



SMR.550 - 38

SPRING COLLEGE IN MATERIALS SCIENCE ON
"NUCLEATION, GROWTH AND SEGREGATION IN MATERIALS
SCIENCE AND ENGINEERING"
(6 May - 7 June 1991)

RADIATION INDUCED ORDERING

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These are preliminary lecture notes, intended only for distribution to participants.

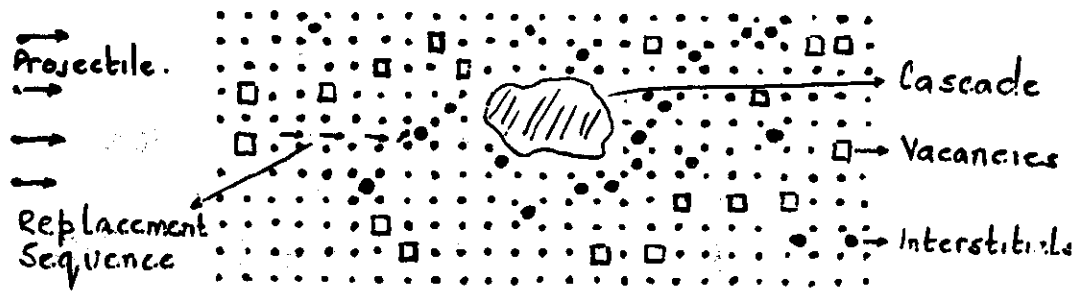
RADIATION INDUCED ORDERING.

1

- 1 A. • Pair Production / Cascade damage
- 2 • Displacements Per Atom (DPA)
- 3 • Differences between electron/ion and neutron irradiation
- 4 • Sink strength / Recombination Rate
- 5 • Replacement sequences
- 6 • Bias
- 7 B • VOID / VACANCY LOOP / INTERSTITIAL LOOP PRODUCTION
- 8 • RATE THEORY - RADIATION DAMAGE
- 9 • LINEAR STABILITY ANALYSIS
- 10 • CRITICAL POINT / SOFT MODE / BIFURCATION
- 11 • DEFECT INTERACTIONS.
- 12 • FLUCTUATIONS.
- 13 C. • Void Lattices
- 14 • Bubble Lattices
- 15 • Defect Walls
- 16 • Spinodal Decomposition
- 17 • Phase Stability of alloys during irradiation.

2.

THE IRRADIATION PROCESS.



Concentration of vacancies / interstitials

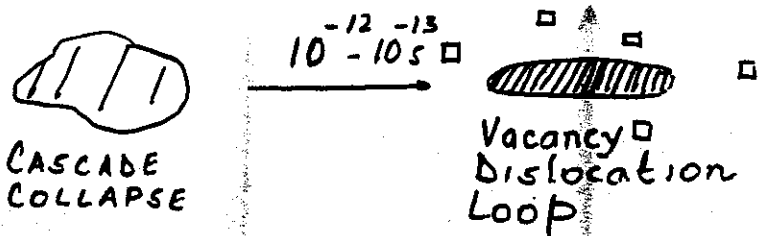
$$v / i$$

Production Rate

$$K \quad (\text{DPA})$$

Recombination Rate

$$\propto v i$$



Cascade efficiency

$$\epsilon$$

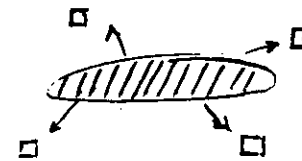
Production Rate for Cascade

$$\epsilon K$$

Vacancy Production Rate

$$K(1-\epsilon)$$

VACANCY LOOP DYNAMICS.

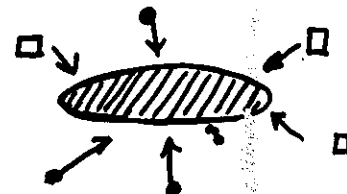


Vacancy Thermal emission Rate

$$\left(\frac{dQ_L}{dt} \right)_{\text{Thermal}} = - D_v C_v^{\text{eq}} p_L$$

$$p_L = \exp \left[\left(\gamma_{sf} + F_{el}(r_L) b^2 / k_B T \right) \right]$$

$$F_{el}(r_L) = (\mu b^2 / 4\pi(1-\nu)) \ln[(r_L + b)/b] / (r_L + b)$$



Interstitial & Vacancy Absorption

$$\left(\frac{dQ_L}{dt} \right)_{\text{Absorption}} = p_L (D_v v - Z_i D_i i)$$

Bias

$$S_L = 2 \pi N_L r_L$$

Density $\rightarrow$ Sink term

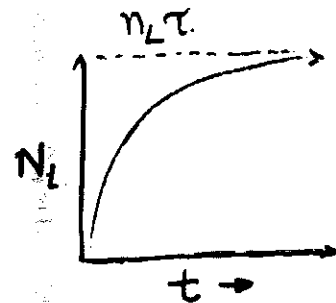
BIAS Z or Z_i or $\Delta Z = Z - 1$



Vacancy loop life time τ

$$\frac{dN_L}{dt} = n_L - \frac{N_L}{\tau}$$

$$1. N_L = n_L \tau (1 - e^{-t/\tau})$$

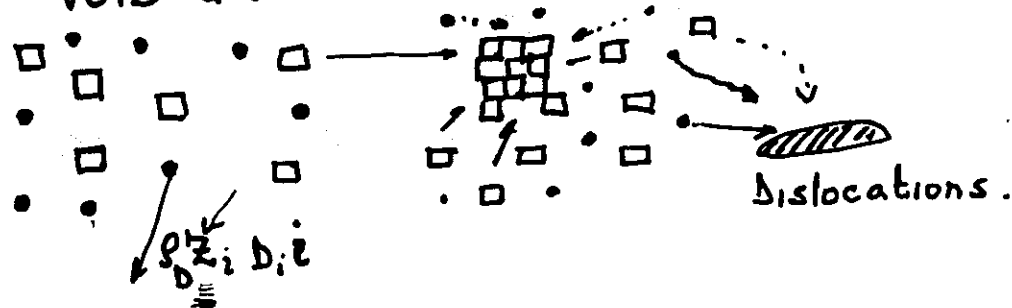


$$\tau = -r_{L0} / \left(\frac{dr_L}{dt} \right)_{r_{L0}}$$

$$Q_L = \pi r_L^2 N_L / b^2 \quad 2. S_L = 2\pi N_L r_L$$

$$\frac{dQ_L}{dt} = \underbrace{\epsilon K}_{\text{Production}} + \underbrace{S_L (D_v v - Z; D_i i)}_{\text{Bias}} - \underbrace{S_L D_v c_v^{eq} p_s}_{\text{Thermal}}$$

VOID GROWTH MECHANISM



$$\frac{dQ_s}{dt} = \underbrace{S_s (D_v v - D_i i)}_{\text{Irradiation term}} - \underbrace{S_s D_v c_v^{eq} p_s}_{\text{Thermal}}$$

$$p_s = \exp \left[\left(\frac{2\gamma}{r_s} - P_g \right) / k_B T \right] \quad S_s = 4\pi N_s r_s \quad Q_s = \frac{4\pi r_s^3}{3 b^3}$$

Growth of bubbles:

$$i=0 \quad v = c_v^{eq}$$

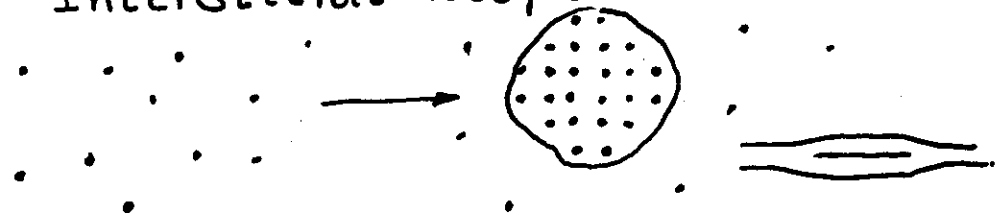
$$\frac{dQ_s}{dt} = S_s D_v c_v^{eq} [1 - p_s]$$

$$p_s > 1 \quad \text{shrinkage} \quad \frac{2\gamma}{r_s} > P_g$$

$$p_s < 1 \quad \text{growth} \quad \frac{2\gamma}{r_s} < P_g$$

$$p_s = 1 \quad \text{Equilibrium} \quad \frac{2\gamma}{r_s} = P_g$$

Interstitial Loops



$$\frac{dQ_I}{dt} = S_I (-D_v v + Z D_i i) + S_I D_i c_v^{eq} p_I$$

$$p_I = \exp \left[- \left(\gamma_{sf} + F_{el}(r_L) b^2 \right) / k_B T \right]$$

$$S_I = 2\pi N_I r_I \quad Q_I = \pi r_I^2 N_I / b^2$$

Point Defect kinetics

$$\beta_v = D_v (\rho_s + \rho_L + \rho_I)$$

$$\beta_i = D_i (\rho_s + Z_i (\rho_L + \rho_I))$$

VOIDS
↓
Neutral
Sinks

DISLOCATIONS
↓
Biased
Sinks:

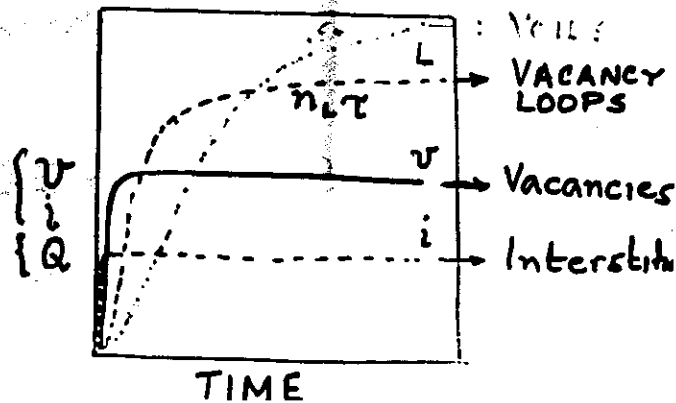
$$\frac{dv}{dt} = k_v - \beta_v v - \alpha v i$$

$$\frac{di}{dt} = \underbrace{k_i}_{\text{Production}} - \underbrace{\beta_i i}_{\text{Sink}} - \underbrace{\alpha v i}_{\text{Recombination}}$$

FAST MODES AND SLOW MODES

$$\frac{dv}{dt} / \frac{di}{dt}$$

$$\frac{dQ}{dt}$$



6.

Steady State Solution

$$\frac{di}{dt} = 0$$

$$\frac{dv}{dt} = 0$$

Sink Dominant Regime

$$\frac{dv}{dt} = k_v - \beta_v v = 0$$

$$\frac{di}{dt} = k_i - \beta_i i = 0$$

$$\bar{v} = \frac{k_v}{\beta_v} = \frac{k_v}{D_v \rho_{\text{terms}}}$$

$D_v \rightarrow \text{Small}$

$$\bar{i} = \frac{k_i}{\beta_i} = \frac{k_i}{D_i \rho_{\text{terms}}}$$

$D_i \rightarrow \text{Large}$

$$\bar{v} \gg \bar{i}$$

Recombination Dominant Regime

$$\frac{dv}{dt} = k_v - \alpha v i = 0$$

$$\frac{di}{dt} = k_i - \alpha v i = 0$$

If $k_v = k_i = k$ $\underline{v = i}$

Provided $(v - i) = 0$ at time $t = 0$

$$v = i = \sqrt{\frac{k}{\alpha}}$$

What do we have to do?

First we look at the problem of production and recombination of defects

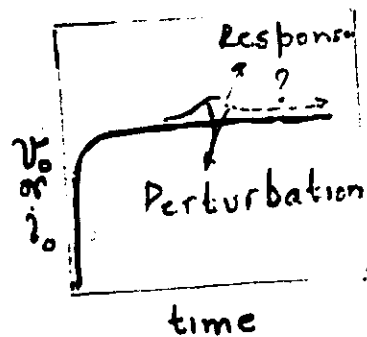
$$\left. \begin{aligned} \frac{dv}{dt} &= K - \alpha v i \\ \frac{di}{dt} &= K - \alpha v i \end{aligned} \right\} \rightarrow \begin{aligned} C_- &= i - v \\ C_+ &= i + v \end{aligned}$$

INITIAL CONDITION $t=0$

$$C_- = i - v = D \quad C_+ = i + v = E$$

What is the nature of the solutions?

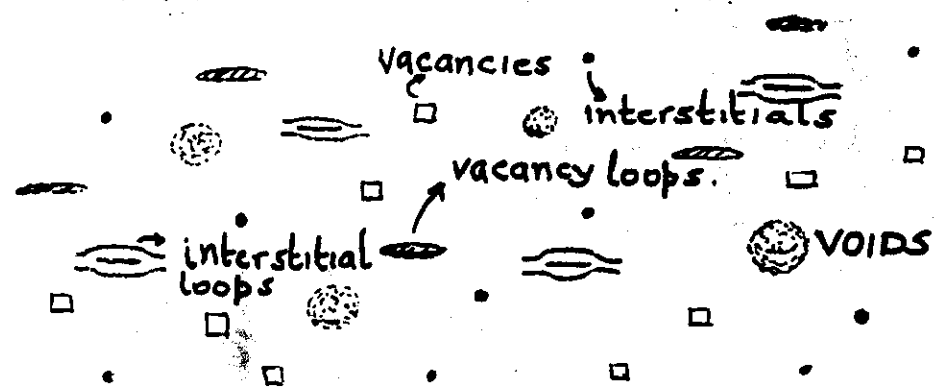
Examine the Steady state solution
→ Stability Aspects



$$\begin{aligned} v &= v_0 + \delta v' \exp \omega t \\ i &= i_0 + \delta i' \exp \omega t \end{aligned}$$

Spatial Dependence $\delta v' = \delta v'' e^{ik \cdot r}$

RADIATION INDUCED ORDERING - 2



$$\frac{dv}{dt} = K(1-\epsilon) - \beta_v v - \alpha v i + TE$$

$$\frac{di}{dt} = K - \beta_i i - \alpha v i$$

$$\beta_v = S_S + S_L + S_I$$

$$\beta_i = S_S + Z(S_L + S_I)$$

$$\frac{dQ_L}{dt} = \epsilon K + S_L(D_v v - Z D_i i) + TE$$

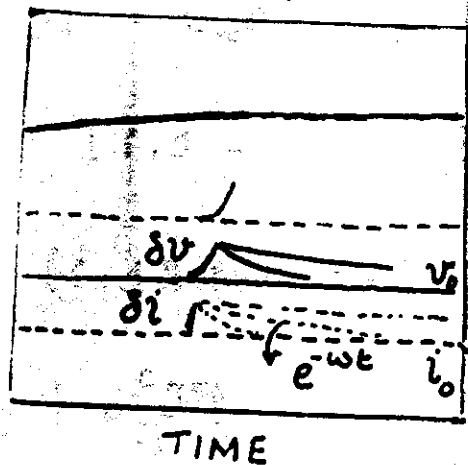
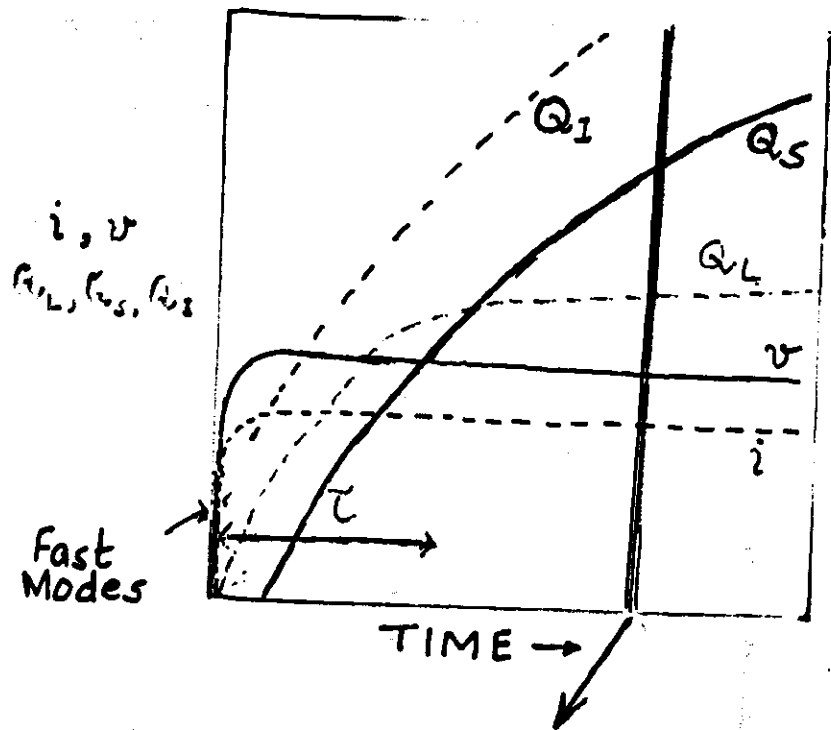
$$\frac{dQ_S}{dt} = S_S(D_v v - D_i i) + TE$$

$$\frac{dQ_I}{dt} = S_I(-D_v v + Z D_i i) + TE$$

$$\begin{aligned} Q_{L,I} &= \pi \Gamma_{L,I}^2 N_{L,I} / b^2 \quad Q_S = \frac{4\pi \Gamma_S^3 N_S}{b^3} \quad \left| \quad \frac{dN_L}{dt} = n_L - \frac{N_L}{\tau} \right. \\ S_{L,I} &= 2\pi N_{L,I} \Gamma_{L,I} \quad S_S = 4\pi N_S \Gamma_S \end{aligned}$$

$$\frac{di}{dt} - \frac{dv}{dt} = \frac{dG_L}{dt} + \frac{dG_S}{dt} - \frac{dG_I}{dt} = 0$$

$$(i-v) = Q_L + Q_S - Q_I = \text{Const}$$



10

$$\frac{di}{dt} = k - \beta_i i$$

$$\frac{dv}{dt} = k - \beta_v v$$

$$i = i_0 + \delta i' e^{\omega t}$$

$$v = v_0 + \delta v' e^{\omega t}$$

$$\omega e^{\omega t} \delta i' = -\beta_i \delta i' e^{\omega t}$$

$$\omega e^{\omega t} \delta v' = -\beta_v \delta v' e^{\omega t}$$

$$\begin{pmatrix} \beta_i + \omega & 0 \\ 0 & \beta_v + \omega \end{pmatrix} \begin{pmatrix} \delta i' \\ \delta v' \end{pmatrix} = 0$$

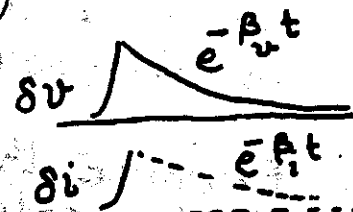
$$\omega = \begin{cases} \omega_1 = -\beta_i \\ \omega_2 = -\beta_v \end{cases}$$

Sink
Dominant

$$k - \beta_i i_0 = 0$$

$$k - \beta_v v_0 = 0$$

Steady State



$$\frac{di}{dt} = k - \alpha v i$$

$$\frac{dv}{dt} = k - \alpha v i$$

$$C_- = i - v = -3 = 0 \text{ to } 10$$

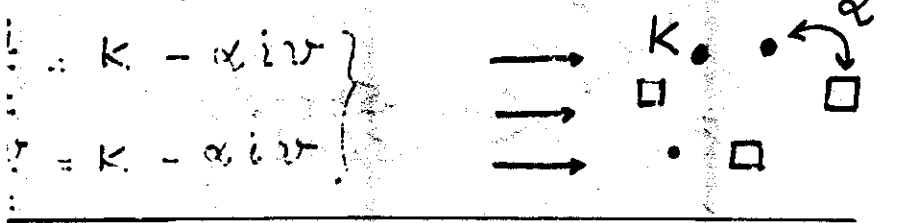
$$C_+ = i + v = 7.6$$

RECOMBINATION
DOMINANT
REGIME



POINT DEFECT CONCENTRATIONS, REACTIVE-LIFETIME BEHAVIOUR, DEFECT REACTIONS

Point Defect Concentrations during irradiation A simple model.



$$= 1 \quad \text{Initial conditions: } t=0 \quad \begin{aligned} C_- = i-v &= D \\ C_+ = i+v &= E \end{aligned}$$

$$\frac{dC_-}{dt} = 0 \quad \frac{dC_+}{dt} = -\left(\frac{\alpha}{2}\right)C_+^2 + (\alpha D^2 + 4K)/2 \quad \text{Riccati Equation}$$

$$C_+ = \left(D^2 + \frac{4K}{\alpha}\right)^{1/2} \frac{E + (D^2 + \frac{4K}{\alpha})^{1/2} \tanh \frac{1}{2} \alpha (D^2 + \frac{4K}{\alpha})^{1/2} t}{E \tanh \frac{1}{2} \alpha (D^2 + \frac{4K}{\alpha})^{1/2} t + (D^2 + \frac{4K}{\alpha})^{1/2}}$$

the asymptotic Limit $t \rightarrow \infty$

$$(v_0) = \left(\frac{D}{2} + \frac{1}{2} \sqrt{D^2 + \frac{4K}{\alpha}}\right), -\frac{D}{2} + \frac{1}{2} \sqrt{D^2 + \frac{4K}{\alpha}}$$

$$\approx \left(\sqrt{\frac{K}{\alpha}}, \sqrt{\frac{K}{\alpha}}\right), D=0, di/dt = dv/dt = 0$$

examine these solutions near a critical point (i_0, v_0)

$$i = i_0 + \delta i' \exp \omega t$$

$$v = v_0 + \delta v' \exp \omega t$$

Linearization of the equations.

$$\frac{di}{dt} = K - \alpha i v \quad i = i_0 + \delta i$$

$$\frac{dv}{dt} = K - \alpha i v \quad v = v_0 + \delta v$$

$$\frac{d \delta v}{dt} = -\alpha i_0 \delta v - \alpha v_0 \delta i \quad v_0 = \frac{D}{2} + \frac{1}{2} \sqrt{D^2 + \frac{4K}{\alpha}}$$

$$\frac{d \delta i}{dt} = -\alpha i_0 \delta v - \alpha v_0 \delta i \quad v_0 = -\frac{D}{2} + \frac{1}{2} \sqrt{D^2 + \frac{4K}{\alpha}}$$

$$\frac{d}{dt} \begin{pmatrix} \delta v \\ \delta i \end{pmatrix} = \begin{pmatrix} -\alpha i_0 & -\alpha v_0 \\ -\alpha v_0 & -\alpha i_0 \end{pmatrix} \begin{pmatrix} \delta v \\ \delta i \end{pmatrix}$$

$$\delta i = \lambda_1 e^{\omega_1 t} \quad \delta v = \lambda_2 e^{\omega_2 t}$$

$$\begin{pmatrix} +\alpha i_0 + \omega & \alpha v_0 \\ \alpha v_0 & \alpha i_0 + \omega \end{pmatrix} \begin{pmatrix} \delta i' \\ \delta v' \end{pmatrix} = 0$$

Solutions exist only if

$$\det \begin{vmatrix} \alpha i_0 + \omega & \alpha v_0 \\ \alpha v_0 & \alpha i_0 + \omega \end{vmatrix} = 0$$

$$\det \begin{vmatrix} \omega + \alpha(-\frac{D}{2} + \frac{1}{2}\sqrt{D^2 + \frac{4K}{\alpha}}) & \alpha(\frac{D}{2} + \frac{1}{2}\sqrt{D^2 + \frac{4K}{\alpha}}) \\ \alpha(-\frac{D}{2} + \frac{1}{2}\sqrt{D^2 + \frac{4K}{\alpha}}) & \omega + \alpha(\frac{D}{2} + \frac{1}{2}\sqrt{D^2 + \frac{4K}{\alpha}}) \end{vmatrix}$$

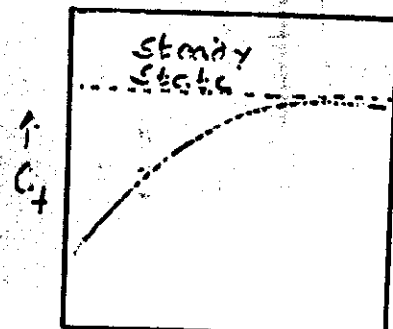
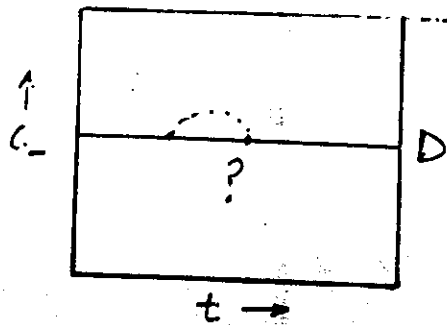
$$\omega_- = 0$$

$$\omega_+ = -\alpha \sqrt{D^2 + \frac{4K}{\alpha}}$$

$$i-v = c_- = D$$

$$i+v = c_+ = (D^2 + \frac{4K}{\alpha})^{1/2} \left[1 - \frac{2(E - \sqrt{D^2 + 4K/\alpha})}{E + \sqrt{D^2 + 4K/\alpha}} \right] \exp(\omega_+ t)$$

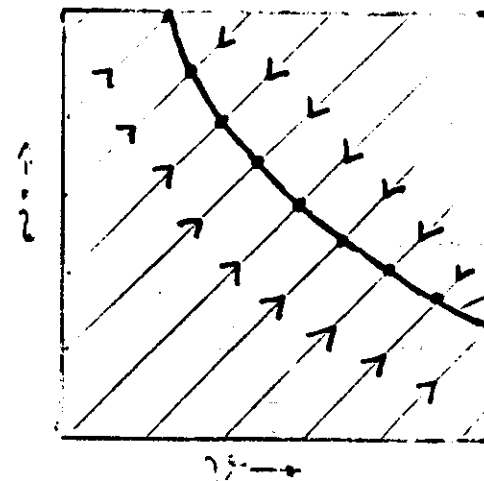
PHASE DIAGRAMS / SOFT MODE / AND
THE PHYSICAL INTERPRETATION
OF THE EQUATIONS.



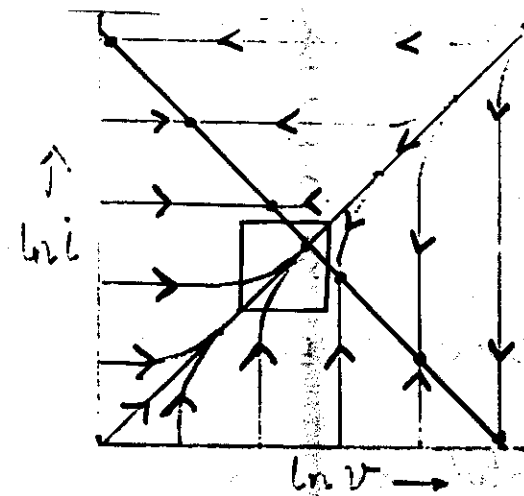
$$c_- = i - v \\ = \text{Constant} \\ \omega_- = 0 \text{ (Soft Mode)!}$$

$$c_+ = i + v \\ \sim e^{\omega_+ t} + (D^2 + \frac{4K}{\alpha})^{1/2}$$

PHASE DIAGRAM



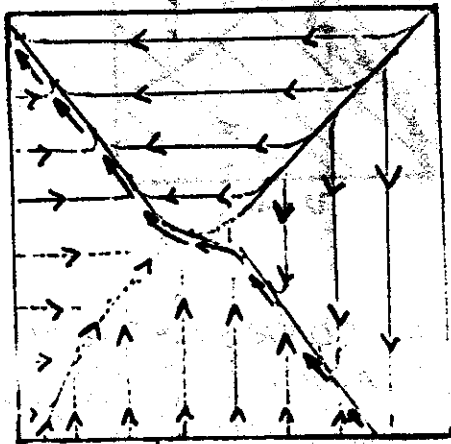
PRODUCTION AND
RECOMBINATION
 $\frac{di}{dv} = +1$



Phase diagram
ln scale

PHASE DIAGRAMS - CONT.

Unequal Production rates and recombination.



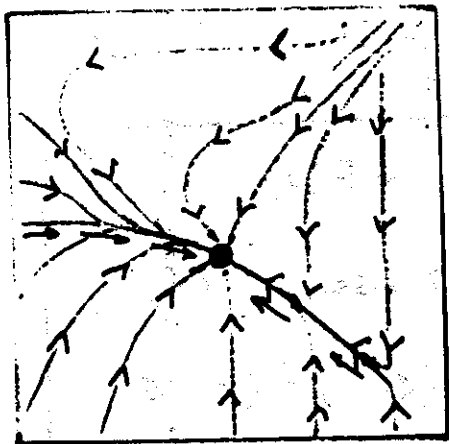
$$\frac{di}{dt} = k_i - \alpha i v$$

$$\frac{dv}{dt} = K_v - \alpha i v$$

$$\frac{dc_-}{dt} = i - v = k_i - k_r$$

$$c_- = (k_i - k_r)t + \Delta$$

unequal Production rates - recombination and loss to sinks.



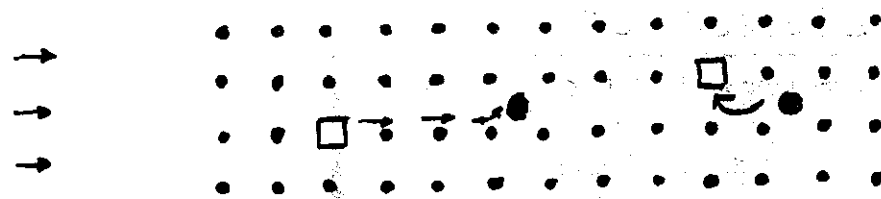
$$\frac{di}{dt} = k_i - \alpha i v - \beta_i i$$

$$\frac{dv}{dt} = k_v - \alpha i v - \beta_v v$$

$$i_0 = \frac{1}{2\alpha\beta_i} \left[(-\beta_i\beta_v + \alpha K_-) + \{(\beta_i\beta_v - \alpha K_-)^2 - 4\alpha\beta_i\beta_v k_i\}^{1/2} \right]$$

$$v_0 = \frac{1}{2\alpha\beta_v} \left[(-\beta_i\beta_v - \alpha K_-) + \{(\beta_i\beta_v + \alpha K_-)^2 - 4\alpha\beta_i\beta_v k_v\}^{1/2} \right]$$

FLUCTUATIONS / CORRELATED AND UNCORRELATED REACTIONS

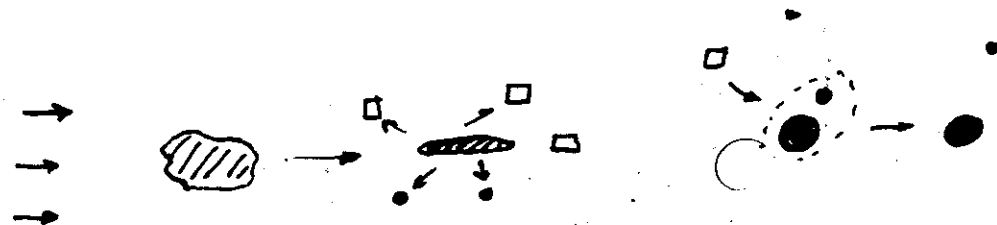


$$c_- = i - v$$

$$(8c_-)^2 = 0$$

$$\frac{dv}{dt} = k - \alpha v i$$

$$\frac{di}{dt} = k - \alpha v i$$



$$c_- = i - v$$

$$(8c_-^2) \neq 0$$

$$\Delta n \neq 0$$

→ Removal of a Constraint.

2. FLUCTUATIONS AND DEVIATION IN CONCENTRATION IN THE PRESENCE OF SOFT MODES

CORRELATED & UNCORRELATED REACTIONS.

1. Correlated reactions

$S \rightarrow$ Source term $\langle S \rangle = 0$

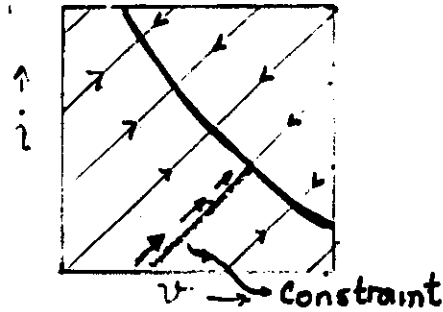
$S \xrightarrow{k} i + v$ Production in pairs

$i + v \xrightarrow{\alpha} S$ Recombination in pairs

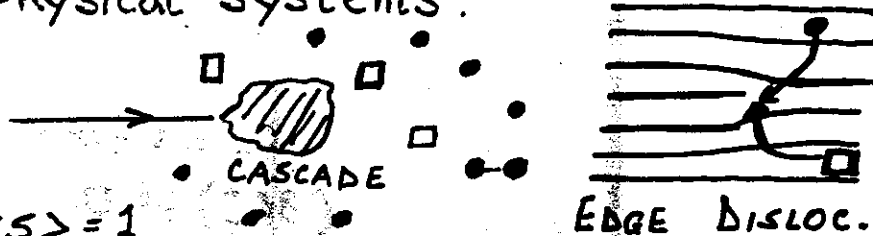
$$\delta i = i - \langle i \rangle, \langle \delta i^2 \rangle = ?$$

$$\delta v = v - \langle v \rangle, \langle \delta v^2 \rangle = ?$$

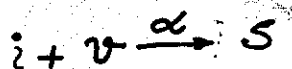
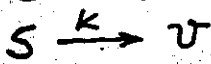
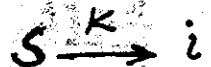
$$C = \begin{pmatrix} \langle \delta v^2 \rangle & \langle \delta v \delta i \rangle \\ \langle \delta i \delta v \rangle & \langle \delta i^2 \rangle \end{pmatrix} = 0$$



2. Uncorrelated reactions - actual Physical systems.



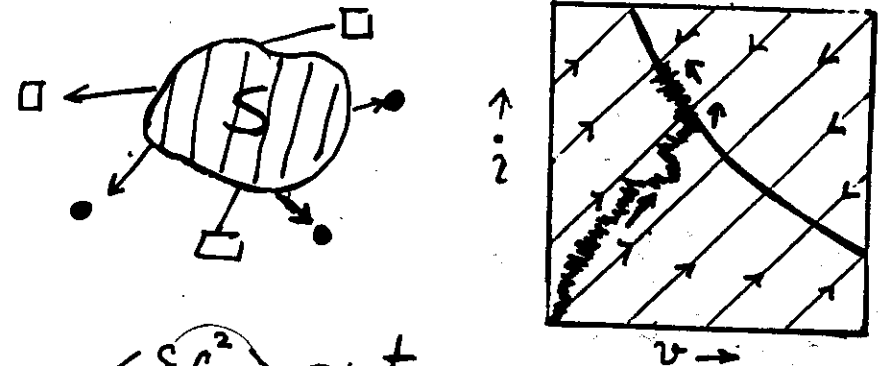
$$\langle S \rangle = 1$$



$$\frac{d\langle i \rangle}{dt} = k\langle S \rangle - \alpha\langle i \rangle\langle v \rangle$$

$$\frac{d\langle v \rangle}{dt} = k\langle S \rangle - \alpha\langle i \rangle\langle v \rangle$$

Uncorrelated Reactions - contd.



$$\langle \delta C_-^2 \rangle \sim t$$

CALCULATION OF THE CORRELATION MATRIX C - KEIZERS' Method (J. Chem. Phys. 63, 1975, 3985)

• Example of correlated reactions.

$$\begin{aligned} \frac{di}{dt} &= k - \alpha i v \\ \frac{dv}{dt} &= k - \alpha i v \end{aligned}$$

$$\begin{aligned} \frac{d\delta i}{dt} &= -\alpha i \delta v - \alpha v \delta i \\ \frac{d\delta v}{dt} &= -\alpha i \delta v - \alpha v \delta i \end{aligned}$$

$$\frac{d\delta S}{dt} = \underline{G} \delta S + F(t)$$

Where $F(t)$ is the noise term

$$\langle F(t) F^T(s) \rangle = \underline{\gamma}(s) \delta(t-s)$$

$\underline{\gamma} \rightarrow$ Strength of the fluctuations - fluctuation dissipation theorem. $\rightarrow$ procedure Keizer

$$\underline{\underline{C}} = \langle \delta \delta^T \rangle = \int_0^t \underline{\underline{E}}(t-s) \underline{\underline{F}} \underline{\underline{E}}^T(t-s) ds$$

In order to conveniently evaluate $\underline{\underline{C}}$ we use

$$e^{\underline{\underline{G}}t} = a_{n-1}(t)\underline{\underline{G}}^{n-1} + a_{n-2}(t)\underline{\underline{G}}^{n-2} \dots a_0(t)\underline{\underline{I}} = \underline{\underline{P}}(\underline{\underline{G}})$$

where $a_n(t)$ are evaluated from the n characteristic equations $p(\lambda_n) = \exp(\lambda_n t)$. For example if

$\underline{\underline{G}}$ is a 2×2 matrix, then

$$\underline{\underline{C}}(t) = \underline{\underline{F}} \int_0^t a_0^2(t-s) ds + (\underline{\underline{G}} \underline{\underline{F}} + \underline{\underline{F}} \underline{\underline{G}}^T) \int_0^t a_0(t-s) a_1(t-s) ds + \underline{\underline{G}} \underline{\underline{F}} \underline{\underline{G}}^T \int_0^t a_1^2(t-s) ds.$$

Evaluation of the terms:

$$\begin{array}{lcl} S \xrightarrow{K} i + v & k=1 & \frac{di}{dt} = K \quad \frac{dv}{dt} = K \\ i + v \xrightarrow{\alpha} S & k=2 & \frac{di}{dt} = -\alpha i v \quad \frac{dv}{dt} = -\alpha i v \end{array}$$

$$\underline{\underline{G}} = -\alpha \begin{pmatrix} i & v \\ i & v \end{pmatrix} \quad \begin{cases} \lambda_1 = 0 \\ \lambda_2 = -\alpha(i+v) \end{cases}$$

$$\underline{\underline{F}} = \frac{\Omega}{\Delta V} \sum_{k=1}^n \nu_{kj} (V_k^+ + V_k^-) \nu_{ki}$$

$\nu_k \rightarrow$ Stoichiometric Co-efficients.

ν_{ki}	$k=1$	$k=2$
$i=1$ i	1	-1
$i=2$ v	1	-1

$$\underline{\underline{F}} = \alpha i v \begin{pmatrix} 1 & 1 \\ 1 & 1 \end{pmatrix}$$

$$\begin{aligned} e^{\lambda_1 t} &= a_0 + a_1 \lambda_1 \\ e^{\lambda_2 t} &= a_0 + a_2 \lambda_2 \\ a_0(t) &= \frac{\lambda_2 e^{\lambda_1 t} - \lambda_1 e^{\lambda_2 t}}{\lambda_2 - \lambda_1} \\ a_1(t) &= \frac{e^{\lambda_1 t} - e^{\lambda_2 t}}{\lambda_1 - \lambda_2} \end{aligned}$$

For the case $\lambda_1 \rightarrow 0$ $|\lambda_1 t| \ll 1$ we get

$$A_1 = \int_0^t a_0^2(t-s) ds = t$$

$$A_2 = \int_0^t a_0(t-s) a_1(t-s) ds = \frac{e^{\lambda_2 t}}{\lambda_2^2} - \frac{1}{\lambda_2^2} - \frac{t}{\lambda_2}$$

$$A_3 = \int_0^t a_1^2(t-s) ds = \frac{e^{2\lambda_2 t}}{2\lambda_2^3} - \frac{2e^{\lambda_2 t}}{\lambda_2^3} + \frac{3}{2\lambda_2^3} + \frac{t}{\lambda_2^2}$$

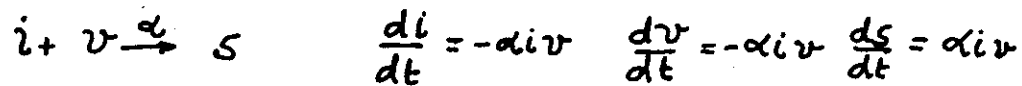
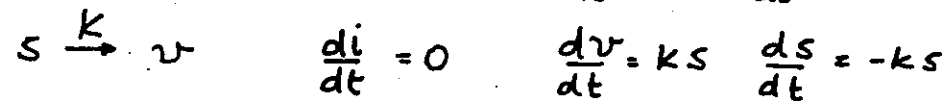
$$\begin{pmatrix} \langle \delta i^2 \rangle & \langle \delta i \delta v \rangle \\ \langle \delta v \delta i \rangle & \langle \delta v^2 \rangle \end{pmatrix} = A_1 \underline{\underline{F}} + (\underline{\underline{G}} \underline{\underline{F}} + \underline{\underline{F}} \underline{\underline{G}}^T) A_2 + \underline{\underline{G}} \underline{\underline{F}} \underline{\underline{G}}^T A_3$$

We can now evaluate $\langle \delta C_-^2 \rangle$

$$\langle \delta C_-^2 \rangle = \langle \delta v^2 \rangle + \langle \delta i^2 \rangle - 2 \langle \delta i \delta v \rangle \equiv 0$$

$\therefore$ Though we have a soft mode $\lambda_1 = 0$ the fluctuations are unable to drive the system due to correlated reactions in this model.

UNCORRELATED REACTIONS.



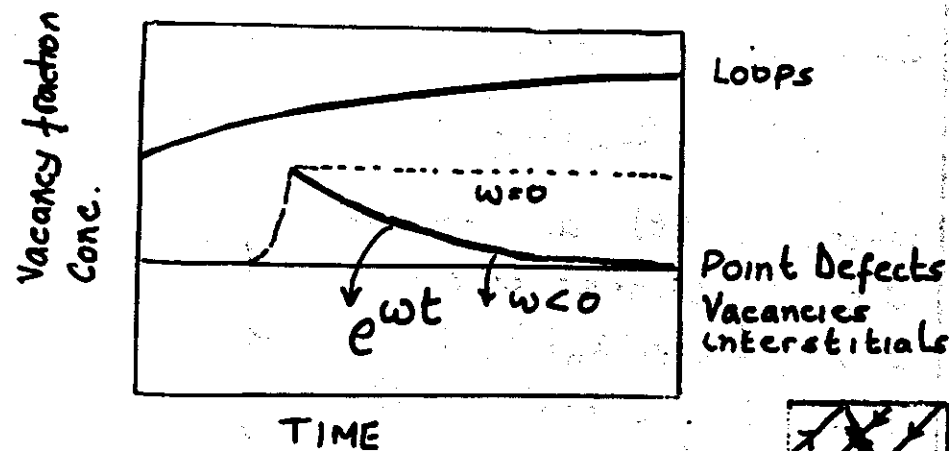
$$\neq = \frac{D}{\Delta V} \sum_k v_{kj} (V_k^+ + V_k^-) v_{ki}$$

$$\langle \delta C_-^2 \rangle = \left(\frac{D}{\Delta V} \right) K t$$

Physical Process:

By introducing the source term S we have introduced one more mechanism for the dissipation of fluctuations. This removes an important constraint on the system.

FLUCTUATIONS & STABILITY



$$\frac{dv}{dt} = K_v - \beta_v v - \alpha v i + D_v \nabla^2 v$$

$$\frac{di}{dt} = K_i - \beta_i i - \alpha v i + D_i \nabla^2 i$$

CORRELATED REACTIONS

$$\beta_i = \beta_v = 0$$

$$\omega_1 = 0, \quad C_- = i - v$$

$$\langle \delta C_- \rangle^2 \equiv 0$$

Relax
Const 1.

UNCORRELATED REACTIONS

$$\omega_1 = 0, \quad C_- = i - v$$

$$\langle \delta C_- \rangle^2 \propto K t$$

Const 2.

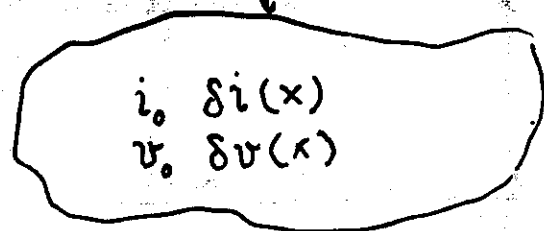
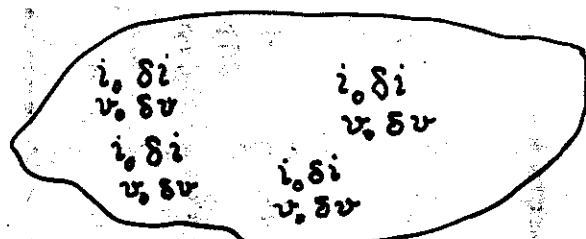
SPATIAL VARIATIONS

$$\beta_i, \beta_v = \text{Const.}$$



3. DIFFUSION AND STABILITY

24

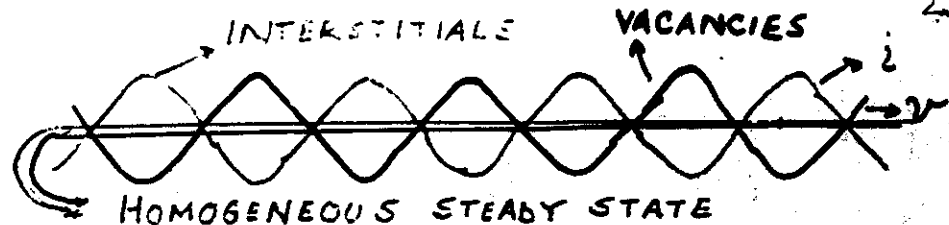


$\delta v(x)$
 $\delta i(x)$
 $i = i_0 + \delta i e^{ikx} e^{wt}$

$$\frac{di}{dt} = k - \beta_i i - \alpha i v + D_i \nabla^2 i$$

$$\frac{dv}{dt} = k - \beta_v v - \alpha i v + \underbrace{D_v \nabla^2 v}_{\text{diffusion term}}$$

β and α are constant



$$\frac{di}{dt} = k_i - \beta_i i - \alpha i v + D_i \nabla^2 i$$

$$\frac{dv}{dt} = k_v - \beta_v v - \alpha i v + D_v \nabla^2 v$$

steady state solution can be obtained from $\frac{d(i,v)}{dt} = k_{(i,v)} - \beta_{(i,v)} - \alpha i v = 0$

Homogeneous Steady State (i_0, v_0)

$$i = i_0 + \delta i(\bar{x}, t) = i_0 + \delta i' e^{wt} e^{ik \cdot \bar{r}}$$

$$v = v_0 + \delta v(\bar{x}, t) = v_0 + \delta v' e^{wt} e^{ik \cdot \bar{r}}$$

$$\frac{d}{dt} \begin{pmatrix} \delta i(\bar{r}, t) \\ \delta v(\bar{r}, t) \end{pmatrix} = \begin{pmatrix} -\beta_i - \alpha v + D_i \nabla^2 & -\alpha i \\ -\alpha v & -\beta_v - \alpha i + D_v \nabla^2 \end{pmatrix} \begin{pmatrix} \delta i(\bar{r}, t) \\ \delta v(\bar{r}, t) \end{pmatrix}$$

substituting

$$\delta i(\bar{r}, t) = \delta i' e^{wt} e^{ik \cdot \bar{r}}$$

$$\delta v(\bar{r}, t) = \delta v' e^{wt} e^{ik \cdot \bar{r}}$$

we get

$$\begin{pmatrix} -\beta_i - \alpha v_0 - D_i k^2 - \omega & -\alpha i_0 \\ -\alpha v_0 & -\beta_v - \alpha i_0 - D_v k^2 - \omega \end{pmatrix} \begin{pmatrix} \delta i' \\ \delta v' \end{pmatrix} = 0$$

A solution exists only if

$$\det \begin{vmatrix} -\beta_i - \alpha v_0 - D_i k^2 - \omega & -\alpha i_0 \\ -\alpha v_0 & -\beta_v - \alpha i_0 - D_v k^2 - \omega \end{vmatrix} = 0$$

or

$$\begin{aligned} \omega^2 + \omega [k^2(D_i + D_v) + (\beta_v + \beta_i + \alpha v_0 + \alpha i_0)] \\ + [k^4 D_i D_v + k^2(D_v \alpha v_0 + D_i \alpha i_0 + D_i \beta_v + D_v \beta_i) \\ + (\beta_i \alpha i_0 + \beta_v \alpha v_0 + \beta_i \beta_v)] = 0 \end{aligned}$$

We observe that all the terms are positive for $k^2 > 0$

1. $\therefore \omega = \begin{cases} \lambda_1 < 0 \\ \lambda_2 < 0 \end{cases}$ and there is therefore no instability possible.

2. The only condition for which $\omega = 0$ is $\beta_i = \beta_v = 0$ and $k^2 = 0$ thus the only stable state is the homogeneous state

3. If we put $\omega = 0$ we get an

equation

$$k^4 D_v D_i + k^2(D_v \alpha v_0 + D_i \alpha i_0 + D_i \beta_v + D_v \beta_i) + (\beta_i \alpha i_0 + \beta_v \alpha v_0 + \beta_i \beta_v) = 0$$

We observe that there are no physical acceptable values of the parameters D, α, β for which a solution $k^2 > 0$ exists.

4. RADIATION INDUCED INSTABILITY - BASIC APPROACH.

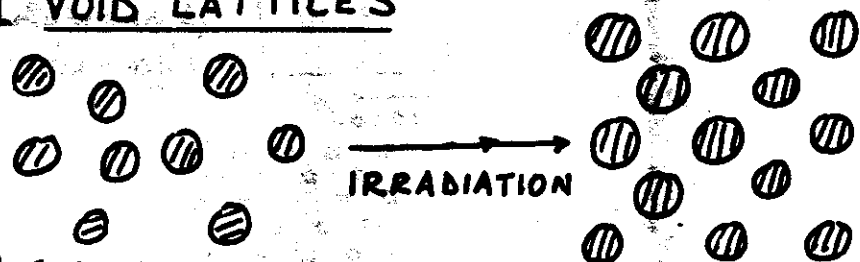
There are two approaches which can be followed to obtain the conditions for which an instability arises

1: Coupling with microstructural Defects

2: More careful examination of the defect diffusion coefficients and fluxes

EXPERIMENTAL OBSERVATIONS ON MICROSTRUCTURAL ORDERING AND SPINOIDAL TYPE ORDERING IN ALLOYS DURING IRRADIATION

1. VOID LATTICES



① SYMMETRY

HAVE THE SAME SYMMETRY AS THAT OF THE
HOST LATTICE

Fe, Ni, Ti, W, Cr	bcc
Al, Ni	fcc
Mg	hcp

2. RADIATION

Produced by cascade producing
radiation → neutrons, ion
seldom formed by electron irradiation

3. DOSE RATE EFFECT

Not very sensitive to dose rate.

④ DOSE DEPENDENCE

There seem to be some evidence
of a critical dose for ordering but
the values are very different from
metal to metal. Also some threshold
gas concentration can influence the
value

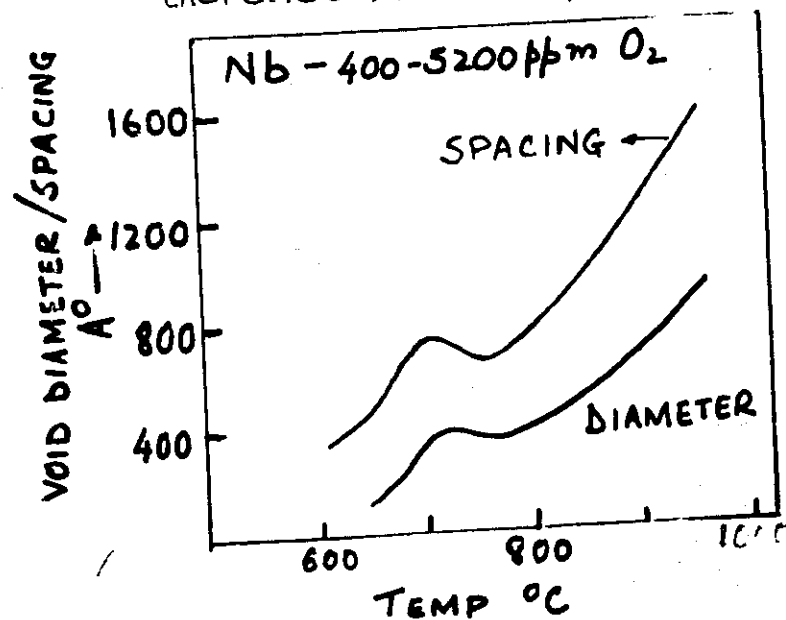
Ni ~ 350-400 dpa
Al ~ 40-80 dpa
Nb, Mo ~ 5 dpa

⑤ Alloy Composition

Weak dependence.

⑥ Temperature dependence

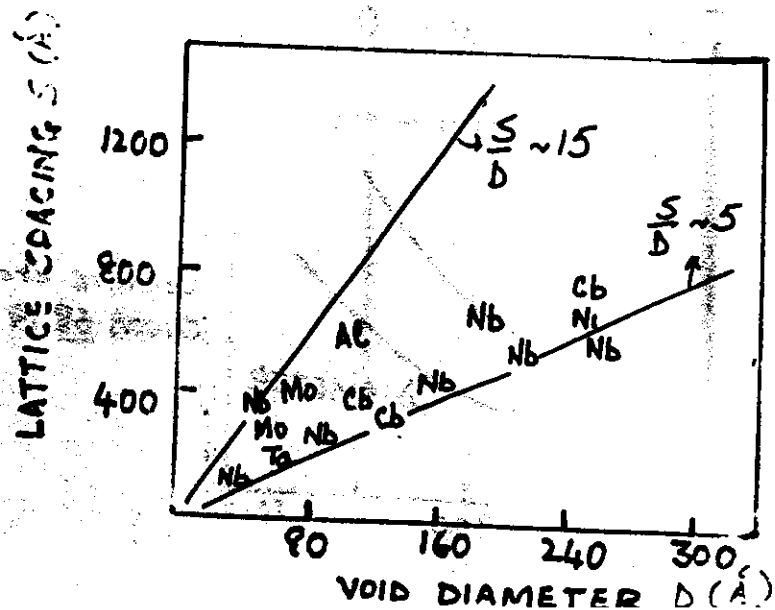
- Very few systematic investigations
- Void diameter & void spacing
increase with temperature.



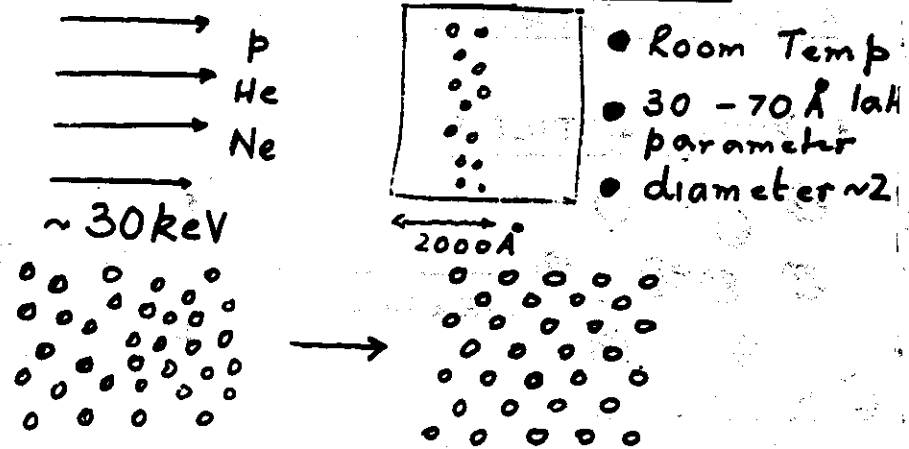
Loomis et al
JNM 68 19 (197)

① SOME TYPICAL DATA ON VOID LATTICES

MATERIAL	PROJECTILE Energy MeV	TEMP. °C	DOSE d.p.a.	SPACING Å	DIAMETER Å
Ta	n > .1	585	20	205	61
W	n > .1	550	15	195	30
	n > .1	880	15	250	40
Mo	n > .1	585	36	265	64
	N 2	870	100	220	40
Nb	n > .1	790	34	665	186
	Ni 5	800	5	350	45
Ni	Ni 5	525	360	620	250
Al	Al .4	50	40	600	100
Mg	n > .1	55	3-6	imperfect	



2 FORMATION OF GAS BUBBLE SUPERLATTICES IN METALS



1. SIMILARITY WITH VOID LATTICE

1. Crystallographic symmetry and orientation is the same as the host matrix
2. Bubble Lattice spacing/diameter $\sim 5-15$
3. Stages of ordering are similar

2. DIFFERENCES WITH VOID LATTICE

1. Lattice spacing & diameter $\sim 1/10$ void Latt.
2. Formed at room temperature
3. Temperature is lower than vacancy migration temperature
4. No strong temperature dependence
5. No significant cascade effects are possible at 30 keV.

SOME TYPICAL BUBBLE LATTICE DATA

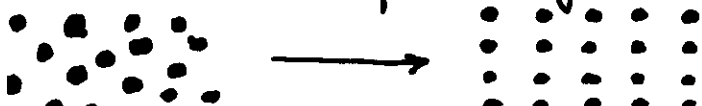
Serial	Projectile Energy KeV	Temp °C	Dose ions/cm ²	Spacing Å	Dia. Å
0	He 36	RT	1×10^{17}	37	} ~24
		500	2×10^{17}	36	
		700	2×10^{17}	35	
	He 25	250	4.5×10^{17}	32	
	He 60	400	5×10^{17}	36	
	Ne 100	550	1×10^{17}	36	—
	He 30	RT	4×10^{17}	77	20
	H 16	RT	13×10^{17}	Not well defined	
316	He 30	RT	4×10^{17}	65	20
	He 30	RT	1.5×10^{18}	90	30

ORDERING IN COMPOUNDS $\text{CaF}_2/\text{SrF}_2$

TEM
→
→
→
→
100 keV

Non-radiative energy transfer produces F[•] & Ca interstitials

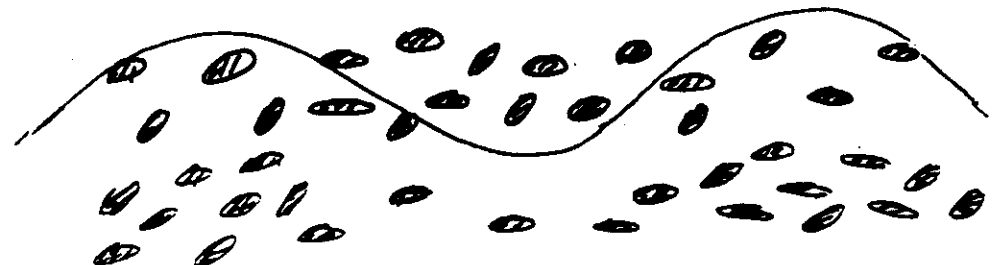
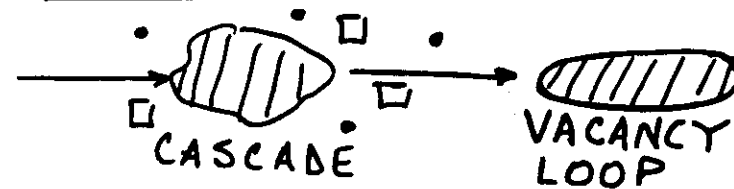
Ca → precipitates are formed which subsequently order



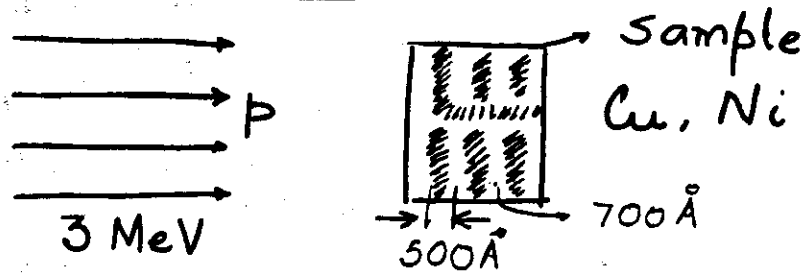
IMPORTANT FEATURES ARE

1. Ca ppts are coherent with CaF_2 structure and form a perfect lattice : A less perfect lattice is formed in SrF_2 because of a small difference in the lattice parameter of Sr and SrF_2 . No ordering is observed in BaF_2 due to incoherent precipitation
2. Supperlattice in the fluorite is simple cubic though the overall structure is fcc. However the fluorite sublattice is simple cubic
3. Room Temp / spacing ~190-280 Å and Diameter 30-40 Å

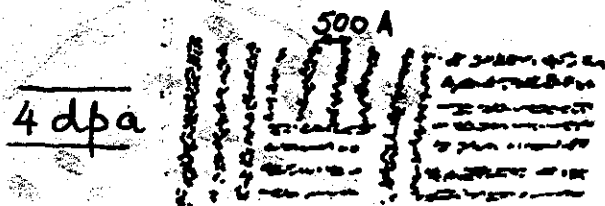
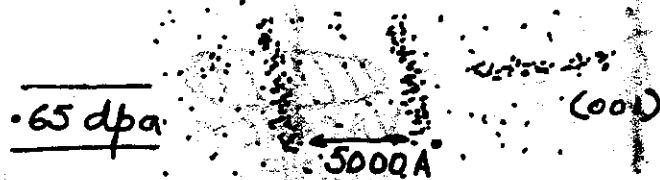
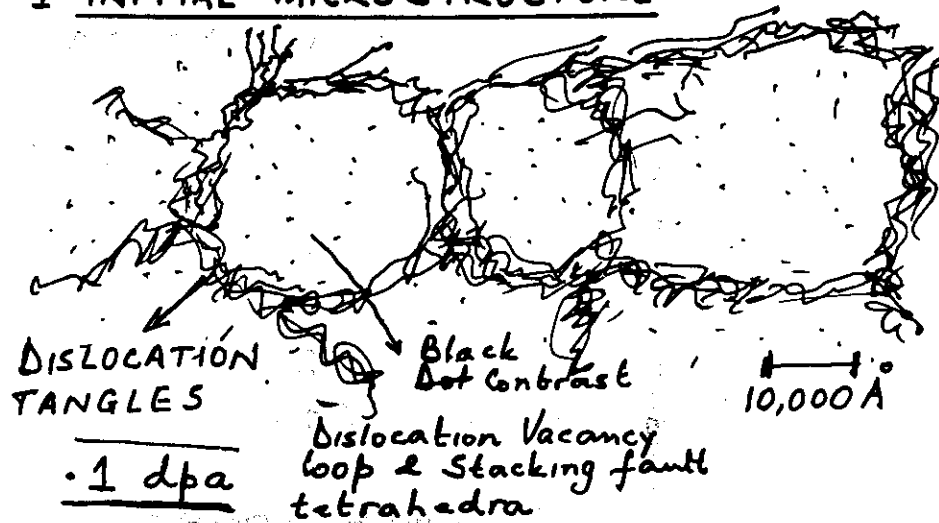
4 ORDERING OF VACANCY LOOPS



5 DISLOCATION PATTERNING DURING IRRADIATION



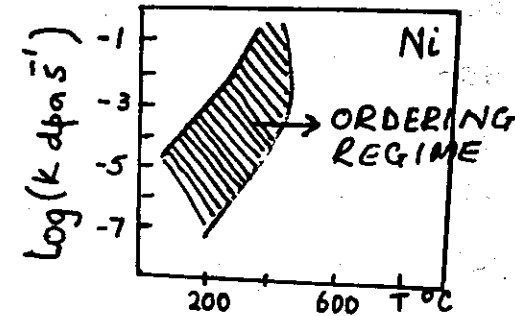
1 INITIAL MICROSTRUCTURE



1 NATURE OF DEFECT WALLS

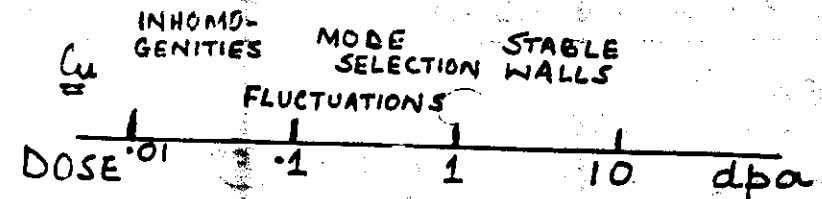
High concentration of dislocations
dislocation loops (Vacancy type.)
stacking fault tetrahedra & a
few voids

2 TEMPERATURE AND DOSE RATE REGIME

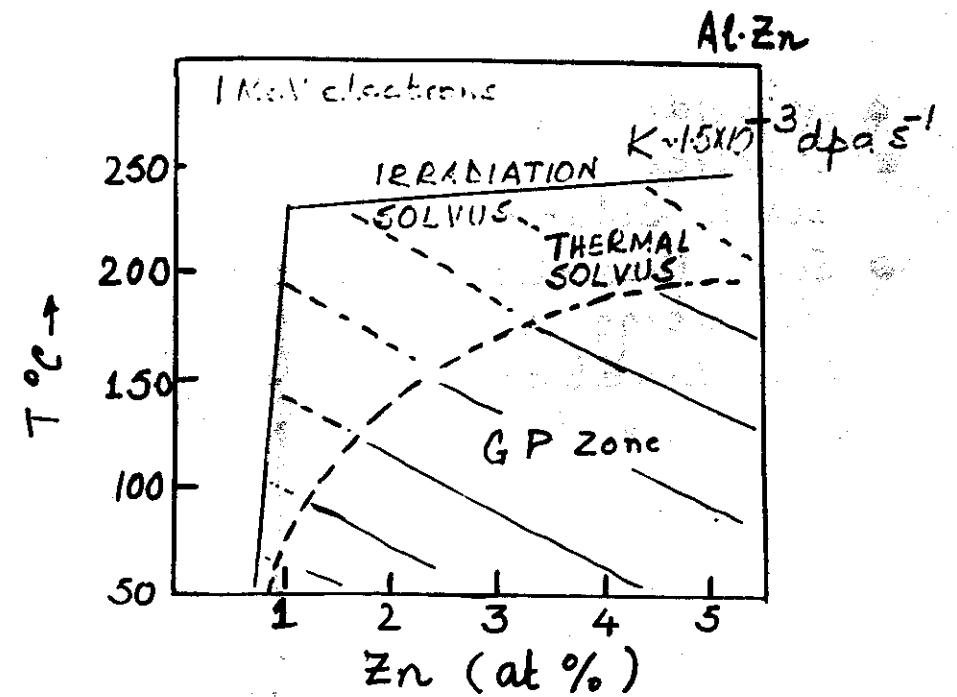
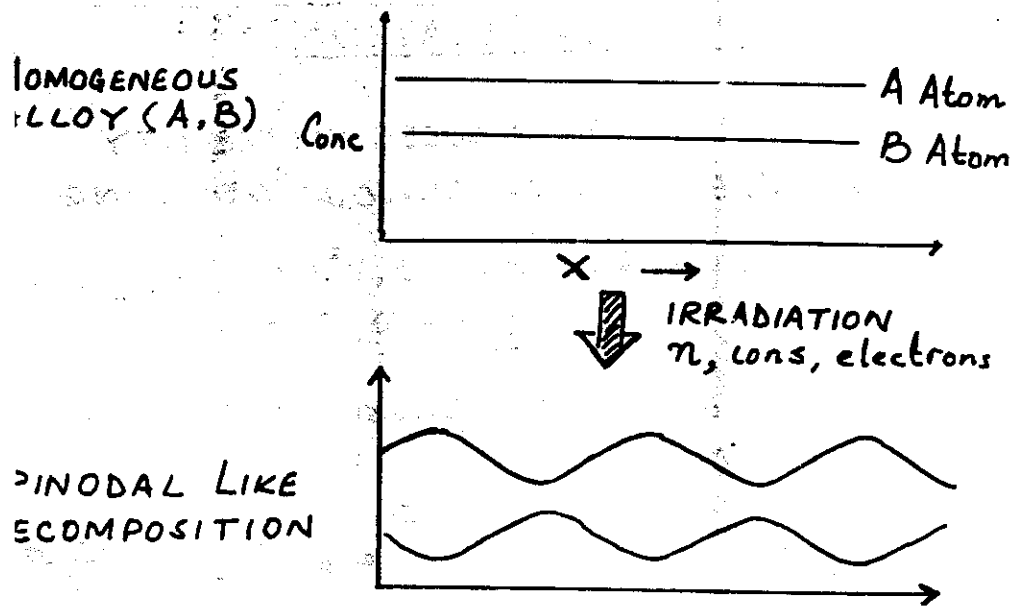


Jager et al
Non-linear Phenomena
in materials Science
Trans Tech Publications
Eds Kubin & Mah.

3 DOSE REGIMES



SPINODAL TYPE ORDERING OF ALLOYS



IMPORTANT SYSTEMS

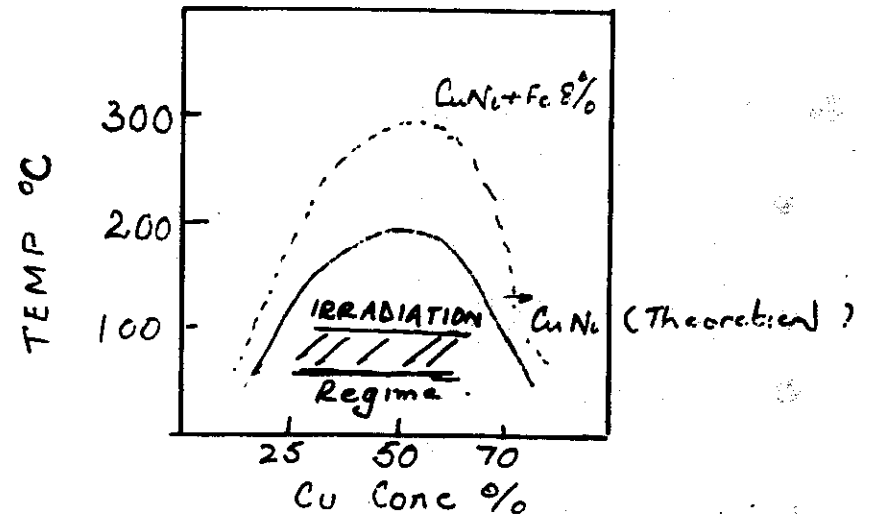
- Al-Zn (2%), Ni Be, Cu Be, W Re
- Cu Ni - Cu Ni Fe
- Fe Ni (35%), Fe Cr-Ni

Experimental results.

Al-Zn Alloy

Suitable quenching and aging treatments produce Zn rich G.P. zones in Al rich Al-Zn alloys. - Vacancies are needed for mass transport which can also be provided by irradiation

Cu Ni



Co-Ni alloy

38

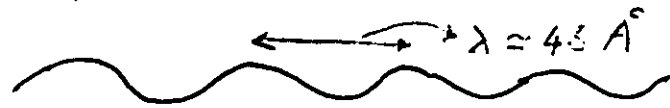
- No evidence for thermal decomposition
- Classical example of a complete solid solution.
- Some theoretical suggestions exist indicating a possible spinodal but kinetics is too sluggish at these low temperatures.

Irradiation

electron irradiation 2-3 MeV

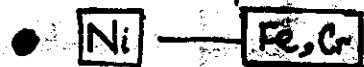
Flux 1.6×10^{20} ions Dose $\sim 1.6 \times 10^{-2}$ dpa

Temperature $100^\circ\text{C} \sim 300^\circ\text{C}$



• Fe-Ni-Cr Alloys Ni $\sim 30-75\%$

- No evidence of thermal spinodal decomposition in these alloys.



- λ sensitive to temperature (400-650°C)
 λ increases with temperature

- 5 MeV Ne^{10+} $\rightarrow$ till 100 dpa / Neutron Irrad.

- Insensitive to displacement rates.

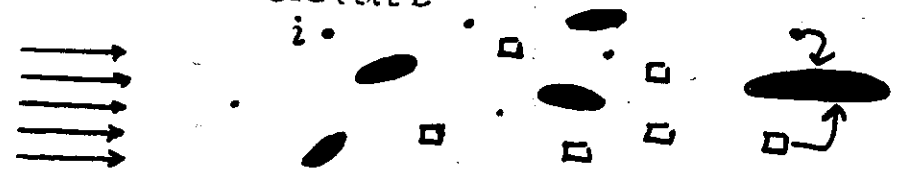
3

B. MICROSTRUCTURAL ORDERING DURING IRRADIATION

1. DEFECT REACTION INDUCED INSTABILITY: EXAMPLE OF SPONTANEOUS ORDERING.

$$\left. \begin{aligned} di/dt &= k_i - \beta_i i + D_i \nabla^2 i \\ dv/dt &= k_v - \beta_v v + D_v \nabla^2 v \end{aligned} \right\} \alpha = 0$$

We examine the physical processes in more detail



$$k_i = K \quad k_v = K(1-\epsilon) \quad K_{\text{vac Loop}} = \epsilon K$$

$$\beta_i = z_i D_i S \quad \beta_v = D_v S$$

Dynamics of the vacancy loop

$$\frac{dQ}{dt} = \epsilon K + (D_v v - z_i D_i i) S$$

$$Q = \frac{4\pi}{3} N r^3$$

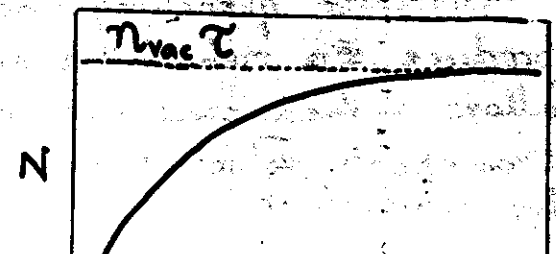
$$S = 2\pi N r$$

$$S = f(Q)$$

$$\delta S = \frac{\partial f}{\partial Q} \delta Q$$

$$= f' \delta Q$$

$$N = \text{Constant} = n_{\text{vac}} \tau$$



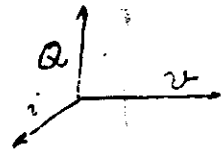
$$\frac{di}{dt} = K - Z_i D_i s i + D_i \nabla^2 i$$

$$\frac{dv}{dt} = K(1-\epsilon) - D_v s v + D_v \nabla^2 v$$

$$\frac{dQ}{dt} = \epsilon K + D_v s v - Z_i D_i s i$$

We immediately observe that one of the modes is soft for $k^2 \rightarrow 0$ since

$$\frac{dQ}{dt} = \frac{d}{dt}(i-v) \quad \nabla^2 \text{ terms} = 0$$



We now examine the stability condition for this equation.

$$\begin{pmatrix} i \\ v \\ Q \end{pmatrix} = \begin{pmatrix} i_0 \\ v_0 \\ Q_0 \end{pmatrix} + \begin{pmatrix} \delta i' \\ \delta v' \\ \delta Q' \end{pmatrix} e^{\omega t} e^{i \mathbf{k} \cdot \mathbf{r}}$$

$$\det \begin{vmatrix} -(Z_i D_i s + D_i k^2) - \omega & 0 & -\frac{K f'}{s} \\ 0 & -(D_v s + D_v k^2) - \omega & -\frac{K(1-\epsilon) f'}{s} \\ -Z_i D_i s & D_v s & -\frac{\epsilon K f' - \omega}{s} \end{vmatrix}$$

Solving the above determinant we get

$$\omega^3 + \omega^2 g_2(k^2) + \omega g_1(k^2) + g(k^2) = 0$$

A soft mode can arise if $g(k^2) = 0$

$$g(k^2) = \left[-\frac{\epsilon K}{s} + \left\{ \frac{(\epsilon Z + (Z-1)) K k^2 + \epsilon K Z s}{(Z s + k^2)(k^2 + s)} \right\} \right] f' = 0$$

$$k^2 = 0$$

$$k_c^2 = \left\{ \frac{(Z-1) - \epsilon}{\epsilon} \right\} s$$

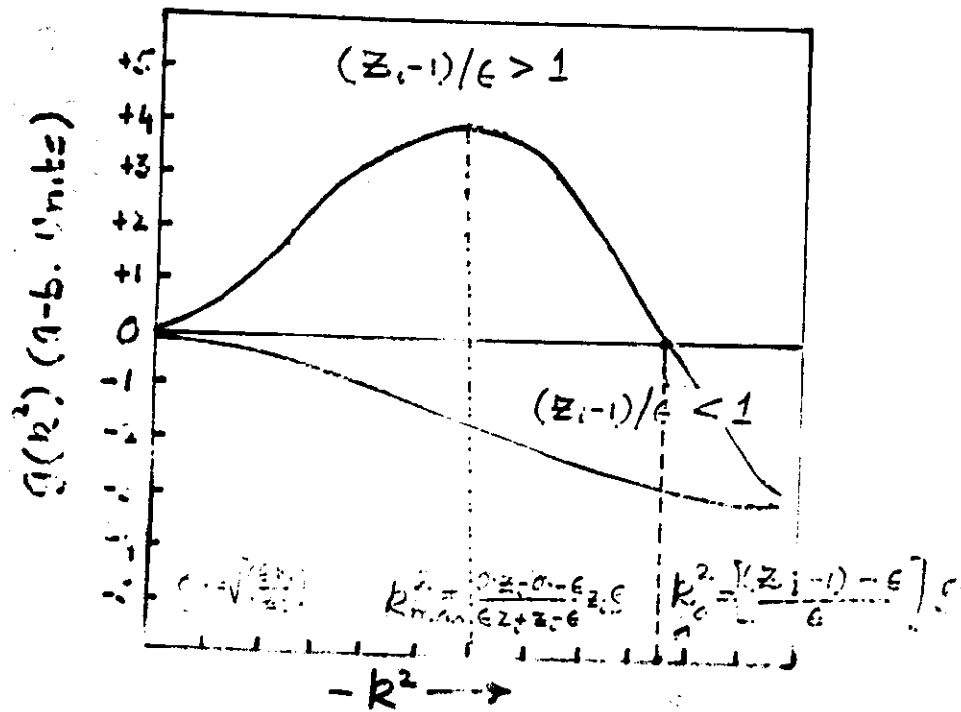
We therefore obtain the condition that a bifurcation to a periodic structure takes place only if

$$\frac{(Z-1)}{\epsilon} > 1$$

$$\begin{aligned} \therefore k^2 > k_c^2 & \quad g(k^2) < 0 & \quad \omega_3 < 0 \\ k^2 < k_c^2 & \quad g(k^2) > 0 & \quad \omega_3 > 0 \end{aligned}$$

$\therefore$ A definite periodic structure will emerge.

42 PLOT OF THE AMPLIFICATION FACTOR $g(k)$



$$(Z_i - 1) \sim 0.08$$

$$\epsilon \sim 0.04$$

$$k_0^2 \sim \frac{1}{2} \rho$$

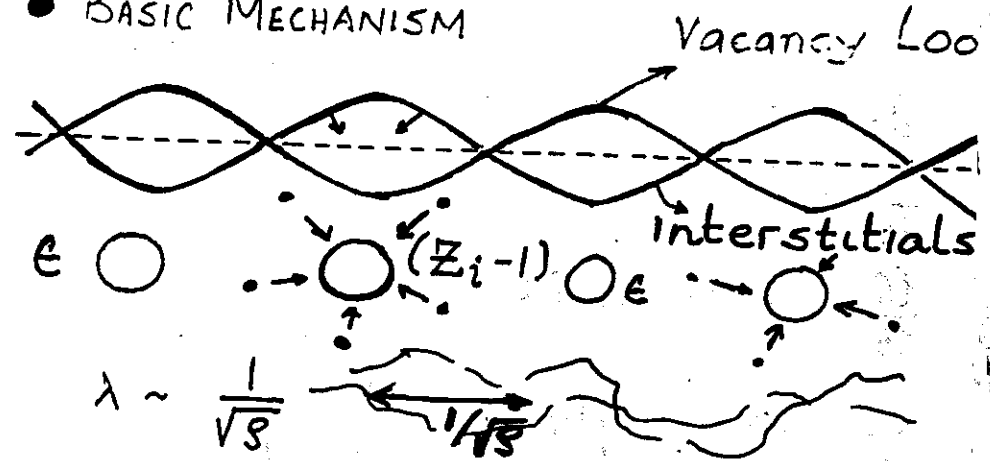
$$\sim 10^{12} \text{ cm}^{-2}$$

$$\rho = \frac{2\pi N}{6} \left| \frac{2 \times 10^{18} \text{ cm}^{-3}}{2 \times 10^{-7} \text{ cm}} \right|^2$$

$$\approx 2 \times 10^{12} \text{ cm}^{-2}$$

$$\lambda \sim \frac{2\pi}{k} \approx \underline{600 \text{ \AA}}$$

• BASIC MECHANISM



• MAIN FEATURES OF THE MODEL

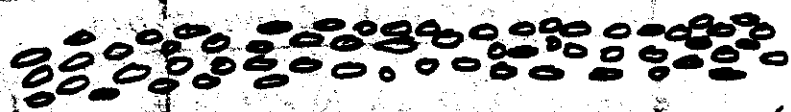
1. Approximate wave length scale
2. Temperature independence
3. Identifies the drive parameter for the bifurcation as $(Z_i - 1)/\epsilon$.

• LIMITATION OF THE MODEL!

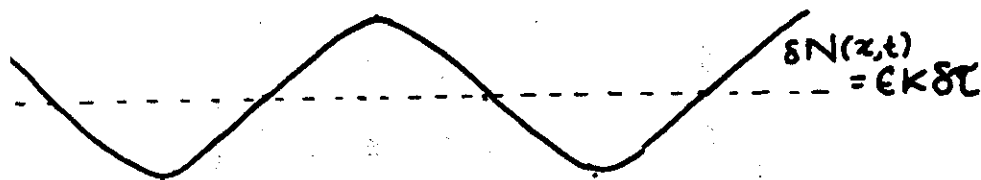
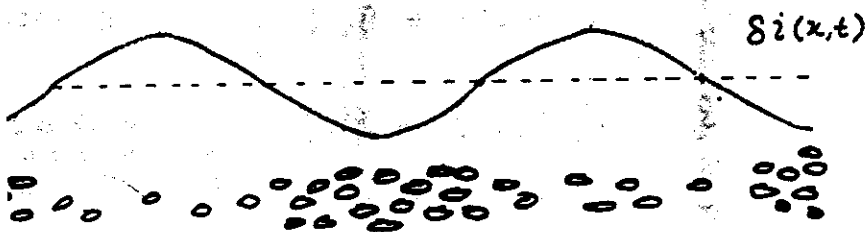
Spontaneous bifurcation which is unphysical.

PHYSICAL MECHANISM

$$i_0 = \frac{k}{\sum D_i S}$$



$$N_0 = (\epsilon K) \tau$$



$$\begin{bmatrix} \delta i \\ \delta v \\ \delta q \end{bmatrix} = \begin{bmatrix} \underline{\underline{G}}(k^2) \end{bmatrix} \begin{bmatrix} \delta i \\ \delta v \\ \delta q \end{bmatrix} \quad \text{Solutions Exist} \quad \text{Det} | \underline{\underline{G}}(k^2) - \underline{\underline{I}} \omega | = 0$$

$$\frac{d}{dt} \begin{bmatrix} \delta p_1 \\ \delta p_2 \\ \delta p_3 \end{bmatrix} = \begin{bmatrix} \underline{\underline{U}}' \underline{\underline{G}}(k^2) \underline{\underline{U}} \end{bmatrix} \begin{bmatrix} \delta p_1 \\ \delta p_2 \\ \delta p_3 \end{bmatrix} \quad \begin{bmatrix} \delta p_1 \\ \delta p_2 \\ \delta p_3 \end{bmatrix} = \underline{\underline{U}} \begin{bmatrix} \delta i \\ \delta v \\ \delta q \end{bmatrix}$$

$$\begin{bmatrix} \omega_1(k^2) & 0 \\ 0 & \omega_2(k^2) & \omega_3(k^2) \end{bmatrix} \rightarrow = 0$$

$$p_i = e^{\omega_i(k^2)t} \quad \text{Mode } \delta p_3 \text{ is soft}$$

$$\text{Det} | \underline{\underline{G}}(k^2) - \underline{\underline{I}} \omega | = 0$$

$$\omega^3 + \omega^2 g_2(k^2) + \omega g_1(k^2) + g_0(k^2) = 0$$

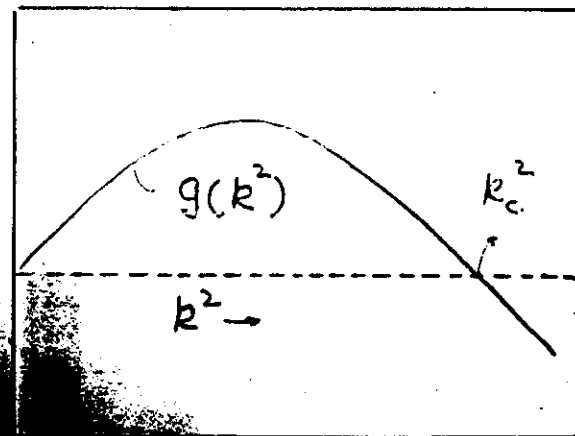
$$g(k^2) = \omega_1 \omega_2 \omega_3$$

$$\omega_1 < 0 \quad \omega_2 < 0 \quad \therefore \omega_1 \omega_2 > 0$$

$$g(k^2) > 0 \quad \omega_3 > 0$$

$$g(k^2) = 0 \quad \omega_3 = 0$$

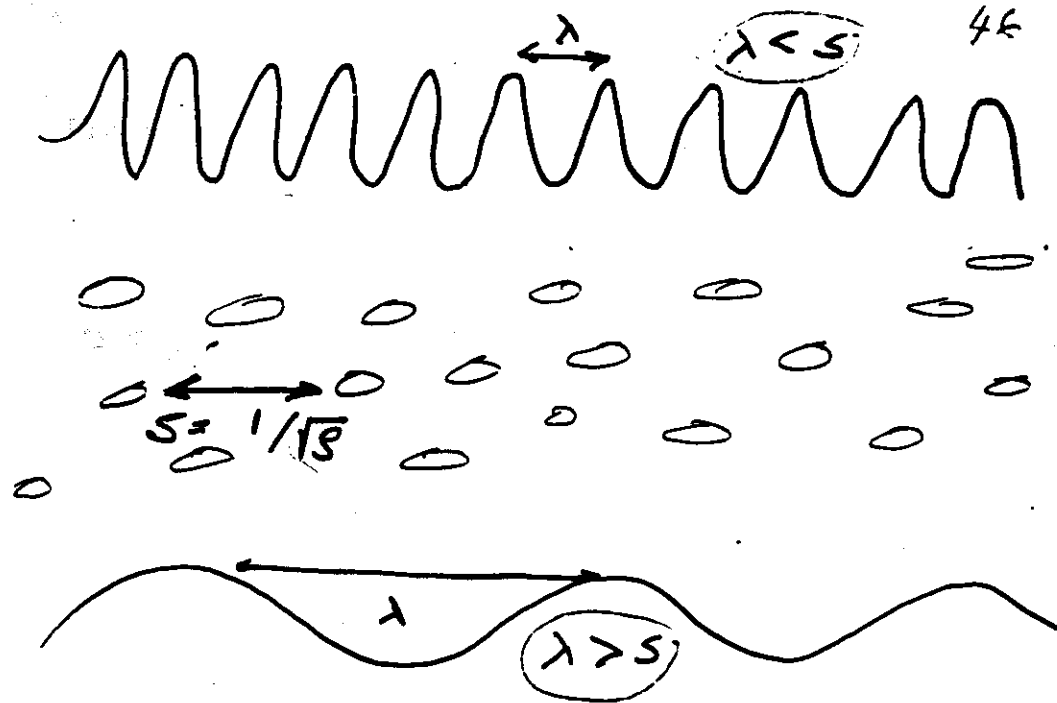
$$g(k^2) < 0 \quad \omega_3 < 0$$



$$\delta G = U_1(k^2) p_1 + U_2(k^2) p_2 + U_3(k^2) p_3$$

$$e^{i \omega_1 t} \quad e^{i \omega_2 t} \quad e^{i \omega_3(k^2) t}$$

$$e^{i \omega_3(k^2) t} e^{i k \cdot r}$$



$$k^2 = \left(\frac{\epsilon - \Delta Z}{\epsilon} \right) \rho$$

$$\epsilon > \Delta Z$$

$$\epsilon > 0$$

$$\frac{dQ}{dt} = \epsilon K + (\Delta_v \bar{v} - z_i D_i i) \rho$$

$$\frac{d\delta Q}{dt} = (\Delta_v \bar{v} - z_i D_i i) \delta \rho + \rho (\Delta_v \delta \bar{v} - z_i D_i \delta i)$$

$$\delta \bar{v} = \frac{K(1-\epsilon) \delta \rho}{\Delta_v \rho (\rho + k^2)} \quad \bar{v} = \frac{K(1-\epsilon)}{\Delta_v \rho}$$

2. EFFECT OF ADDITIONAL SINK TERMS ON THE BIFURCATION CONDITION.

- ① Point Defects ② Point Defects & Vacancy Loop
③ Point Defects, Vacancy loops, Voids & Dislocation

$$\left. \begin{aligned} \frac{di}{dt} &= K_i - \beta_i i - \alpha_i v + D_i \nabla^2 i \\ \frac{dv}{dt} &= K_v - \beta_v v - \alpha_v i + D_v \nabla^2 v \end{aligned} \right\} \begin{aligned} K_i &= K \\ K_v &= K(\epsilon) \end{aligned}$$

$$\left. \begin{aligned} \beta_i &= D_i \rho_s + z_i D_i (\rho_L + \rho_D) \\ \beta_v &= D_v (\rho_s + \rho_L + \rho_D) \end{aligned} \right\} \begin{aligned} S &\rightarrow \text{Voids} \\ L &\rightarrow \text{Vac. Lo} \\ D &\rightarrow \text{Disloc.} \end{aligned}$$

$$\frac{dQ_s}{dt} = \rho_s (D_v v - D_i i)$$

$$\frac{dQ_L}{dt} = \epsilon K + \rho_L (D_v v - z_i D_i i)$$

$$\frac{dQ_D}{dt} = \rho_D (z_i D_i i - D_v v)$$

Before doing the stability analysis we examine the nature of the homogeneous solution

$$\frac{di}{dt} - \frac{dv}{dt} = \frac{dQ_s}{dt} + \frac{dQ_L}{dt} - \frac{dQ_D}{dt}$$

$$\omega \rightarrow 0 \quad k^2 \rightarrow 0$$

we set

$$\frac{di}{dt} = \frac{dv}{dt} = 0 \quad \text{and substitute in}$$

We can now eliminate the fast modes using the relations

$$\dot{z}_0 = \frac{K}{D_i B} \quad v_0 = \frac{K(1-\epsilon)}{D_v A}$$

$$\delta z = \frac{-K(\delta s_s + z \delta s_L + z \delta s_D)}{D_i B (B + k^2)}$$

$$\delta v = \frac{-K(1-\epsilon)(\delta s_s + \delta s_L + \delta s_D)}{D_v A (A + k^2)}$$

$$A = (s_s + s_L + s_D)$$

$$B = s_s + z(s_L + s_D)$$

Evaluating δQ_s , δQ_L and δQ_D we get

$$\begin{bmatrix} Q_s \\ Q_L \\ Q_D \end{bmatrix} = \begin{bmatrix} (s_L + s_D)R_{(1)} + k^2 R_{(0)} & -s_s R_{(1)} & -s_s R_{(1)} \\ -s_L R_{(1)} & s_s R_{(1)} + s_D R_{(2)} + k^2 R_{(1)} & -s_L R_{(2)} \\ -s_D R_{(1)} & -s_D R_{(2)} & s_s R_{(1)} + s_L R_{(2)} + k^2 R_{(1)} \end{bmatrix} \begin{bmatrix} \frac{\partial s_s}{\partial Q_s} \delta Q_s \\ \frac{\partial s_L}{\partial Q_L} \delta Q_L \\ \frac{\partial s_D}{\partial Q_D} \delta Q_D \end{bmatrix}$$

where

$$R_{(n)} = \frac{k}{A(A + k^2)} - \frac{z_i^n K(1-\epsilon)}{B(B + k^2)}$$

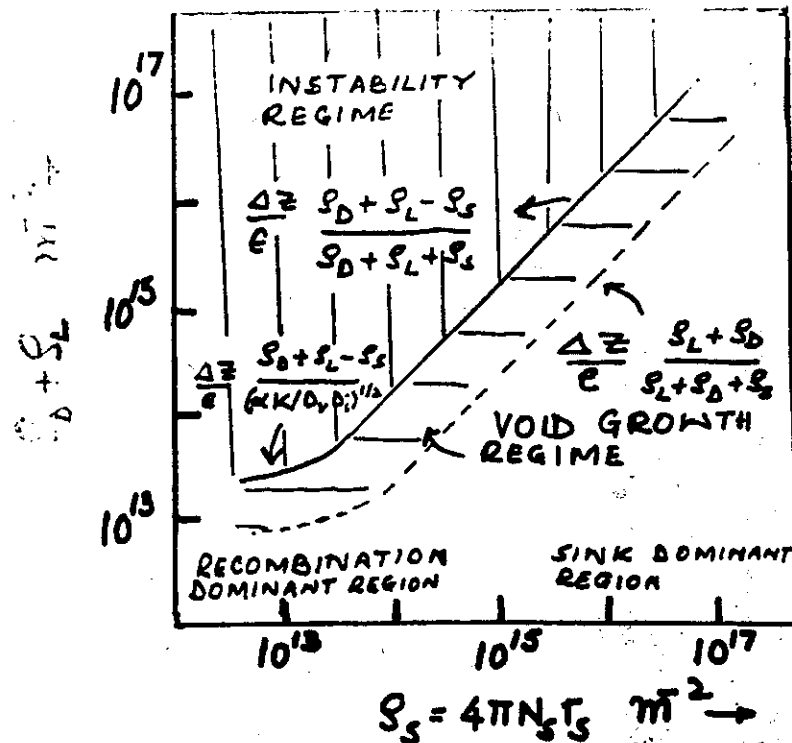
$$R_{(1)} = (z_i - 1)K \frac{(s_L + s_D)^2 - s_s^2 - k^2 s_s}{AB(A + k^2)(B + k^2)} - \frac{\epsilon K}{A(A + k^2)}$$

CONDITION FOR SPATIAL INSTABILITY.

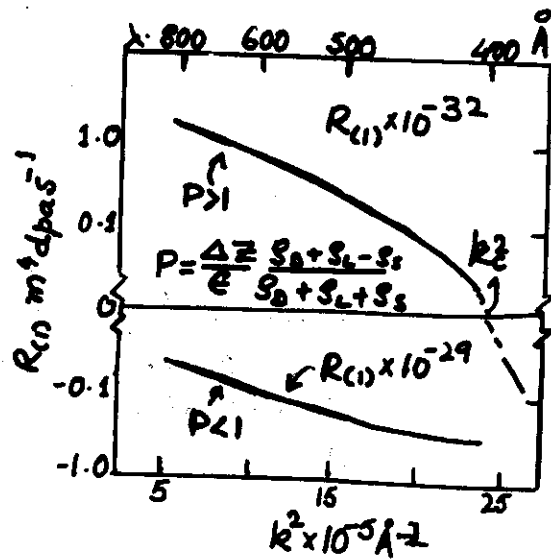
$$R_{(1)} > 0$$

$$\frac{z_i - 1}{\epsilon} \frac{s_D + s_L - s_s}{s_D + s_L + s_s} > 0$$

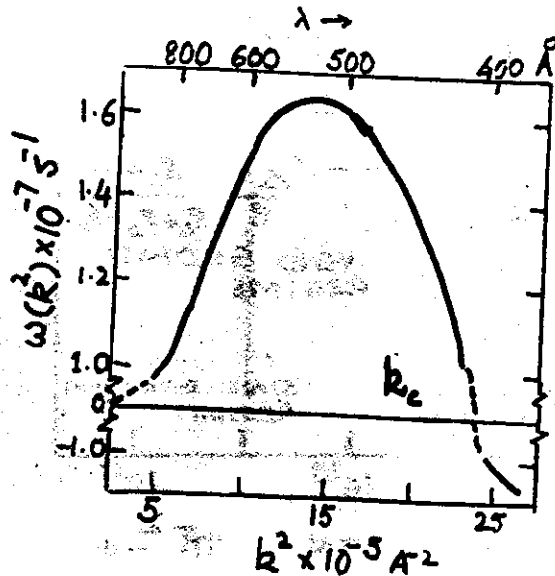
- We observe that by addition of neutral sinks like voids there is a minimum threshold dislocation density required and no spontaneous bifurcation occurs.



PLOT OF $R(k)$ AS A FUNCTION OF k^2 50



PLOT OF ONE OF THE ROOTS $\omega(k^2)$



EFFECT OF TEMPERATURE ON VOID LATTICE FORMATION. — inclusion of thermal emission terms

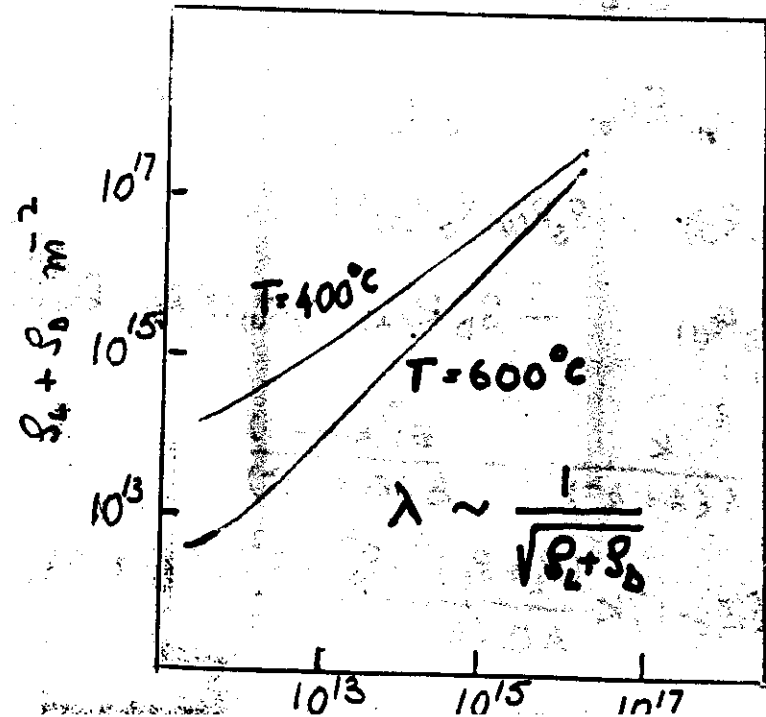
$$v_{th} = e^{-E_f/kT}$$

$$\left(\frac{dQ_s}{dt}\right)_{th} = -S_s D_v v_{th} e^{[(\frac{2\gamma}{r_s} - P_g)b^3/k_0T]}$$

$$\left(\frac{dQ_L}{dt}\right)_{th} = -S_L D_v v_{th} e^{[(\gamma_{sf} + E_{el}(r_L))b^2/k_0T]}$$

$$\left(\frac{dQ_R}{dt}\right)_{th} = +S_R D_v v_{th}$$

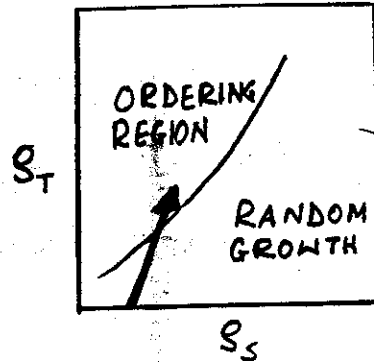
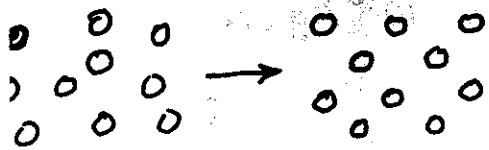
$$K_v = K(1-\epsilon) + K_{th}$$



SUMMARY OF THE RESULTS FOR VOID LATTICE FORMATION

Control Parameter

$$\frac{\Delta \epsilon}{\epsilon} = \frac{s_D + s_L - s_S}{s_D + s_L + s_S} > 1$$



Wave Length or Spacing $\lambda \sim (\text{sink strength})^{-1/2}$

Temperature dependence of λ
 $\lambda \uparrow$ as $T \uparrow$

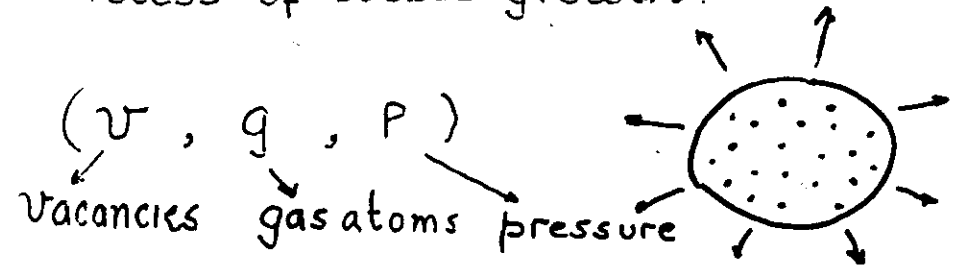
Dose Rate dependence
weak dependence

Critical Dose
Decided by the control parameter

2. RELATED MODELS FOR ORDERING - BUBBLE SUPERLATTICE FORMATION.

- Bubble lattice is formed by implantation of 2.0 - 40 keV (H/He) ions
- At these energies there is no cascade production and hence no vacancy loop. We have to therefore look for an alternate process for an instability to develop giving rise to a periodic structure.

Process of bubble growth.



$$\text{Volume } V_b = v \Omega = \frac{4\pi}{3} r_i^3$$

$$\left. \begin{array}{l} \text{Surface} \\ \text{Energy} \end{array} \right\} T_i = S(v)T = 4\pi r_i^2 T$$

Let us assume that the bubble acquires an additional δg gas atoms which results in further overpressurization.

$$(v, g + P) + \delta g \rightarrow (v, g + \delta g, P + \delta P)$$

The over pressurized bubble relaxes by giving out an interstitial and increases the volume by Ω .

$$(v, g + \delta g, P + \delta P) \rightarrow (v+1, g + \delta g, P) + i$$

The new volume and surface energy become

$$V_b(v+1) = \frac{4\pi}{3} r_2^3 \quad \& \quad T_2 = T_1 + \delta T = 4\pi r_2^2 T$$

Conserving the energy, we have (E_i^F = interstitial formation energy)

$$V_b(v) \delta P = \delta g k_B T = \Omega P + \delta T + E_i^F$$

$$\therefore \delta g = \frac{\Omega P + \delta T + E_i^F}{k_B T}$$

Let a fraction αK gas atoms be injected in the metal per second.

Then

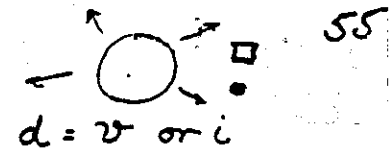
$$\frac{dg}{dt} = \alpha K$$

$\therefore$ No of interstitials produced per sec. will be

$$E = \frac{1}{K \delta g} \frac{dg}{dt} = \frac{\alpha k_B T}{\Omega P + \delta T + E_i^F}$$

Elast Factor:

$$Z_d = 1 - \frac{3\mu_d \delta P^2}{56 k_B T G^2}$$



μ_d = elastic polarizability $\begin{cases} \mu_v \approx -15 \text{ eV} \\ \mu_i \approx -150 \text{ eV} \end{cases}$

$$\delta P = \left\{ \left(P - \frac{2\gamma}{r} \right) \right\} \approx 2 \text{ GPa}$$

$$G = \text{shear modulus} \approx 10^{10} \text{ kg/m}^2$$

$$k_B T \approx 0.025 \text{ eV}$$

$$Z_v = 1.013 \quad Z_i = 1.13$$

$$\Delta Z \approx 11\% \quad (\text{Wolfer et al ASTM STP 570 (1975) 233})$$

STABILITY ANALYSIS

$$\frac{dv}{dt} = K - D_v \nabla^2 v + D_v \nabla^2 v$$

$$\frac{di}{dt} = K(1+\epsilon) - Z D_i \nabla^2 i + D_i \nabla^2 i$$

$$\frac{dq}{dt} = (D_v v - Z D_i i) \nabla^2$$

$$Q = Q_0 + \delta Q e^{i k \cdot r} e^{i \omega t}$$

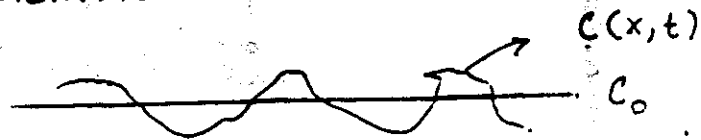
$$\omega(k^2) = -\frac{EK}{S_0} + \left[\frac{(Z\epsilon + Z - 1)Kk^2 + \epsilon Z K S_0}{(Z S_0 + k^2)(S_0 + k^2)} \right]$$

B PERIODIC DECOMPOSITION IN ALLOYS

- RADIATION ENHANCED PROCESSES
- RADIATION INDUCED MECHANISMS

Reaction Controlled Diffusion Controlled

RADIATION ENHANCED PROCESSES



Local Conc. fluctuation.

$$\delta c(x,t) = [c(x,t) - c_0]$$

$$\delta c(x,t) = \sum_{\underline{k}} A(\underline{k},t) e^{i\underline{k} \cdot \underline{x}}$$

Equation of motion for the amplitudes is $\frac{d}{dt} A(\underline{k},t) = \alpha(\underline{k},t) A(\underline{k},t)$

$\alpha(\underline{k},t)$. Amplification factor.

$$\therefore A(\underline{k},t) = \exp \left\{ \int_0^t \alpha(\underline{k},t') dt' \right\} \cdot A(\underline{k},0)$$

If we have a multicomponent system

$$\frac{d}{dt} A_j(\underline{k},t) = \sum_{\underline{l}} S_{jl} A_l(\underline{k},t)$$

S_{jl} → Stability matrix.

behaviour of the alloy will be given by the maximum eigenvalue of the stability matrix S

$$\alpha_{\text{eff}}(\underline{k}) = \max \omega_j(\underline{k})$$

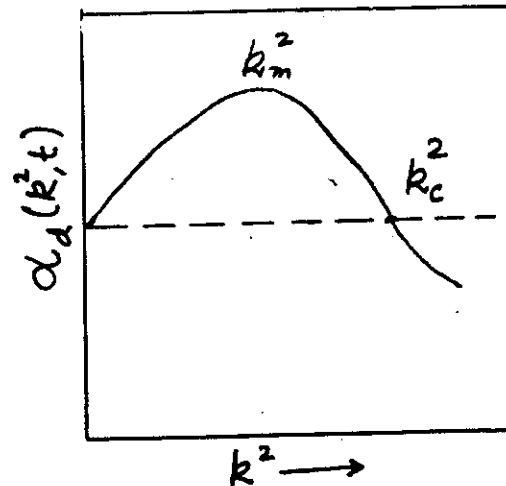
$$S = \underbrace{S_d}_{\text{diagonal}} + \underbrace{S_{nd}}_{\text{non diagonal}}$$

$$\alpha_{\text{eff}} = \alpha_d + \alpha_{nd}$$

Contribution from off diagonal terms.

Important for irradiation conditions.

Thermodynamic Model for spinodal decomposition



Cahn $\alpha_d = a k^2 (k_c^2 - k^2)$

Cook $= a(k,t) k^2 (k_c^2 - k^2)$

Langer $= a(k,t) (k_c^2(t) - k^2)$

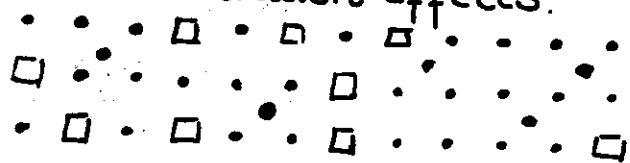
$$\alpha_{\text{cahn}} = -M_{\text{therm}} k^2 (f'' + 2Gk^2)$$

$M_{\text{therm}} = G, D_{\text{therm}}$ G is the gradient energy term

INFLUENCE OF RADIATION

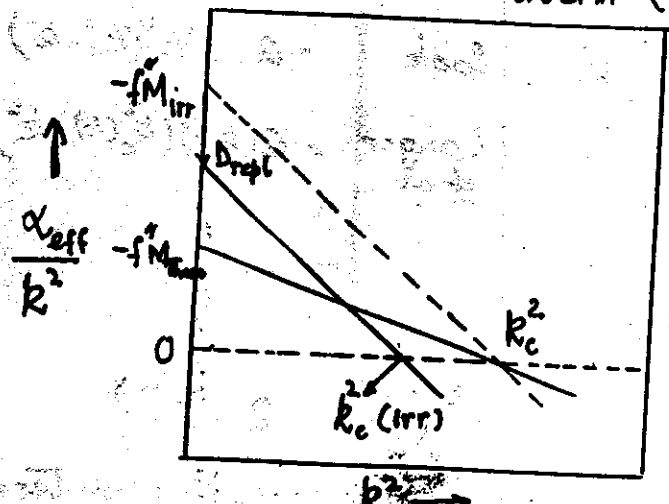
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Defect concentration effects.



(1) Free energy f and f'' is not significantly influenced by the irradiation produced defects. This is because the stored energies in the defect configurations is small as compared to the thermal phase change energy involving the atoms.

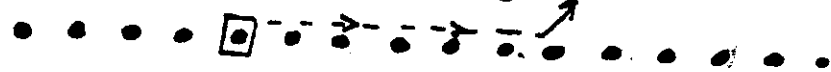
(2) However defects significantly influence the mobility due to additional transport mechanisms
 $M_{irr} = M_{therm} (D_{irr}/D_{therm})$



$$\frac{\alpha_{cahn}}{k^2} = -M_{therm} (f'' + 2KR^2)$$

λ_{max}
 Unaffected!

Replacement mixing



$$D_{repl} = \bar{C}_B D_{repl}^A + C_A D_{repl}^B$$

$$= \frac{1}{6} b^2 K_{repl} \langle n \rangle$$

$$D_{repl}^{A,B} = \frac{1}{2} b^2 \sum_{\alpha=1}^2 \cos^2 \alpha T_{\alpha}^{A,B} \quad T_{\alpha}^{A,B} = K_{repl} \sum_{n=1}^{\infty} n p(n) p(\alpha, n)$$

$$D_{repl}^{ions} \approx 3 \cdot 10^{-14} \text{ cm}^2 \text{ K} \quad D_{repl}^{elec} \sim 1 D_{repl}^{ions}$$

$$\left(\frac{dSC(x,t)}{dt} \right)_{repl} = D_{repl} \nabla^2 SC(x,t)$$

$$\frac{d}{dt} A(k,t) = -D_{repl} k^2 A(k,t)$$

$$\frac{\alpha_{eff}}{k^2} = \frac{1}{6} D_{repl} - M_{irr} \{ f'' + 2Gk^2 \}$$

Unlike radiation enhanced diffusion replacement mixing can shift k_c^2 and k_{max}^2 to lower values of k or larger λ . Thus only longer wavelengths may be stable.

Stability of the alloy depends on Temp T Through $M_{irr}(T) \propto f''(T)$. The phase boundary can be determined.

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Compositional changes by defect reactions

Consider an A, B alloy then

$$\xi_A = \frac{n_A}{n_A + n_B} \quad \xi_B = \frac{n_B}{n_A + n_B}$$

$$n_A + n_B + n_V = n_L = \text{const}$$

$$\frac{d}{dt} \xi_A = \frac{\xi_B \frac{d}{dt} n_A - \xi_A \frac{d}{dt} n_B}{n_A + n_B}$$

Changes in n_A and n_B can be due to the following processes.

1) Production of a Frenkel Defect

$$\left. \frac{d}{dt} n_{A,B} \right|_{\text{production}} = -K_{A,B}^v$$

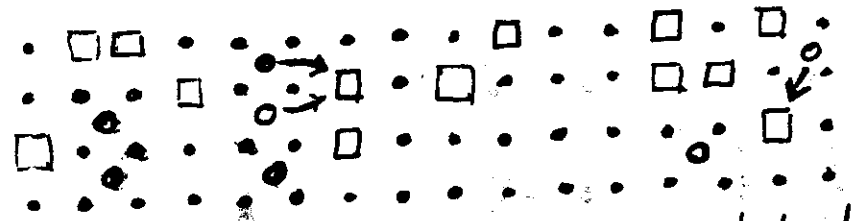
$$\text{where } K_{A,B}^v = \sigma_{A,B} \Phi \xi_{A,B}$$

2) Recombination

i_A, i_B interstitial concentration
and $v = n_V/n_L$ Vacancy Concentration

$$\frac{d}{dt} n_{A,B} = \alpha_{A,B} i_{A,B} v$$

$\alpha_{A,B}$ → Recombination Constant.

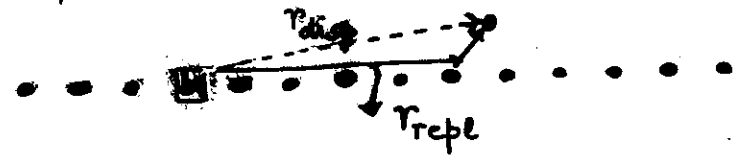


Recombination of A interstitial in A-rich region is more probable than B-interstitial in A-rich region

$$\alpha_{A,B} = \alpha_{A,B}^0 + \gamma_{A,B} \xi_{A,B}$$

This can give rise to uphill diffusion processes.

(3) Replacement chain mixing.



$$K_{A,B}^i(\underline{r}) = \Phi \int \sigma_{A,B} \xi_{A,B}(\underline{r}') \omega(\underline{r} - \underline{r}') d\underline{r}'$$

$$\omega(\underline{r} - \underline{r}') \propto \delta(\underline{r}_{\text{disp}} - |\underline{r} - \underline{r}'|)$$

$$K_{A,B}^i(\underline{r}) = (1-m) K_{A,B}^v(\underline{r}) + m \langle K_{A,B}^v \rangle$$

$$m = \cos(k r_{\text{disp}})$$

(4) Interstitial exchange reactions ⁶²



$$\left. \frac{d}{dt} i_A \right|_{\text{exch.}} = -v_A \xi_B i_A + v_B \xi_A i_B = - \left. \frac{d}{dt} i_B \right|_{\text{exch.}}$$

$$\left. \frac{d}{dt} \xi_{A,B} \right|_{\text{exch.}} = - \left. \frac{d}{dt} i_{A,B} \right|_{\text{exch.}}$$

(5) Diffusion coupling

$$\frac{d}{dt} v_{\text{diff}} = D_v \nabla^2 v; \quad \left. \frac{d}{dt} i_{A,B} \right|_{\text{diff}} = D_{i_{A,B}} \nabla^2 i_{A,B}$$

Rate equations ⁶

$$\frac{d}{dt} \xi_A = -\xi_B (K_A^v - \alpha_A i_A v) + \xi_A (K_B^v - \alpha_B i_B v)$$

$$\frac{d i_A}{dt} = K_A^i - \alpha_A i_A v + D_{i_A} \nabla^2 i_A + \left. \frac{\partial}{\partial t} i_A \right|_{\text{exch.}}$$

$$\frac{d i_B}{dt} = K_B^i - \alpha_B i_B v + D_{i_B} \nabla^2 i_B + \left. \frac{\partial}{\partial t} i_B \right|_{\text{exch.}}$$

$$\frac{d v}{dt} = K_A^v + K_B^v - (\alpha_A i_A + \alpha_B i_B) v + D_v \nabla^2 v$$

$$\approx S = \begin{bmatrix} -D_{i_A} k^2 - \alpha_A \bar{i} - v_A \xi_B & v_B \xi_A & -\alpha_A \bar{i}_A & z_A - \Delta z_A \\ v_A \xi_B & -D_{i_B} k^2 - \alpha_B \bar{i} - v_B \xi_A & -\alpha_B \bar{i}_B & -(z_B - \Delta z_B) \\ -\alpha_A \bar{v} & -\alpha_B \bar{v} & -D_v k^2 - (\alpha_A \bar{i}_A + \alpha_B \bar{i}_B) & (z_A - z_B) \\ (\alpha_A \bar{v} + v_A) \xi_B & -(\alpha_B \bar{v} + v_B) \xi_A & \alpha_A \bar{i}_A \xi_B - \alpha_B \bar{i}_B \xi_A & -(z_A \xi_B + z_B \xi_A) \end{bmatrix}$$

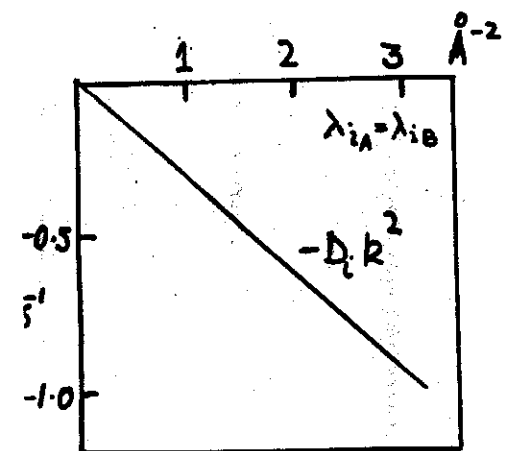
where

$$z_A = \frac{\partial}{\partial \xi_A} K_A^v - \bar{i}_A \bar{v} \frac{\partial \alpha_A}{\partial \xi_A} + v_A \bar{i}_A + v_B \bar{i}_B$$

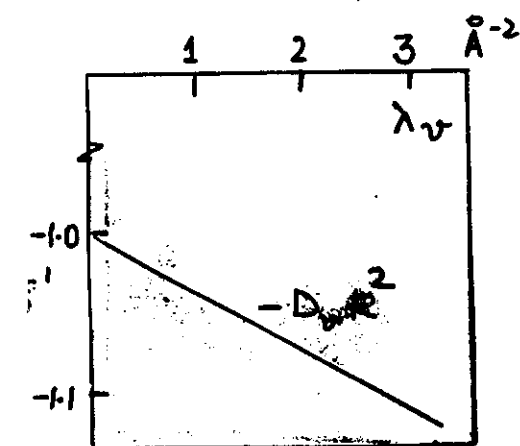
$$\Delta z_A = \frac{\partial}{\partial \xi_A} (K_A^v - K_A^i)$$

$$|S| = \lambda_{i_A} \cdot \lambda_{i_B} \cdot \lambda_v \cdot \lambda_{\xi_A}$$

k^2 - Dependence of the eigenvalues. ⁶⁴

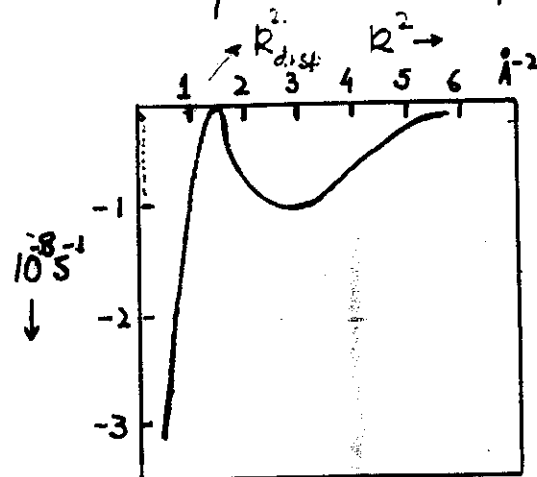


$$D_{iA} = D_{iB}$$

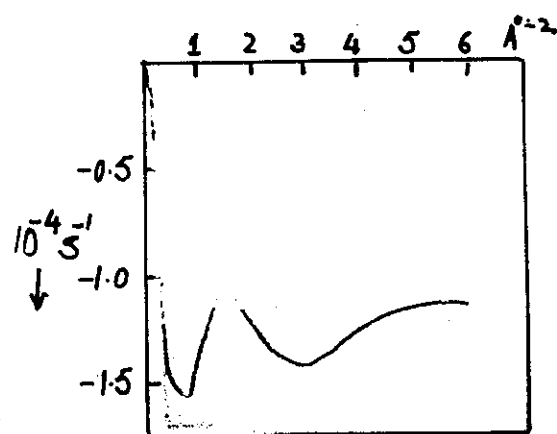


$$D_{iA} = D_{iB}$$

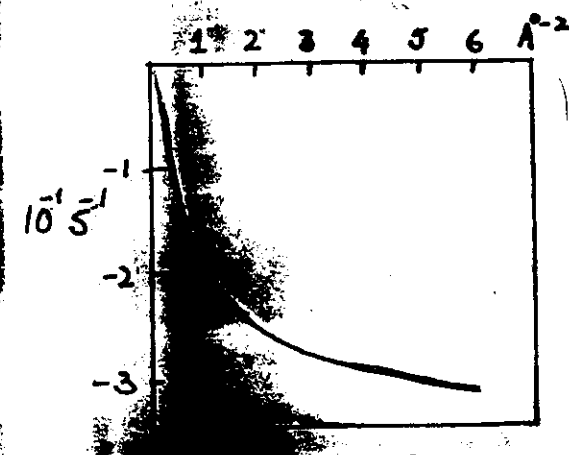
k^2 - Dependence of the eigenvalue λ_{ξ_A}



$$\nu_A = \nu_B = 0$$



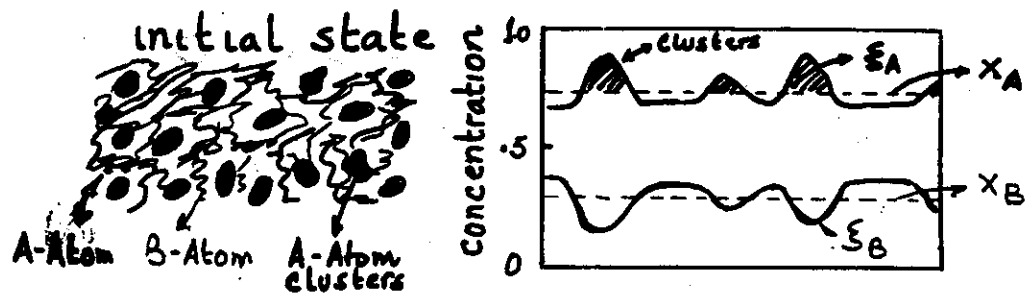
$$\nu_A = 0 \quad \nu_B \neq 0$$



$$\nu_A \neq 0 \quad \nu_B \neq 0$$

A MODEL FOR CuNi

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$$\eta = \frac{1}{\Omega_s} \int_{\Omega_s} (\xi_A - X_A) \xi_A d\Omega$$

$$\frac{dI_A}{dt} = K_A - (\alpha_A + \gamma_A \eta) I_A V$$

$$\frac{dI_B}{dt} = K_B - (\alpha_B + \gamma_B \eta) I_B V$$

$$\frac{d\eta}{dt} = D_A (Z_A \eta + \eta_s) I_A - D_B (Z_B \eta + \eta_s) I_B$$

This shows a similar coupling as in the case of vacancy loops.

$$\lambda \sim 60 \text{ \AA}$$