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SMR/481 - 5

EXPERIMENTAL WORKSHOP ON
 HIGH TEMPERATURE SUPERCONDUCTORS AND RELATED MATERIALS
 (ADVANCED ACTIVITIES)

(26 November - 14 December 1990)

" Critical currents and their physical problems in high T_c oxide superconductors "

presented by:

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Critical currents and their physical problems
 in high T_c oxide superconductors

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REFERENCES

Introduction

- A.M. Campbell and J.E. Evetts, Adv. Phys. 21, 199 (1972)
 A.I. Larkin and Yu.N. Ovchinnikov, J. Low Temp. Phys. 34, 409 (1979)
 E.H. Brandt and U. Essmann, Phys. Stat. Sol. B 144, 13 (1987)
 J.W. Ekin, K. Salama, V. Selvamanickam, preprint
 C.P. Bean, Phys. Rev. Lett. 8, 250 (1962)
 C.P. Bean, Phys. Rev. Lett. 36, 31 (1964)
 R.W. Rollins, H. Küpfer and W. Gey, J. Appl. Phys. 45, 5392 (1974)

Granularity

- J.R. Clem, Physica C 153-155, 50-55 (1988)
 G. Deutscher and K.A. Müller, Phys. Rev. Lett. 59, 1745 (1987)
 H. Küpfer, I. Apfelstedt, R. Flükiger, C. Keller, R. Meier-Hirmer, B. Runtsch, A. Turowski, U. Wiech and T. Wolf, Cryogenics 28, 650 (1988)
 T. Wolf, W. Goldacker, B. Obst, G. Roth and R. Flükiger, J. Crystal Growth 96, 1010 (1989)
 H. Küpfer, C. Keller, T. Wolf, I. Apfelstedt, R. Meier-Hirmer, U. Wiech and R. Flükiger, Progr. High Temp. Superconductivity 18, 172 (1989)
 M. Däumling, J. Seuntjens and D.C. Larbalestier, Nature 346, 332 (1990)
 K. Salama, V. Selvamanickam, L. Gao and K. Sun, Appl. Phys. Lett. 54, 2352 (1989)
 C.M. Gyorgy, R.B. van Dover, K.A. Jackson, L.F. Schneemeyer and J.V. Waszczak, Appl. Phys. Lett. 55, 283 (1989)

C. Keller, H. K pfer, R. Meier-Hirmer, U. Wiech, V. Selvamanickam and K. Salama, Cryogenics 30, 401 (1990)

M. Murakami, S. Gotoh, N. Koshizuka, T. Tanaka, T. Matsushita, S. Kambe and K. Kitazawa, Cryogenics 30, 390 (1990)

Relaxation

P.W. Anderson and Y.B. Kim, Rev. Mod. Phys. 36, 39 (1964)

M.R. Beasley, R. Labusch and W.W. Webb, Phys. Rev. 181, 682 (1969)

D. Dew-Hughes, Cryogenics 28, 674 (1988)

C.W. Hagen, R.P. Griessen and E. Salomons, Physica C 157, 199 (1989)

C. Keller, H. K pfer, R. Meier-Hirmer, U. Wiech, V. Selvamanickam and K. Salama, Cryogenics 30, 410 (1990)

V.V. Moshchalkov, A.A. Gippius, A.A. Zhukov, H.H. Nyan, V.I. Voronkova and V.K. Yanovskii, Physica C 165, 62 (1990)

T. Fujiyoshi, K. Toko, T. Matsushita and K. Yamafuji, Jpn. J. Appl. Phys. 28, L 1906 (1989)

Y. Xu, M. Suenaga, A.R. Moodenbaugh, D.O. Welch, Phys. Rev. B40, 10882-10890 (1989)

M.P. Maley et. al., Phys. Rev. B42, 2639-2642 (1990)

D.O. Welch, to be published in IEEE Trans. on Magnetics (1990)

J.Z. Sun, C.B. Eom, B. Lairson, J.C. Bravman and T.H. Geballe, to be published in Phys. Rev. Lett. (1990)

C.W. Hagen and R.P. Griessen, Phys. Rev. Lett. 62, 2857 (1989)

A. Gurevich, H. K pfer and C. Keller, to be published

high field high current application:

$$10^5 \text{ A/cm}^2 \quad (B \leq 10 \text{ T}, 77 \text{ K}) \quad \rho < 10^{-7} \mu\Omega \text{ cm}$$

Introduction

critical currents in Type II supercond.
measurement techniques of j_c

1). granularity

detection of weak supercond. coupling
in a ceramic (intergrain)
in a single crystal (intragrain)

2). low pinning energy

problems of flux pinning
investigation in High T_c s.c.

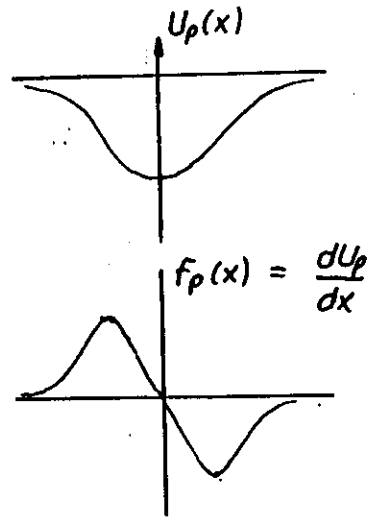
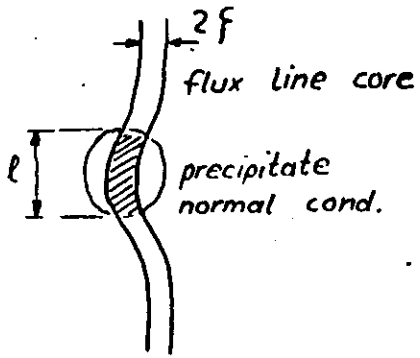
3). thermal relaxation

time dependent decay of j_c
 B and T dependence of activation energy

4). reversible region between B_{irr} and B_{c2}

measurement of B_{c2} and B_{irr}
correlation between defect structure and B_{irr}

Pinning potential $U_p \approx \frac{1}{2} \mu_0 H_c^- \bar{l} f' l$



elementary pinning force f_p
 $f_p^{\max} \quad 10^{-10} \div 10^{-15} \text{ N}$

summation of f_p

- interaction energy between flux lines $U_{ij} \ll U_p$

$F_p = N f_p$

direct summation ("strong pins")

N : defect density

- $U_{ij} > U_p$

$F_p = \sqrt{\frac{N f_p^2}{V}} \sim N^2 f_p^4$

collective pinning ("weak pins")

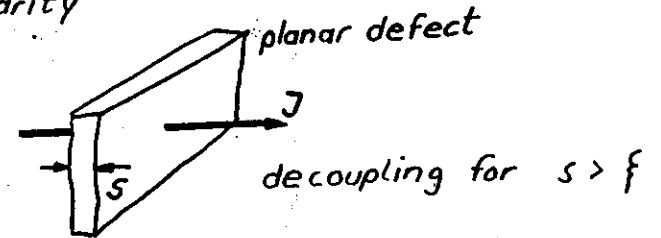
V : correlation volume

Larkin and Ovchinnikov

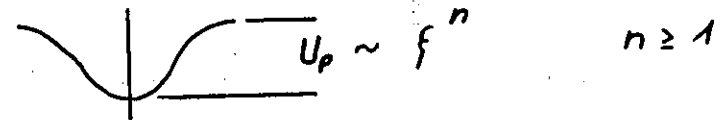
| | low T_c | high T_c |
|-----------------|-----------|------------|
| ξ | 50 Å | 5 Å |
| operation temp. | 4 K | 77 K |

defect dimension similar

- 1) granularity



- 2) low pinning energy



- 3) thermal activation

$U_0(U_p)/k_B T (77 \text{ K}) \leq 10$

$U_0(U_p)/k_B T (4 \text{ K}) \geq 100$

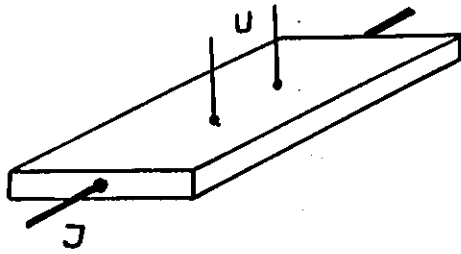
- 4) reversible region between B_{irr} and B_{c2}

$U_0(U_p)/k_B T (B_{irr}) \approx 1$

$J_c = 0$ at $B_{irr} < B < B_{c2}$

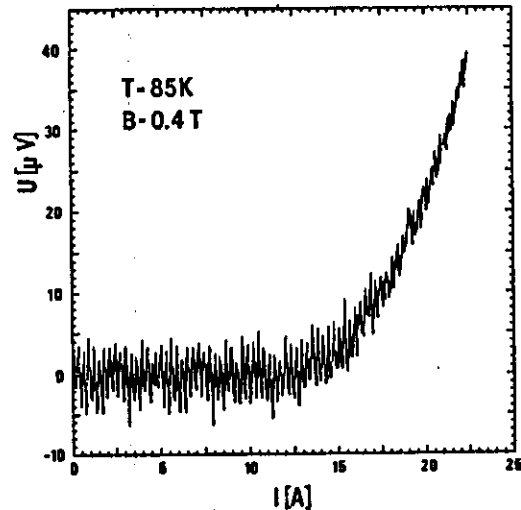
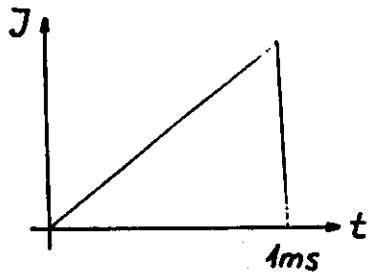
measurement techniques for j_c

1) resistive dc : transport j_c

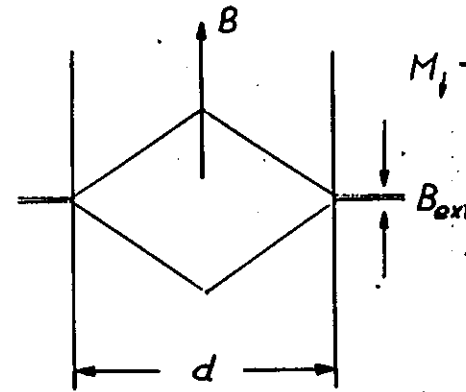


- $E \geq 0.1 \mu\text{V/cm}$
short sample length
- both cooled (4K or 77K)
high contact resistance J. Ekin

bulk specimen : current pulse



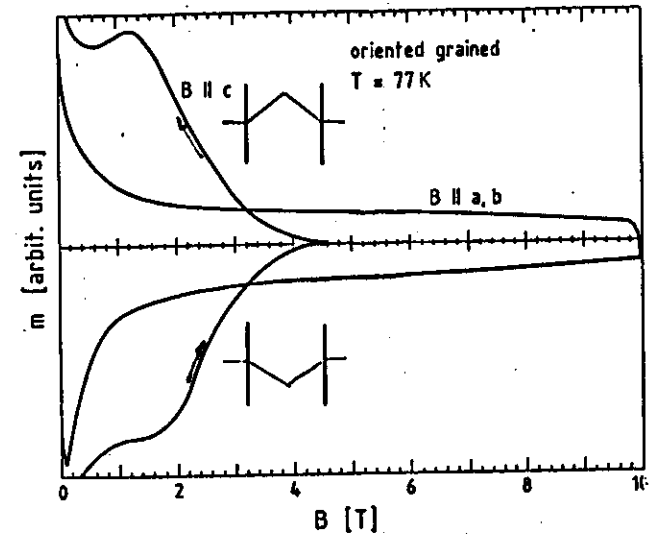
2) dc magnetization : shielding j_c



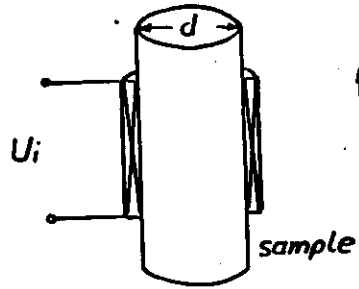
$$M_i - M_f = \Delta M = \frac{1}{3} j_c d$$

c.P. Bean

- $E \approx \frac{d}{2} \frac{dB}{dt} \geq 10^{-4} \mu\text{V/cm}$
- Temperature variabel
- scaling length d
- flux jumps at low T

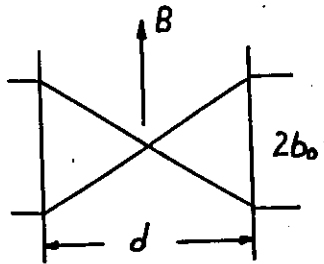


inductive measurements : shielding j_c



$$B_{dc} + b_0 \sin \omega t$$

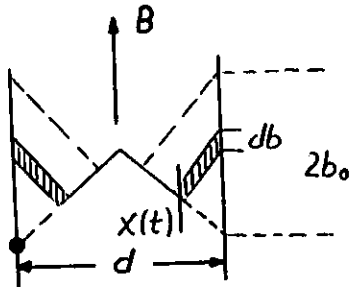
3) ac susceptibility



$$\chi'' \sim \int_{-T/4}^{T/4} U_i dt$$

maximum of χ'' at $j_c = 2b_0 / \mu_0 d$

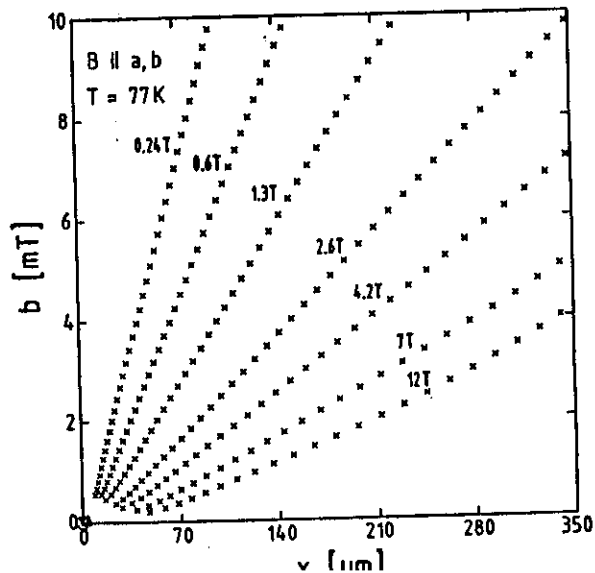
4) flux profile



$$U_i \sim \frac{d\phi}{dt} \sim \frac{db}{dt} x(t)$$

$$j_c = \frac{1}{\mu_0} \frac{db}{dx}$$

Rollins et al.



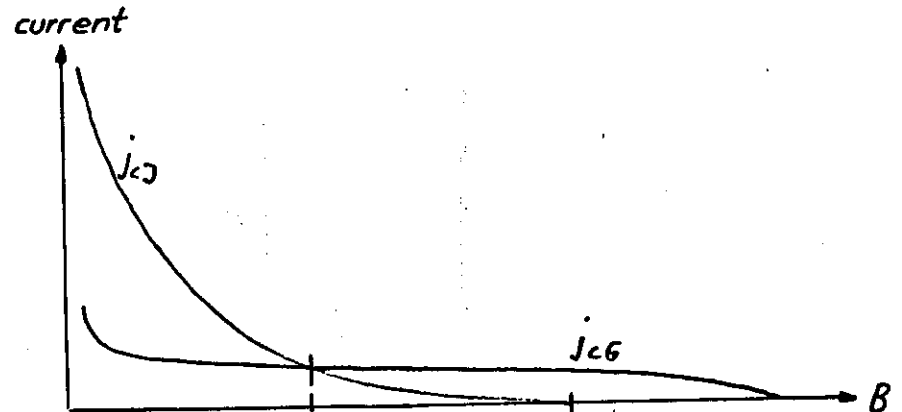
decoupling of the superconducting wave function at planar defects

J.R. Clem
Deutscher and Müller

two current systems :

intragrain (bulk) current (pinning) j_{cG}

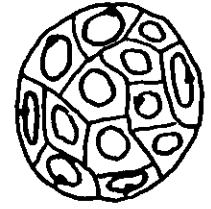
intergrain (weak link) " (coupling energy) j_{cJ}



not granular



granular



$$j_{cT} = j_{cS} = j_{cG} \quad j_{cT} = j_{cJ} < j_{cS} \quad j_{cT} = 0$$

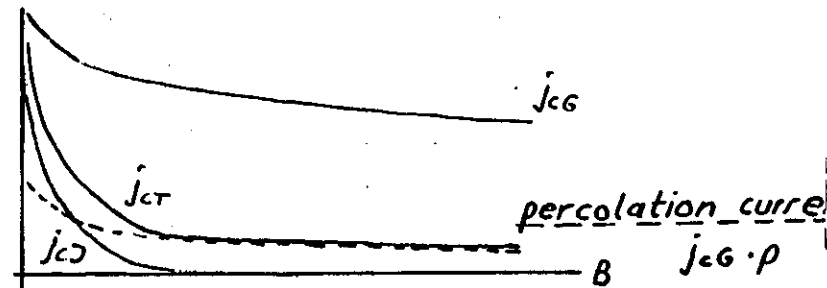
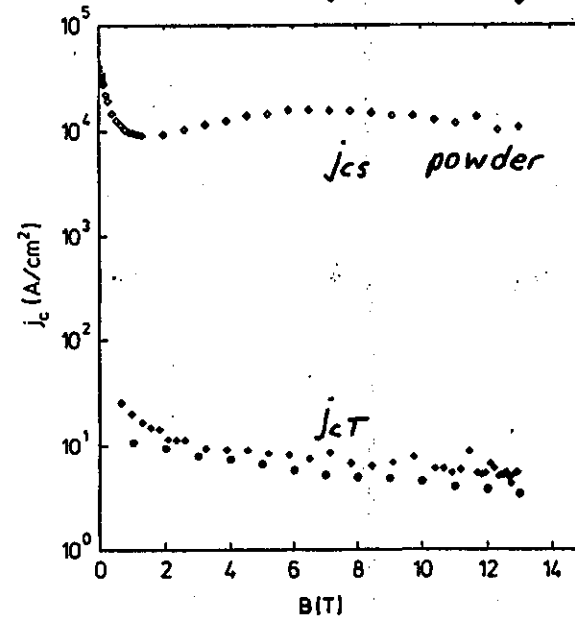
$$j_{cS} = j_{cG}$$

detection of granularity (two current systems)

polycrystalline $YBa_2Cu_3O_{7-x}$

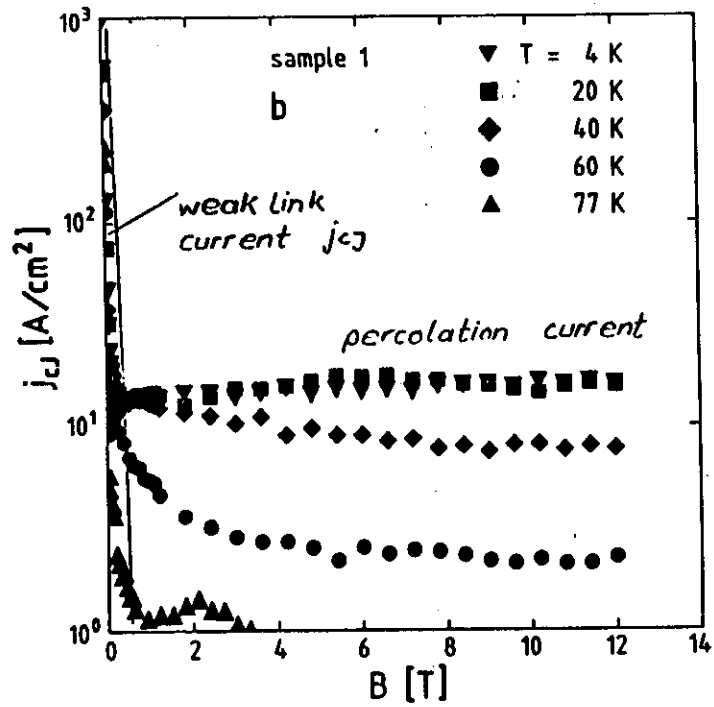
- transport {
 - 1) B and T dependence of j_{CT}
 - 2) $j_{CT}(B)$ history
- dc shielding {
 - 3) comparison between j_{CT} and j_{CS}
 - 4) magnetization at $B < B_{c2}$
 - 5) j_{CS} (particle size)
 - 6) comparison between $j_{CS}(\Delta m)$ and $j_{CS}(\Delta B)$
- ac shielding {
 - 7) "fish tail" in $j_{CS}(B)$
 - 8) ac susceptibility
 - 9) flux profile
- 10) HF penetration depth
- intragrain weak links

comparison between j_{CT} and j_{CS}



shielding current measurement requires d , but $d(B, T)$ due to distribution of weak link properties

intergrain critical current density j_{cT} at high fields



weak link character ($j_{c3} \sim B^{-1}$) vanishes at $B \geq 0.5T$

percolative current via superconduct. grain bound.

- no correlation of j_{cT} between low and high fields
- similar field dependence of j_{cT} and $j_{cS} = j_{c0}$
- $j_{cT}(\vec{B} \perp \vec{j}) \approx j_{cT}(\vec{B} \parallel \vec{j})$
- ac susceptibility: grain specific at high fields

YBa2Cu3O6.9 polycrystalline

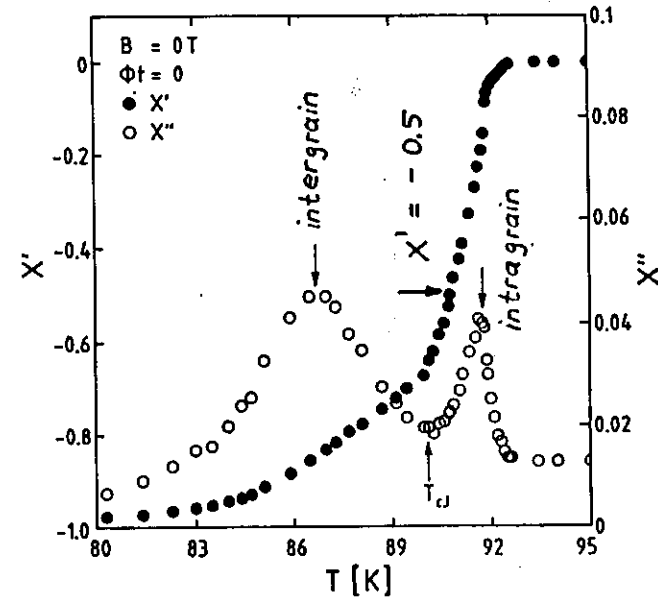
density = 95 %

grain size = $25 \mu\text{m}$ twin spacing = $0.12 \mu\text{m}$

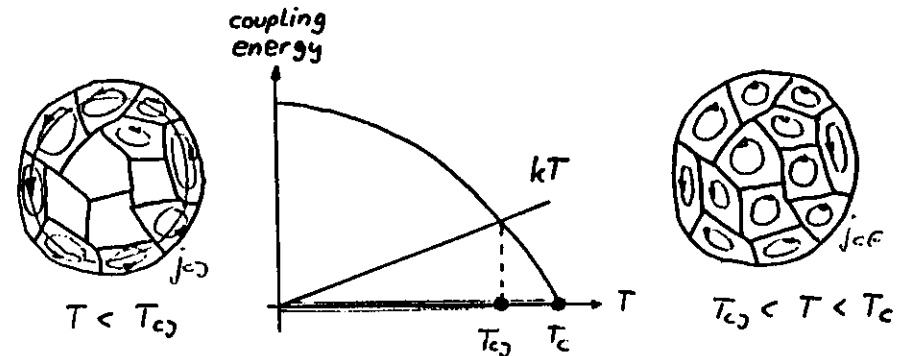
$\rho(100\text{K}) = 200 \mu\Omega\text{cm}$ $\rho(300\text{K}) = 600 \mu\Omega\text{cm}$

ac susceptibility

11 Hz, 1 Oe

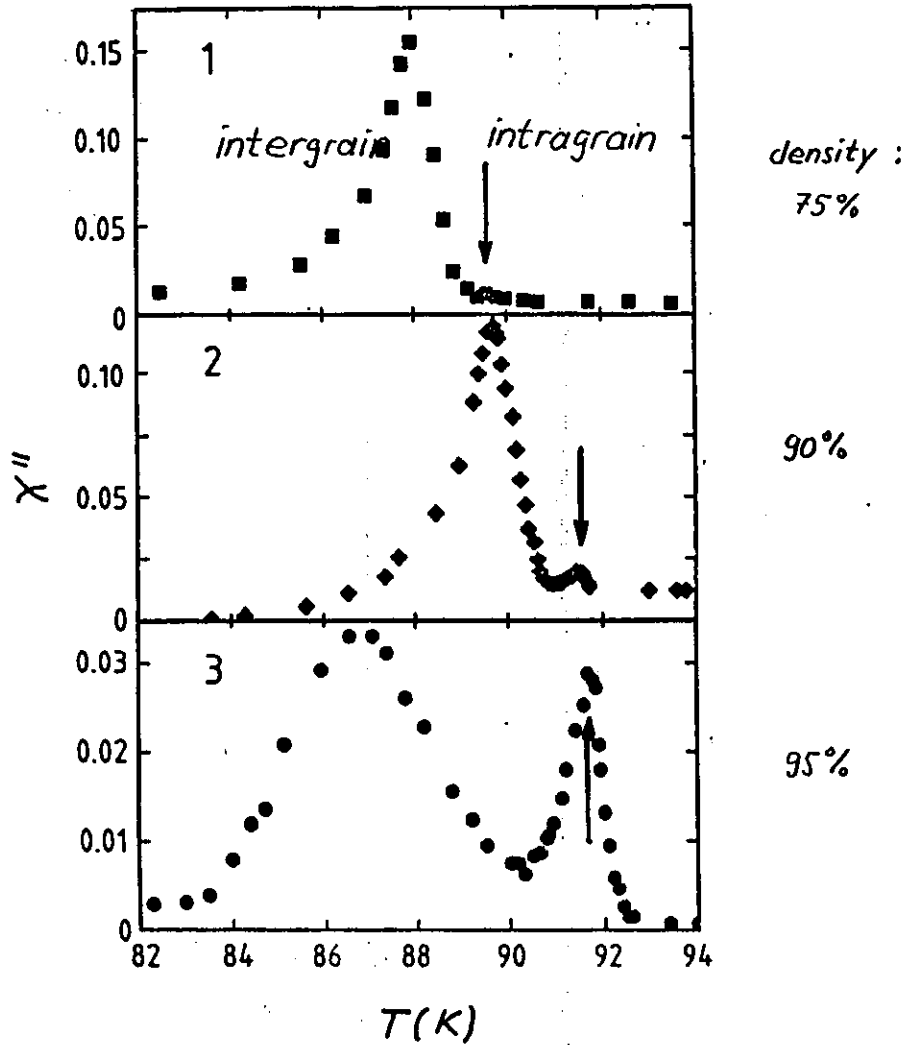


superconducting grains weakly coupled at boundaries

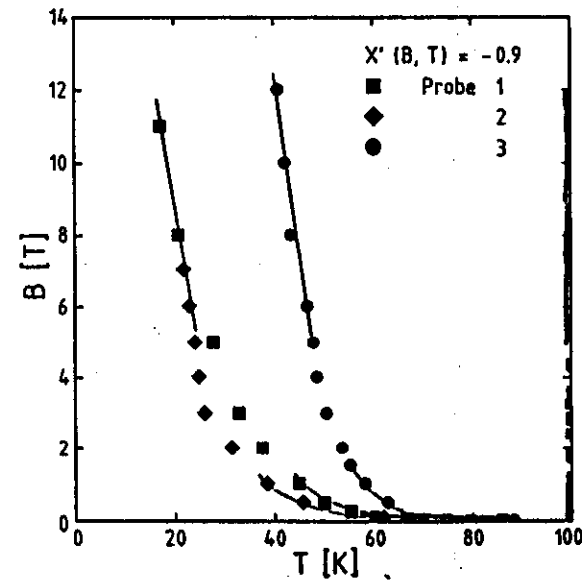
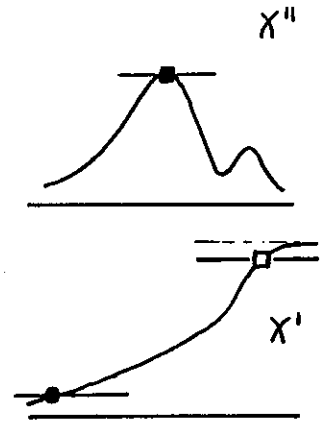
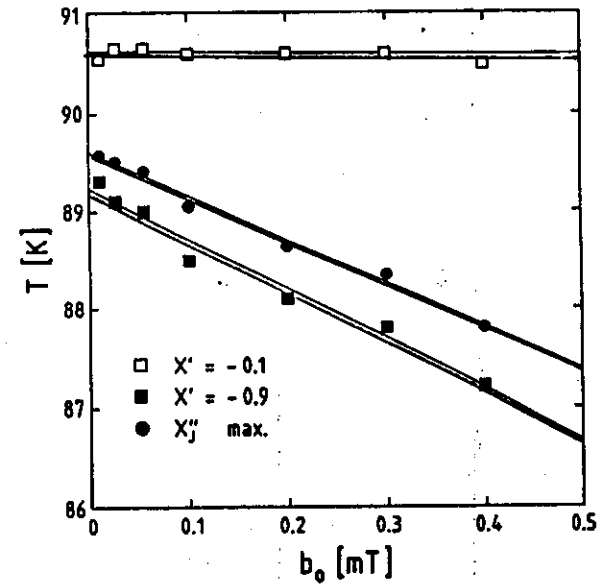


"double peak" structure depends on :
 difference between j_{c6} and j_{c7} , density, ...

variation of b_0, B_{dc}



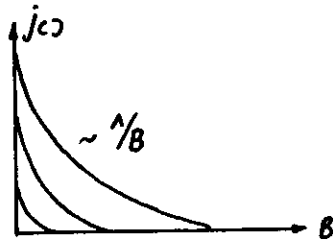
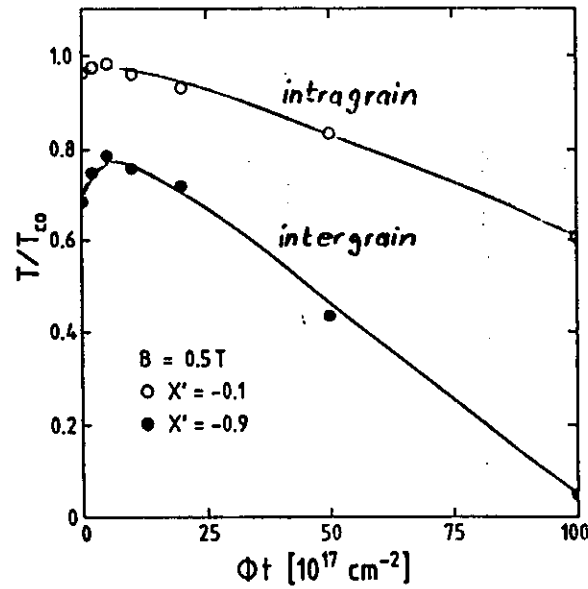
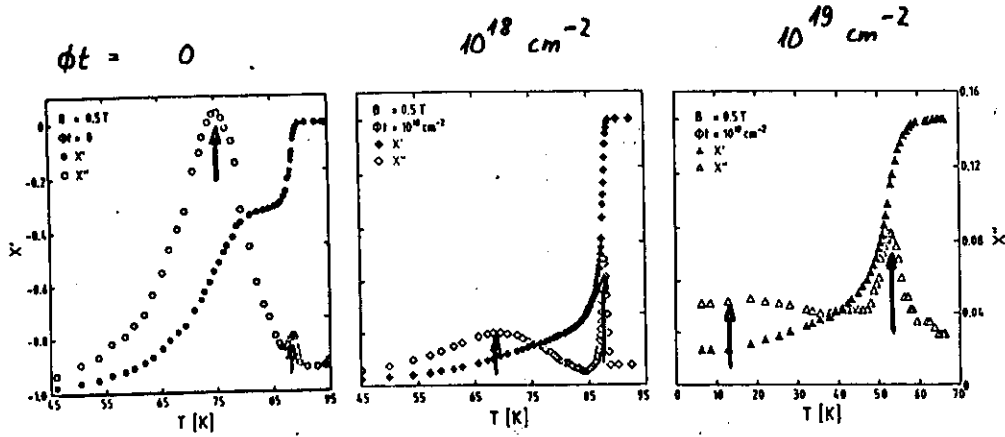
high density, second phase content small.
 ("intergrain volume" small) \Rightarrow
 only intragrain peak detectable



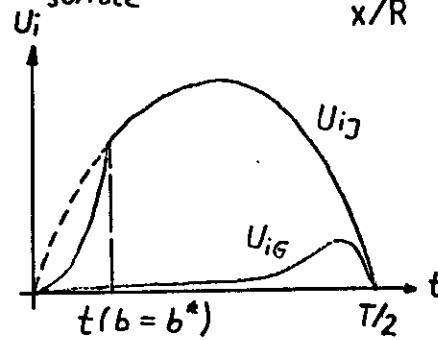
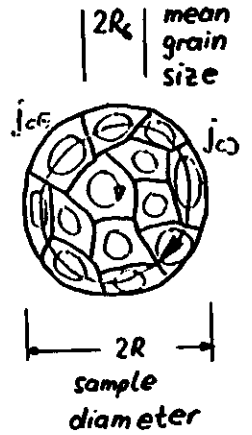
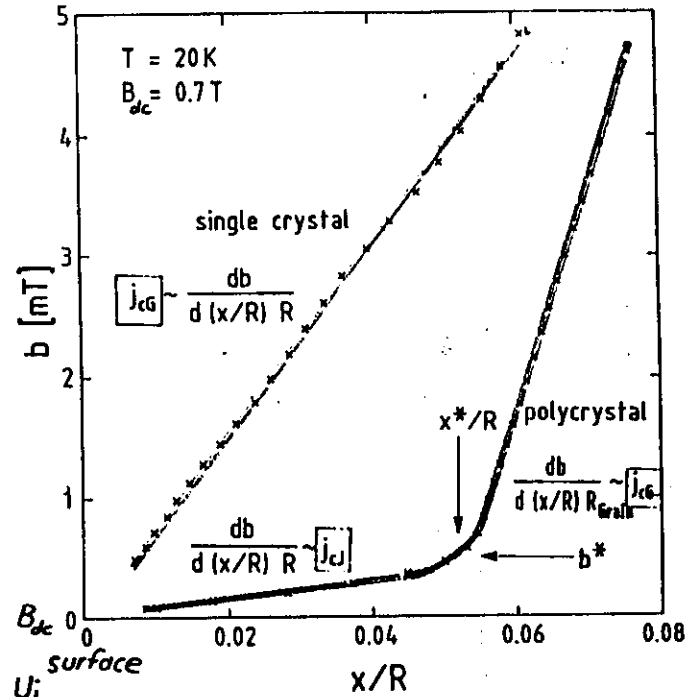
weak link
 character of
 $X'(B, T)$ vanishes
 at higher fields

grain specific
 $\frac{dB_{irr.}(T)}{dT}$

flux profile measurement of inter- and intragrain
current density



distribution of \bar{j}_c becomes pronounced in applied B



$$U_i = U_{iJ} + U_{iG}$$

$$j_{cG} R_g > j_{c0} R$$

radiation sensitivity of junction properties is larger than of grain properties especially in $B \neq 0$

$$b < b^*$$

$$U_i \approx U_{iJ} \gg U_{iG}$$

$$j_{c0} = \frac{db}{d(x/R)R} \frac{A_2}{A}$$

$$b > b^*$$

$$U_{iG} = U_i - U_i^{NC} A_2/A$$

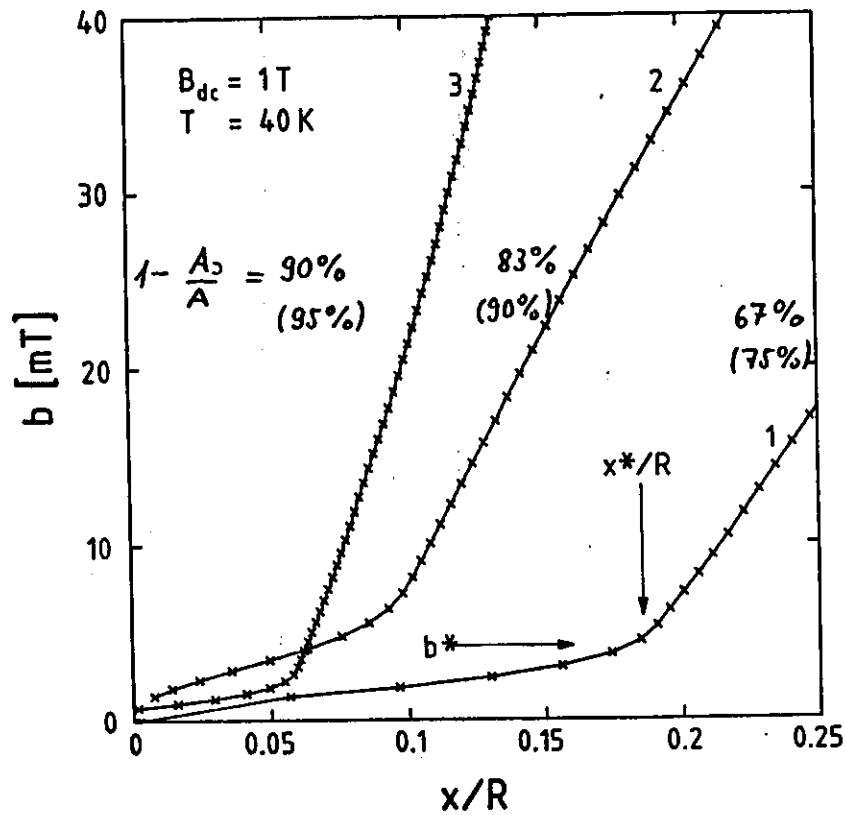
$$j_{cG} = \frac{db}{d(x/R)R_g}$$

intergrain volume V_3

from flux profile measurement:

$$A_j = A - \pi(R - x^*)^2$$

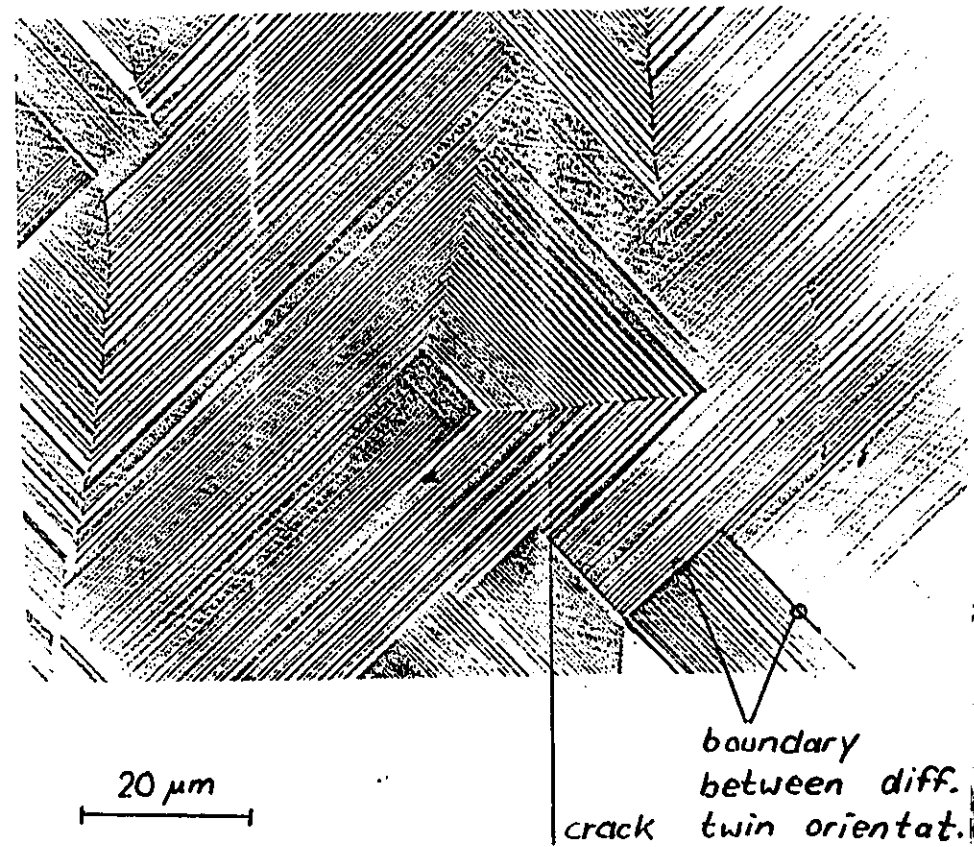
$$\frac{A_j}{A} = \frac{x^*}{R} \left(2 - \frac{x^*}{R} \right) = \frac{V_3}{V}$$



YBa₂Cu₃O_{7-x} single crystal

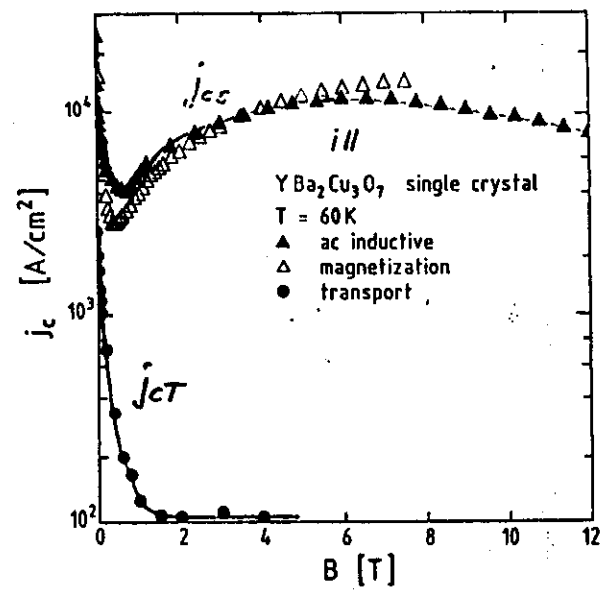
possible intragrain weak links:

- twin boundaries
- stacking faults
- region of different twin orientation
- oxygen deficient regions
- microcracks



$YBa_2Cu_3O_x$ single crystal (grown in Al_2O_3 crucible)

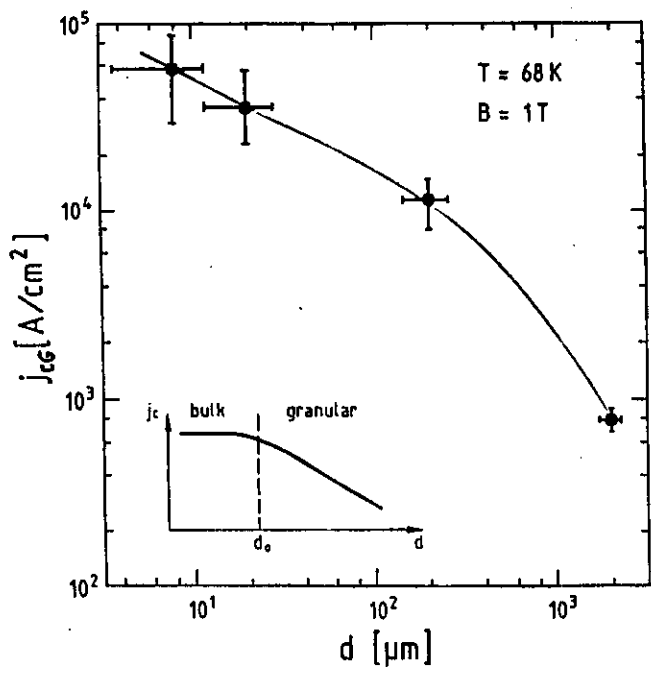
$T_c = 84 K$ $\rho(100K) = 0.15 m\Omega cm$ $RRR = 3.5$
 Wolf et al.



transport current <<
shielding current

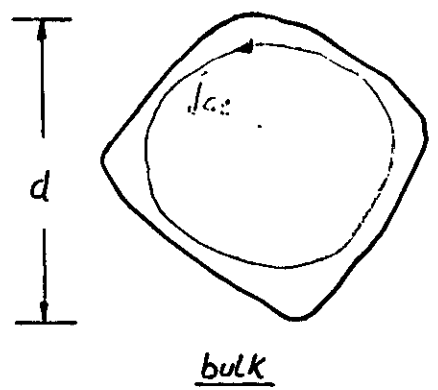
$$\frac{j_{cT}}{j_{cs}} (B=0) = 10^{-1}$$

$$\frac{j_{cT}}{j_{cs}} (B > 1T) = 10^{-2}$$



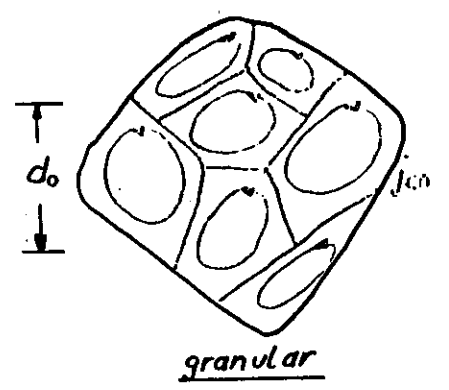
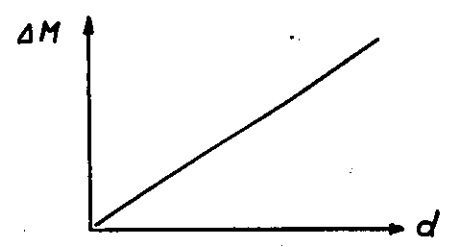
increase of the
 shielding current
 with decreasing
 powder particle sized

intragrain junctions
 determine the
 transport current

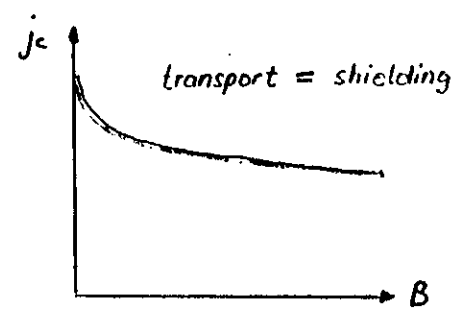
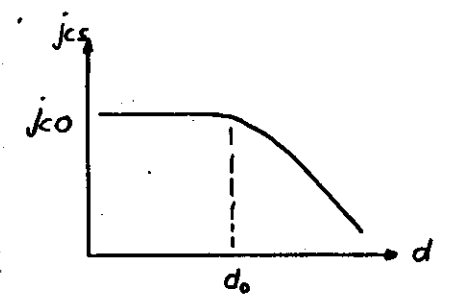
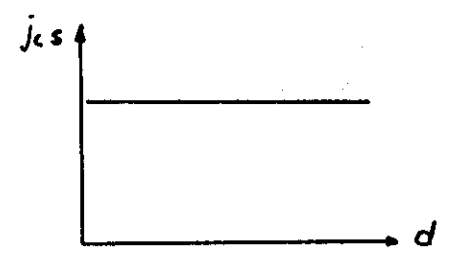
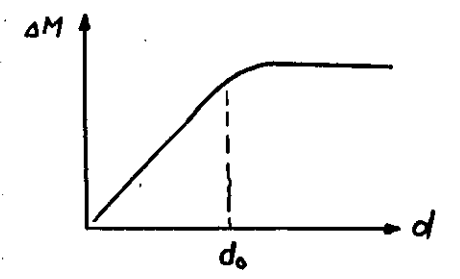


bulk

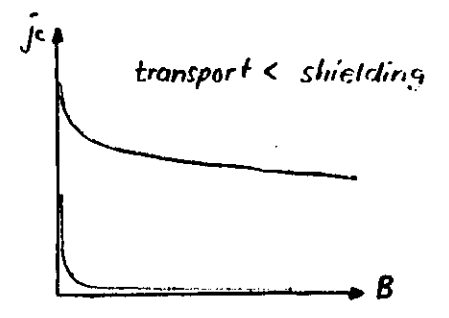
$$\Delta M \sim j_{cs} d$$



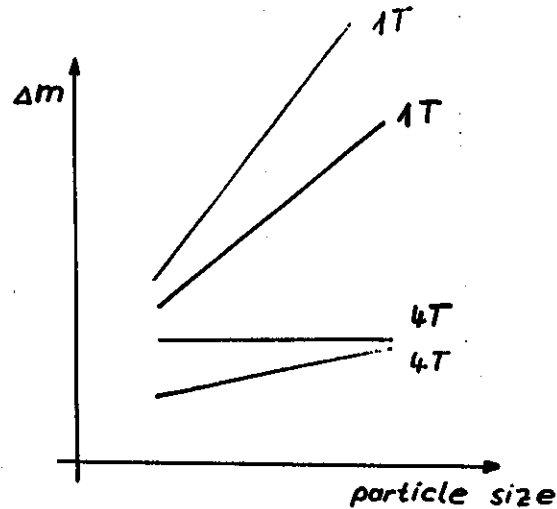
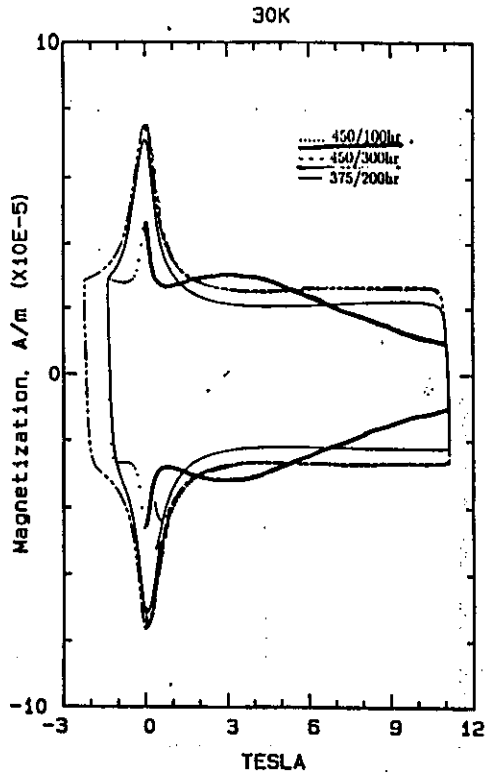
granular



transport = shielding



transport < shielding



$\delta = 0.05$

oxygen deficient region with lower T_c , B_{c2} determine granularity and pinning

with increasing oxygen content granularity vanishes and j_{cs} decreases

Δm does not extrapolate to zero: even at 0T

$j_{cs} > j_{ct}$

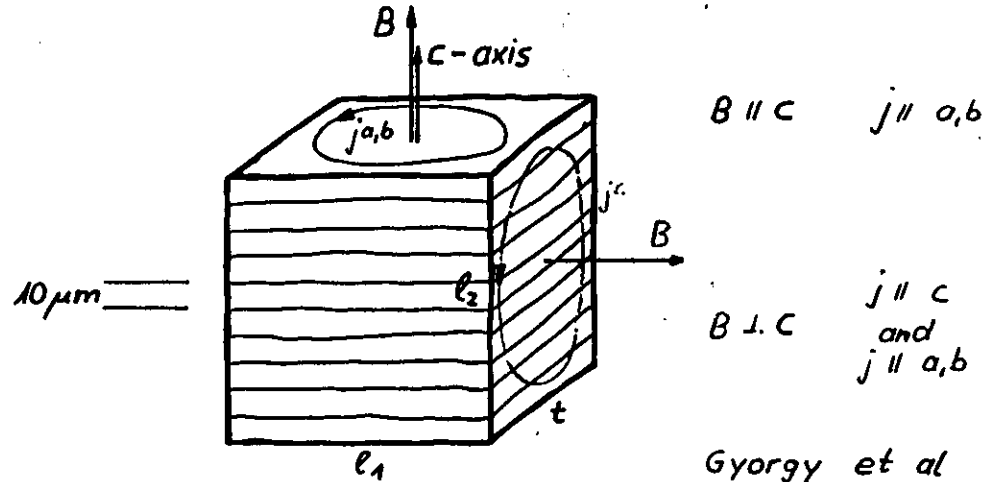
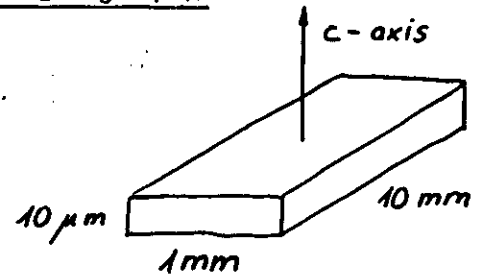
oriented . grained $YBa_2Cu_3O_{7-x}$

platelike grains are c-axis oriented

$\approx 20\%$ 211 phase size of precip. $\geq 5\mu m$

twin spacing $\approx 0.1\mu m$

preparation : Salama et al.



$l_1 \approx l_2 \approx 3.6 \text{ mm}$

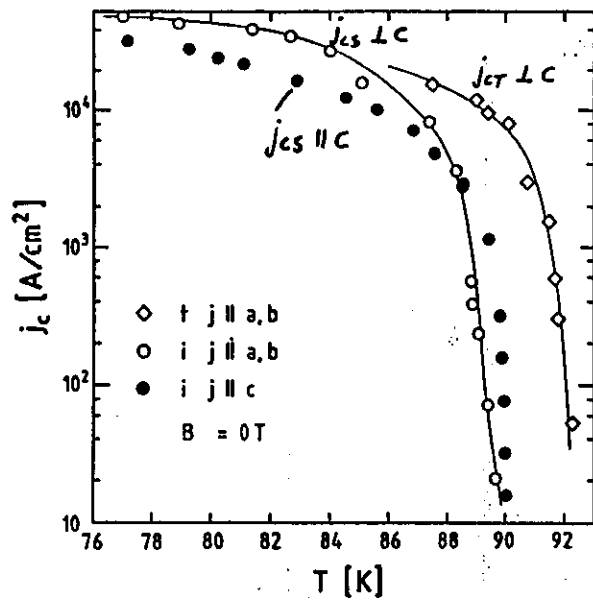
$t \approx 0.8 \text{ mm}$

$\Delta m_{B \perp c} \approx \frac{j_{ct}}{2} \left(1 - \frac{t j_{cs}}{3 l_2 j_{a,b}} \right)$

in both geometries { same demagnetization
 same aspect ratio t/l

Gyorgy et al

comparison between j_{cT} and j_{cs} in $B \perp a,b$

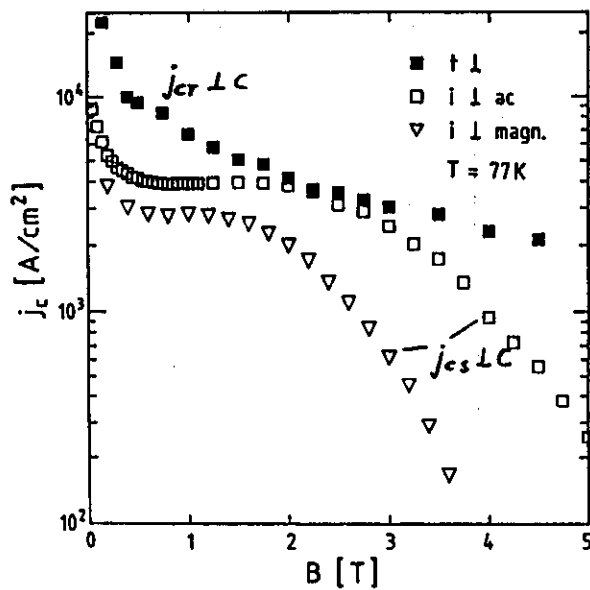


$j_{cT} \approx j_{cs}$

different samples

" criteria

" misalign-
ment between
 B and c -axis



no intragrain
weak links in
 $j_{|| a,b}$

Murakami et al

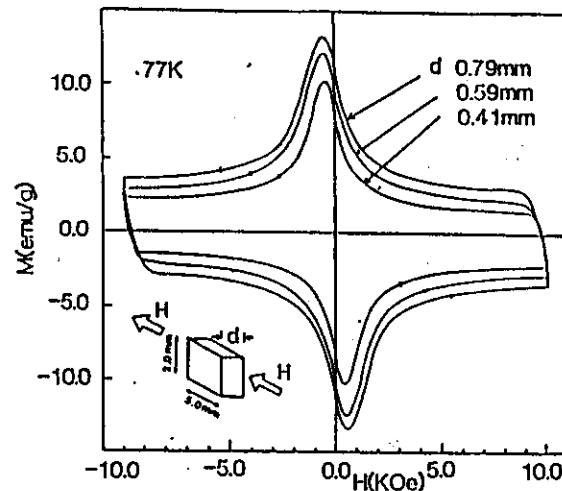


Fig. 4. Magnetization curves for YBaCuO quenched and melt processed. These data were taken as reducing the sample size.

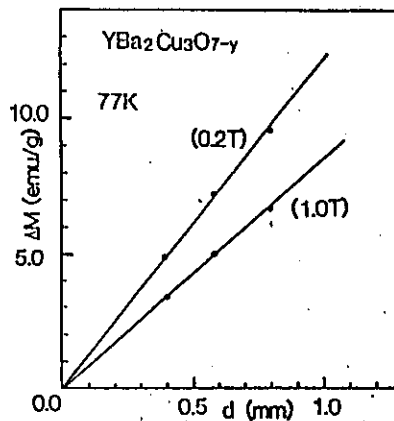


Fig. 5. Relationships between ΔM and d .

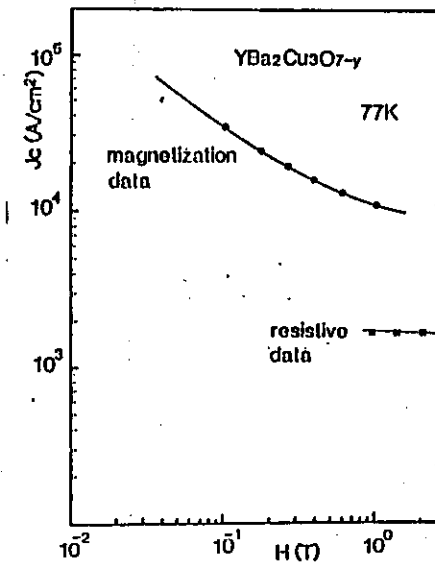
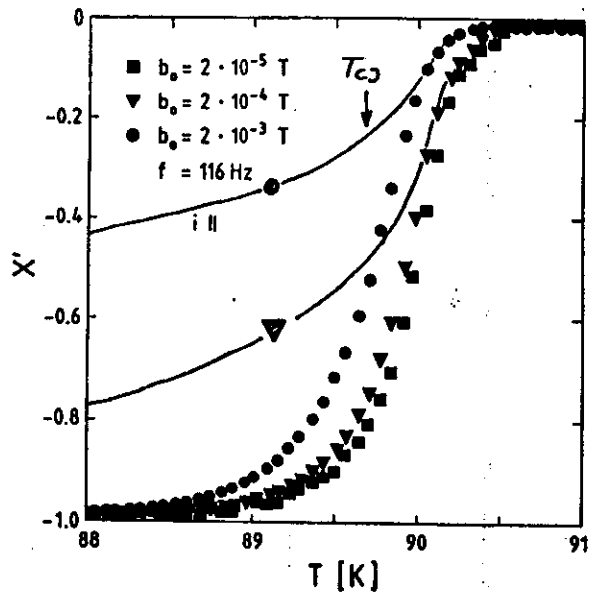


Fig. 6. Magnetic field dependence of critical current density.

Inter - and intragrain junctions ?

shielding current passes grain boundaries: $B \perp C$



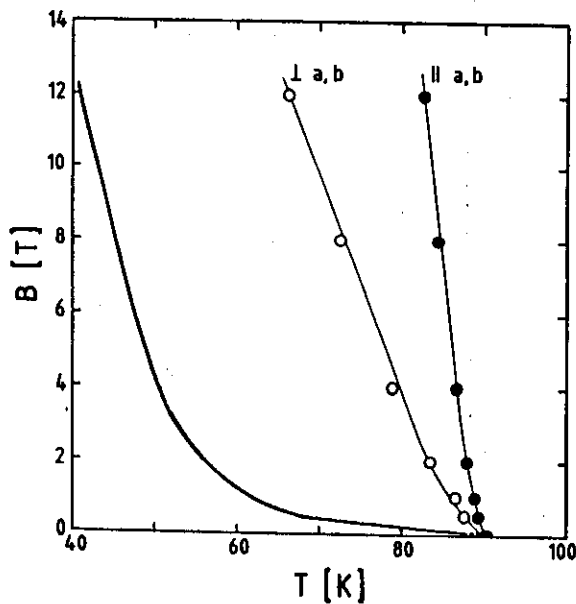
$$\frac{dT}{db_0} \left[\frac{K}{mT} \right]$$

granular

- 10

oriented grained

- 0.2



$$X' = -0.5$$

11.6 Hz

10^{-4} T

○ $j \parallel a,b$

● $j \parallel c$
grain boundaries passed

$B \parallel a,b$ $j \parallel c$

ac measurements do not show:

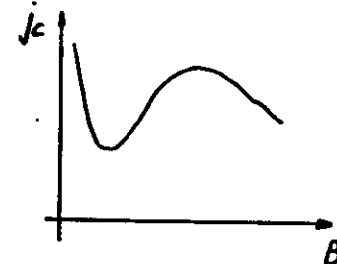
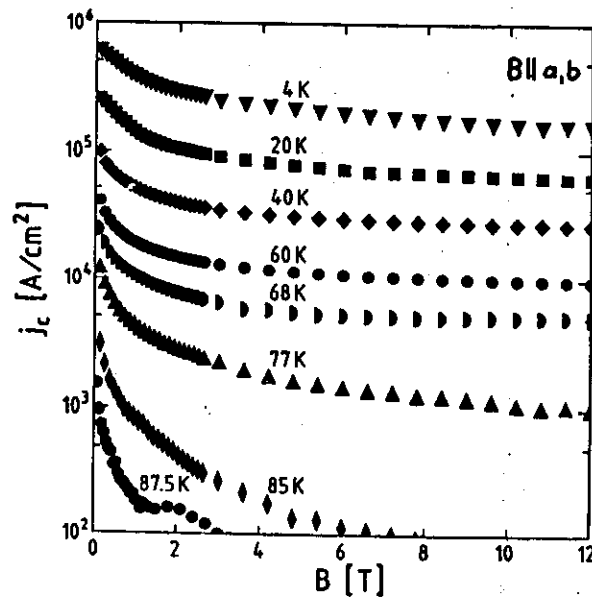
- two maxima in X''
- pronounced shift of $X''(T)$ with b_0
- two j_c values in flux profile

using grain size instead sample size:

$$j_c \parallel c \approx 5 \cdot j_c \parallel a,b$$

no decoupling at grain boundaries

decoupling within a grain ?



no "fish tail" effect in $j_c(B)$

investigation of granularity

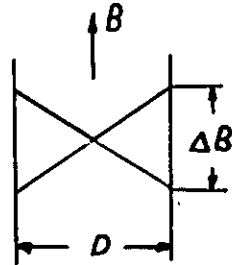
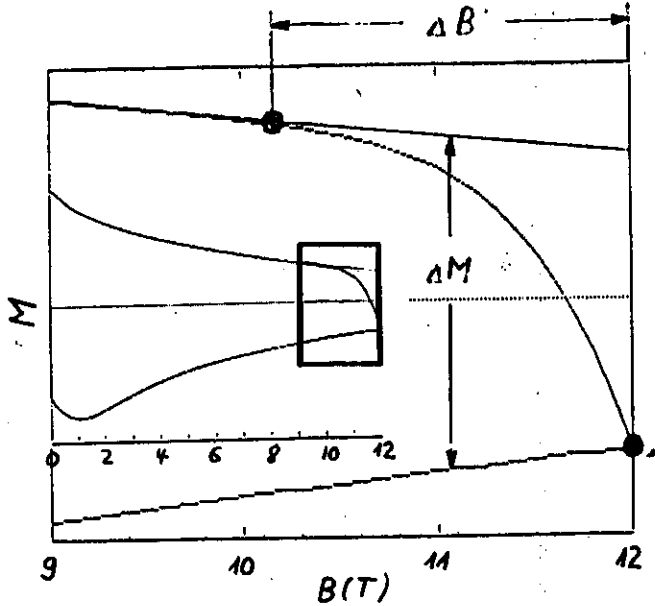
from deviation of uniform bulk current flow

T. Matsushita

critical state model:

$$j_c(\Delta M) \approx \frac{3 \Delta M}{D}$$

$$j_c(\Delta B) \approx \frac{\Delta B}{\mu_0 D}$$



$$\frac{j_c(\Delta B)}{j_c(\Delta M)} \approx \frac{\Delta B}{3 \mu_0 \Delta M} \approx 1$$

no surface pinning
correct geometry
 $j_c = \text{const. within } \Delta B$
 $D > \lambda'$

D cancels, no information about granularity?

$$\frac{\Delta B}{3 \mu_0 \Delta M} > 1$$

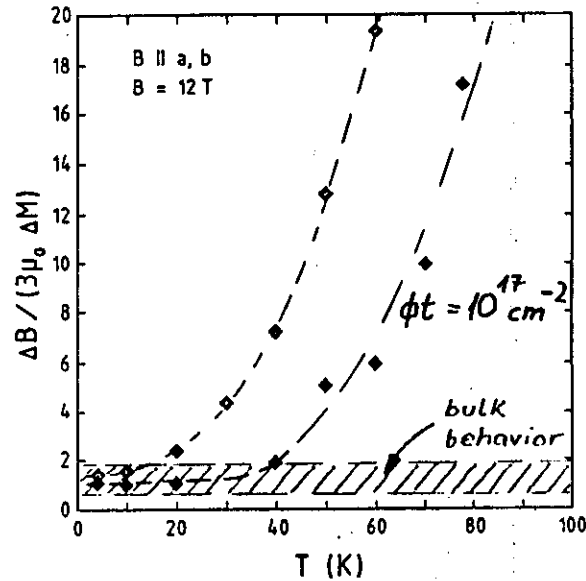
in the case of partial decoupling, inhomogeneities
percolation: nonuniform current

flow, precursor of complete decoupling

$$\underline{B \perp a, b} \quad \underline{j \parallel a, b}$$

$$\frac{\Delta B}{3 \mu_0 \Delta M} \approx 1 \quad \text{up to } 77 \text{ K, } 4 \text{ T}$$

$$\underline{B \parallel a, b} \quad \underline{j \parallel c}$$

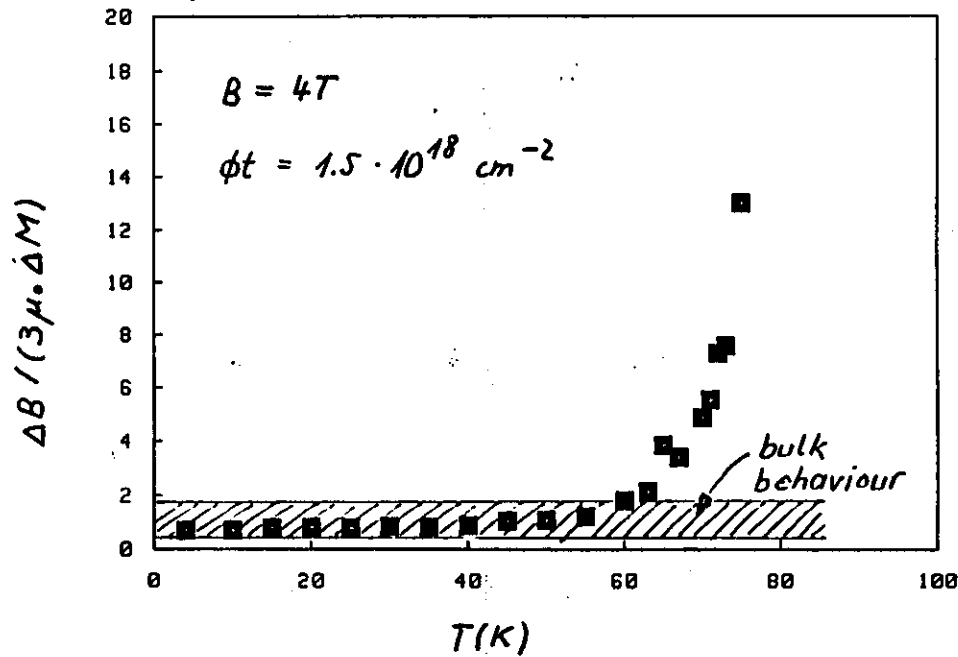


increasing deviation from homogeneous bulk current flow with B and T:
planar defects
oxygen def. regions

- addition oxygenation
 - fast neutron irradiation
- } decrease $\frac{\Delta B}{\Delta M}$

nonuniform current flow results from oxygen def. and inhomog. ox. distribution

$j_{|| a, b}$ shows granular features at higher ϕt



intragrain weak links

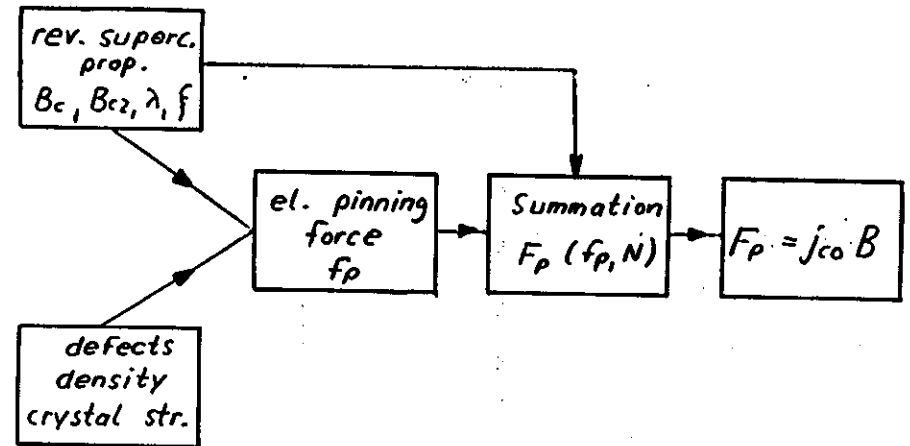
- oxygen deficient regions
- stacking faults ?
- microcracks ?

intergrain weak links

grain boundaries

2) Flux pinning

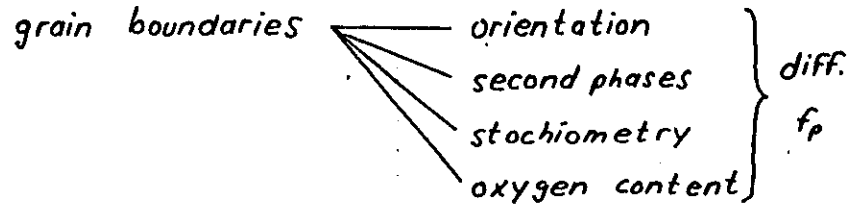
improvement of j_c requires identification of the defect structure which is responsible for the measured j_c



- usual way:
- (1) measurement of $j_c(B, T)$ and rev. properties and characteriz. of defect structure
 - (2) estimation of $f_p(B, T)$
 - (3) estimation of $F_p(B, T)$ (summation problem)
 - (4) comparison of $F_p(B, T)$ with measured $j_c(B, T)$
 - (5) variation of the defect structure $F_p(N, \text{defects})$

Problems

- Large number of different defects



twin boundaries

stacking faults

precipitates

microcracks

porosity, microvoids

stoichiometry fluctuation

oxygen deficient regions

point defects

dislocations

- variation of a certain defect structure changes other superconducting properties

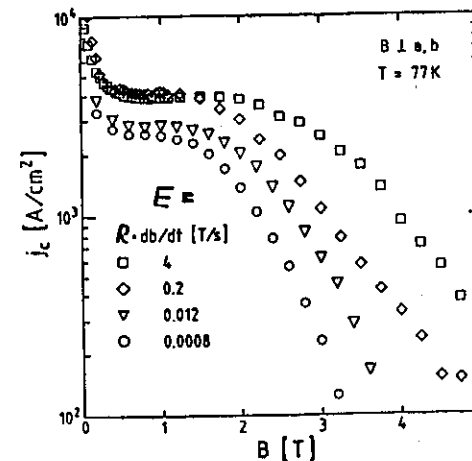
- high T_c specific pinning mechanisms:
 - intrinsic pinning
 - pinning from point defects
- anisotropy of the superconducting prop.
 - " of the defect structure

$$\left. \begin{array}{l} j \parallel a, b \quad B \parallel a, b \\ j \parallel a, b \quad B \parallel c \\ j \parallel c \quad B \parallel a, b \end{array} \right\} \text{for } j \perp B$$

- contribution from surface pinning in thin films

- measurements of $j_c(B, T)$ are

influenced by relaxation, especially at higher T and B , but pinning models are based on unrelaxed current \bar{j}_c



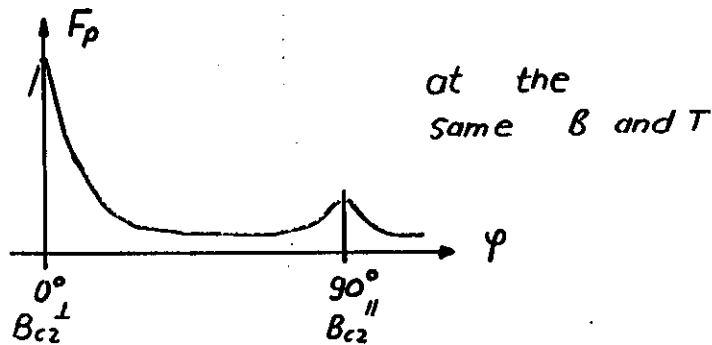
- comparison of the pinning strength in different geometries requires to eliminate the influence of different B_{c2} , \mathcal{L} and different reduced field B/B_{c2}

$$F_p(B, T) = K(\text{defect specific}) \underbrace{B_{c2}^m(T)}_{\text{temp. dependence}} \underbrace{f(B/B_{c2})}_{\text{field dependence}}$$

$$\frac{K^{\parallel}}{K^{\perp}} = \frac{F_p^{\parallel}(T_1, B_1/B_{c2}^{\parallel})}{F_p^{\perp}(T_2, B_2/B_{c2}^{\perp})}$$

$$\text{with } [B_{c2}^{\parallel}(T_1)]^{m^{\parallel}} = [B_{c2}^{\perp}(T_2)]^{m^{\perp}}$$

$$f(B_1/B_{c2}^{\parallel}) = f(B_2/B_{c2}^{\perp})$$



$$\left(\frac{B_{c2}^{\parallel}}{B_{c2}^{\perp}}\right)^m \approx \begin{cases} 50 & \text{Y-Ba-Cu-O} \\ 10^3 & \text{Bi-Sr-Ca-Cu-O} \end{cases}$$

variation of the defect structure

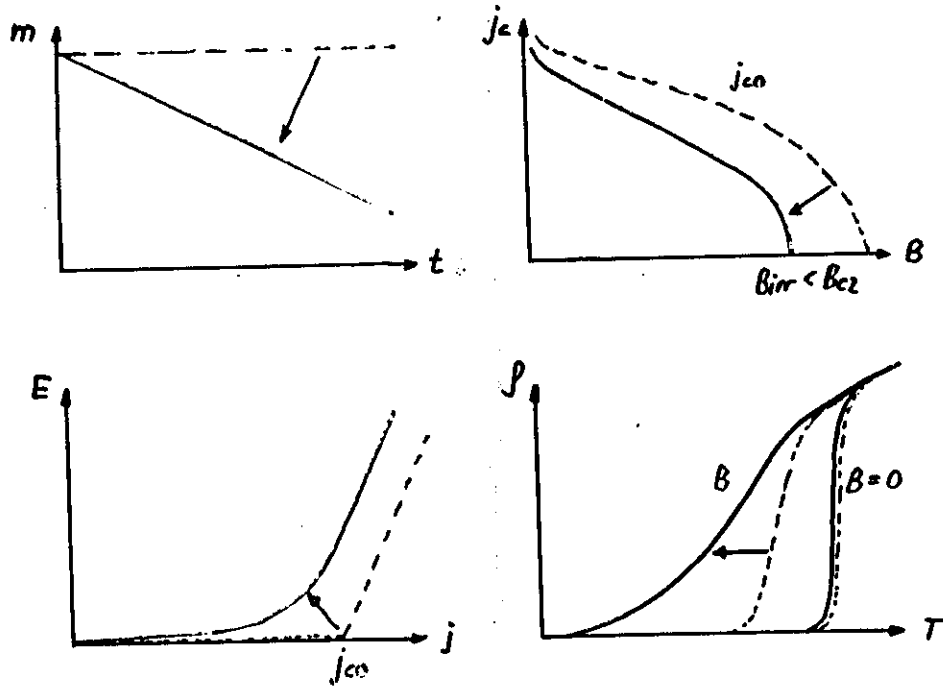
- (1) texture
melt processing
- (2) 211 inclusion from liquid
- (3) radiation damage
- (4) doping with Fe, Zn, ...
- (5) oxygenation
- (6) alloying with Ag
- (7) precipitates

granularity and pinning cannot strictly separated. The same defect may act as pinning centre and as weak link in different geometry, B or T region

A.M. Campbell

3) RELAXATION

experimental observations:

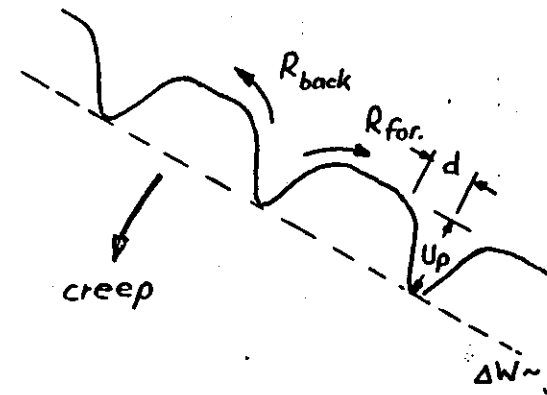


- time dependent decay of the current by frustration or junction resistivity

● time dependent decay by thermally activated flux creep

- fluctuation, flux lattice melting, ...

thermally activated flux creep:



U_0 : activation energy

U_p : pinning pot.

d : potential width

V : flux bundle volume

Ω_0 : attempt frequ.
($10^6 \div 10^{12}$ Hz)

$$U(t) = U_0 - \Delta W(t)$$

$$\Delta W = j B d V$$

$$U(j) = U_0 \left(1 - \frac{j}{j_{c0}}\right)$$

Anderson

Beasley et al.

Campbell and Evetts

Dew Hughes

Griessen

hopping rate:

$$R_{for.} = \Omega_0 e^{-\frac{(U_0 - \Delta W)}{kT}}$$

$$R_{back} = \Omega_0 e^{-\frac{(U_0 + \Delta W)}{kT}}$$

$$R_{res.} = 2 \Omega_0 e^{-\frac{U_0}{kT}} \sinh \frac{\Delta W}{kT}$$

$$\vec{E} = \vec{v} \times \vec{B} \quad \text{electric field due to flux movement}$$

$$E = R_{res.} a_0 B \quad \begin{array}{l} a_0: \text{flux line distance} \\ = \text{hopping distance} \end{array}$$

$$U_0 \gg kT$$

$$(1) j_{co} \geq j \geq j_{co} \frac{kT}{U_0} \quad (\Delta W \gg kT)$$

$$E = a_0 B \Omega_0 e^{-\frac{U_0}{kT} (1 - \frac{j}{j_{co}})} \sim e^j$$

$$(2) j \leq j_{co} \frac{kT}{U_0} \quad (\Delta W \leq kT)$$

$$E = 2 \frac{U_0}{kT} \frac{a_0 B \Omega_0}{j_{co}} e^{-\frac{U_0}{kT}} j \sim j$$

Maxwell, Bean : $E = -\frac{1}{2} \mu_0 R \frac{dM}{dt}$

$$(1) M = M_0 \left(1 - \frac{kT}{U_0} \ln \left(1 + \frac{t}{\tau_0} \right) \right) \sim \ln t$$

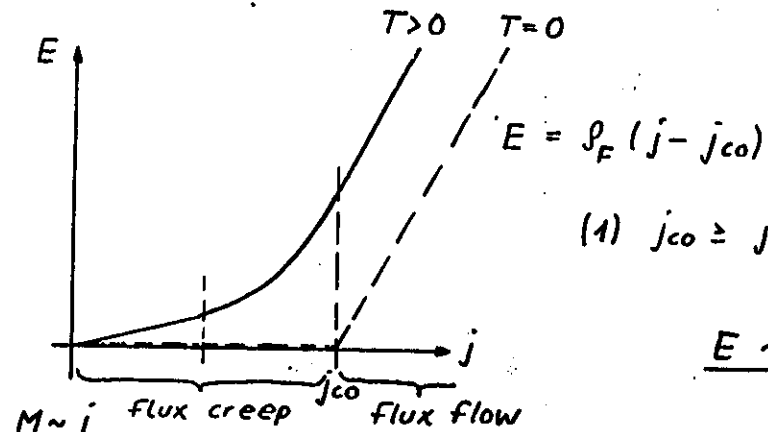
$$\tau_0 = \frac{1}{6} \mu_0 R^2 \frac{kT}{U_0} \frac{j_{co}}{E_0}$$

$$(2) M = M_0 \frac{kT}{U_0} e^{-2(t - \tau_c)/\tau_c} \sim e^{-t}$$

$$\tau_c = \tau_0 e^{U_0/kT}$$

Keller et al.

$$U_0 \gg kT$$



$$(1) j_{co} \geq j \geq j_{co} \frac{kT}{U_0}$$

$$E \sim e^j$$

$$M \sim j \sim \ln t$$

$$(2) j \leq j_{co} \frac{kT}{U_0}$$

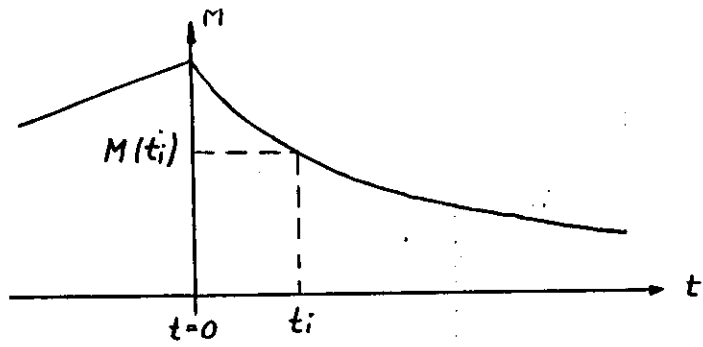
$$E \sim j$$

$$M \sim j \sim e^{-t}$$

$$U_0 \approx kT$$

$$E = 2 \frac{U_0}{kT} \frac{a_0 B \Omega_0}{j_{co}} e^{-U_0/kT} j$$

$$M \sim j = j_{co} e^{-t/\tau_s}$$



$$\frac{U_0}{kT} = \frac{M(t=0)}{dM/dlnt} = \frac{1}{\underbrace{\left[1 - \frac{kT}{U_0} \ln\left(1 + \frac{t_i}{\tau_0}\right)\right]}_{\text{decay within } 0 \leq t \leq t_i}} \frac{M(t_i)}{dM/dlnt}$$

$$\tau_0 \approx \Omega_0^{-1} \approx 10^{-12} \div 10^{-6} \text{ s}$$

$$\tau_0 = \frac{1}{6} \mu_0 R^2 \frac{kT}{U_0} \frac{j_{co}}{E_0}$$

typical values for $YBa_2Cu_3O_{7-x}$ ($B \ll B_{irr}$)

$$\frac{U_0}{kT} = 10$$

$$j_{co} = 10^4 \text{ A/cm}^2$$

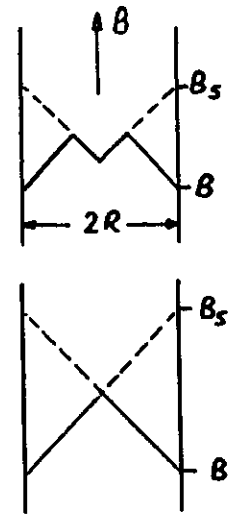
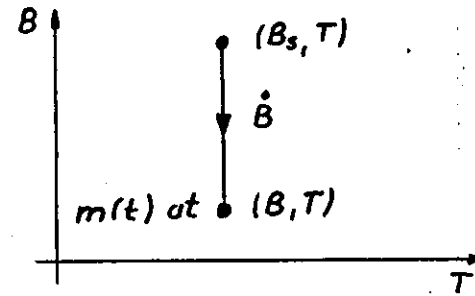
$$E_0 = 1 \mu\text{V/cm}$$

$$R = 1 \text{ mm}$$

result in $0.1 \text{ s} \lesssim \tau_0 \lesssim 10 \text{ s}$

$m(t)$ measured at B, T depends on

(1) history: $B_s, T_s, \dot{B}, \dot{T}$



Matsushita et al.

Moshchalkov et al.

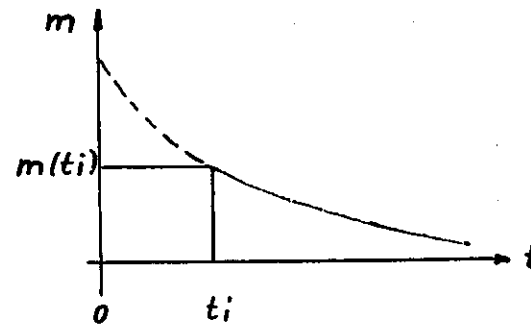
$$(B_s - B) > \mu_0 j_c 2R$$

$$(B_s - B) \approx 3 T$$

$$\dot{B} \approx 0.012 \text{ T/s}$$

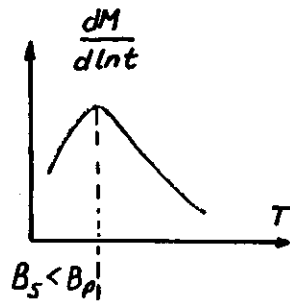
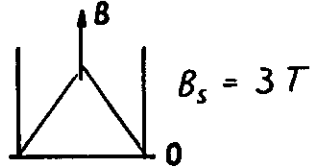
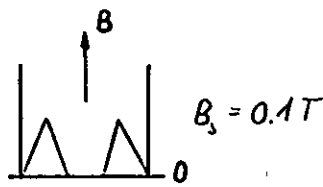
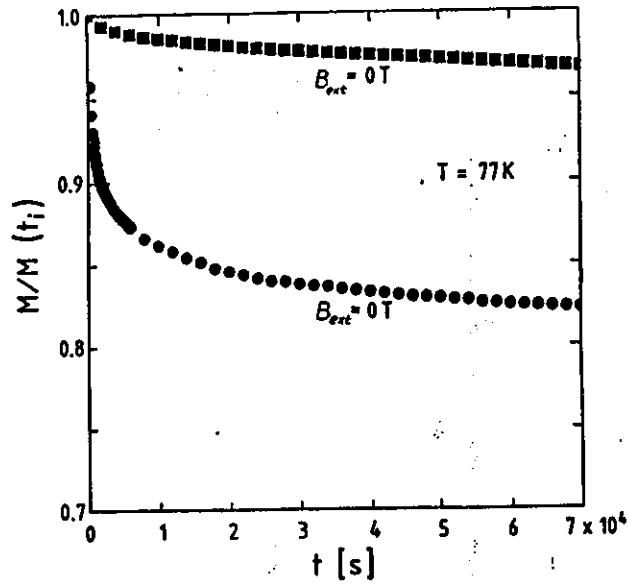
not sufficient in the vicinity of B_{irr}

(2) time t_i

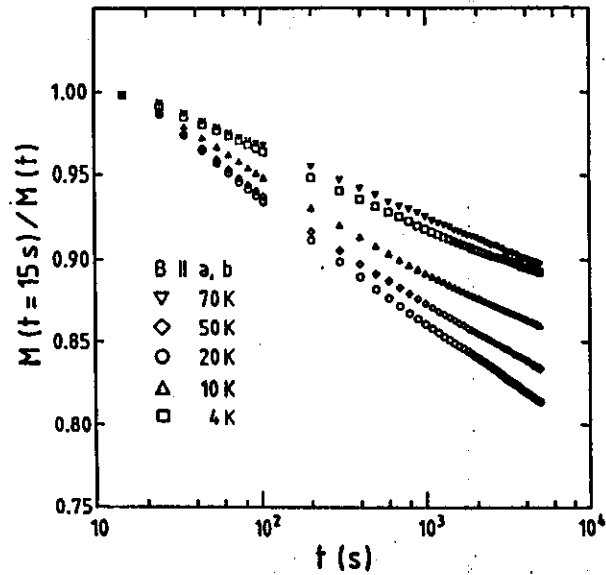
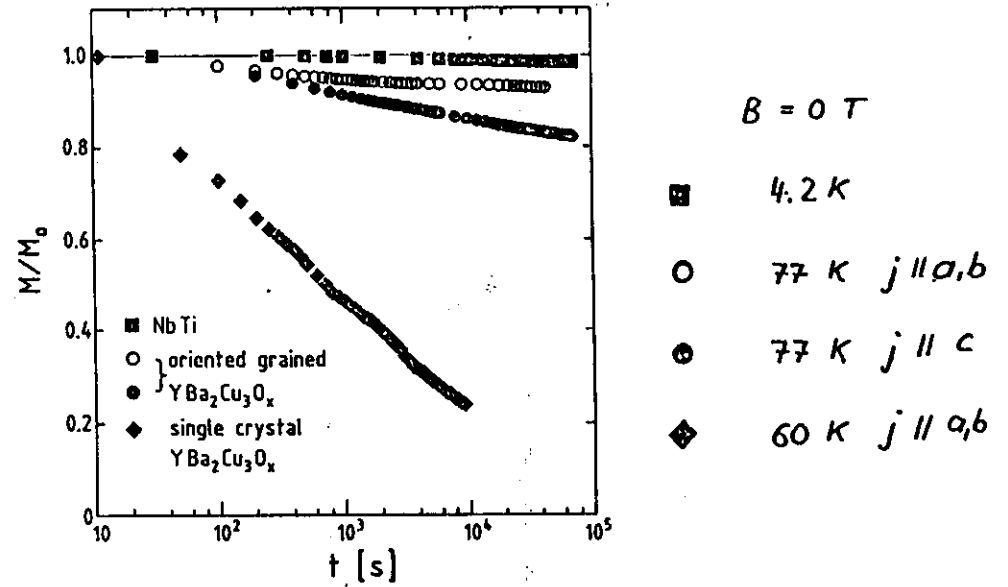


first measurement of m at t_i

$$t_i \approx 15 \text{ s}$$



Comparison with Nb 49 wt. % Ti
 $T_c \approx 9K$
 $j_c(4.2K, 0T) \approx 3.5 \times 10^5 A/cm^2$



deviations
 from a logar.
 decay of M :

- (1) $t \leq 50s$
- (2) in the vicinity
 of B_{irr}
 ($U_0 \geq kT$)

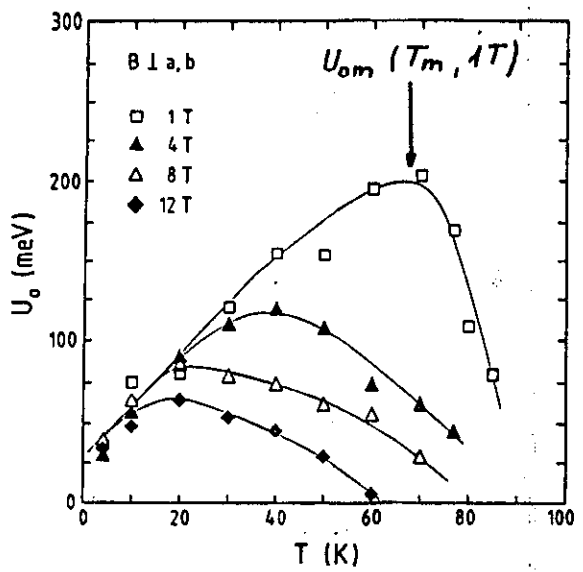
| | U_0 / kT | $U_0 [meV]$ | $\rho [\mu\Omega cm]$ |
|----------------------------------------------------------------------------------------------------|------------|-------------|-----------------------|
| oriented grained $j c$ YBa ₂ Cu ₃ O _x | 41 | 272 | |
| 77K $j a,b$ | 55 | 365 | 10^{-9} |
| Nb 49 Ti | 4K | 487 | 10^{-12} |
| | 7.5 K | 182 | |
| YBa ₂ Cu ₃ O _x 60K single crystal with intragrain junctions | 8 | 41 | |

$$\frac{U_0}{kT} = - \frac{M(t_i)}{dM / d \ln t}$$

$$\frac{1}{\left[1 - \frac{kT}{U_0} \ln \left(1 + \frac{t_i}{\tau}\right)\right]} < 2$$

expected $U_0(B, T)$ from

- (1) $j_c(B, T)$ similar as $U_0(U_p)$
- (2) low T_c superconductors

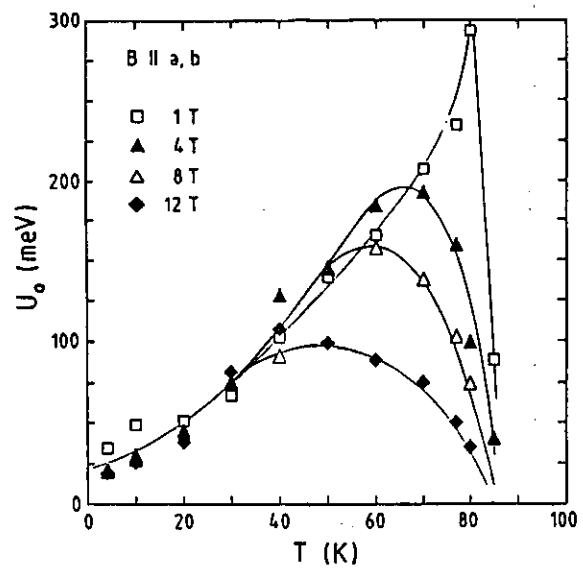


$U_0(B, T)$ behaviour :

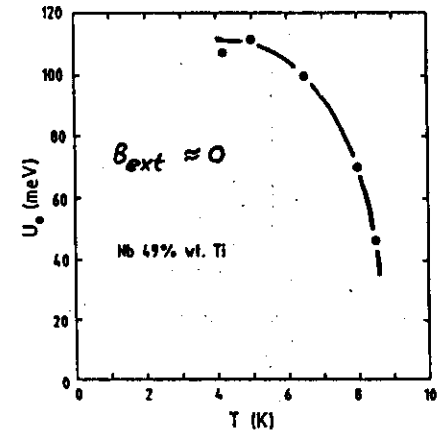
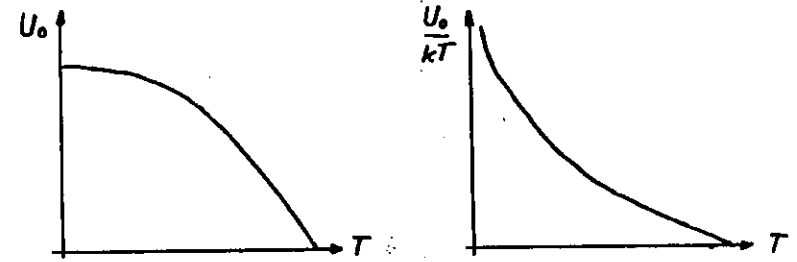
(1) $U_0(T)$ increases with T up to $U_{0m} (T = T_m)$

(2) T_m and U_{0m} decrease with B

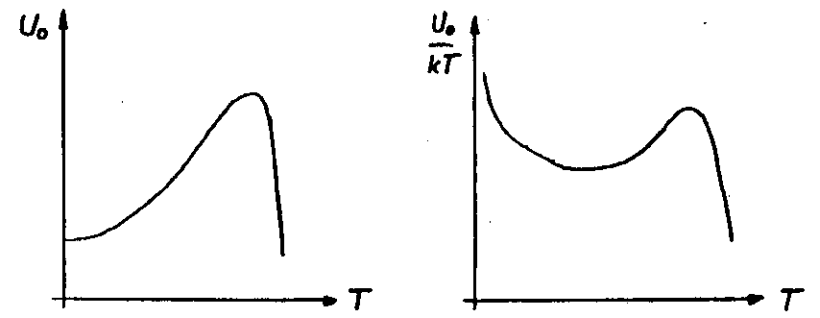
(3) above T_m
 U_0 approaches T_{irr} at which $U_0 \approx kT$



$T \approx T_m$
 $30 \lesssim \frac{U_0}{kT} \lesssim 80$
 comparable with low T_c supercond.



observed



explanations for $U_0(T)$

(1) distribution $F(U_0)$ neglected

(a) single vortex activation

→ (b) flux bundle activation

Fujiyoshi et al.

→ (c) nonlin. $U(j)$ relation Suenaga et al.
Maley et al.
Welch

(2) distribution $F(U_0)$ dominant

(a) j_c distribution, $E \sim j^n$ Sun et al.

→ (b) exp. time scale determines measured part of $F(U_0)$
Hagen and Griessen

→ (c) percolation path in $F(U_0)$
Gurevich et al.

→ (3) change of pinning structure
Keller et al.

$U_0(T)$ due to collective interaction

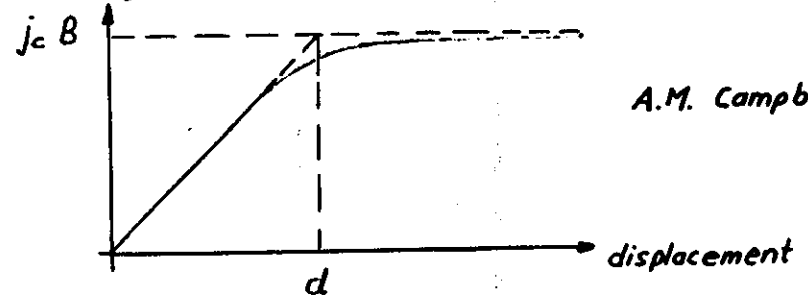
$U_0 \neq U_p \Rightarrow$ thermal activation of flux bundles

$$j_c B d = \hat{U}_0 = \frac{U_0}{V}$$

$V =$ flux bundle volume

$d =$ reversible displacement

restoring force



A.M. Campbell

$$V \sim \frac{T}{dM / d \ln t}$$

$V \hat{=} V_c$ correlation volume of the collective theory

Kes et al., Matsushita

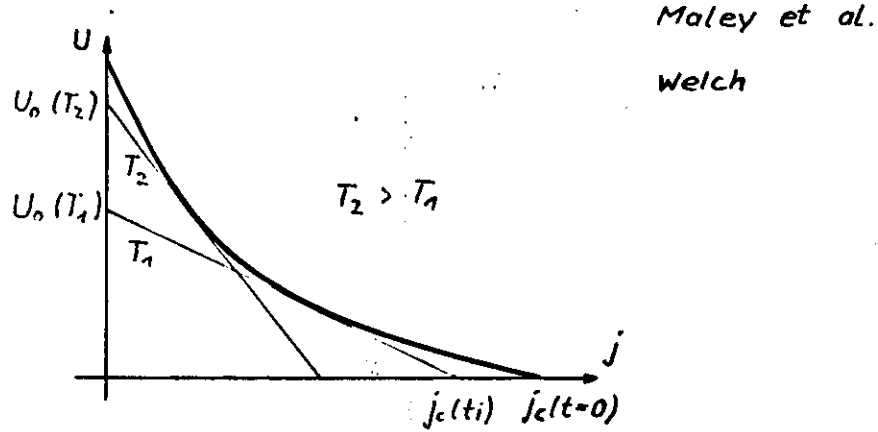
$V_c(T) \sim \tilde{L}_c(T)$ nonlocal limit

$$\tilde{L}_c(T) \sim \left(\frac{B_{c2}^2(T)}{\lambda^4(T) f_p^2(T)} \right)^{1/3} \quad \text{Larkin and Ovchinnikov}$$

$$f_p \sim B_c^2 \xi^m \quad \text{core interaction}$$

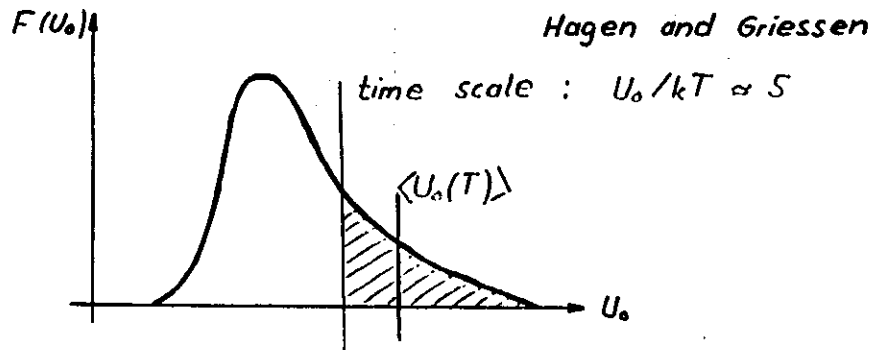
$$V(T) \sim \tilde{L}_c(T) \text{ increases for } m < 2$$

(1c) nonlinear $U(j)$ relation



decay of j_c from $j_c(t=0)$ to $j_c(t_i)$ increases with T

(2b) experimental time scale determines measured part of $F(U_0)$

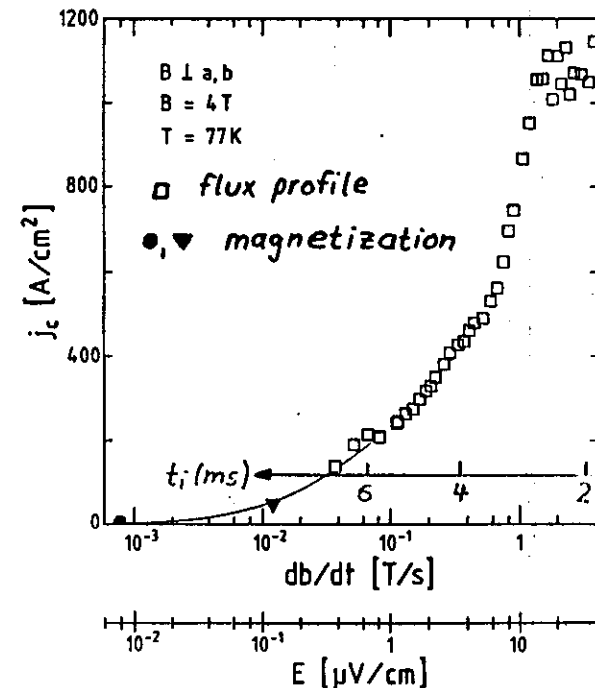


statistically independent relaxation
 $\langle U_0(T) \rangle$ increases with T

both explanations predict :

- non log. time dependence of M
- variation of $M(t_i) \sim j_c(t_i)$ with time scale

flux profile measurement in dependence on $db/dt \sim b_0 f$ shift t_i into ms regime



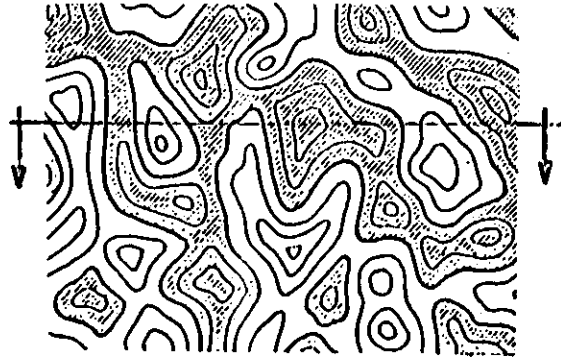
above $U_{0m}(T_m)$
 rapid decay of j_c

but
 j_c independent of t_i below $U_{0m}(T_m)$

60K, 10T :
 $j_c = (10200 \pm 200) \text{ A/cm}^2$

(2c) distribution of U
but relaxation not statistically independent

Gurevich et al.



2 dim. map $U(\vec{r})$

$$p(\vec{r}) = p_0 e^{-U(\vec{r})/kT}$$

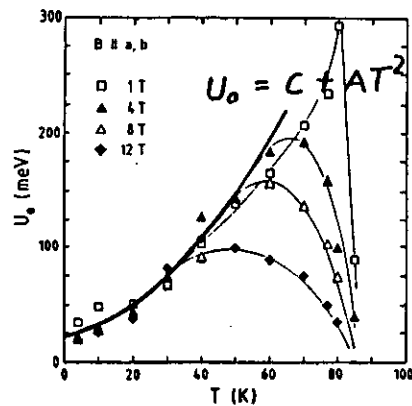
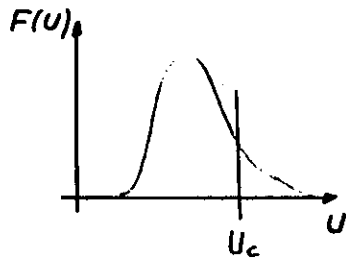
thermally activated
depinning occurs
first at the
percolation threshold
 U_c



from eff. medium theory and arbitrary $F(U, B, T)$:

$$U_0(B, T) = U_c y(B, T) - \frac{\tau^2 T^2}{6 y(B, T)} \left. \frac{dF/dU}{F} \right|_{U=U_c}$$

for low T $y(B, T) = y(B, 0) \approx \text{const}$



(3) change of pinning structure

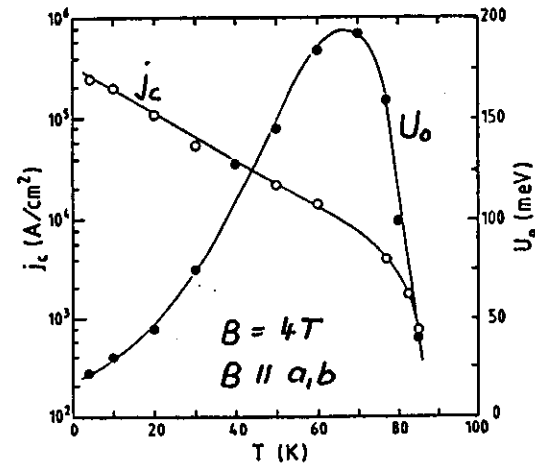
Keller et al.

oxygen deficient regions become
normal conducting with B and T

Däumling et al.

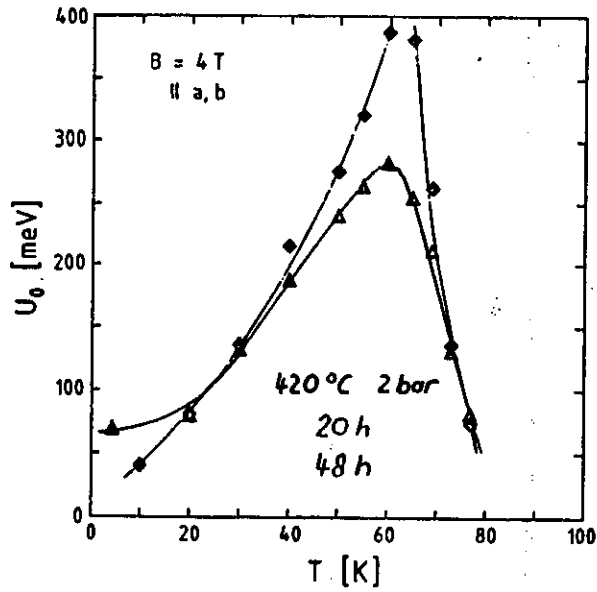
$U_0(T) \uparrow$: additional pinning and
restricted flux movement

$j_c(T) \downarrow$: decrease with T is less
fast than without add. def.



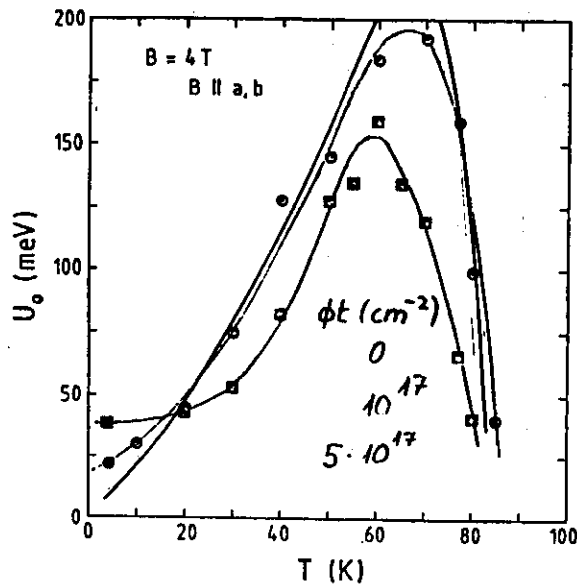
above the
maximum of U_0
 j_c decreases
faster than
 $e^{-\alpha T}$

oxygen loading and n -irradiation
result in similar change of U_0



U_0 :
increase at low T
decrease at T_m

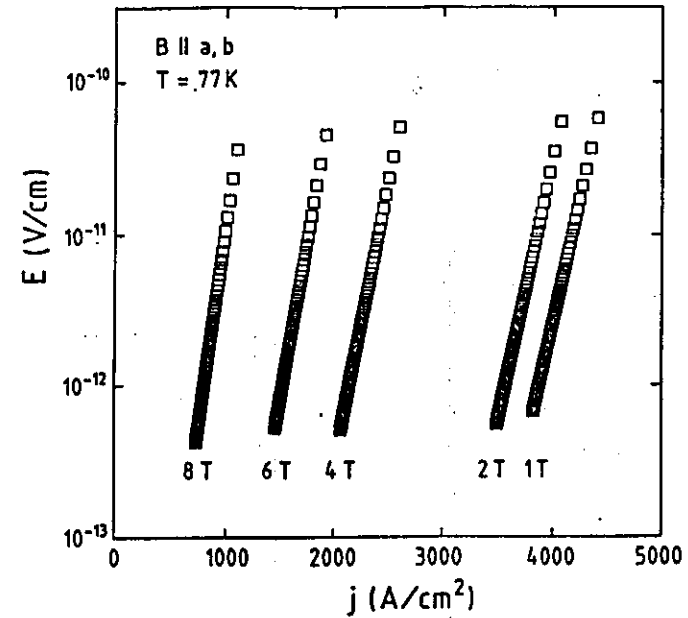
more homogeneous
oxygen distribution
reduces percolation
threshold and
restricted flux
movement



Resistivity from magnetic relaxation:

$$j(t) = \frac{M(t)}{2R}$$

$$E = \frac{1}{2} \mu_0 R \frac{dM}{dt}$$



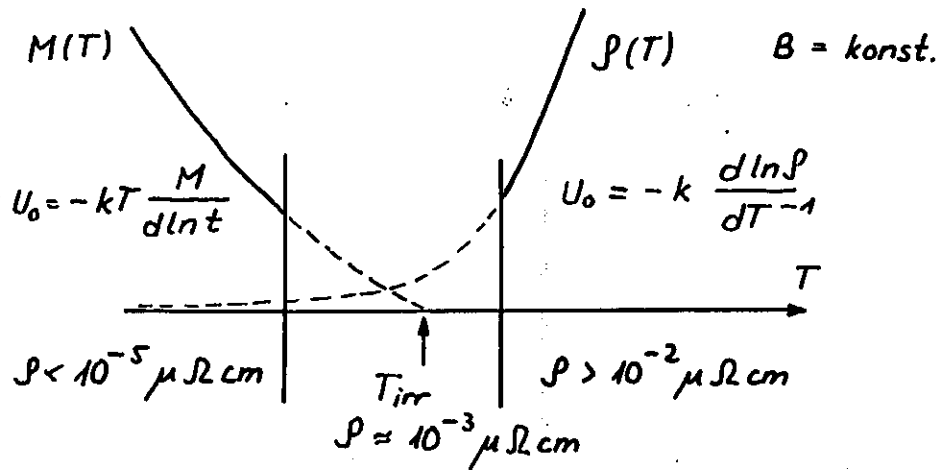
$E \sim e^j$ in accordance with $\ln t$ decay
of M

$$10^{-4} \mu\text{V/cm} \geq E \geq 10^{-6} \mu\text{V/cm}$$

$$10^{-5} \mu\Omega \text{ cm} \geq \rho \geq 10^{-11} \mu\Omega \text{ cm}$$

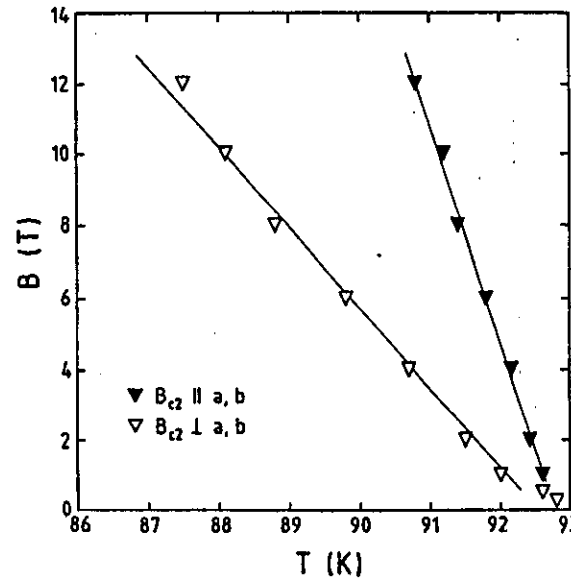
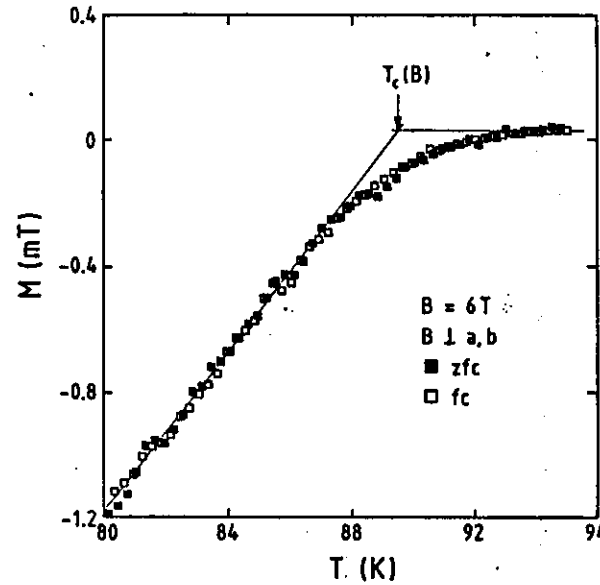
comparison between U_0 from magnetic relaxation and resistive transition

$$\rho(B, T) = \rho_0 e^{-U_0/kT} \quad \text{Palstra et al.}$$



$B_{c2}(T)$ from reversible magnetization

Finnemore
Welp et al.

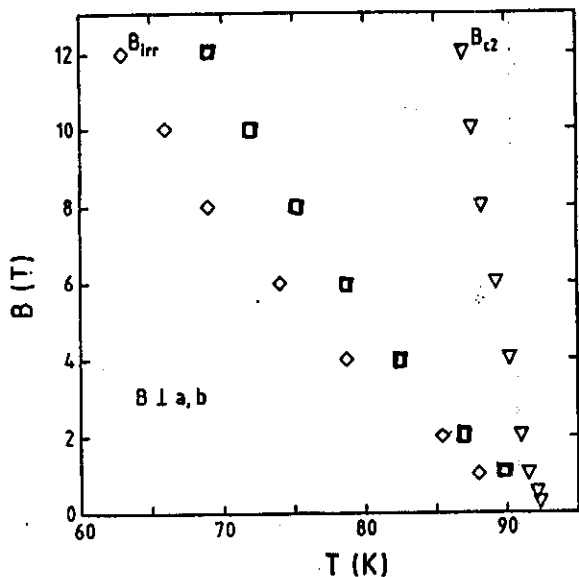


| | $M(t)$ | $\rho(T)$ |
|--------|--------------------------------|--------------------------------|
| B | $< B_{irr}$ | $> B_{irr}$ |
| j | $> 10^2 \text{ A/cm}^2$ | $< 1 \text{ A/cm}^2$ |
| E | $< 10^{-3} \mu\text{V/cm}$ | $> 10^{-1} \mu\text{V/cm}$ |
| $E(j)$ | $\sim e^j$ | $\sim j$ |
| ρ | $< 10^{-5} \mu\Omega\text{cm}$ | $> 10^{-2} \mu\Omega\text{cm}$ |
| U_0 | $< 0.4 \text{ eV}$ | $> 1 \text{ eV}$ |

?

| | $B \parallel a, b$ | $B \parallel c$ |
|-------------------------------|--------------------|-----------------|
| $\frac{dM}{dT}$ (mT/K) | 0.034 | 0.122 |
| $\frac{dB_{c2}}{dT}$ (T/K) | -6.1 | -2.4 |
| $B_{c2}(0)$ (T) | 388 | 152 |
| $f(0)$ (nm) | 0.57 | 1.44 |
| $B_{c1}(0)$ (mT) | 13 | 41 |
| λ | 293 | 92 |

$B_{irr}(T)$. from magn. or ac measurement

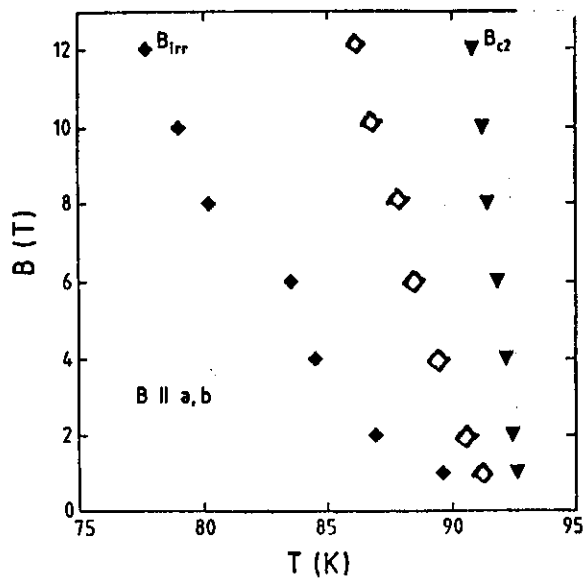


criteria for $B_{irr}(T)$:

$$E \approx 1 \mu V/cm$$

$$\rho \approx 10^{-3} \mu\Omega cm$$

decreasing sensitivity shifts $B_{irr}(T)$ to higher values :



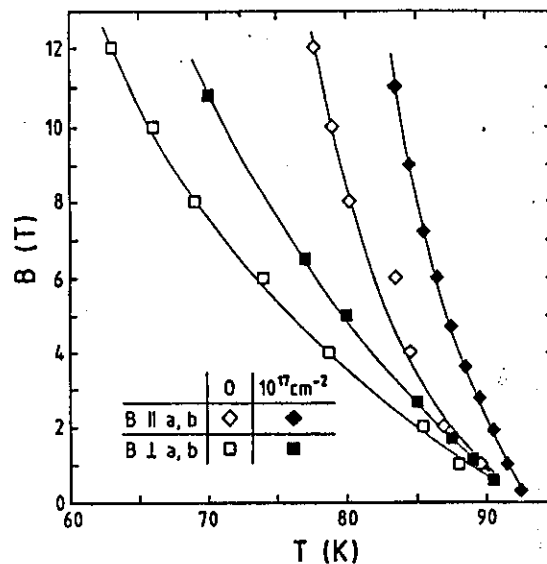
$$E \approx 10 \mu V/cm$$

$$T = 77 K$$

$$B \perp a, b$$

$$B_{c2} \approx 38 T$$

$$B_{irr} \approx 4 \div 8 T$$



$$B_{irr}(T) = \beta (1 - T/T_c)^n$$

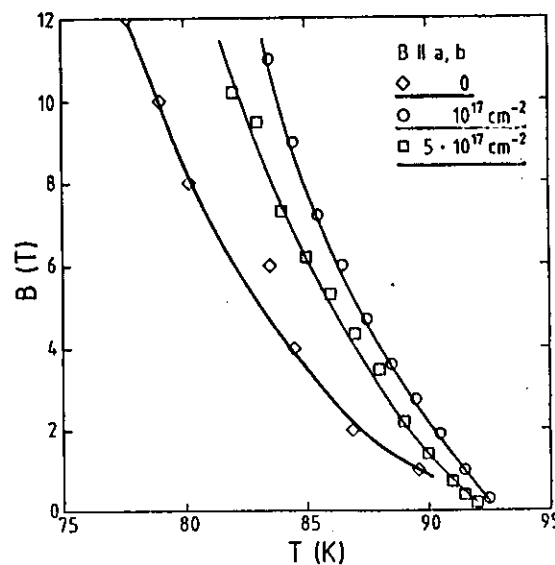
Yeshurun & Malozemoff

$$B \perp a, b \quad n \approx 1.3$$

$$\beta(\phi t = 0) \approx 51 T$$

$$\beta(10^{17}) \approx 66 T$$

minor increase of B_{irr} inspite large increase of j_c



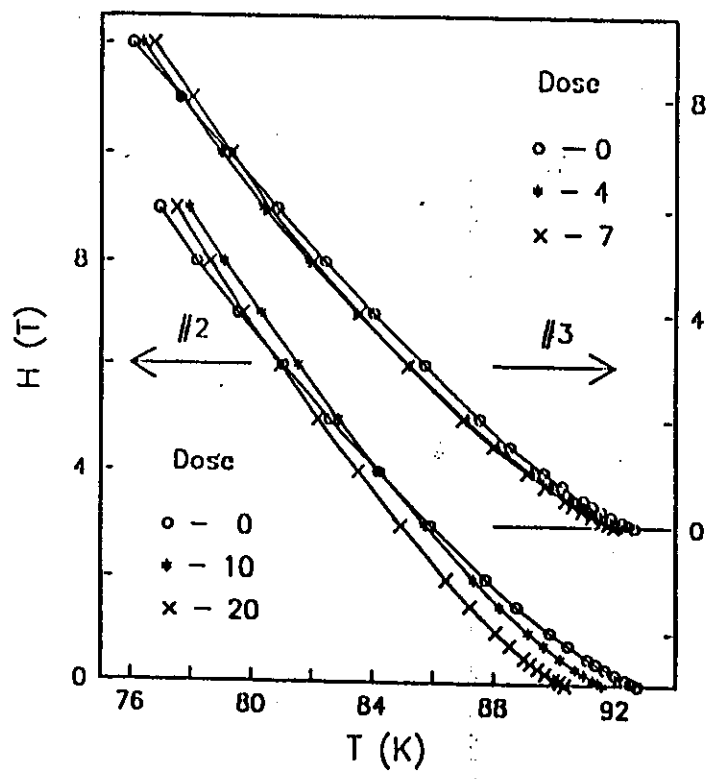
• flux melting at about $B_{irr}(T)$

• pinning before and after irr. caused by weak el. forces

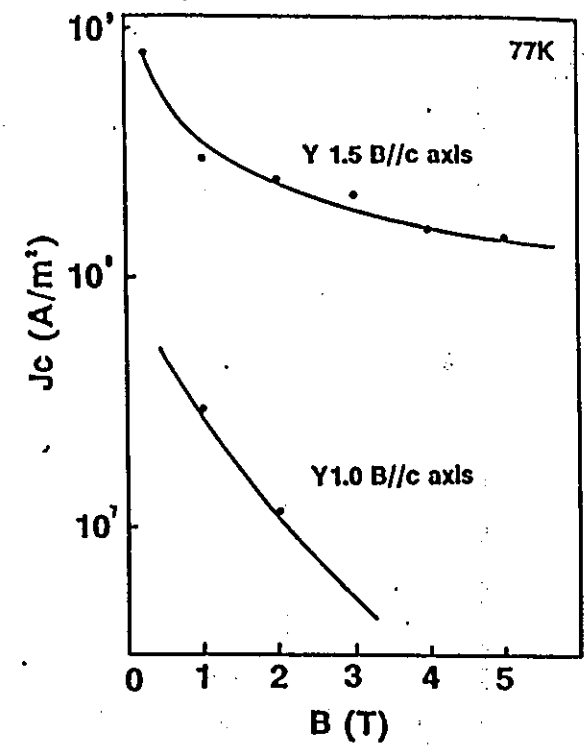
in agreement with force - displacement curves

A. M. Campbell

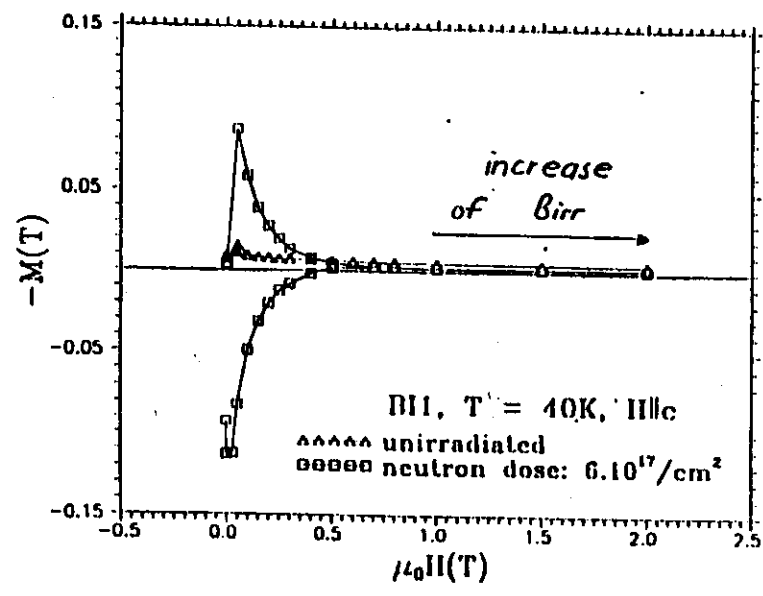
Murakami et al.



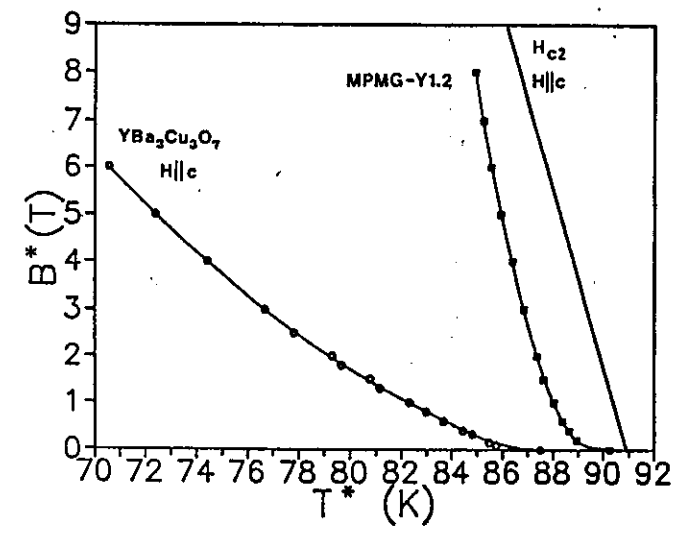
Civate et al.
 $\frac{J_c}{J_{c0}} \approx 100$
 at 77K, 1T
 but
 no increase
 of B_{irr}



MPMG $YBa_2Cu_3O_7$
 with different
 content and size
 of 211 inclusions
 "strong pinning cente
 Y 1.5 25%
 1 μm
 $B_{irr}(77K) \approx 10T$
 increase of U_0



$Bi_2Sr_2CaCu_2O_{10}$
 single crystal
 Weber et al.



Sagdahl et al.