraction (four circles diffractometer)

... neutron powder scattering - Rietveld

:: electron microscopy.

Superconducting phase in Y-Ba-Cu-O system

a. phase relations in YBCO system & stable phases: $YBa_2Cu_3O_{7-x}$

b. $T_c \geq 90K$, A.c. χ or Meissner effect, anisotropic resistance measurement,

c direction: $m \rightarrow smi \rightarrow sp$

a-b plane: $m \rightarrow sp$

(4)

c. Symmetry & space Group: Orthorhombic . Pmmn ⑤

d. phase transition : Tr. $\xrightarrow{760^\circ\text{C}}$ Or.

Lattice constants change anisotropically with the T_g and Oxygen content ,

$a \uparrow$, $b \downarrow$, $c \uparrow$ ($V \uparrow$) when O change from 7.0 to 6.5 (O as b, $c \uparrow$ when O from 6.5 to 6.0.

Two plateaus : 90K (7.0 - 6.75), 60K (6.75 - 6.45)

e. DTA, TG: No thermal effect $\xrightarrow{\text{at } 750^\circ\text{C}}$ low enthalpy
weight loss on heating (max. at 500°C)
weight gain on cooling in air

f. Structure of 1-2-3 compound (Or)

1. triple layers perovskite
2. one dimensional Cu-O chain in $\bar{b}$ axis

3. Basic data : (see table)

g. Microstructure : inhomogeneous composition
Twin, domain, stacking fault,
Amorphous.

h. Structure of Non-superconducting $\text{YBa}_2\text{Cu}_3\text{O}_6$

⑤

tetragonal . s.g. P4/mmm ⑥

{ oxygen amount : $6.5 \geq x \geq 6.0$
distribution of oxygen : Vacant in b axes.
 $\rightarrow$ No Cu-O-Cu-O chain

No twin, No domain

composition fluctuation

Semiconductivity, until 4.2K

Common structure features of $\text{YBa}_2\text{Cu}_3\text{O}_7$ & $\text{YBa}_2\text{Cu}_3\text{O}_6$
triple layers $\left\{ \begin{array}{l} 3 \text{ subcell of perovskite} \\ \text{Ba-Y-Ba-Ba-Y ordering along } \bar{c} \\ \text{Cu-O bond length } \sim 1.95\text{\AA} \end{array} \right.$

Uncertain things about 1-2-3 :

1. Oxidation state or valence of Cu ion.
 Cu^+ , Cu^{2+} , Cu^{3+} or $\text{Cu}^{2+}/\text{Cu}^{3+}$ mixture
2.3 ? XAES, XPS, ESR.
2. dimensionality : 1, 2, or 3 ?
3. exact role of oxygen ?
4. role of defects ?
Low critical current density
 10^3 amps/cm^2 (bulk sample) $\rightarrow 10^4 \text{ A/cm}^2$
5. isotopic effect : $\text{O}^{18} \rightarrow \text{O}^{16}$, $\text{Cu}^{63} \rightarrow \text{Cu}^{65}$

⑥

for $\text{La}_{2-x}\text{Sr}_x\text{CuO}_{4-y}$ $\Delta T \sim 0.5 \text{ K}$ ⑦
 $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ $\Delta T \sim 0.5 \text{ K}$ 0.18

for $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$
 no obvious isotopic effect for Cu
no support BCS. No against BCS.

6. c parameter jump when $O = 6.5$
 Very small jump of c parameter due to
 the Bond length of Cu-O bond along c axes
 Is it the true reason for 90K superconductivity?

7. Partial ordering of oxygen vacancies along
 b axes,

It is not confirmed by others.

⑦

Superconducting phases in La-Ba-Cu-O system ⑧
 ABO₃ type $\text{LaBa}_2\text{Cu}_3\text{O}_{7-x}$ (80K)
 K₂NiF₄ type $\text{La}_{2-x}\text{Ba}_x\text{CuO}_{4-y}$ (30-40K)

$\text{LaBa}_2\text{Cu}_3\text{O}_{7-x} - \text{LaBaCu}_2\text{O}_{5-x}$ (or $\text{La}_2\text{Ba}_2\text{Cu}_6\text{O}_{15-x}$)

Solid solution

structure change continuously.

T_c from 80K to 40K be adjusted

→ La & Ba ordering not necessary

$\text{La}_{2-x}\text{Ba}_x\text{CuO}_{4-y}$:

La_2CuO_4 stoichiometric compound
 — semiconductivity until 4.2 K

$\text{La}_2\text{Cu}_{1-x}\text{O}_{4-y}$ non stoichiometric
 — superconductivity at 30-40K

only can synthesize at high temperature
 and high oxygen pressure.

La_2CuO_4 by adding Ba
 Orthorhombic → tetragonal

Cu-O chain ?

⑧

Substitution of Y

most of rare earths except of Ce, Pr, Pm, T
the other 3⁺, 4⁺ elements.

F-O compound: 150K ? ~~X~~

Substitution of Ba, (mono; di-; tri-valence)

Cu is absolutely necessary.

Four criteria for searching of new or high T_c superconductors:

- Zero resistivity
- .. Meissner effect
- ... high stability
- :: high reproducibility

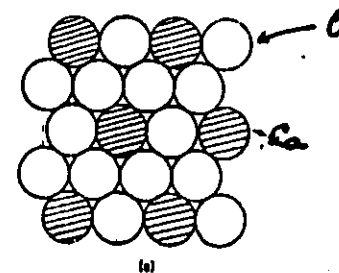
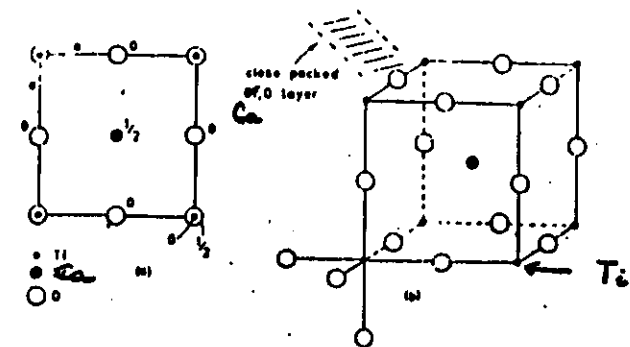
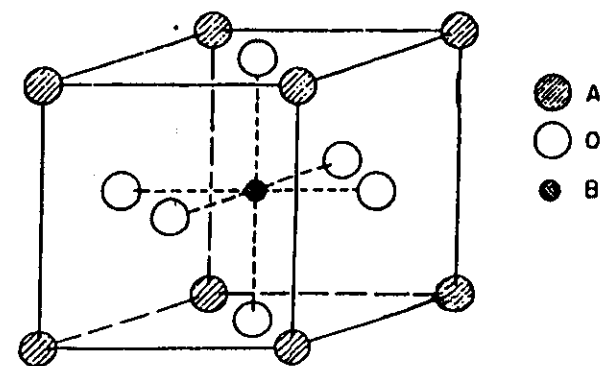


Fig. 1



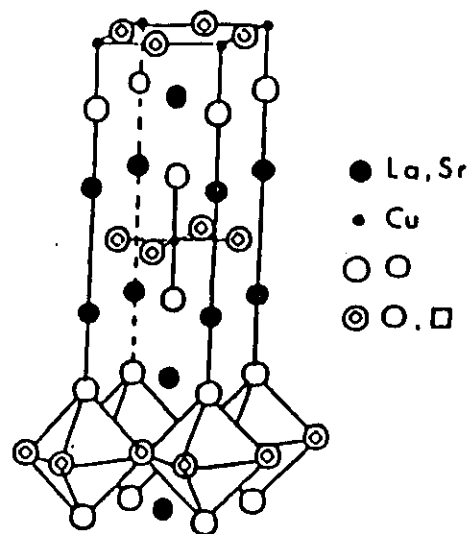
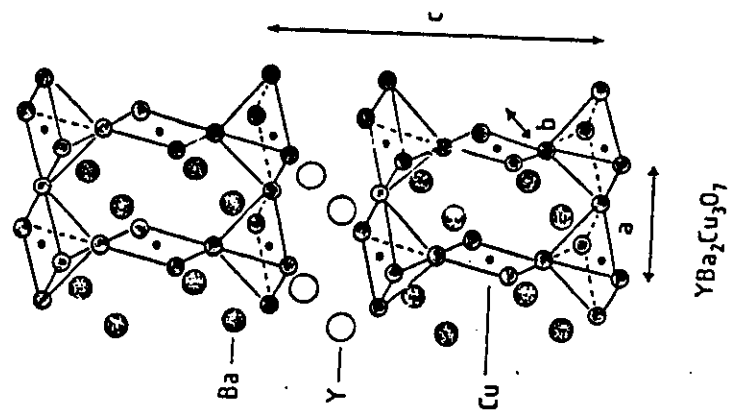
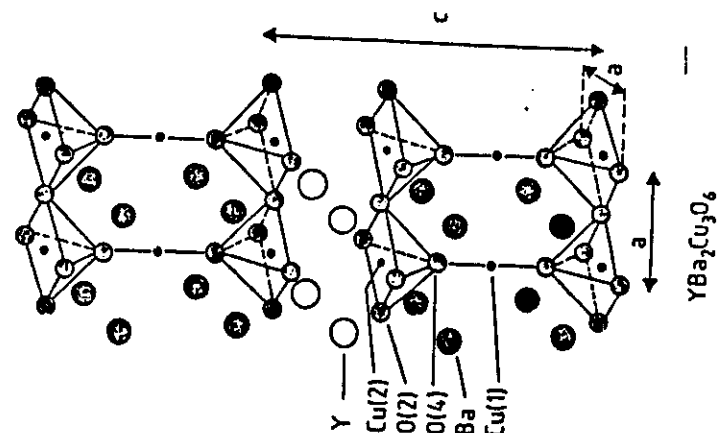
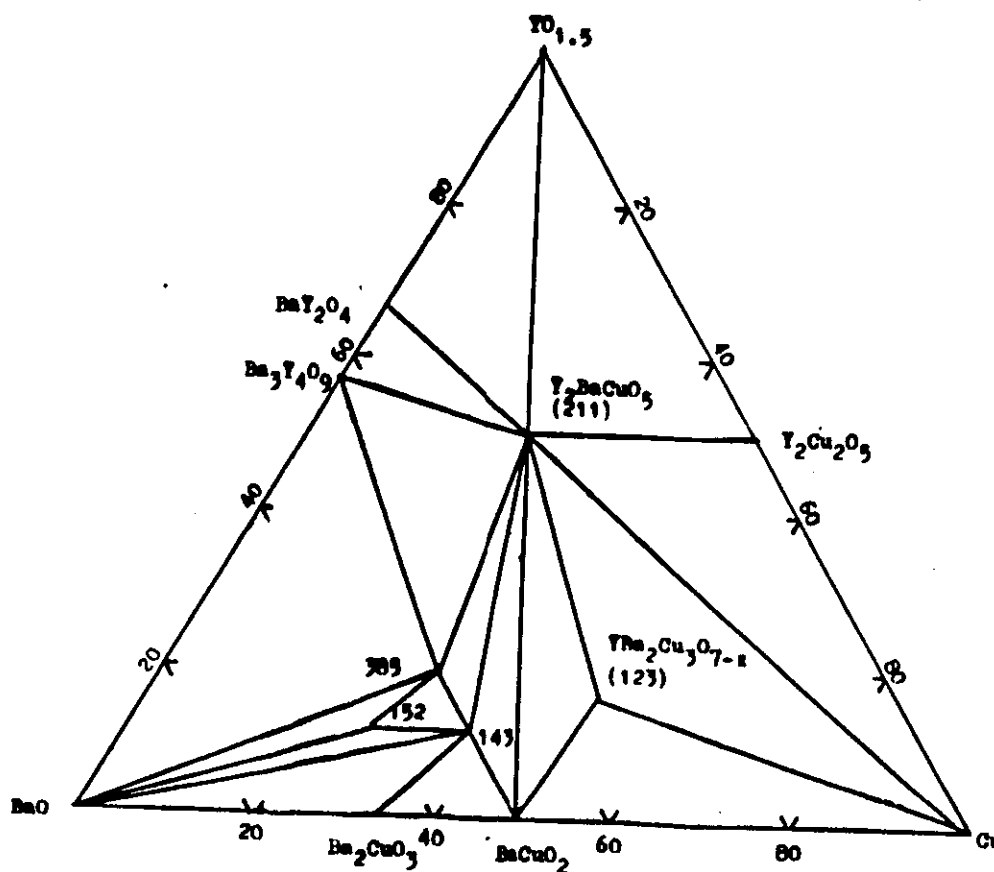


Fig 3 a,

11



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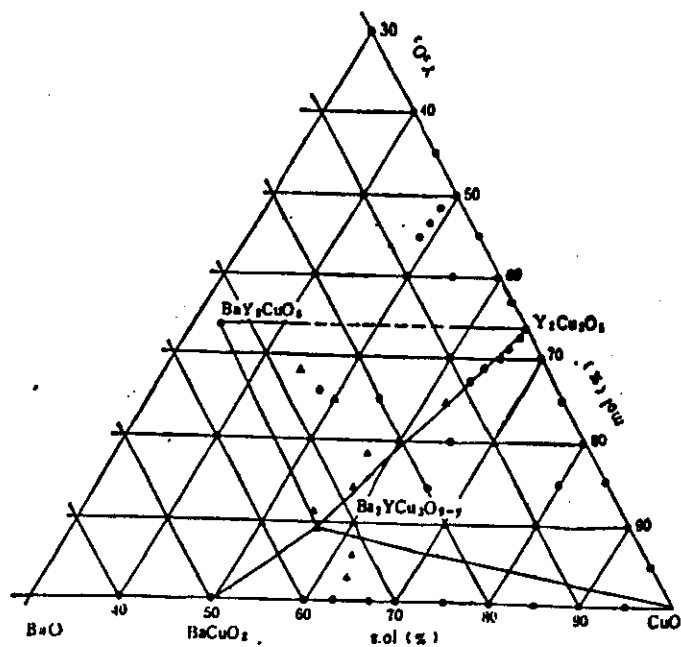
Hinks & Philips Lab.

(13)

The stable phases in Y_2O_3 -BaO-CuO system

Compound	Symmetry	Lattice constants (Å)
$BaCuO_2$	Cubic	$a = 18.86$
$Ba_2CuO_3^*$	?	
$Ba_2Y_2O_5$	Tetragonal	$a = 3.3771, c = 11.852$
$Ba_2Y_2O_4$	Orthorhombic	$a = 10.388, b = 12.00, c = 3.448$
$Y_2Cu_2O_5$ (green)	Orthorhombic	?
Y_2BaCuO_5 (211) (green)	Orthorhombic	$a = 7.123, b = 12.163, c = 5.649$
$YBa_2Cu_3O_{7-x}$ (123) (black)	Orthorhombic superconducting phase	$a = 3.823, b = 3.829, c = 11.61$
$YBa_2Cu_3O_6$ (high Temp.) phase	Tetragonal	$a = 3.8648, c = 11.763$
$YBa_4Cu_3O_{8.5}$	?	
$YBa_5Cu_3O_{8.5}$	?	
$Y_3Ba_8Cu_5O_{17.5}$	?	
+ Y_2O_3		
+ CuO		
+ BaO		

(14)



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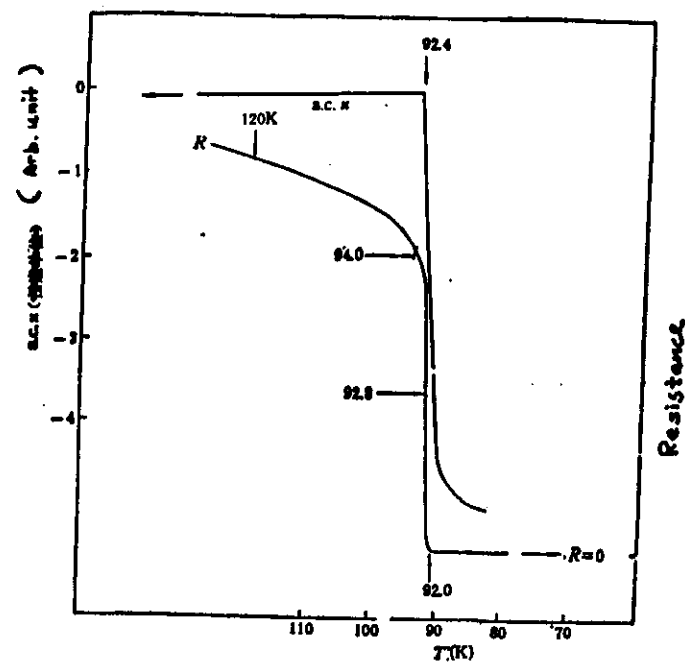
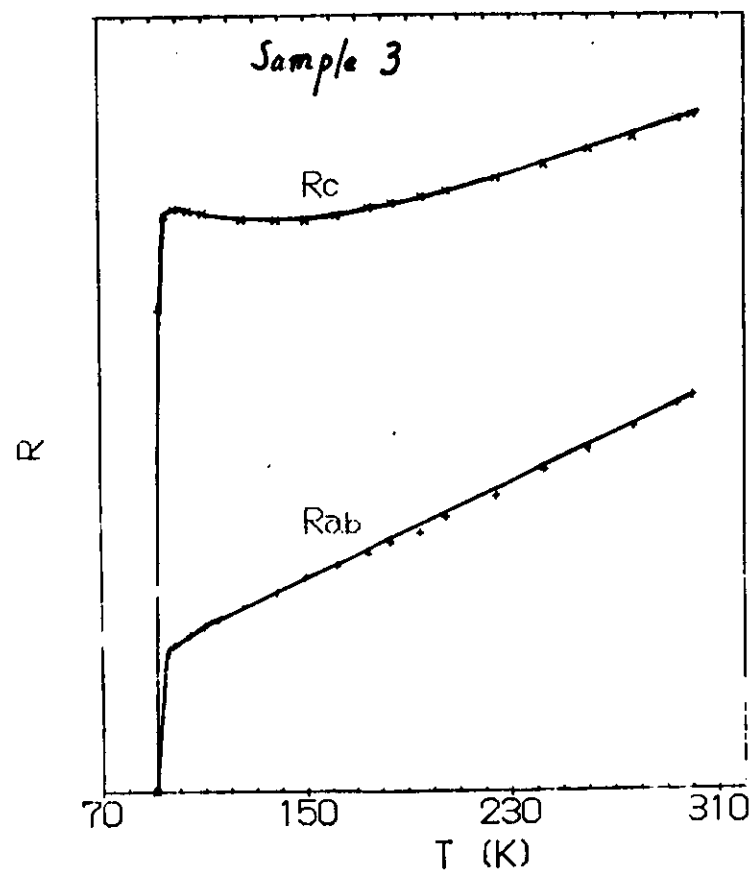


Fig 5.

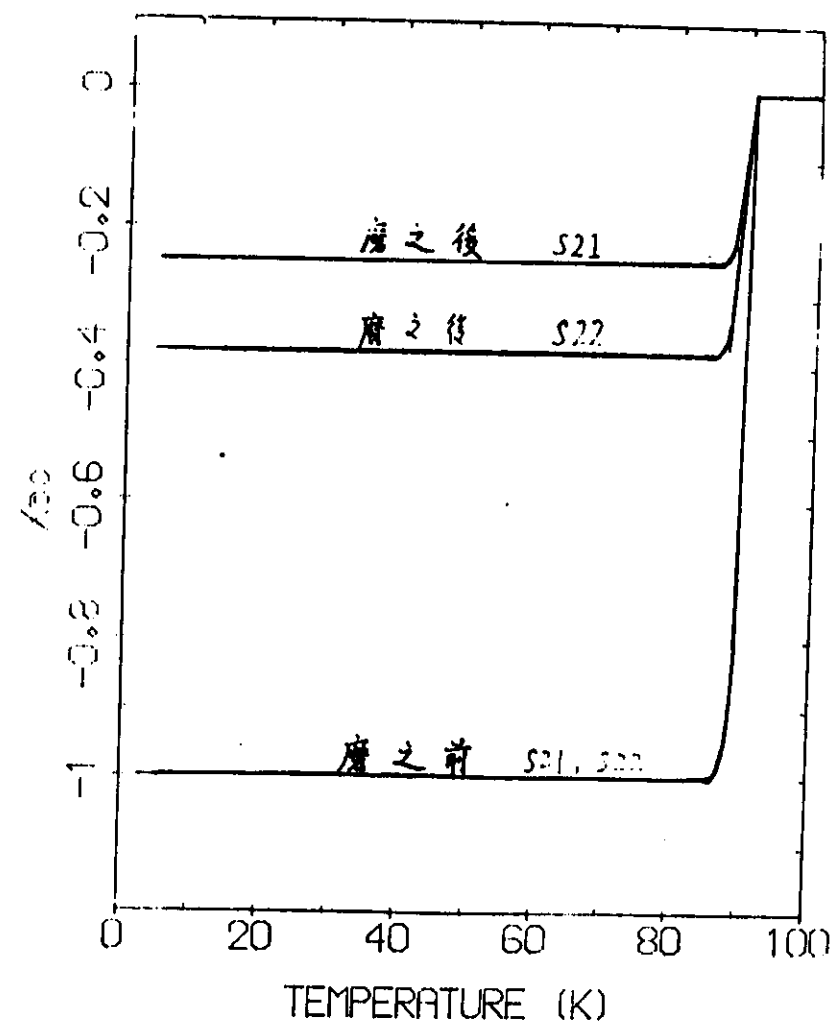
R vs T & A.C. R vs T
for the bulk sample.

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Anisotropic resistivity vs temperature curves of the small single crystal.



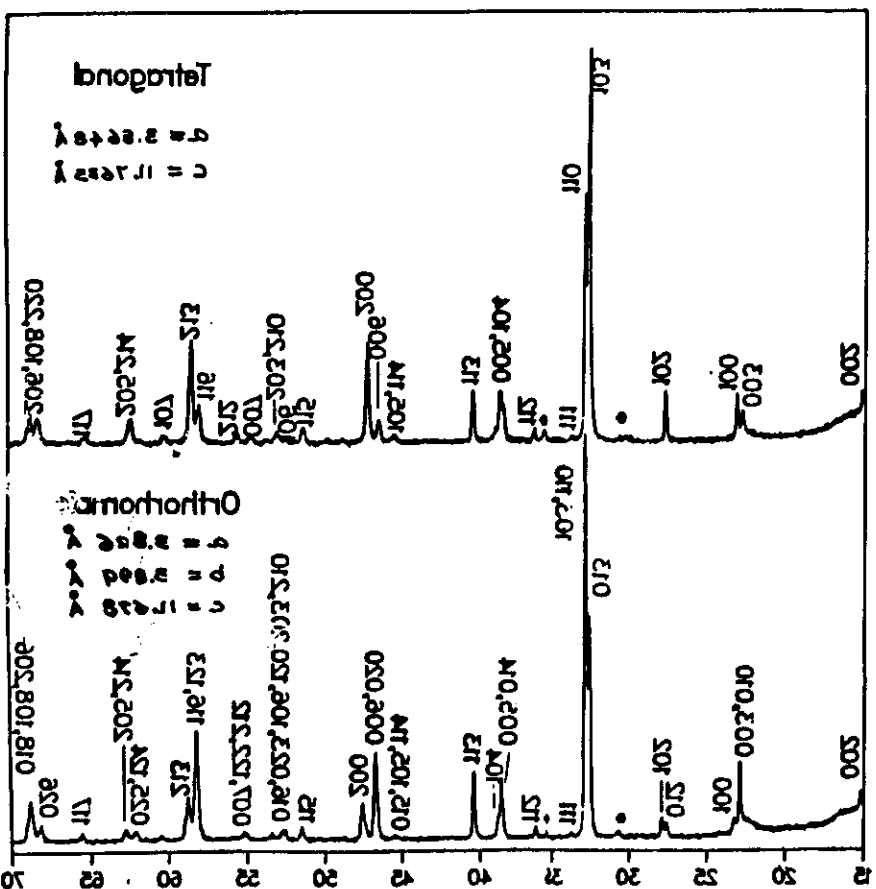
(17)



(18)

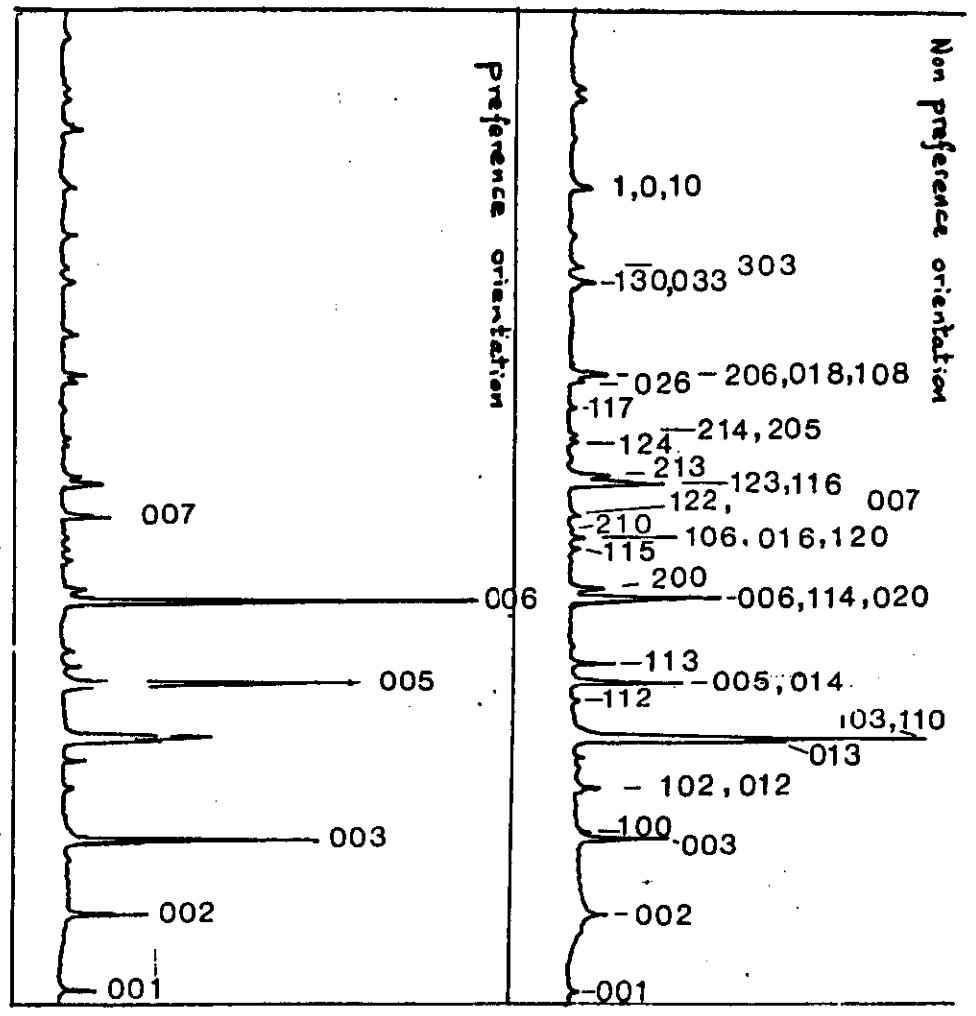
After M. M. A. 7012A

3-LIHT-S



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Institute of Physics unpublished data



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fig 4. d

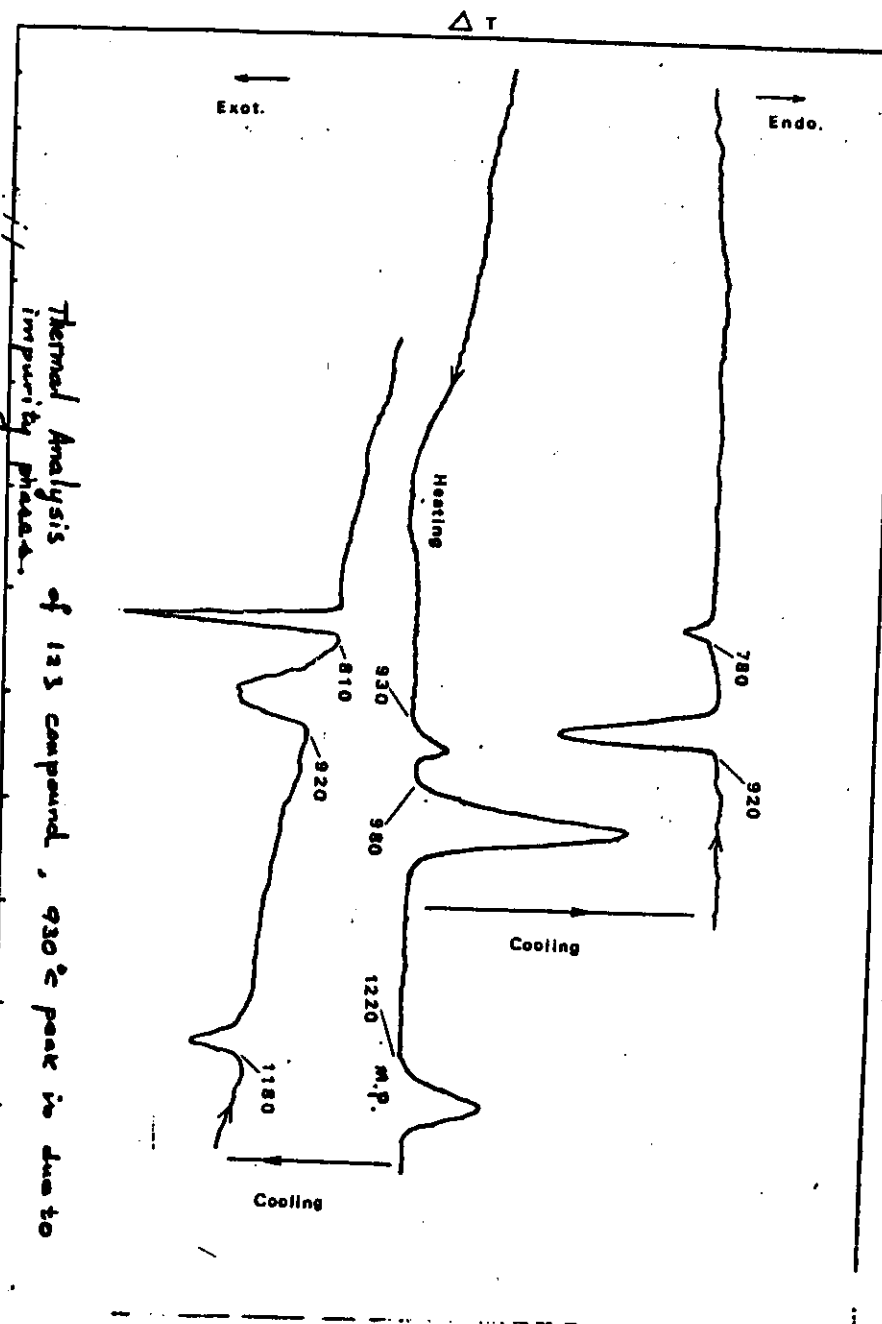
γ $\text{Ba}_2\text{Cu}_3\text{O}_{7-x}$ x-ray data.

h	k	l	$d_{\text{OBS}}(\text{\AA})$	$1/I_0 (\%)$
0	0	2	5.844	2
0	0	3	3.893	11
1	0	0	3.822	3
0	1	2	3.235	3
1	0	2	3.198	5
0	1	3	2.750	60
1	0	3	2.726	100
1	1	0		
1	1	1	2.653	2
1	1	2	2.469	3
0	0	5	2.336	11
1	0	4	2.321	3
1	1	3	2.232	13
0	2	1	1.946	23
0	0	6		
2	0	0	1.911	10
1	1	5	1.775	3
0	1	6	1.741	2
0	2	3		
1	0	6	1.734	2
1	2	0		
2	0	3		
2	1	0	1.716	2
1	2	1		
1	2	2	1.662	1

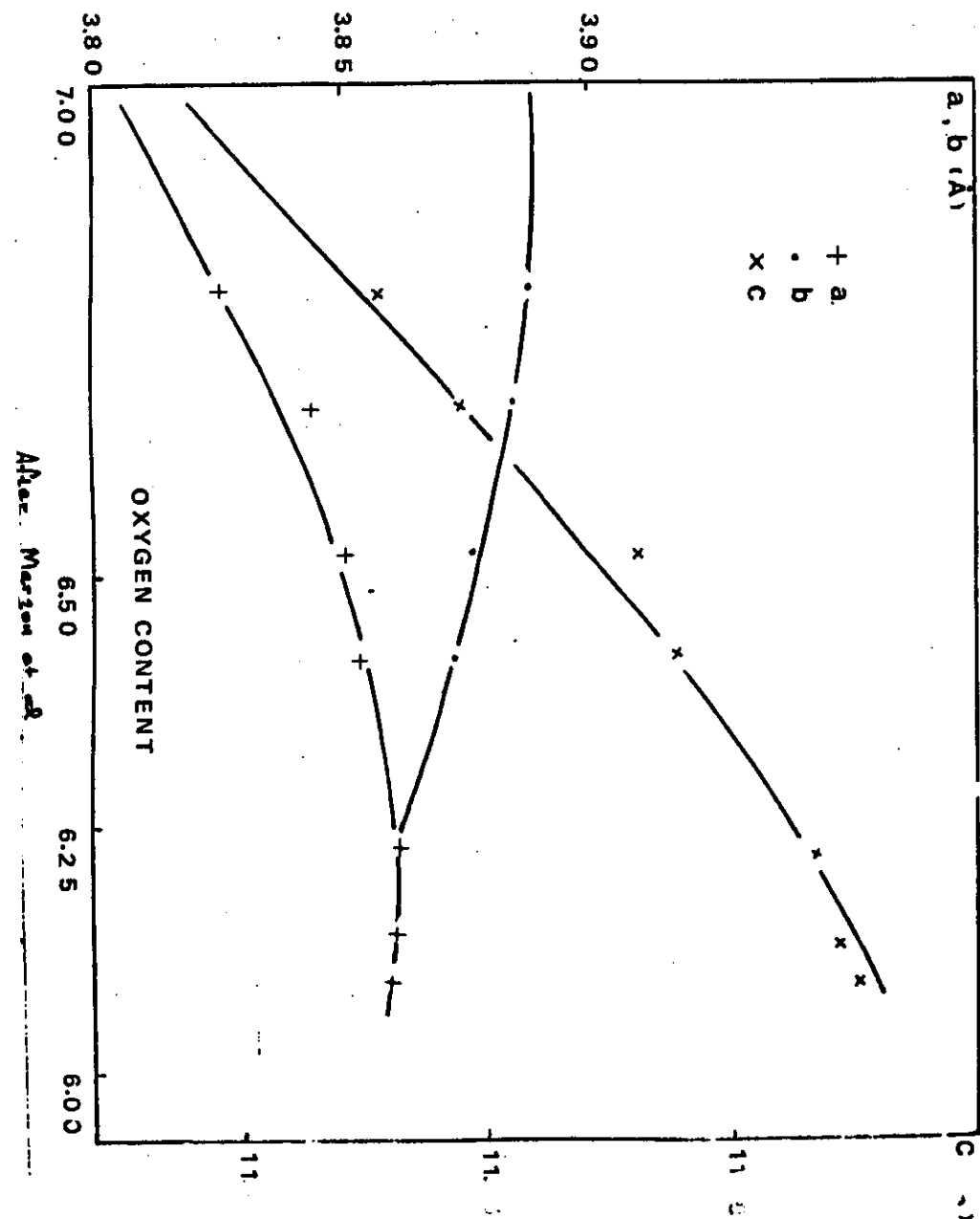
(21)

Thermal Analysis of 123 compound, 930°C peak is due to impurity phase.

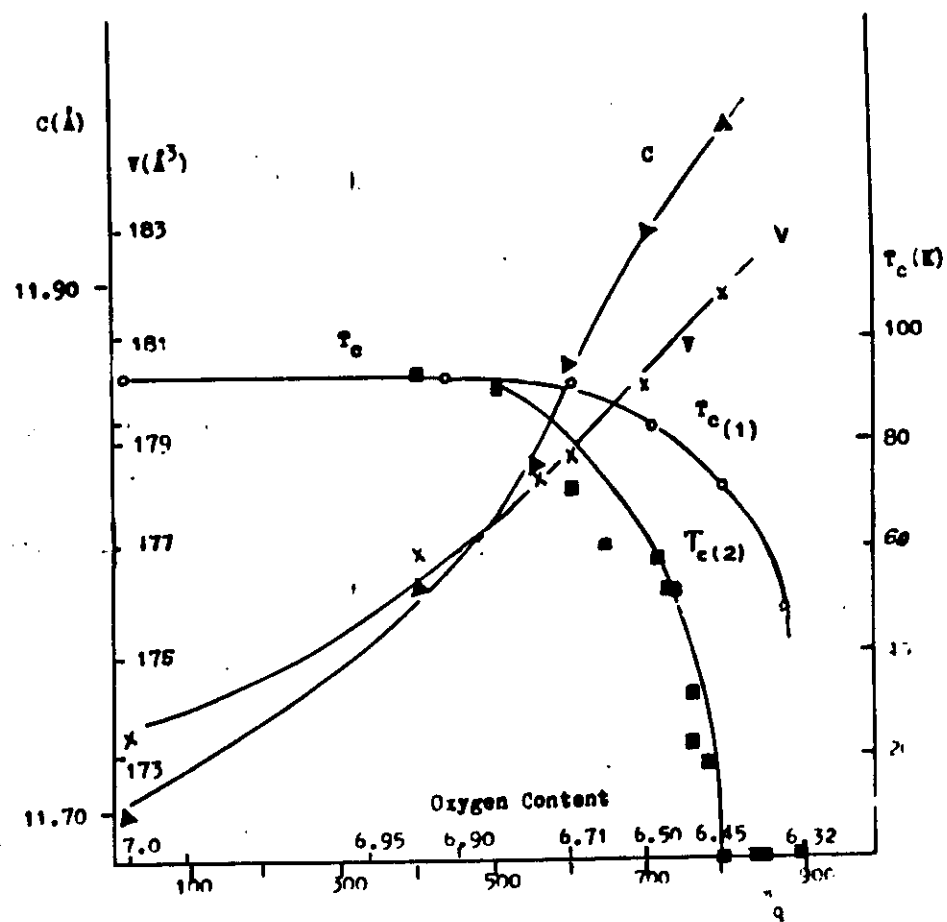
Temperature



(22)

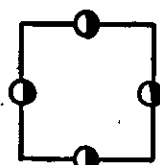


(23)



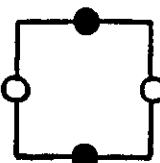
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(24)



Observed Average

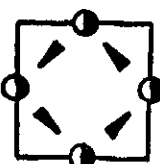
partial occupancy



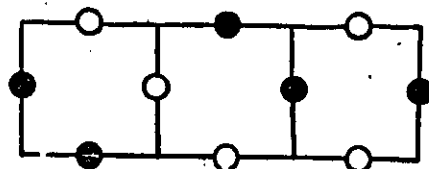
vacancy

Orthorhombic Twins

partial occupancy



Dynamic Disorder



Static Disorder

The models for twin

Table 11: Atomic parameters

		YBa ₂ Cu ₃ O ₆ (this work)	YBa ₂ Cu ₃ O ₆ (Santoro et al.) (14)	YBa ₂ Cu ₃ O ₇ (Beno et al.) (13)	YBa ₂ Cu ₃ O ₇ (Capponi et al.) (14)
		a=3.865(1) Å	a=3.8570(1) Å	a=3.8231(1) Å	a=3.8206(1) Å
		b=3.8663(1) Å	b=3.8663(1) Å	b=3.8663(1) Å	b=3.8651(1) Å
		c=11.852(3) Å	c=11.8194(3) Å	c=11.6809(2) Å	c=11.6757(4) Å
Atom	Parameter	P4/nmm	P4/nmm	Pmmn	Pmmn
Y	x	.5	.5	.5	.5
	y	.5	.5	.5	.5
	z	.5	.5	.5	.5
	u ₁₁	.0053(2)			
	u ₂₂	.0053(2)	u _{iso} =.0092(5)	u _{iso} =.0058(5)	u _{iso} =.0073(14)
	u ₃₃	.0086(3)			
	occ.	1	1	1	1
Ba	x	.5	.5	.5	.5
	y	.5	.5	.5	.5
	z	.1951(2)	.1952(2)	.1843(3)	.1841(4)
	u ₁₁	.0085(1)			
	u ₂₂	.0085(1)	u _{iso} =.0063(5)	u _{iso} =.0068(6)	u _{iso} =.0075(14)
	u ₃₃	.0100(1)			
	occ.	1	1	1	1
Cu(1)	x	0	0	0	0
	y	0	0	0	0
	z	0	0	0	0
	u ₁₁	.0139(4)			
	u ₂₂	.0139(4)	u _{iso} =.0127(5)	u _{iso} =.0063(6)	u _{iso} =.0048(14)
	u ₃₃	.0113(5)			
	occ.	1	1	1	1
Cu(2)	x	0	0	0	0
	y	0	0	0	0
	z	.3609(1)	.3607(1)	.3556(1)	.3549(3)
	u ₁₁	.0043(2)			
	u ₂₂	.0043(2)	u _{iso} =.0062(4)	u _{iso} =.0037(5)	u _{iso} =.0065(9)
	u ₃₃	.0122(4)			
	occ.	1	1	1	1

②

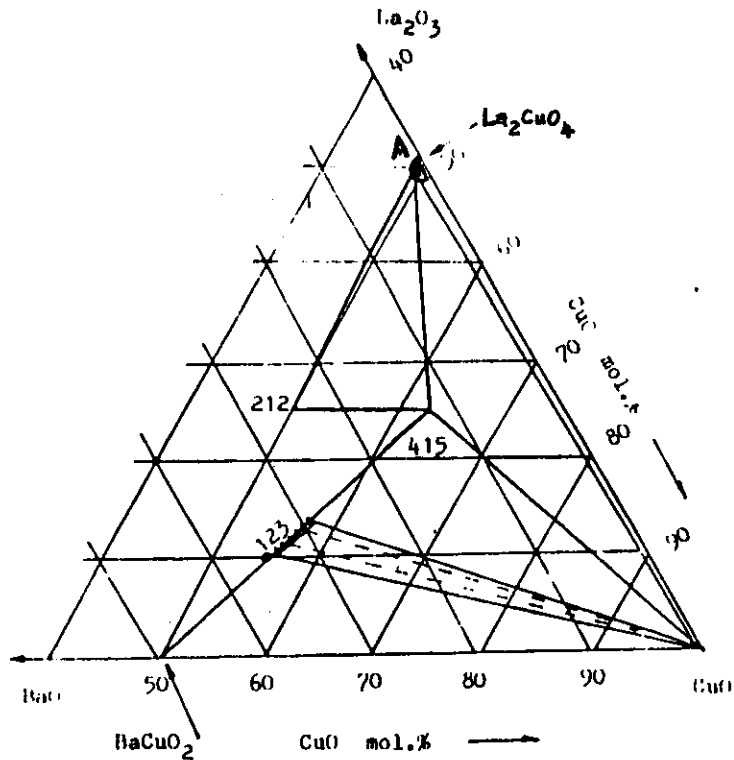
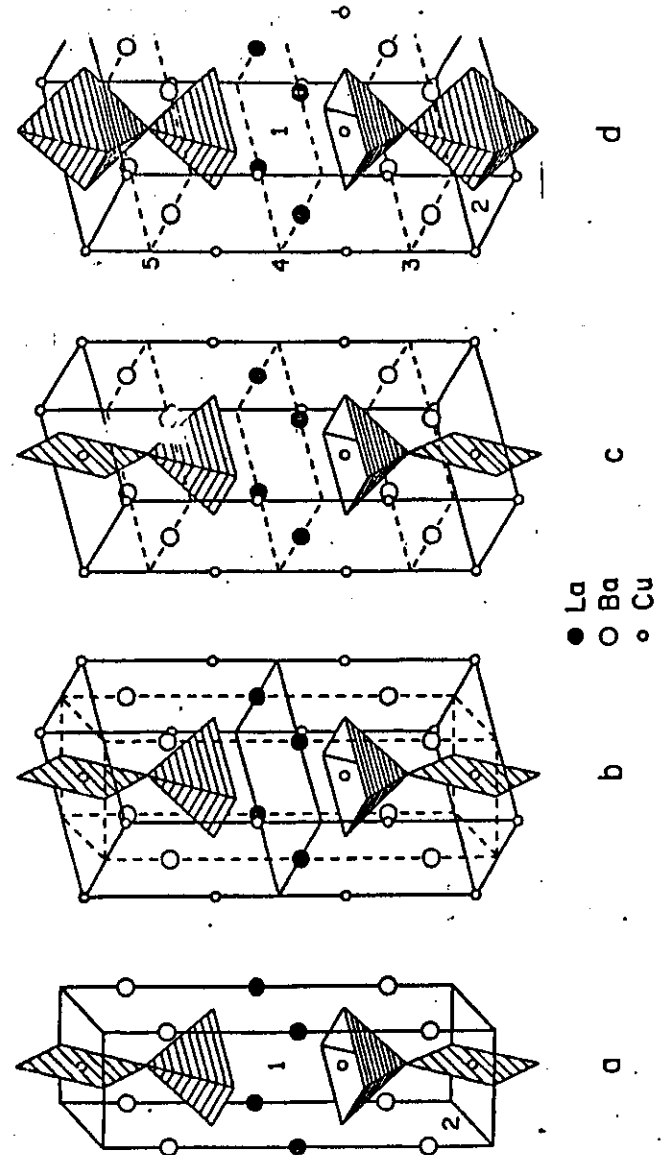


Fig. 2 The phase diagram of La-Ba-Cu-O system in CuO-rich region. 212 and 415 refer to the non-superconducting phase compositions $\text{La}_2\text{BaCu}_2\text{O}_6$ and $\text{La}_4\text{BaCu}_5\text{O}_{14.7}$, respectively. The phase at place A is $\text{La}_{2-x}\text{Ba}_x\text{CuO}_4$ ($0.4 \leq x \leq 0.07$). The hatched line indicates the region of solid solution which has an ideal composition $\text{LaBa}_2\text{Cu}_3\text{O}_y$.

*After Che et al. and Dong et al.
(Inst. of Phys)*

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(a) 123 compound.
(b) $a = \sqrt{2}a_p$



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