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Raman spectroscopy and electron probe microanalysis

Lecture 1: Raman spectroscopy- simplified theory, instrumentation, analytical applications to materials science

Lecture 2: Electron probe microanalysis- theory, instrumentation, application to high temperature corrosion. Raman spectroscopy phase determination in crystal growth of high temperature superconductors, characterisation of surface phases on high temperature superconductors.

Lecture 3: More advanced Raman spectroscopy- polarisation measurement, factor group analysis, lattice defect characterisation, thin film stress measurement.

SMR.550 - 19

SPRING COLLEGE IN MATERIALS SCIENCE ON
"NUCLEATION, GROWTH AND SEGREGATION IN MATERIALS
SCIENCE AND ENGINEERING"
(6 May - 7 June 1991)

ELECTRON PROBE MICROANALYSIS AND RAMAN SPECTROSCOPY

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United Kingdom

These are preliminary lecture notes, intended only for distribution to participants.

NOTICE, MARCH 31, 1928

SUPPLEMENT--Recent Scientific and Technical Books

NOT TO BE REMOVED FROM

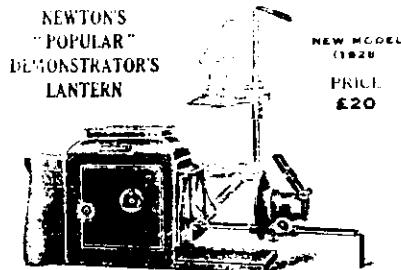


A WEEKLY JOURNAL OF SCIENCE

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Of Nature trusts the mind which builds for age."—WORDSWORTH.

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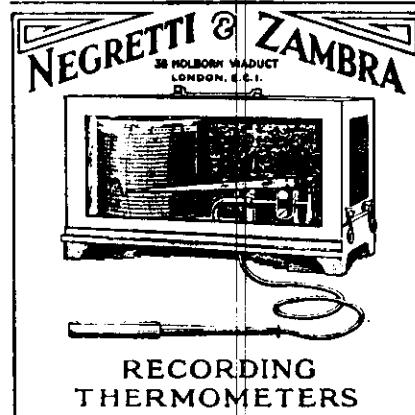
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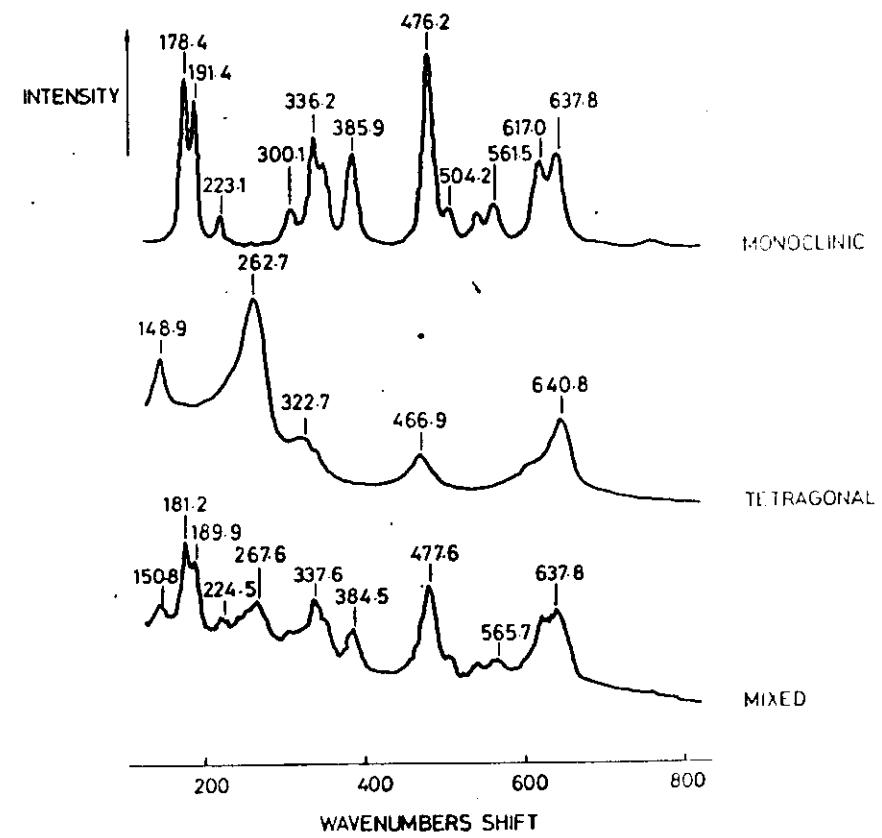
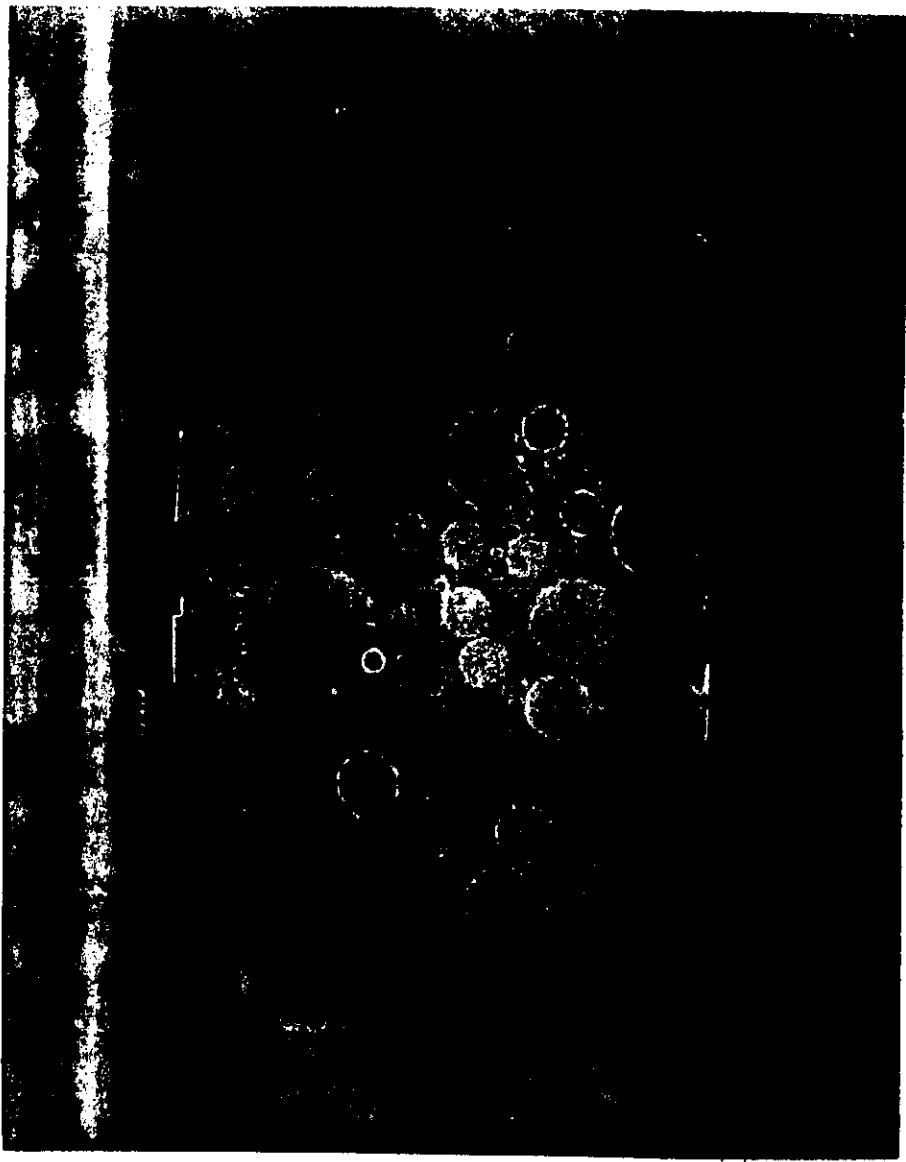
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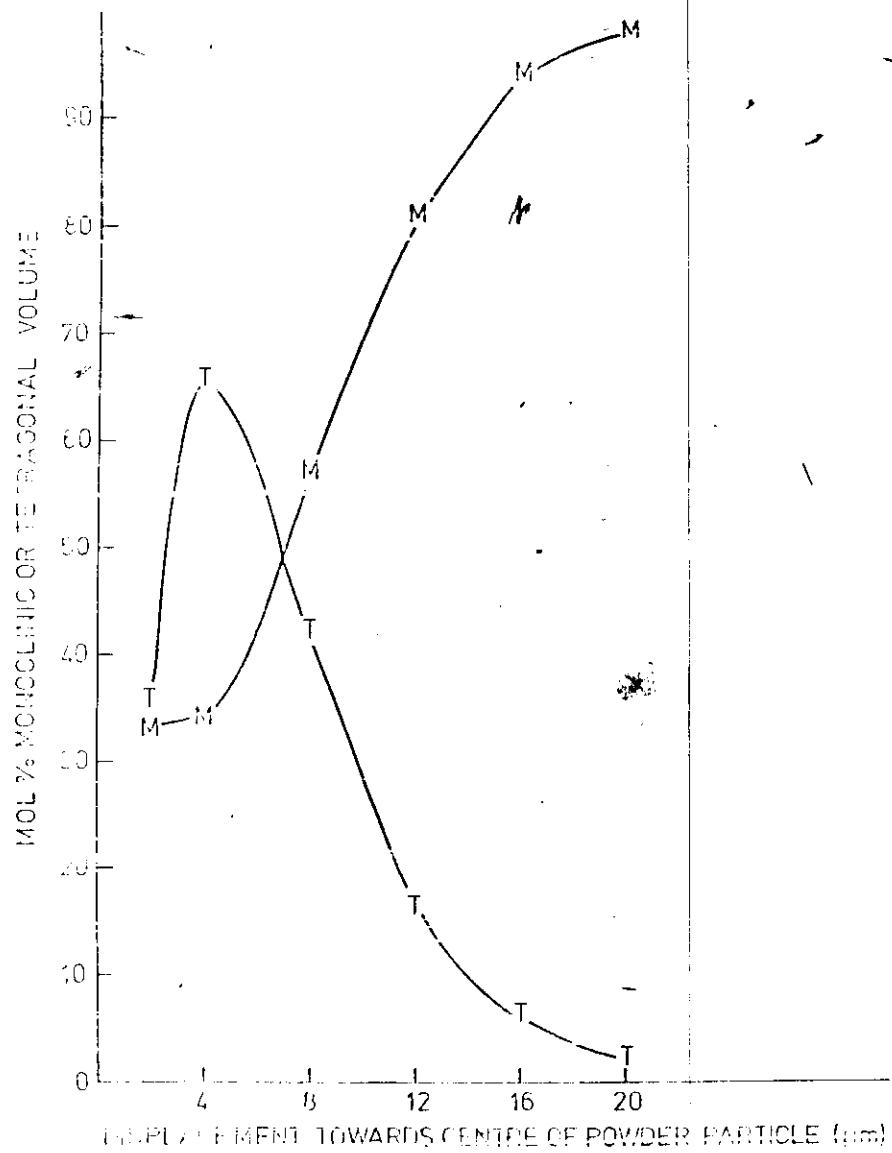
14 COMMERCIAL STREET, LEEDS.



2
RAMAN SPECTROSCOPY
KEY POINTS

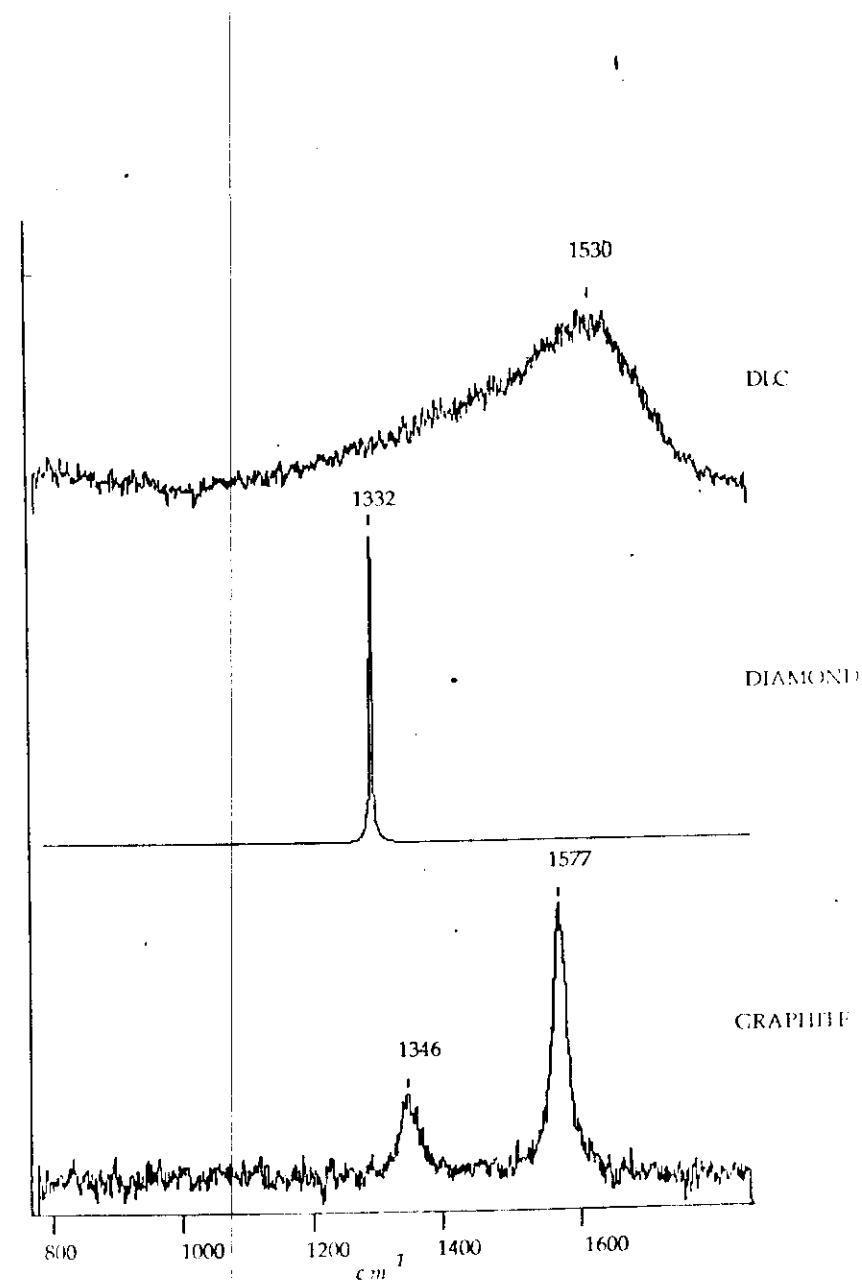
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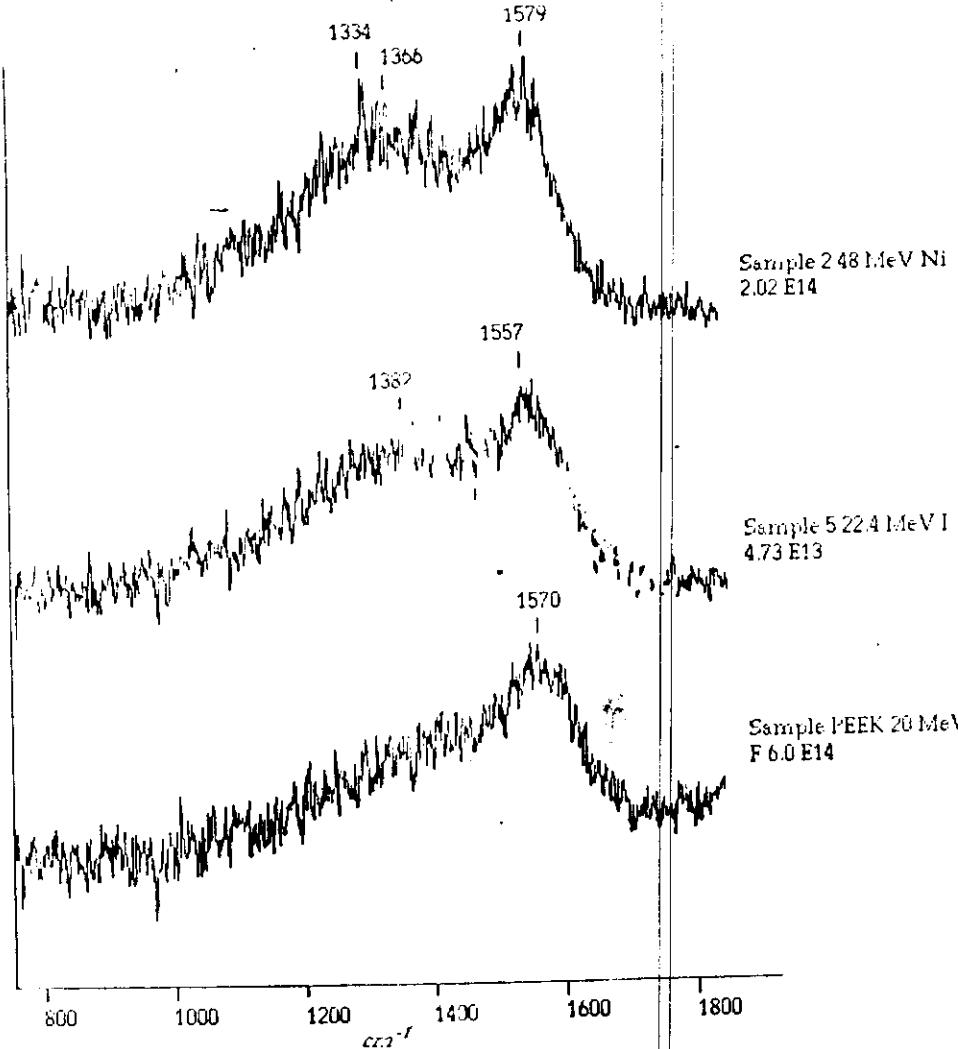


RAMAN ANALYSIS OF 10% wt Y_2O_3 SOL/GEL PRECURSOR FLOT SHOWS THAT PERCENTAGES OF MONOCLINIC AND TETRAHEDRICAL PHASES BASED ON NORMALISED RAMAN INTENSITIES AT THE 291 cm^{-1} (III) AND 981 cm^{-1} (II) BANDS.

5



Raman spectra of diamond, graphite and diamond like carbon 'DLC'.



Raman spectra of irradiated polymers as a function of dose and energy, showing sp^2 and sp^3 and diamond sp^3 bonding.

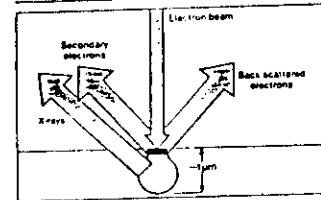
ELECTRON PROBE MICROANALYSIS KEY POINTS

- quantitative elemental analysis for $Z > 4$ (WD) and $Z > 11$ (ED)
- better than 1 micron spatial and depth resolution
- point, line and image maps possible
- nominally non-destructive
- used in moderate vacuum
- almost all materials are suitable for EPMA

EMPA

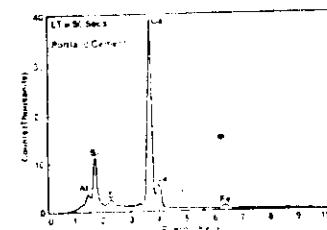
Electron Microprobe Analysis

PROCESS



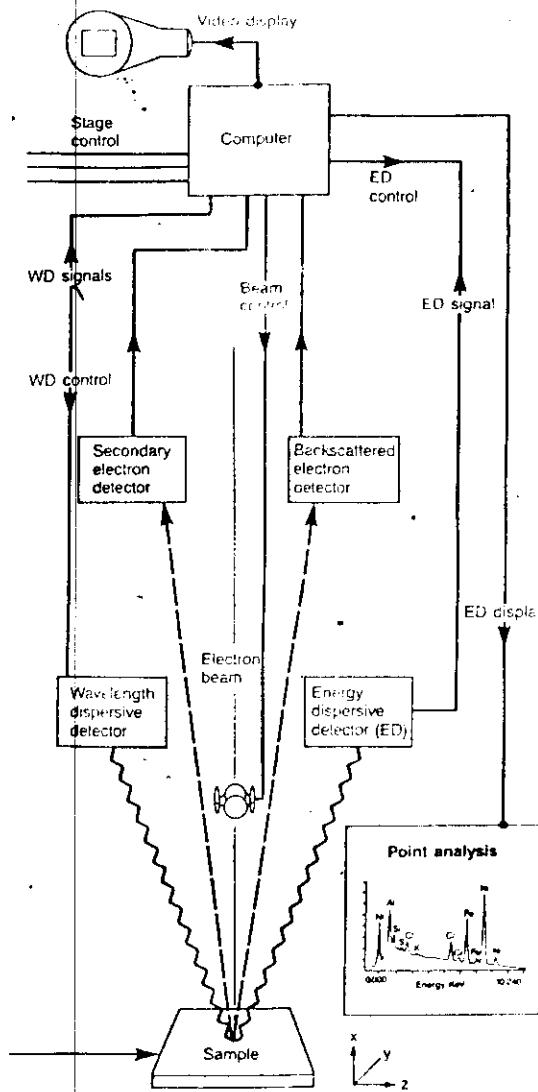
AN EXAMPLE

EMPA is used to investigate elemental composition in solids eg cements. Cement can be used for intermediate radioactive waste encapsulation.



EMPA

- A JEOL 8600 scanning microprobe
- A Cameca CAMEBAX scanning microprobe
- Fully quantitative analysis
- Generation of multi-element line scans
- Digital mapping of elemental distributions
- Automated feature detection



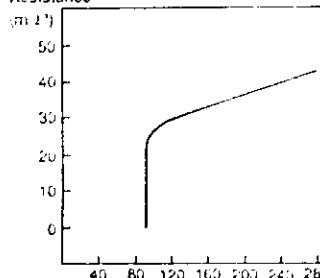
Schematic diagram of a Cameca Camebax electron microprobe

PROPERTIES OF SUPERCONDUCTORS

Zero Resistance

When superconductors are cooled below their critical temperature they have zero electrical resistance.

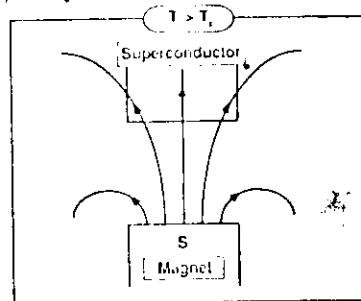
Resistance



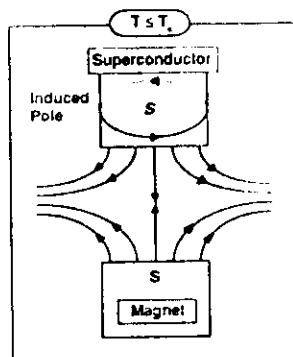
The temperature dependence of the electrical resistance of a sample of $\text{YBa}_2\text{Cu}_3\text{O}_7$ - a zero resistance state is achieved at about 90K.

The Meissner Effect

When the temperature (T) of a superconductor is above its critical temperature (T_c) an external magnetic field will pass through it.



When a superconductor is cooled below its critical temperature it exhibits perfect diamagnetism, it is repelled by a magnetic field.



When a superconductor is cooled to its critical temperature, or below, it exhibits perfect diamagnetism - The Meissner Effect - an external magnetic field will induce an electric current to circulate within the superconductor. This circulating current generates a magnetic field in such a direction that it opposes the force that is creating it (Lenz's Law). As there is zero electrical resistance within the superconductor the induced magnetic field and the electric current will remain until the temperature of the superconductor is greater than its critical temperature. The external field and the induced magnetic field are in opposite directions and so repel each other, hence the levitation.

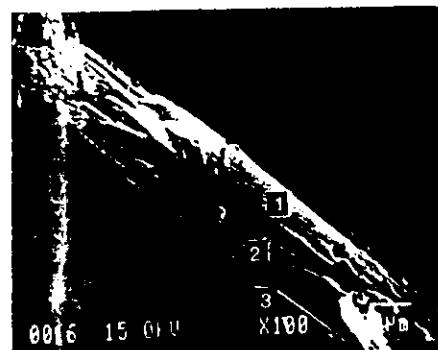
8



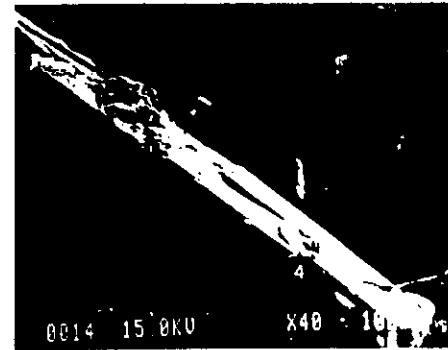
Sample 1



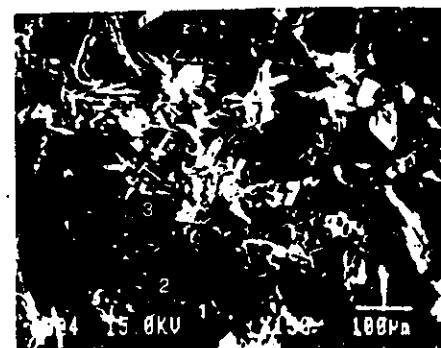
Sample 3



Sample 4

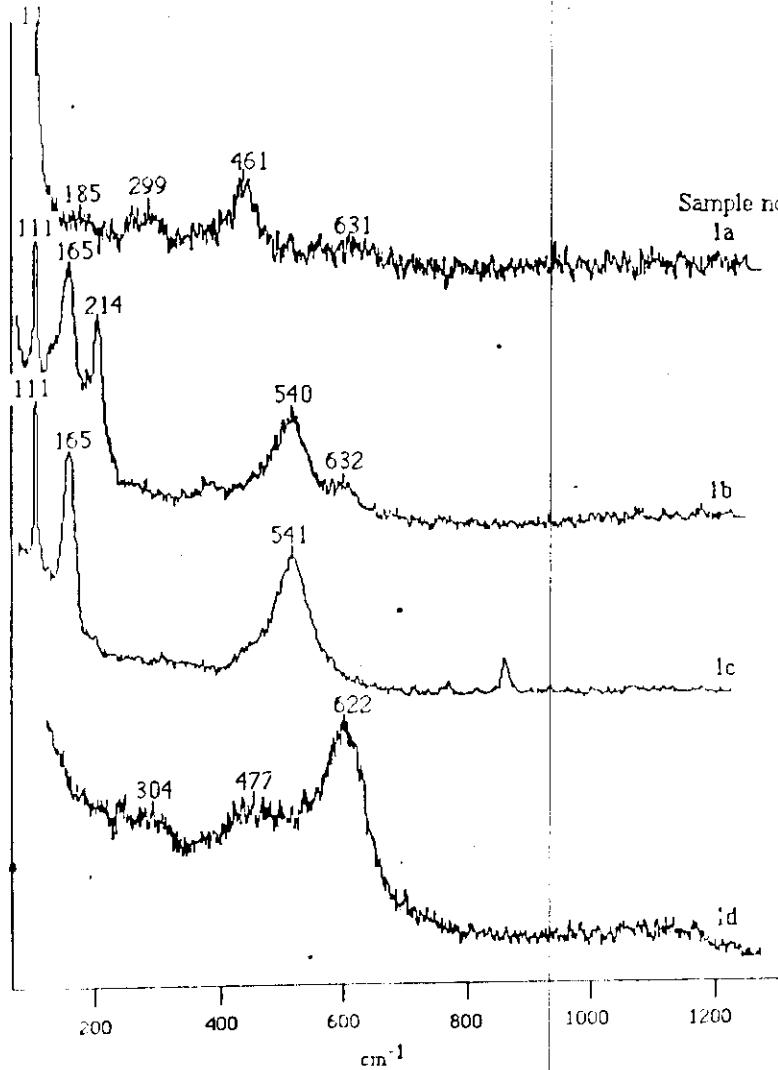


Sample 4

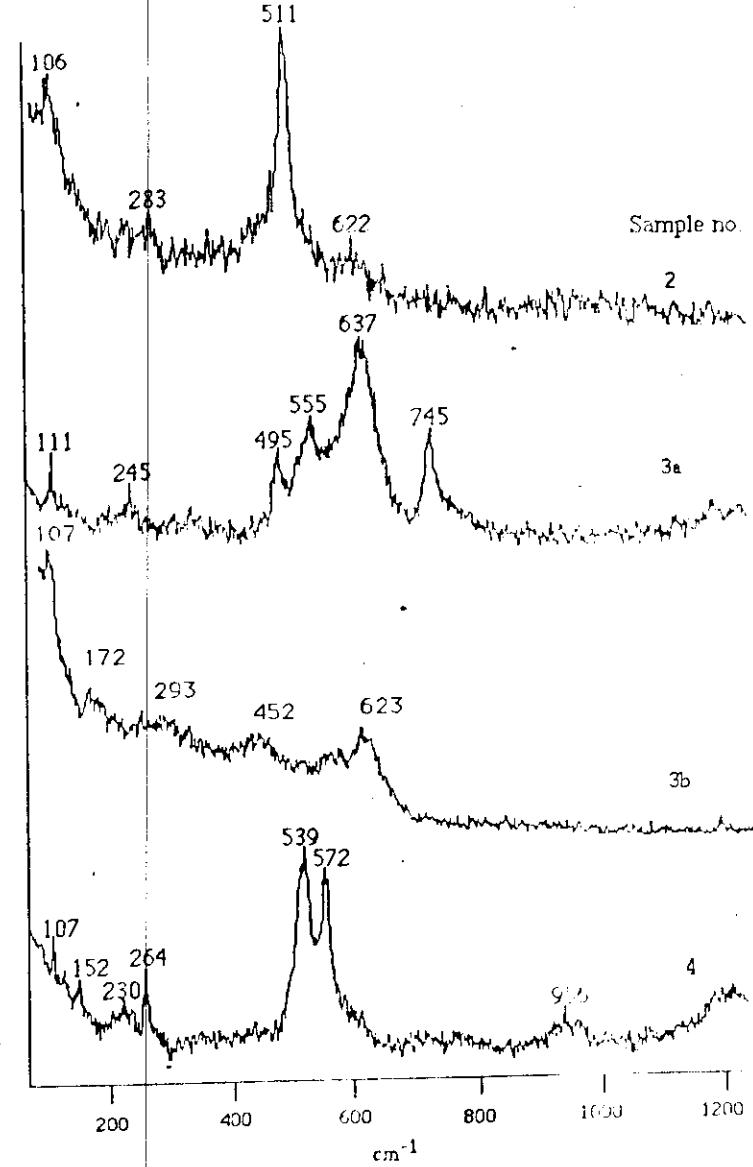


Sample 6

Secondary electron images of samples showing positions of EMPA measurements.



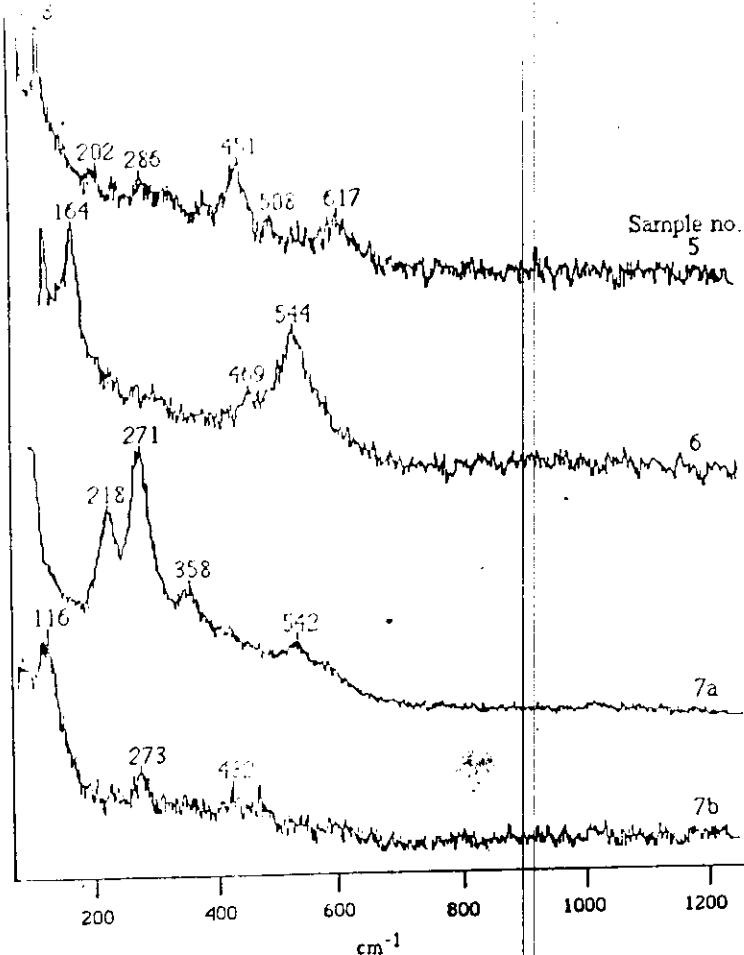
A single sample obtained from a CuO flux was analysed in four different positions. The Raman spectra from each position are significantly different.



Raman spectra of crystals 2, 3 and 4, obtained from CuO flux. Samples 3 and 4 were from the same batch, sample 2 from another.

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Table 1
Summary of results with chemical phases



Raman spectra of three crystals obtained by growth from a KCl flux. Samples 6 and 7 were from the same batch whereas sample 5 was from a separate batch. Spectrum (b) from sample 7 was recorded with a defocussed spot, averaging a larger area.

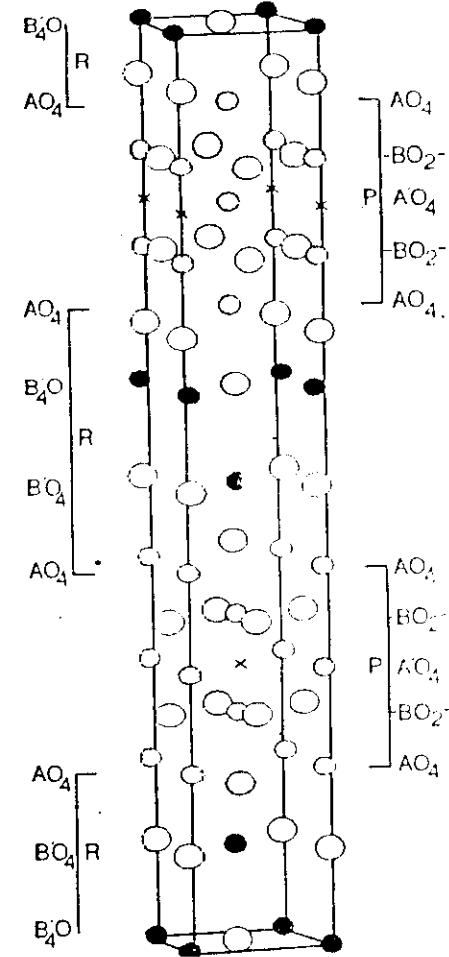
Sample no./Point	EMPA Atom % Bi/Sr/Ca/Cu/O	Total w/%	Phase	Sample no./Point	RAMAN Frequency (cm ⁻¹)
1/1	32:12:3:12:41	93		1/a	113, 299, 461, 631
1/2	32:10:2:12:44	91		1/b	111, 165, 541
1/3	14:14:5:14:52	95	Bi ₂ Sr ₂ CaCu ₂ O ₈	1/c	111, 165, 541
1/4	30:12:3:9:47	90		1/d	304, 477, 622
3/2	14:10:5:13:57	93	Bi ₂ Sr ₂ CaCu ₂ O ₈	3/b	107, 172, 293, 452, 623
3/3	14:14:5:14:52	95	Bi ₂ Sr ₂ CaCu ₂ O ₈		
3/4	18:17:7:18:40	93			
4/3	0:13:14:30:43	90	SrCaCu ₃ O ₄	4/a	107, 152, 230, 264, 539, 572, 956
4/4	0:16:15:33:37	92			
5				5/a	108, 451, 617
6/2	20:11:12:16:41	90		6/a	164, 544
7				7/a	218, 271, 358, 542
7				7/a	116, 273 (defocussed)

-Group theory of $\text{Bi}_2\text{Sr}_2\text{Ca}\text{Cu}_2\text{O}_8$

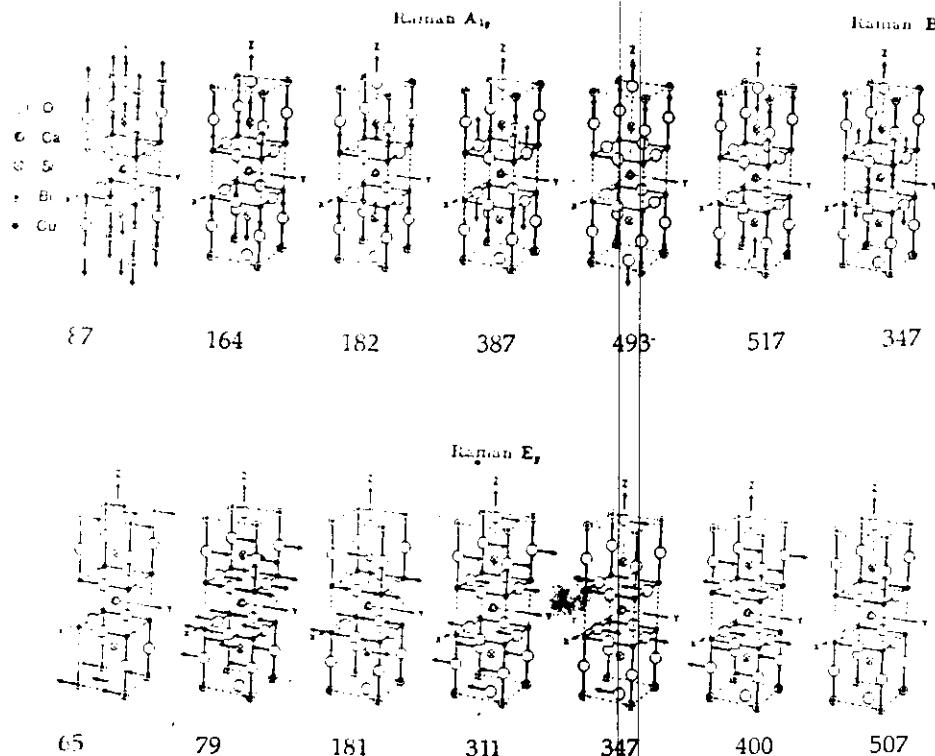
- Factor group- D_{2h}
- Symmetry- Distorted tetragonal
- Raman active vibrational modes-

$$6A_{1g} + B_{1g} + 7E_g$$

A' (Ca) ○
 A (Sr) ○
 B' (Bi) ●
 B (Cu) ○
 Oxygen ○
 oxygen vacancy ✕



Structural unit cell of $\text{Bi}_2\text{Sr}_2\text{Ca}\text{Cu}_2\text{O}_8$
showing perovskite octahedra (P) and rock
salt octahedra (R)



PHYSICAL REVIEW B

VOLUME 39, NUMBER 4

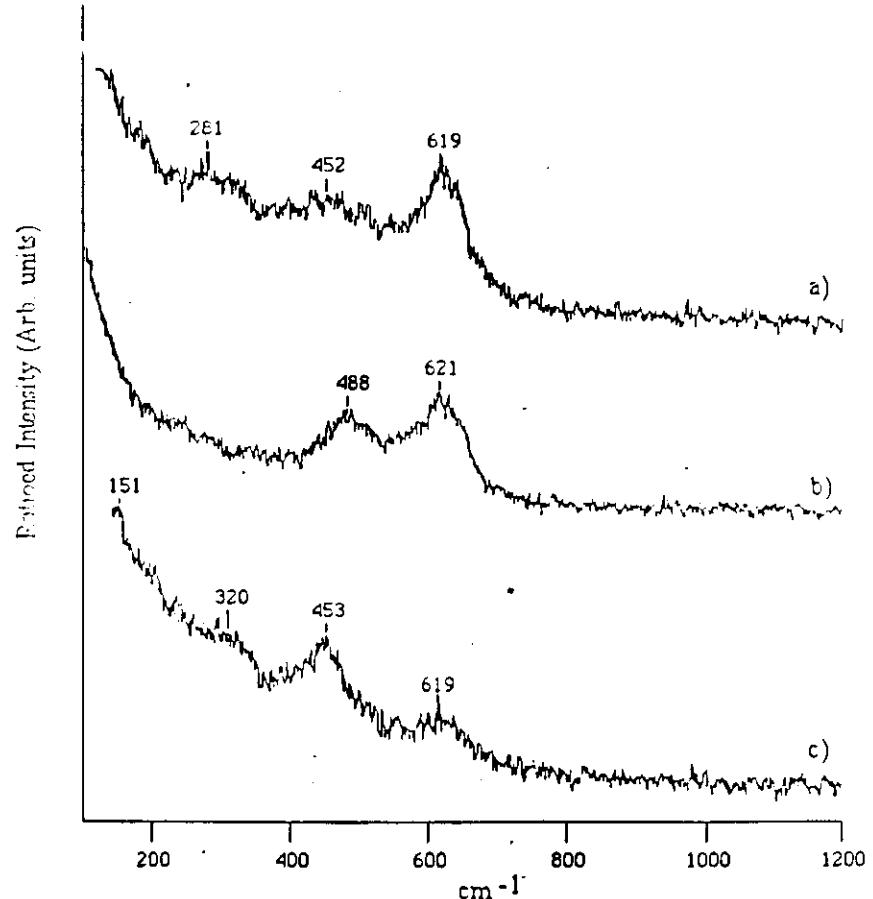
1 FEBRUARY 1989

Calculation of Raman- and infrared-active modes of $\text{Bi}_2\text{CaSr}_2\text{Cu}_3\text{O}_8$

J. Prade,* A. D. Kulkarni, and F. W. de Wette
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Universität Regensburg, D-8400 Regensburg, Federal Republic of Germany

W. Kress
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(Received 6 September 1988)

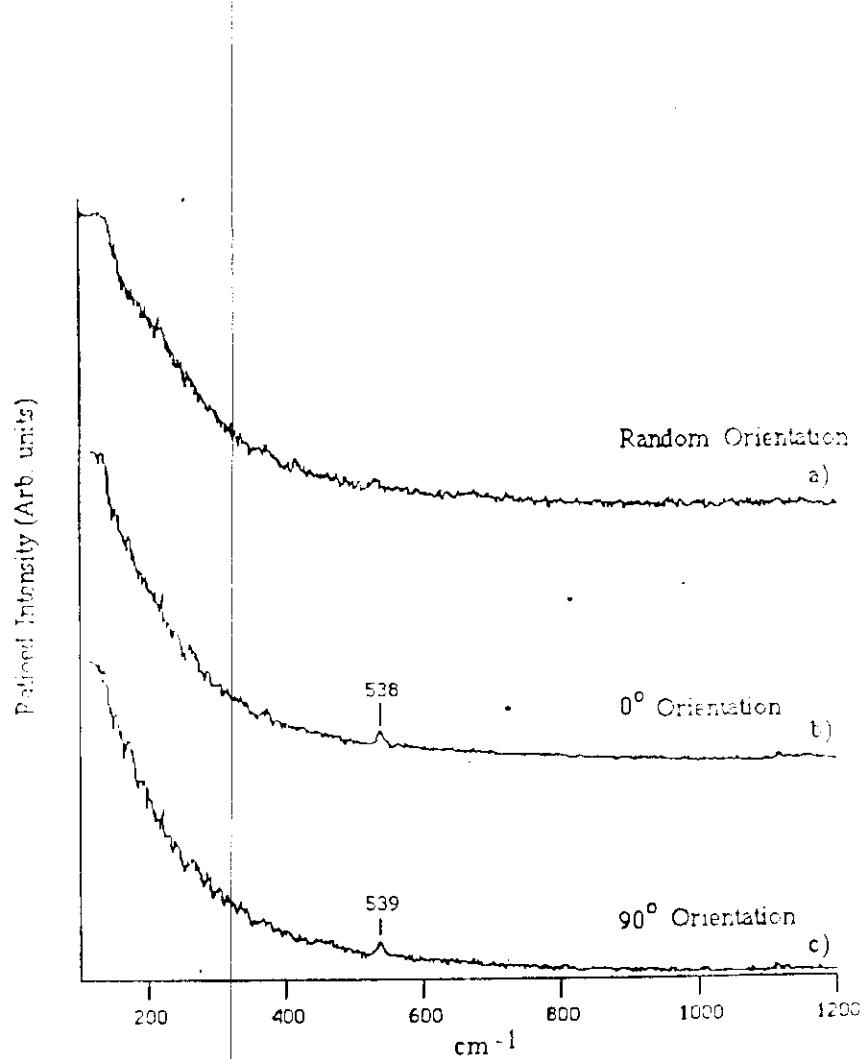


Raman excitations observed after exposure to laboratory atmosphere of single crystal face. Curves (a), (b), (c) were obtained from different, but randomly chosen, locations.

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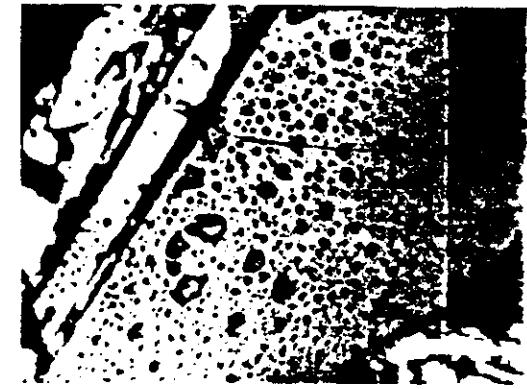
Micrograph of freshly cleaved crystallite face, analysed by Raman microscopy (length bar=80 μm).



Raman spectra for a freshly cleaved single crystal at
(a) random ab plane orientation, and (b) and (c) 90° rotation
in the ab plane with respect to a crystal edge (see
Figure 2).



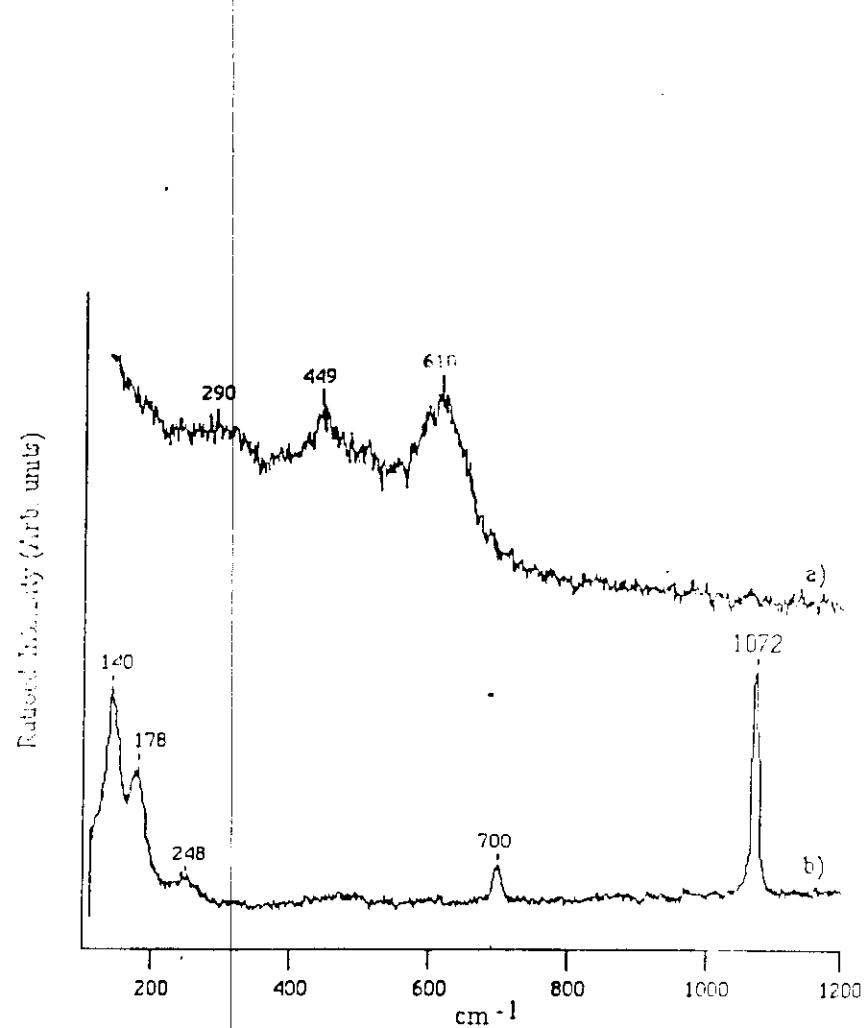
Micrograph of crystallite face after 0.5 hr saturated steam attack.



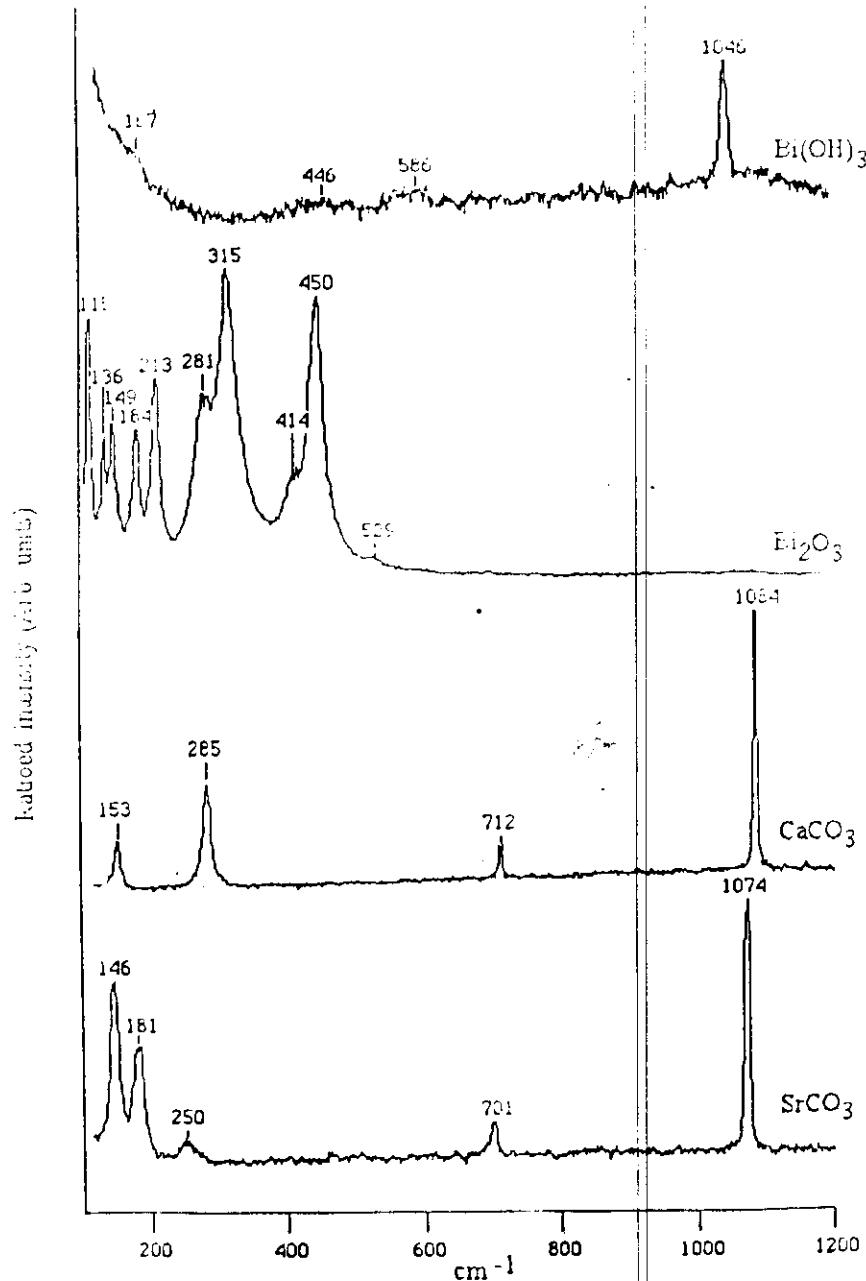
Micrograph of crystallite face after 10 hrs of saturated steam attack.



SEM image of crystallite face after 10hrs of saturated steam attack.



Raman features observed during the early (a) and late (b) stages of degradation by saturated steam.



THEORY OF RAMAN SCATTERING

$$\underline{P} = \underline{\alpha} \cdot \underline{e}$$

\underline{P} induced dipole vector
 $\underline{\alpha}$ polarisability tensor
 \underline{e} laser electric vector

Generally,,

x	a b c	x
y	d e f	y
z	g h i	z

-For phonon scattering, α is always symmetric

-Every vibrational mode has a particular scattering tensor, whose symmetry reflects that of the mode

The results of Raman analysis of model degradation compounds.
 The sharp feature at 1046 cm^{-1} in the spectrum of $\text{Bi}(\text{OH})_3$, is due to nitrate contamination.

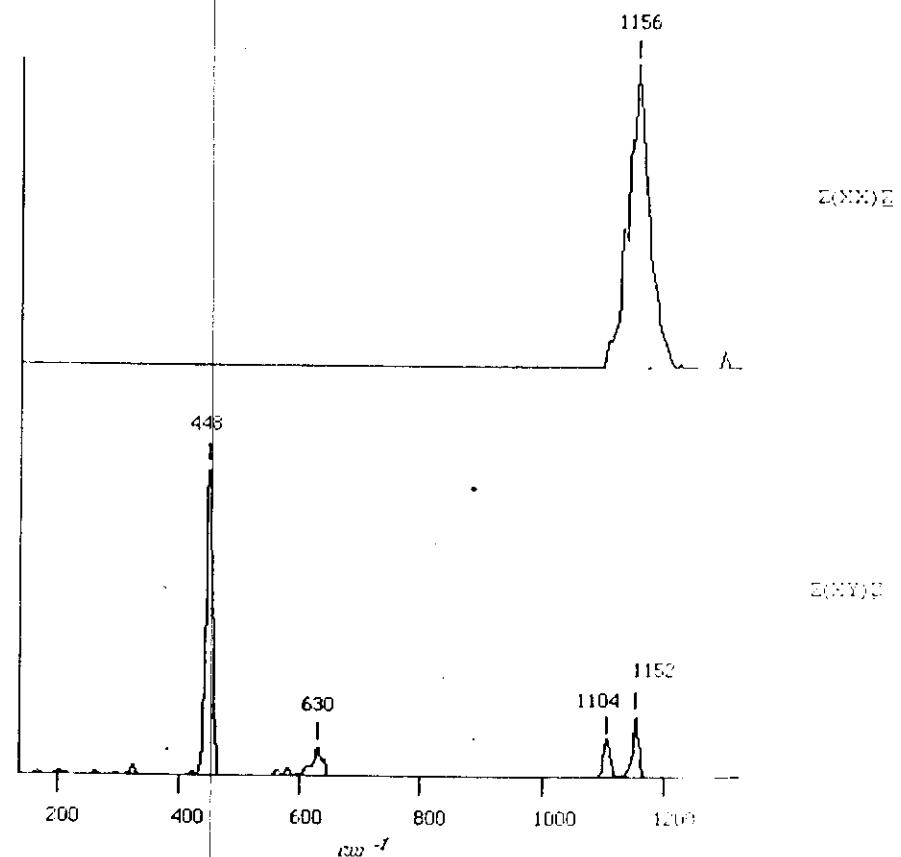
EXPERIMENTAL DETAILS

- Align crystal with respect to laboratory axes and laser beam electric vector with respect to the Raman microscope
- PORTO notation defines the experiment:

A (B C) D

A- direction of incident laser beam
 D- direction of collection of scattered light
 B- polarisation of laser electric vector
 C- polarisation of scattered light analyser

- We can now select the elements of the polarisability tensor detected and hopefully identify the Raman vibrations as specific modes.



Polarised Raman spectra of single crystal uranium dioxide. Porto notation to show the experimental conditions.

URANIUM DIOXIDE

STRUCTURE: Fluorite

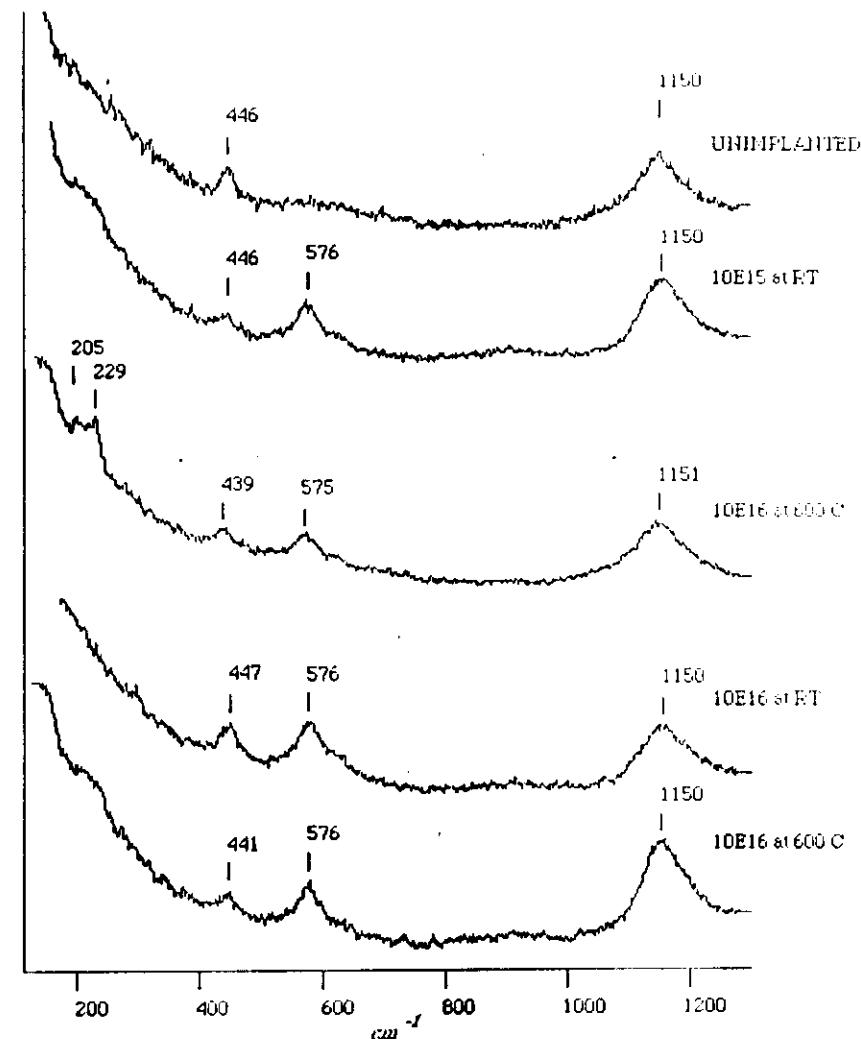
SPACE GROUP: O_h^5

Group theory : $T_{2g} + T_{1u}$ vibrational modes

Scattering tensor α :

$$T_{2g} = \begin{matrix} 000 & 00d & 0d0 \\ 00d & 000 & d00 \\ 0d0 & d00 & 000 \end{matrix}$$

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Raman spectra of Kr ion implanted uranium dioxide pellets. Implantation was carried out at the indicated doses (cm^{-2}), at 200 KeV energy and at the temperatures given. The calculated range of the implant is 400 Å.

3. RESULTS

Figure 4 shows the measured dispersion curves for the symmetric directions [001], [330], and [633], or, in group-theory notation (see Appendix I), Δ , Σ , and Λ . The frequencies of some selected normal modes are given in Table I. The experimental uncertainty is generally ~ 2 or 3% in frequency, but may be considerably larger for the branches of highest frequency.

TABLE I

Frequencies of selected normal modes of vibration in uranium dioxide at 296°K (No space group symbol is shown for the T_{1g} modes at $q = 0$ which are split by the macroscopic electric field)

Wave vector coordinates (k_1, k_2, k_3)	Mode	Frequency (units 10^7 c.p.s.)
(0, 0, 0)	(Longitudinal) T_{1g}	16.7 ± 0.60 13.42 ± 0.16
	(Transverse)	8.53 ± 0.11
		446 cm^{-1}
(0, 0, 1, 0)	M_1	15.2 ± 1.0
	M_1'	13.68 ± 0.21
	M_2	6.98 ± 0.12
	M_3	5.27 ± 0.05
	M_3'	5.27 ± 0.05
(0, 1, 0, 1, 0)	$S_1(01)$	16.8 ± 0.7
	$S_1(02)$	14.0 ± 0.5
	$S_2(01)$	9.83 ± 0.29
	$S_2(02)$	6.49 ± 0.11
	$S_3(01)$	4.64 ± 0.07
	$S_3(02)$	4.24 ± 0.10
(0, 5, 0.5, 0, 5)	T_1'	17.2 ± 1.2
	T_1''	12.6 ± 0.4
	T_2'	12.2 ± 0.4
	T_2''	10.57 ± 0.27
	T_3'	5.21 ± 0.10
	T_3''	2.73 ± 0.04

It is interesting to note that the longitudinal acoustic ($\Delta_1(\Lambda)$) and the longitudinal optic (Δ_1') branches are accidentally degenerate at the zone boundary (M) within the limits of experimental error, a fact which impeded precise frequency measurements in this region.

The steeply rising Δ_1' branch shows that there is no gap in the frequency distribution between the optic and the acoustic modes such as one might expect as a result of the large difference in mass between oxygen and uranium. Detailed calculations of the distribution (Fig. 5, Section 4) show that there is probably no gap in any part of the frequency spectrum.

The initial slopes of the acoustic branches are consistent with independent measurements of the elastic constants (Wachtman *et al.* 1965), and the frequencies of the TO and LO modes at $q = 0$ are consistent with the measured static (Briggs 1964) and high-frequency (Ackermann *et al.* 1959) dielectric constants (ϵ_0 and ϵ_∞ respectively) through the relation (Lyddane *et al.* 1941)

$$(3) \quad \left(\frac{\nu_{TO}}{\nu_{LO}} \right)^2 = \frac{\epsilon_0}{\epsilon_\infty}$$

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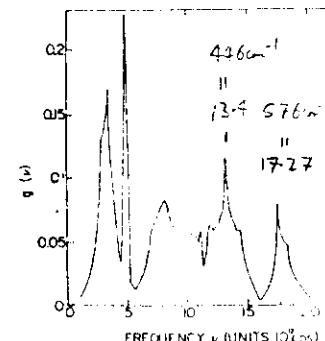


FIG. 5. The frequency distribution function calculated from the best-fit shell model (Model III). The statistical fluctuations in the curve (which is, in fact, a histogram) are too small to be shown in this figure.

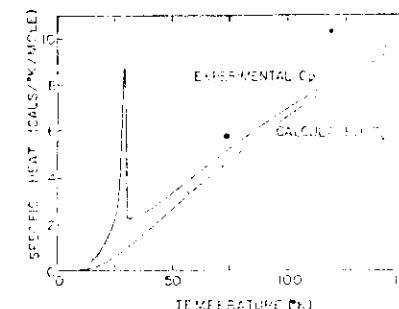


FIG. 6. The measured total specific heat, C_p , and the calculated lattice specific heat, C_s , of UO_2 . The observed anomaly at 90°K is due to antiferromagnetic ordering.

measurements. A surprising feature, as shown by Fig. 6, is the magnitude of the magnetic specific heat above the antiferromagnetic ordering temperature, T_c . This suggests that there is considerable short-range magnetic order well above T_c . The contribution of the normal modes to the entropy at 298°K is calculated to be 16.12 cal/K mole, whereas that measured by Lave *et al.* (1952) is 18.63 cal/K mole. The discrepancy between the calculated specific heat, C_s , and the measured C_p at 250°K is 0.60 cal/K mole. The discrepancy probably arises from the difference between C_p and C_s , the antiferromagnetic contribution to the specific heat, and the errors in the measurements and the

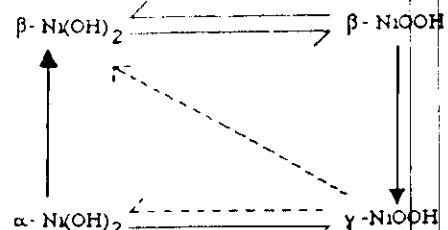


Fig. 1. Nickel oxyhydroxide electrode reaction scheme.

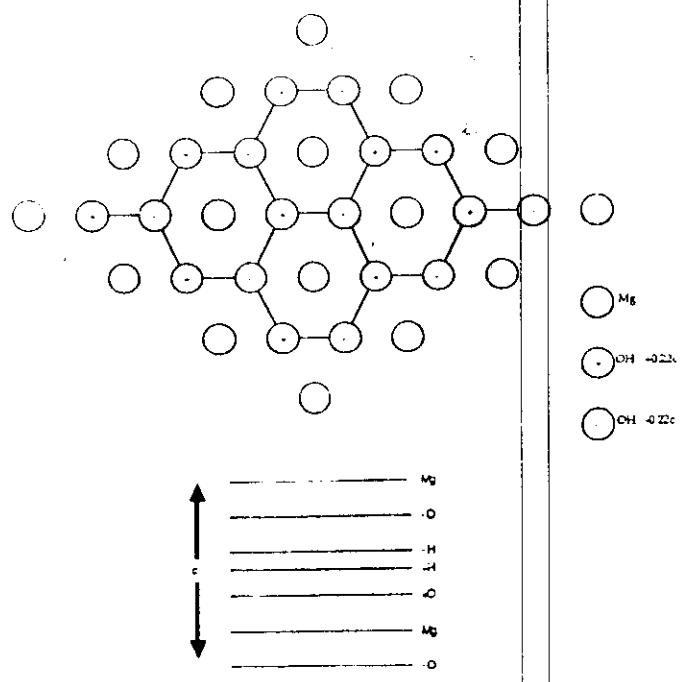
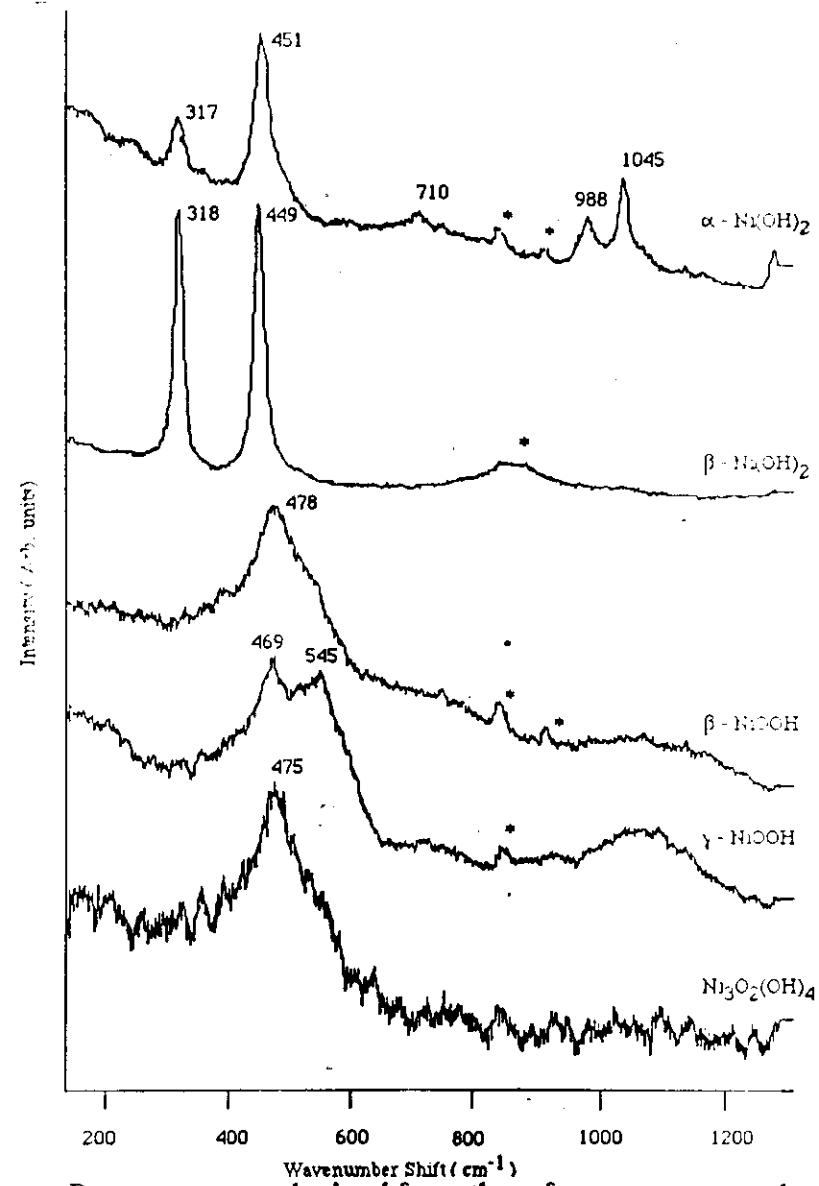
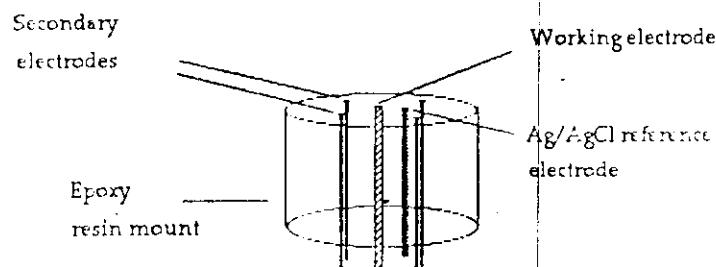


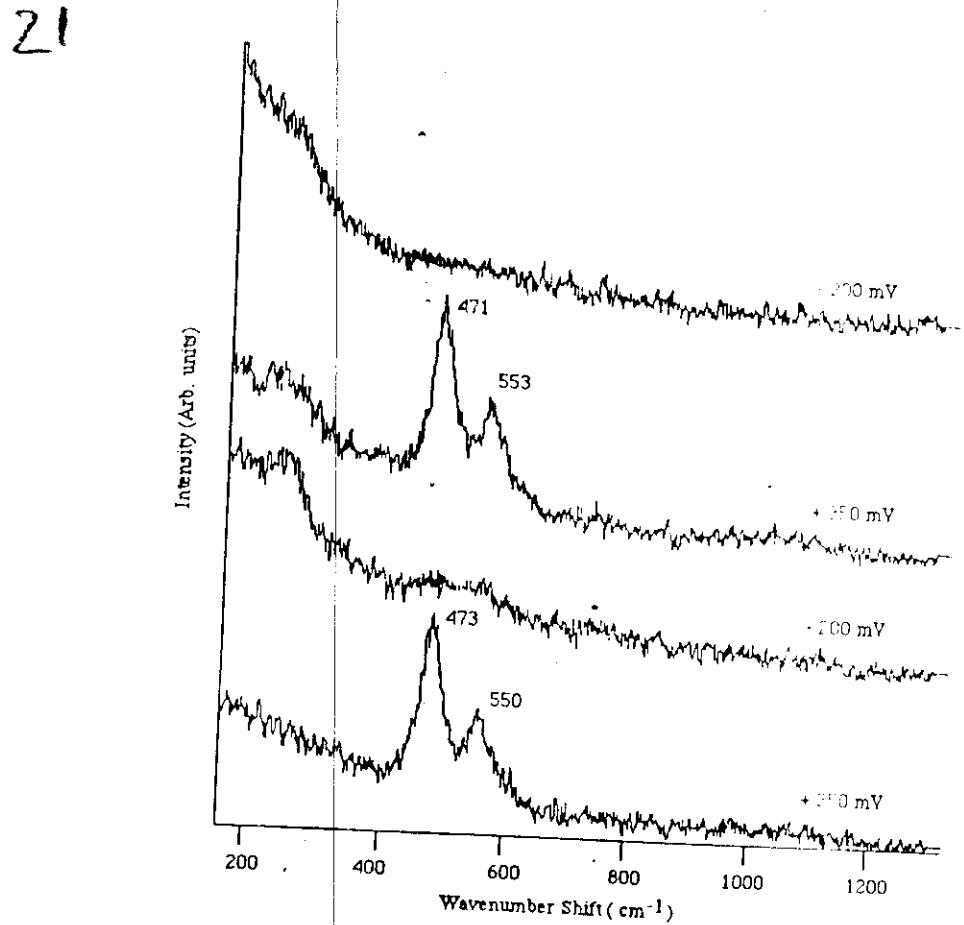
Fig. 2. Graphical representations of the common Brønsted structure $M_6(OH)_8$, showing the hexagonal arrangement of the hydroxyl ions around the metal ion and the layer geometry.



Raman spectra obtained from the reference compounds
 $\alpha\text{-Ni(OH)}_2$, $\beta\text{-Ni(OH)}_2$, $\beta\text{-NiOOH}$, $\gamma\text{-NiOOH}$
and $Ni_3O_2(OH)_4$ * LRM objective bands.



Electrochemical cell for **insitu** Raman measurements



Raman spectra obtained from the surface of a cycled Nickel oxyhydroxide electrode (NOE) with an electrolyte of 8 M KOH (aq).

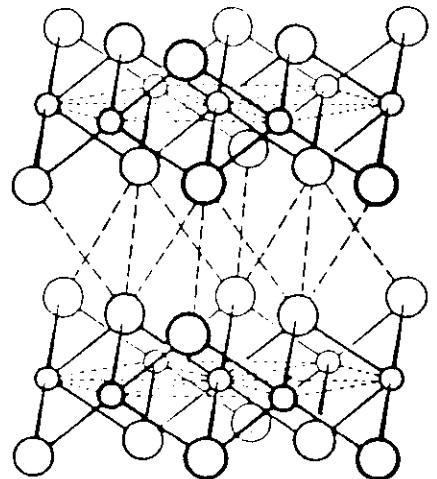


FIG. 9. Portions of two layers of the CdI₂ structure. The small circles represent the metal atoms.

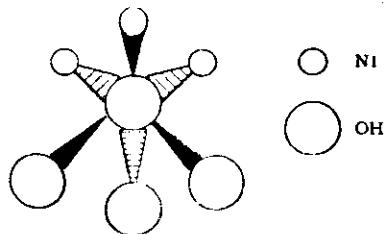


FIG. 10. Unit cell derived from the CdI₂ structure.

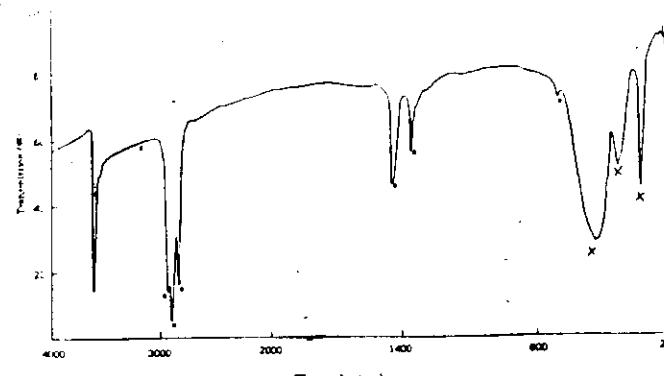


FIG. 5. Infrared spectrum of β -Ni(OH)₂. (0) Bands due to nujol.

C_{3v}	E	$2C_3$	$3\sigma_g$	$x^2 - y^2, z^2$
A_1	1	1	1^z	
A_2	1	1	$-1^z R_z$	
E	2	-1	0 ($x, y, z; R_x, R_y, R_z$)	$(x^2 - y^2, xy)(xz, yz)$

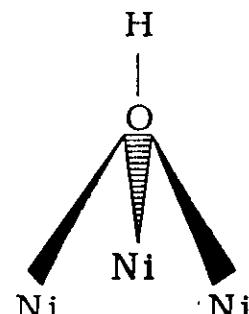


FIG. 11. The spectroscopic unit cell.

This gives the reducible representations, χ' :

C_{3v}	E	$2C_3$	$3\sigma_g$
χ_{Vib}^E	15	0	3
χ_{Trans}^E	3	0	1
χ_{Rot}^E	3	0	1

This reduces to give the irreducible representations, Γ' :

$$\Gamma_{\text{Vib}} = 4A_1 + A_2 + 5E$$

$$\Gamma_{\text{Rot}} = A_2 + E$$

$$\Gamma_{\text{Trans}} = A_1 + E$$

Now,

$$\Gamma_{\text{Vib}} = \Gamma_{\text{Vib}} + \Gamma_{\text{Rot}} + \Gamma_{\text{Trans}}$$

$$\rightarrow \Gamma_{\text{Vib}} = 3A_1 + 3E.$$

TABLE IV. Mode assignments for Raman and infrared bands of β -Ni(OH)₂.

Frequency (cm ⁻¹)	Activity	Symmetry type	Assignment
3630	IR	A_1	Asymmetric δ_{OH}
3580	IR, Raman	A_1	Symmetric δ_{OH}
553	IR	E	Deformation δ_{OH}
450	IR, Raman	A_1	ν_{HO}
350	IR	E	OH libration
318	Raman	E	Ni-OH lattice