



INTERNATIONAL ATOMIC ENERGY AGENCY
UNITED NATIONS EDUCATIONAL, SCIENTIFIC AND CULTURAL ORGANIZATION
INTERNATIONAL CENTRE FOR THEORETICAL PHYSICS
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UNITED NATIONS INDUSTRIAL DEVELOPMENT ORGANIZATION



INTERNATIONAL CENTRE FOR SCIENCE AND HIGH TECHNOLOGY

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H4.SMR/676-25

**SECOND SCHOOL ON THE USE OF SYNCHROTRON
RADIATION IN SCIENCE AND TECHNOLOGY:
"JOHN FUGGLE MEMORIAL"**

25 October - 19 November 1993

Miramare - Trieste, Italy

EXAFS

**A. Fontaine
L.U.R.E.
Orsay, France**

III

EXAFS

Model : single scattering
of the photoelectron

Data analysis

Data Collection

Examples.

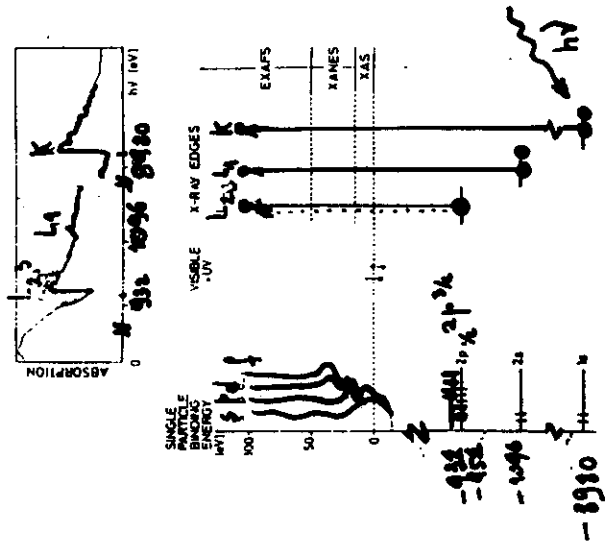


Fig. 2 Description schématisée des transitions optiques et de structure de cœur dans une large gamme d'énergie (J. Guilleminot, 1999).

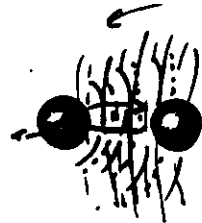
"Transition" à une "particule"

$$\mu \sim \sum_f |\langle f | \vec{R} \cdot \vec{E} | i \rangle|^2 \delta(E_f - E_i + \hbar\omega)$$

$|f\rangle = |f_0\rangle + \text{ondulatos (rétro) diffusion}$

Règles de sélection $\Delta l_{if} = \pm 1 \quad \Delta m = 0$

EXAFS SIGNAL
for



Back scattering

$$f_b(\pi, k) = \frac{1}{2ik} \sum_l (2l+1) (e^{2i\delta_l} - 1) P_l(-1)$$

Coming back on the central atom

$$\frac{e^{ikR_j}}{R_j} e^{i\int_0^k \delta_j(k') dk'}$$

$$\chi(k) = \frac{\mu - \mu_0}{\mu_0} = \frac{-|f_b(\pi, k)|}{\mu_0} \sin(2kR_j + \theta_j)$$

[2δ_j + θ_j]

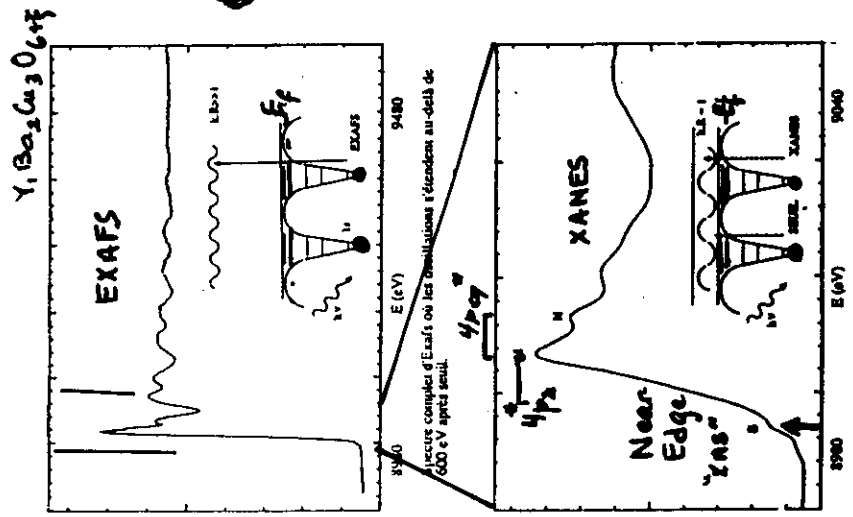
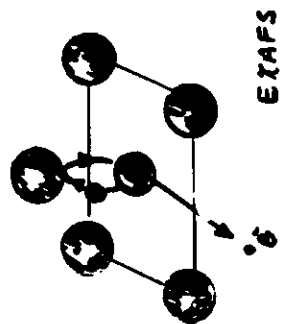
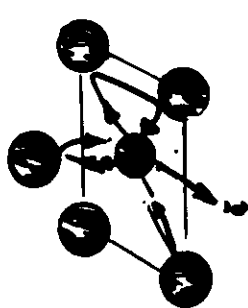


Fig. 3 Séparation schématisée d'un spectre d'absorption en deux zones.

Cu. Seuil K



EXAFS



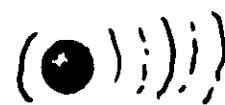
XANES

III. 3

Heuristic derivation

$$\mu = \sum_f |\langle i | \vec{R} \cdot \vec{E} | f \rangle|^2 \delta(E_f - E_i - \hbar\omega)$$

ISOLATED ATOM



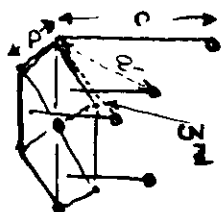
$$|f_0\rangle \sim \frac{Y_{l,0}(\frac{\vec{R}}{R}) h_l^l(kR) e^{i\delta_l^f(k)}}{2kR_j}$$

≡ OUT GOING WAVE
~ plane wave @ R_j

SURROUNDED EMITTER

$$|f\rangle = |f_0\rangle (1 + \chi)$$

III.3 c DERBYE-WALLER DAMPING

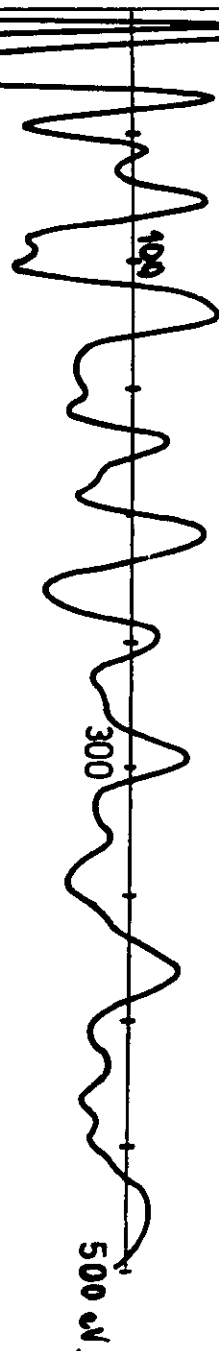


$$a = 2.912 \text{ \AA} \quad a = 2.66 \text{ \AA}$$

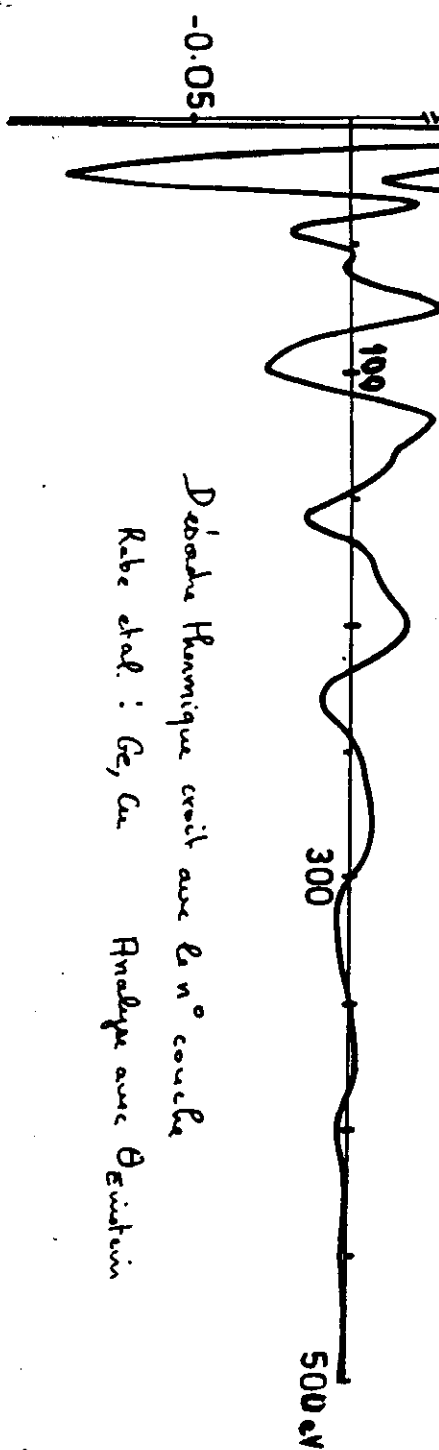
$$c/a = 1.86$$

$u_3 = 3.925 \text{ \AA}$ at 25 K.
X-ray diffraction data

Zn: 25 K



Zn: 300 K



Debye-Henriette crest and E_n could be
Rabe et al.: Ge, Cu Analysis and Distribution

23

$$\chi(k) = \frac{\mu}{\mu_0}$$

$$\chi(k) = -\frac{1}{k} \sum_j \frac{1}{R_j^2} \left| f_j(\pi, k) \right| \sin(2k R_j + 2\phi_j(k) + \theta) \cdot e^{-2\sigma_j^2 k^2} \cdot e^{-\frac{R_j}{\lambda}}$$

Powder $\sum M_j$
or
cubic lattice shell

$$\cdot \frac{\hbar^2 k^2}{2m} = E - E_0 \quad \text{origin of the kinetic energy}$$

$$\cdot e^{-\frac{R_j}{\lambda}} \quad \text{mean free path of the photoelectron}$$

$$\cdot e^{-2\sigma_j^2 k^2} \quad \text{D.N. static or dynamic disorder}$$

ADVANTAGES of EXAFS.

(I) SELECTIVITY of the "ELECTRON" SOURCE

$$\text{Cu } K \quad 8979.8 \quad \text{Zn } K \quad 9659 \quad \text{Eu } L_{III} \quad 6977 \text{ eV.}$$

(II) GOOD ACCURACY IN BOND LENGTH DET
0.01 \AA

(III) CHEMICAL IDENTIFICATION of NEIGHBORS
& COORDINATION NUMBERS
 ΔZ large enough, $\pm \frac{1}{4}$ in best cases
(out of 12)

NEEDS

PHASE SHIFTS
& BACKSCATTERING AMPLITUDES

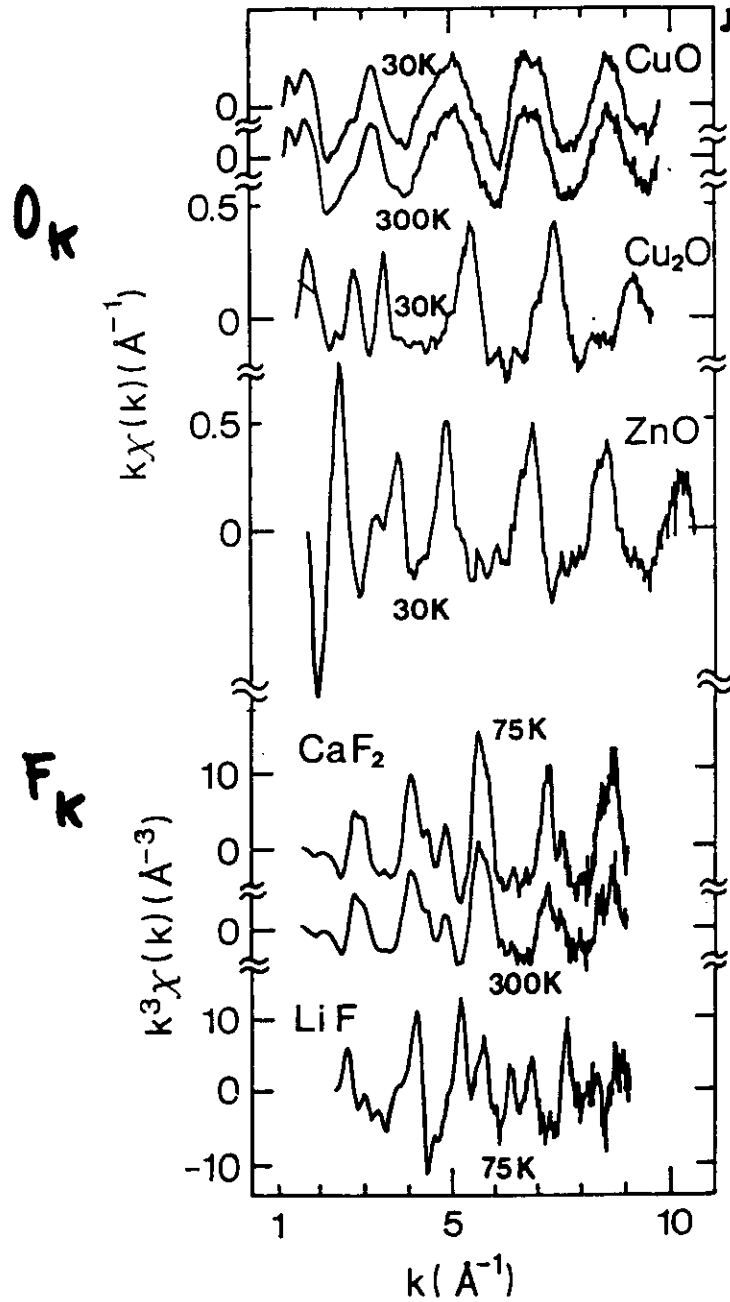
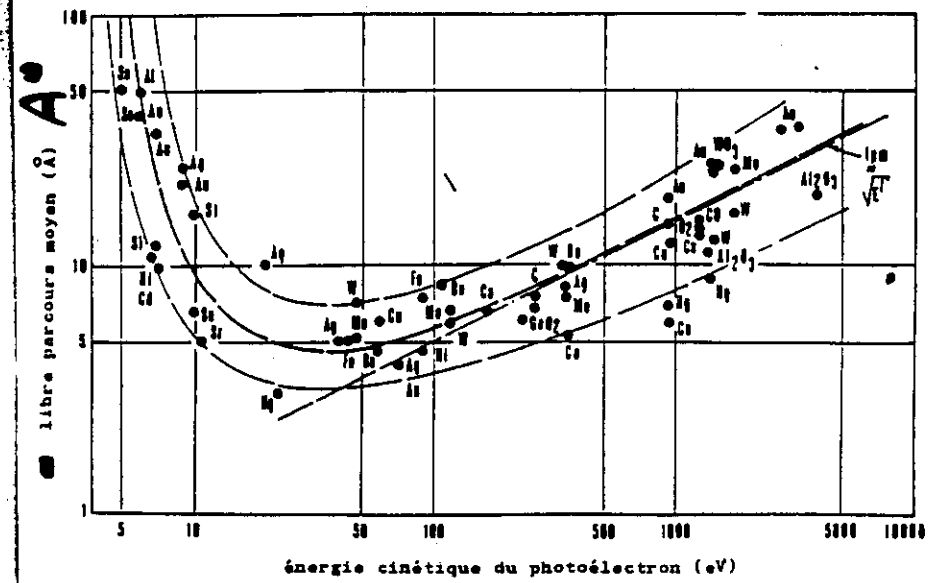


Fig. 3

Mean free path of the photoe



→ eV

$\lambda \propto k$
 $E_{kin} > 50 \text{ eV}$

Figure 1.6 - d'après C.R. Brundle (référence / 3 /).

Libre parcours moyen des électrons dans un solide en fonction de leur énergie cinétique.

"universal" ?

$\lambda \sim k$ a.u.

SUMMARY

EXAFS $\approx$ SPHERICAL LEED $\approx$ DIFFRACTION OF LOW ENERGY ELECTRON

where the source of $\bar{e}$ is
INSIDE THE SAMPLE
ON A SELECTED ATOM

EXAFS AND OTHER SCATTERING TECHNIQUES

◆ X-RAY SCATT. PAIR CORRELATION FUNCT.
Binary alloy $\begin{matrix} AA \\ AB \\ BB \end{matrix}$

Bad News EXAFS does not see long distances

Good News EXAFS is accurate to evaluate local ORDER

◆ NEUTRONS $b_{\text{O}} > b_{\text{Fe}}$
GOOD CONTRAST
EXAFS $f(\pi, \lambda)$ & PHASE

◆ X-Ray WEAKER INTERACTION THAN $\bar{e}$
STRONGER ——— THAN γ

Hard X-rays Sample "REASONABLE VOLUME"

The Iro-sro controversy

Along the previous sections, we have described the debate that occurred between the supporters of the l.r.o. (long-range order) and s.r.o. (short-range order) theories.

We could cite Hanawalt (1932): While it is true that the structure shown by XAFS: the rôle of the immediate surroundings in XAFS: differs in details from that displayed by the same substance in the solid state, yet the outstanding observation is the high degree of similarity between them. This fact probably means that the same explanation should be applicable to both of them. [or Coster (1935):

"As has been shown in several papers from this laboratory, the fine structure of X-ray absorption edges, in accordance with Kronig's theory, is intimately related to the type of crystal lattice in which the absorbing element occurs. Besides, however, it has been demonstrated that the immediate surrounding in the crystal lattice of the atoms in question is also of great importance."

SUCCESS

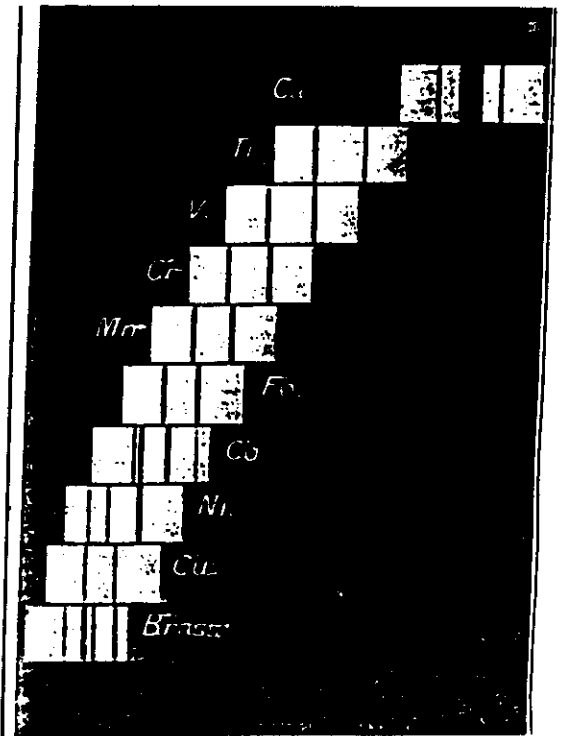
1934

W. HARTREE, D. HARTREE

Self consistent calculations of the wave functions to test KRONIG & PETERSEN'S MODEL with a MACHINE made of 20 BUNDS OF REELAND

GeCl₄: checked the crystal oscillation and reported "The discrepancy could be removed by assuming R_{GeCl} at $2.07 \text{ \AA}$ INSTEAD OF $(2.10 \pm 0.02 \text{ \AA})$ as known after WIERL". Actual value is $2.08 \text{ \AA}$!!!

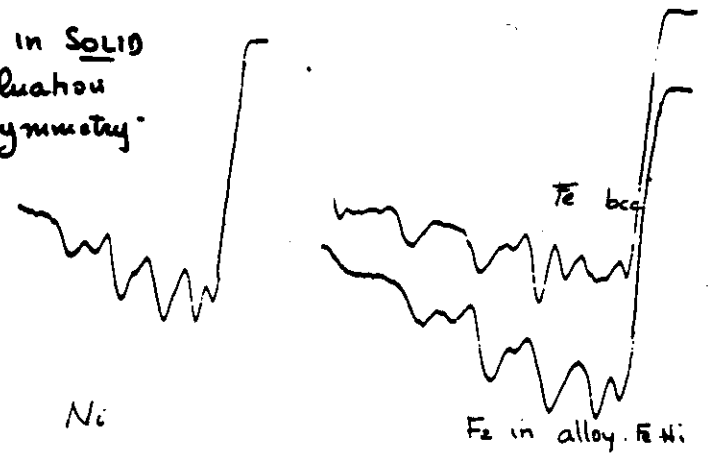
Coster, Kramer: AsCl₃ same success after modern reevaluation.



MOSELEY (1913)
Emission Lines $K\alpha, K\beta$ Ca $\rightarrow$ Zn

X-A - Fine Structure, Kronig Oscillations

Fine structure in SOLID
Successful evaluation of the size symmetry



Veldkamp (1933)

or Smoluchowski, struck by the analogy between the spectra

of bulk copper and disordered AuCu₃ (1935b): "Die Erklärung dieser Tatsache ist darin zu suchen, daß die in der Theorie gebrauchten Approximationsmethoden und der Begriff einer Bragg'schen Reflexion in diesem Energiegebiet der herausgeworfenen Elektronen nicht mehr brauchbar sind, und daß, wie öfters betont, die Schwankungen des Absorptionskoeffizienten hier mehr durch Atom- als durch Gittereigenschaften verursacht sind (The explanation of this fact must be sought, that the approximation method and the concept of Bragg reflection required by the theory is no longer necessary in that domain of the photoelectron energy. and that, as often stressed, the absorption coefficient structures depend here more on the atomic than on the lattice characteristics)."

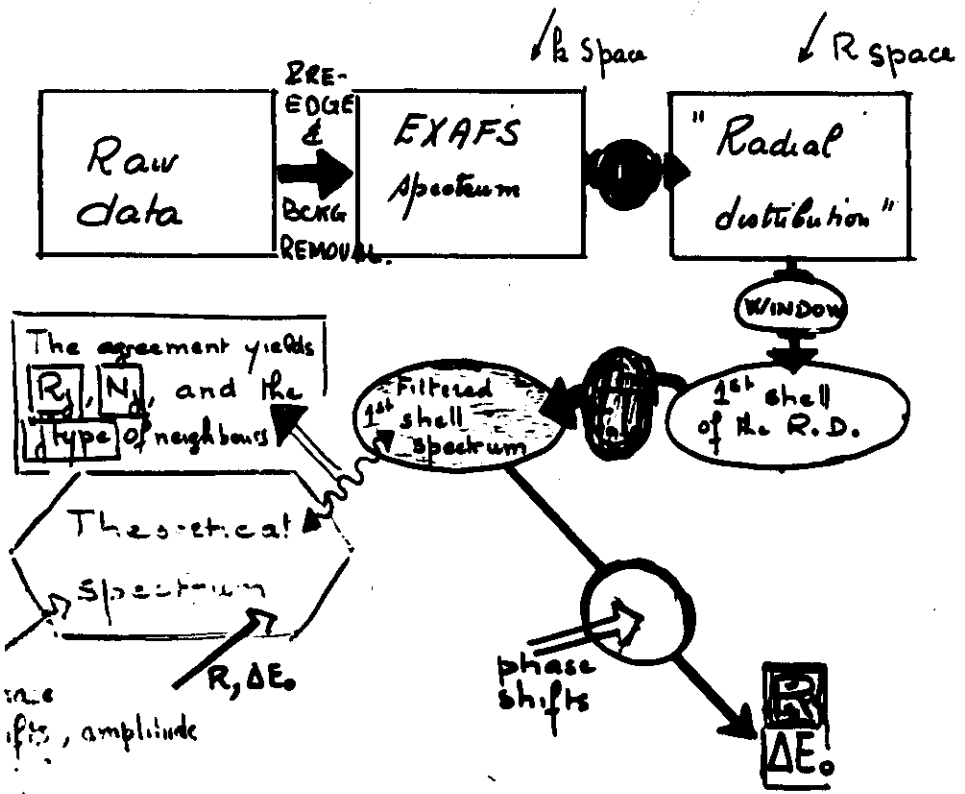
Strangely enough, this controversy was the opposition of two theories which were originally conceived by the same man.

Kronig

III.5 DATA ANALYSIS

a/ SCHEME

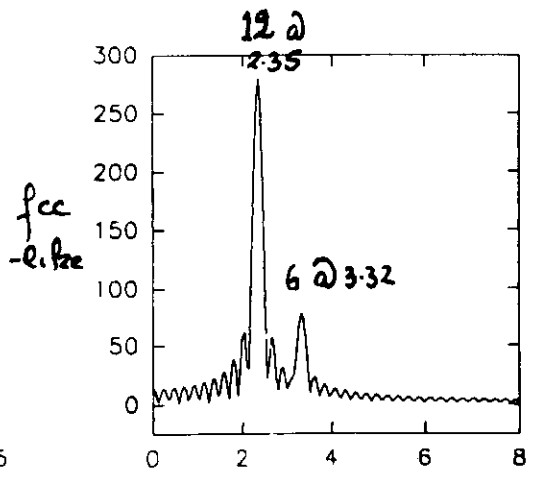
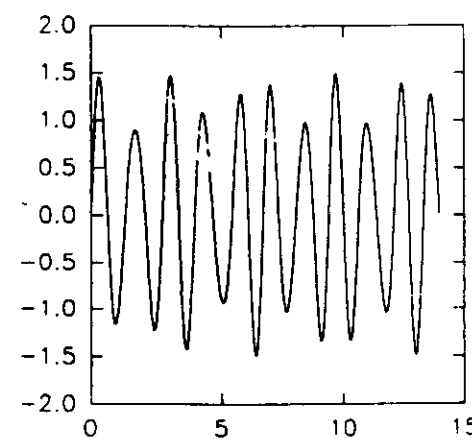
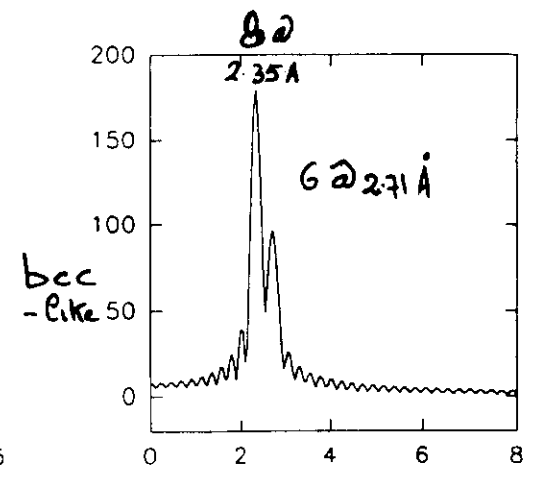
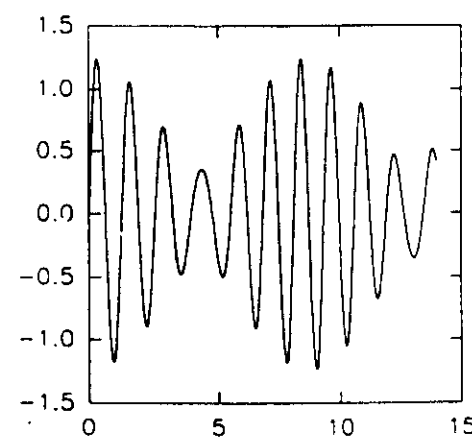
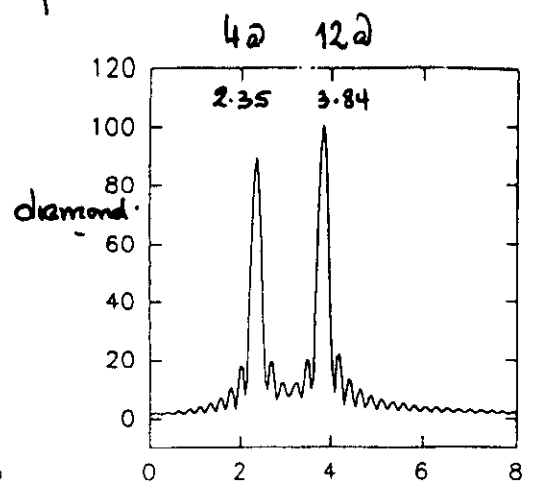
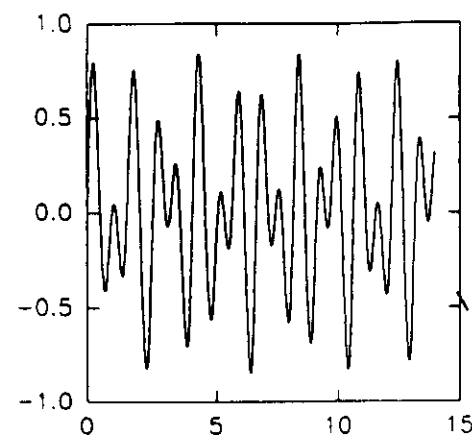
$$\begin{cases} \Delta E_0 = E_0 - E_{\text{threshold}} \\ \frac{\hbar^2 k^2}{2m} = E - E_0 \end{cases}$$

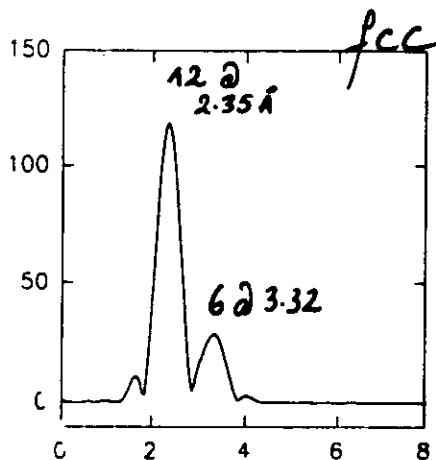
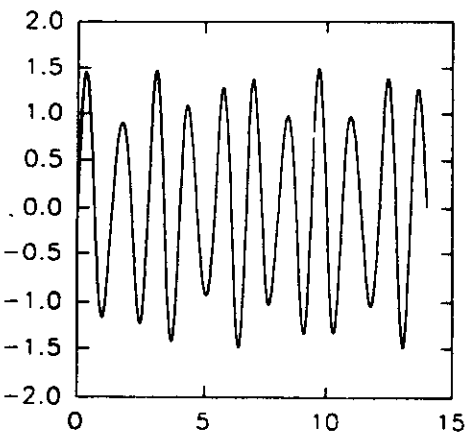
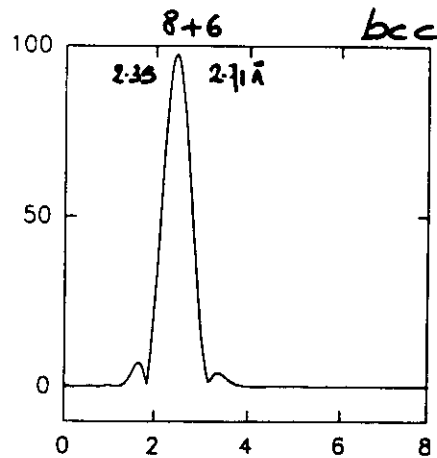
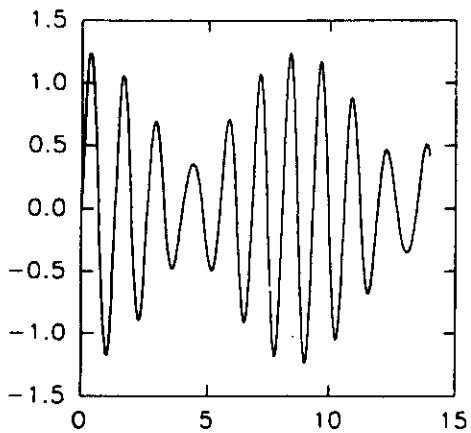
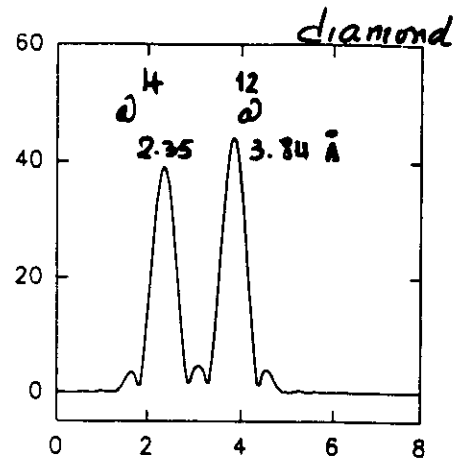
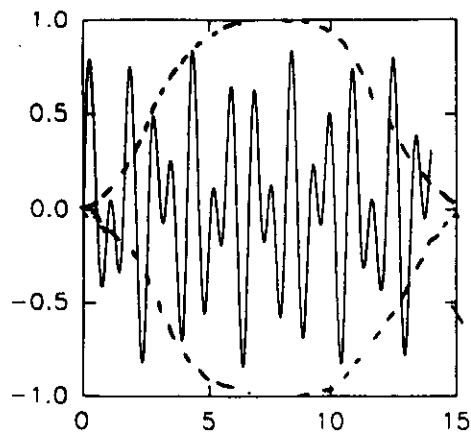


HETEROGENEOUS SHELL
HOMOGENEOUS SHELL

fitting

hypothetical comparison





III S.2

Phase shift & amplitudes

* Theoretical evaluations

Excellent for phase shifts

Not that Good for AMPLITUDES

* Best procedure to evaluate f_{AB} for the AB pair

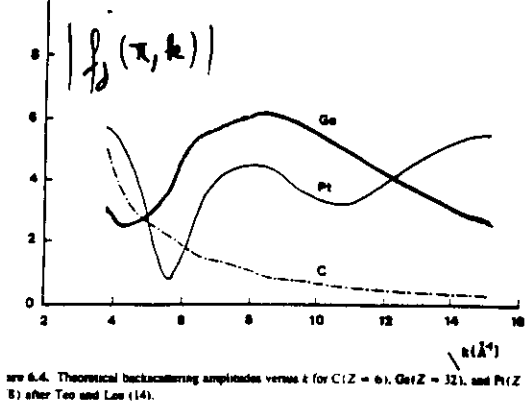
a/ Measure a well-known sample with a simple lattice with a structure close to the UNKNOWN sample with A as absorber B as backscatterer

b/ Extract phase shift $\phi(k)$ and amplitudes $|f(k, R)|$

c/ Transfer into the data analysis of the unknown sample.

* KEEP STANDARDS & INVESTIGATED SAMPLES AS CHEMICALLY CLOSE AS POSSIBLE

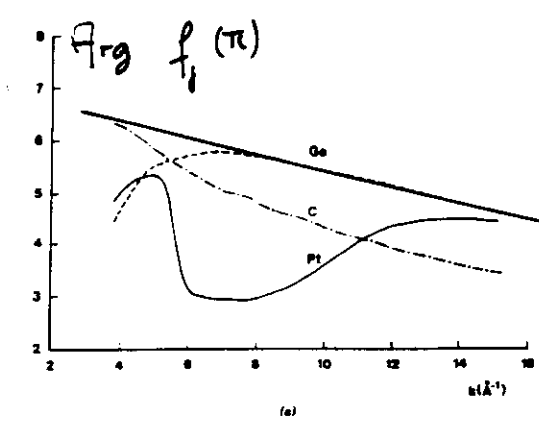
* WELL-CONDITIONED PROBLEM



≠ amplitudes

Total phase
 $2kR_j + \psi(k)$
 $\psi(k) = 2\delta_1(k) + \theta_j(k)$

• $\langle \frac{1}{2} \frac{d\psi(k)}{dk} \rangle \lesssim 0.40$
 low Z elements
 high Z : $\psi(k)$ non monotonic.



average slope
 -0.15 Å^{-1}
 $\frac{d\theta_j}{d(2k)} \sim 0.05 \text{ Å}^{-1}$

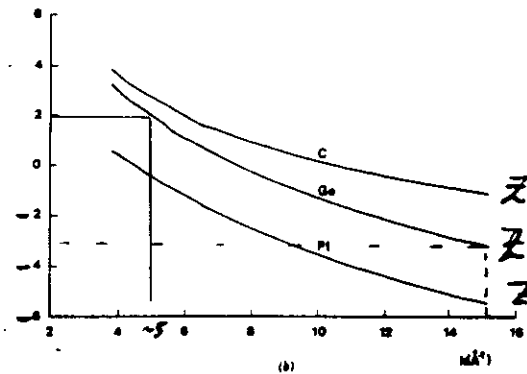
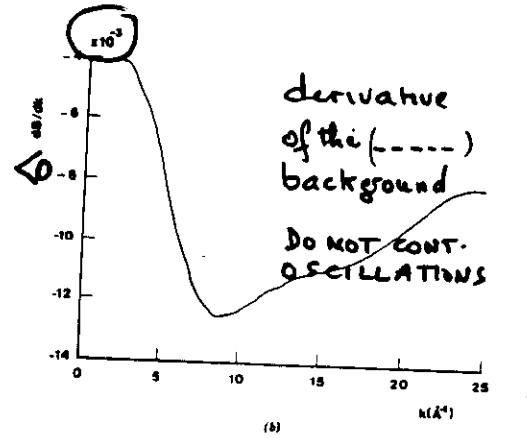
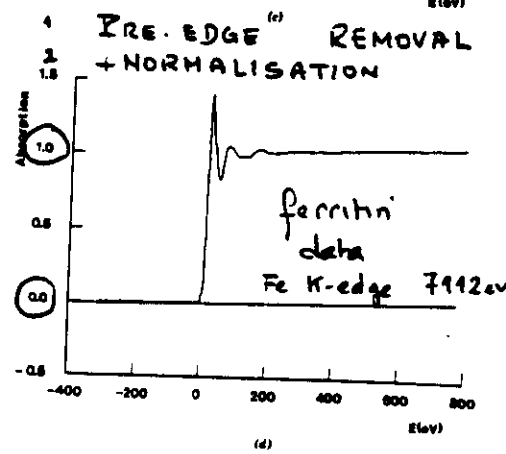
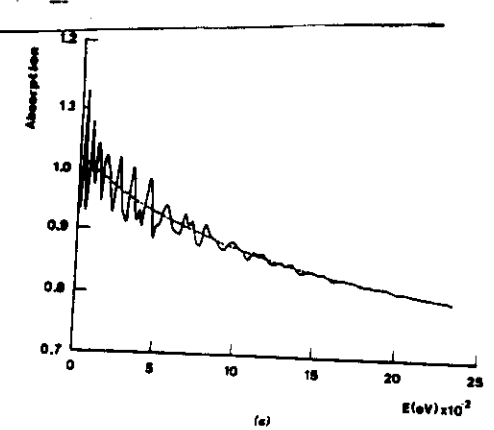
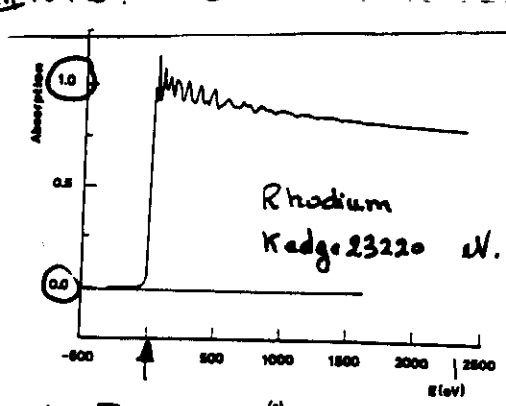


Figure 6.5. (a) Theoretical scattering phase shifts for carbon, germanium, and platinum versus k . Theoretical control atom phase shifts for carbon, germanium, and platinum. All calculations are in Teo and Lee (14).

$\bar{Z} = 6$ C
 $\bar{Z} = 32$ Ge
 $\bar{Z} = 78$ Pt

• $\langle \frac{1}{2} \frac{d(2\delta_1)}{dk} \rangle = -0.5 \text{ Å}^{-1}$

Phase = $2kR_j + 2\delta_1(k) + \theta(k) \Rightarrow \text{VERY GOOD EVALUATION}$



3 Choice of E_0 as the inflection point.

Figure 6.2. (a) The normalized data of Fig. 6.1c for rhodium metal showing on an expanded scale for the data above the edge. The calculated background curve used to extract the EXAFS is shown as a dashed line. (b) The derivative of this background curve versus the photoelectron wave vector k . (c) The normalized EXAFS for rhodium metal plotted versus k .

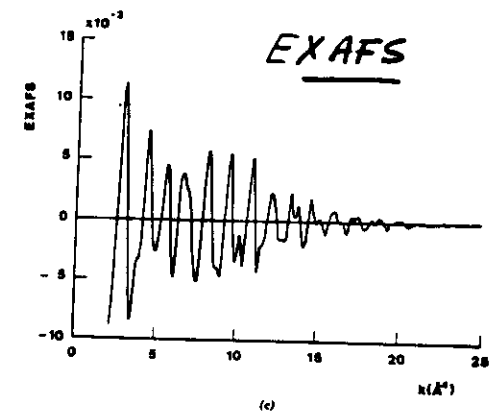
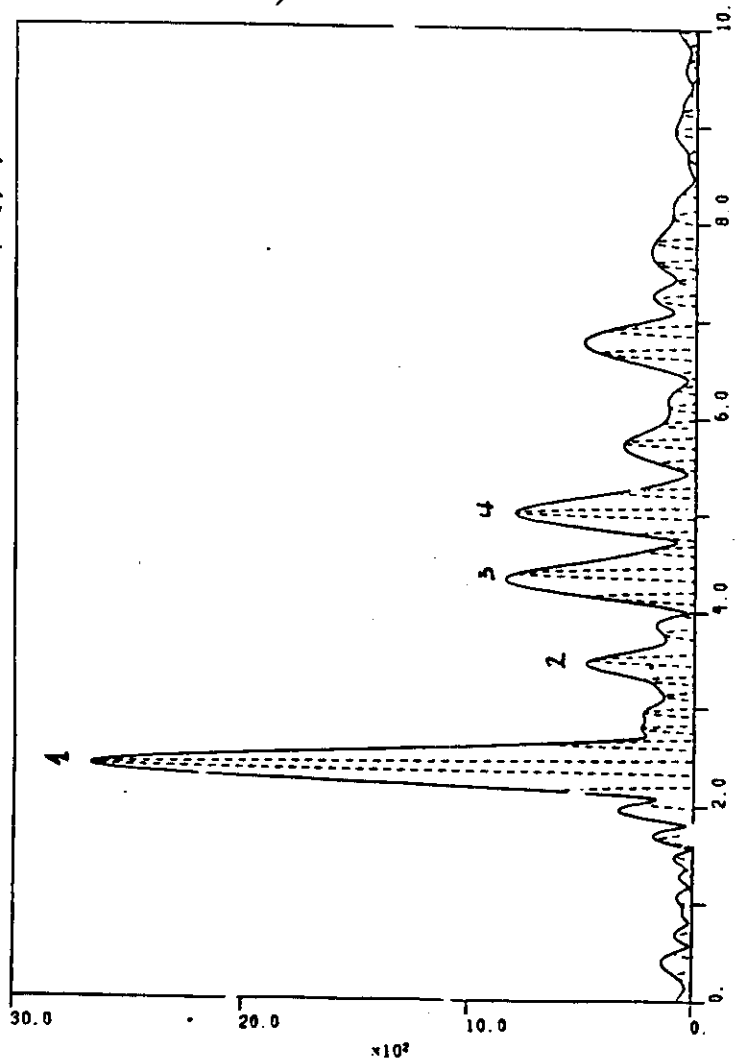


Figure 6.2. (Continued)

D. SAYERS, B.A. BUNKER
 Ch 6. EL Khatib, Zinn



PBK TS: 1- 12 GLTCHS: 0- 0- 0- 0- 0- 0- 0- 0-

0: 0- 0: 0- 0- 0- 0- EG= 0.0 CBK PTS: 74- 399 SH=0.

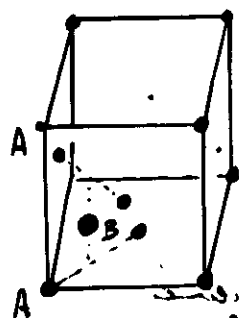
2620 WE=1.0000 r.SEG= 0 WT=3.00 FH= 0 TR: 2.60-18.40 RM=10.00 DR=.010

Rhodium Foil. (c) (Data SS2L)
D. SAYERS. NCSU. line 15 Feb 81.

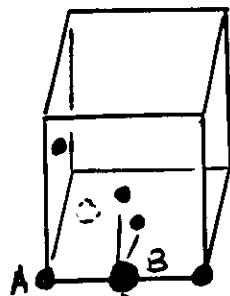
shell #	k-range	λ	N	R	$\Delta\sigma^2, 10^4 \text{ \AA}^2$	$E_0(\text{eV})$
1	5-17	2	12	2.69	2.23	-0.24
1	6-16	"	11.8	2.69	2.13	.42
2	5-17	"	4.77	3.80	1.96	-0.06
2	6-16	"	4.78	3.80	1.97	-0.24
3	5-17	"	24.7	4.67	3.12	-0.4
3	6-16	"	25.1	4.67	3.21	-0.4
4	5-17	"	12.1	5.38	3.14	+0.4
4	6-16	"	12.78	5.38	3.38	+0.2

$$\chi(k) = \sum_j \frac{N_j}{k R_j^2} e^{-2i k R_j^2} e^{-\frac{R_j}{\lambda(k)}} |f_j(\pi, k)| \sin(2kR_j + \gamma)$$

$$\gamma = 2\delta + \text{Arg}[f_j(\pi, k)]$$



TETRAH. (4)
Zinc Blende



Rocksalt OCTAH. (6)

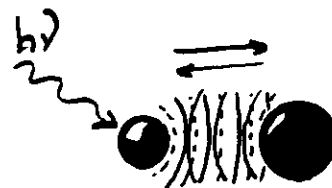
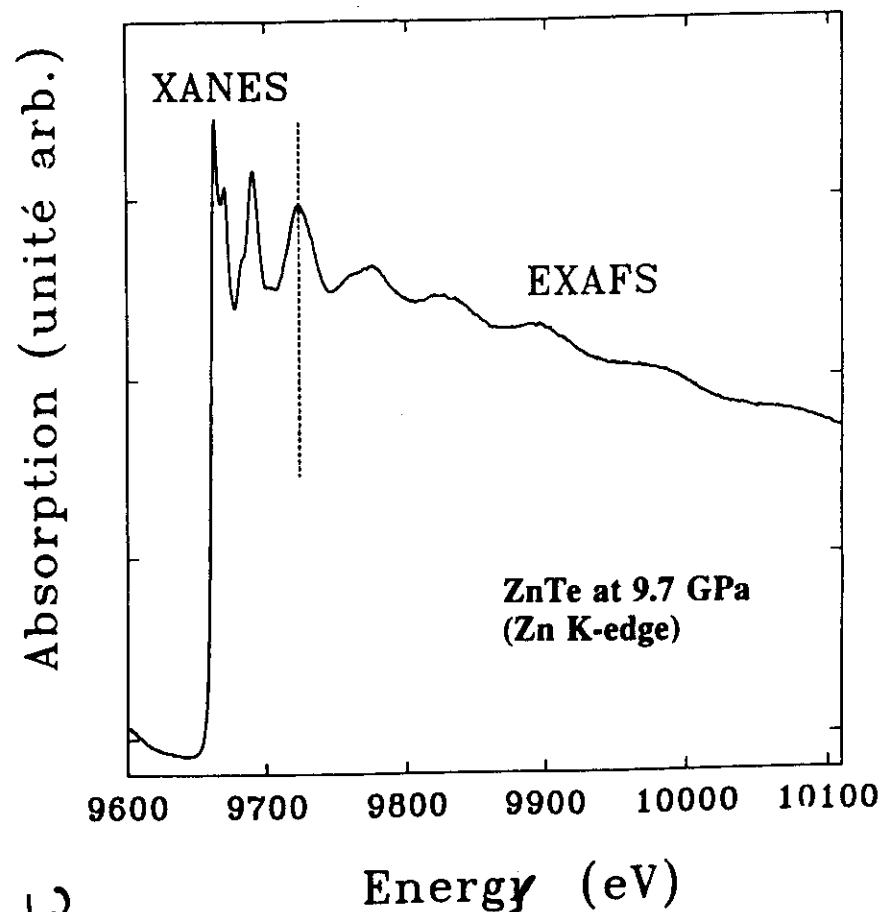
- Departure of $f_j(\pi, k)$ from atomic-like behavior

- NON SPHERICITY of the atomic potential

- Electrons on the A-B bond

- Smaller HP Cell. Go To A LARGER Number of LOW Z el's

X-ray absorption spectroscopy at high pressures



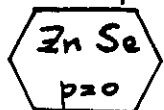
Simple "model"

$$\chi(k) = \sum_i \frac{N_i}{k R_i^2} A_i(k) \sin(2kR_i + \phi_i(k)) e^{-2k^2 \sigma_i^2} e^{-\frac{2R_i}{\lambda(k)}}$$

Zn-Se under high pressure



EXAFS spectra



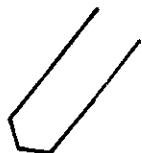
EXAFS ANAL.

PHASE SHIFTS

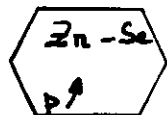
$\phi_{\text{Zn-Se}}$ $\phi_{\text{Se-Zn}}$

AMPLITUDE

$A_{\text{Zn-Se}}$ $A_{\text{Se-Zn}}$



FITTING ROUTINE



R_d
 N_d
 $\Delta\sigma_d^2$
Local Environment

Pressure-dependent EXAFS

~ Perturbative Approach.



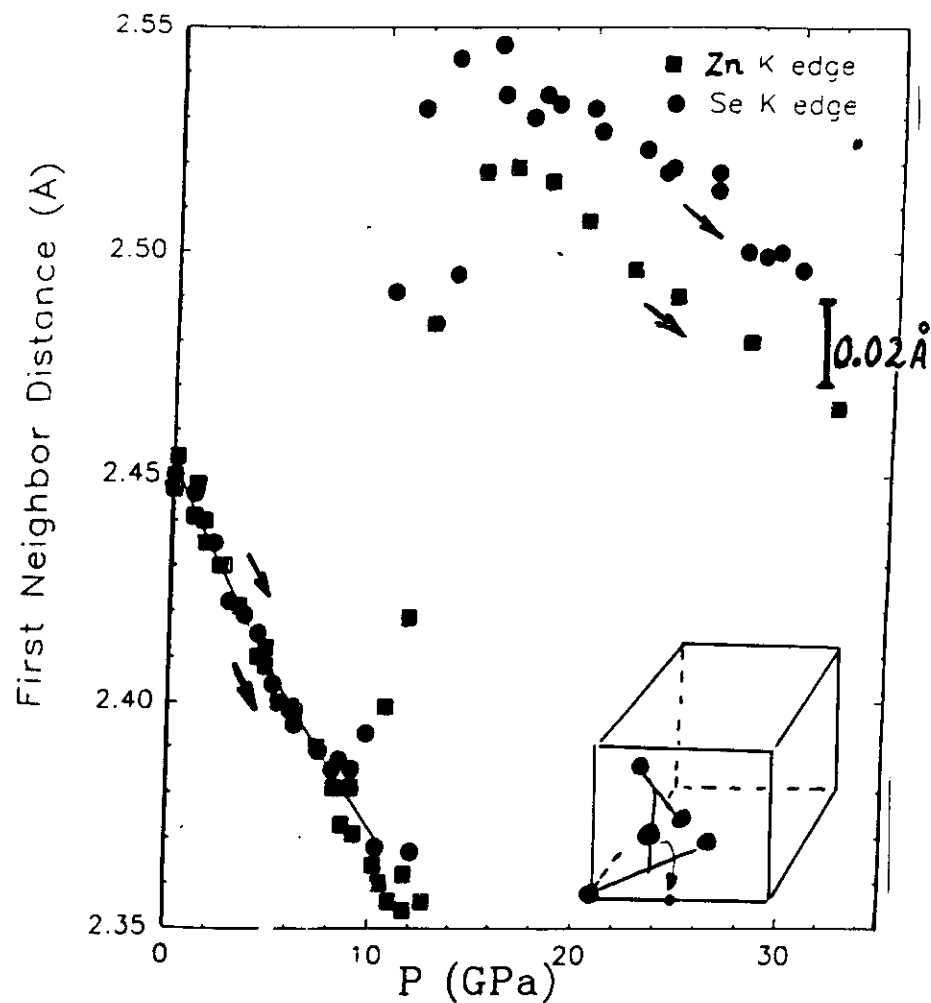
Znc. Blende

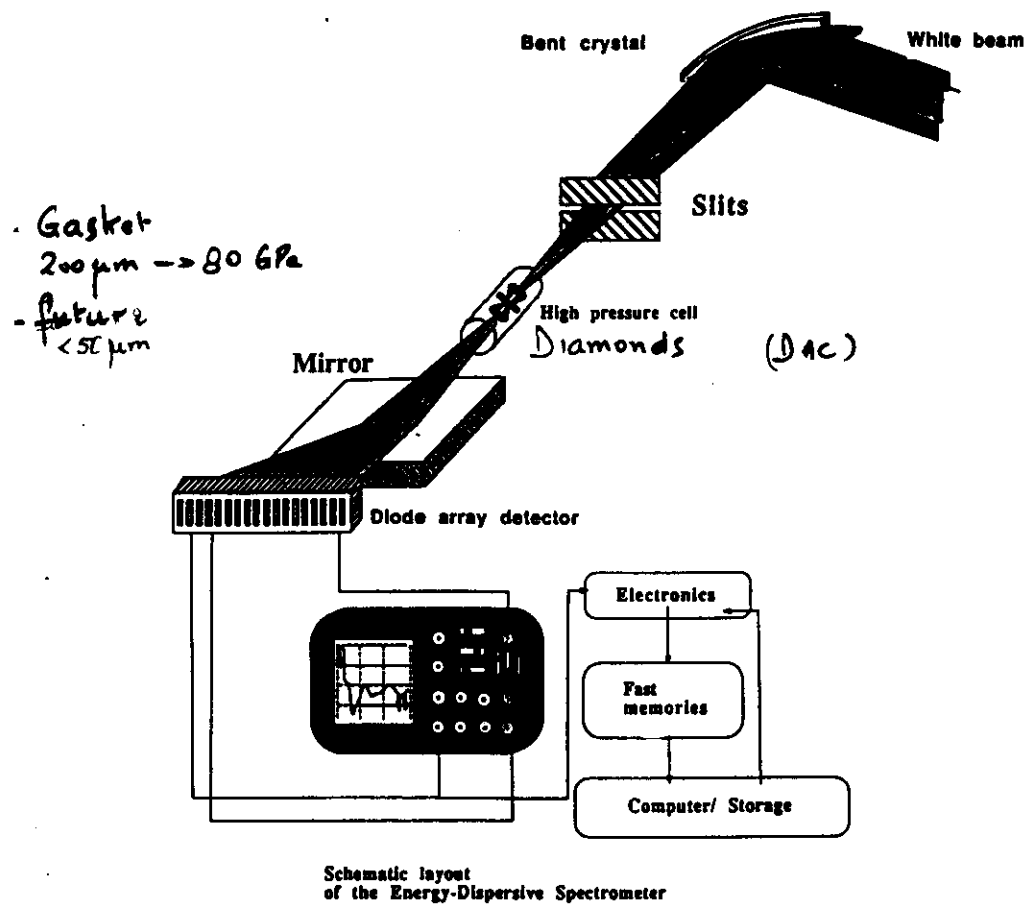
11 GPa

Rock salt

A. San Miguel
Thesis June 1993

ZnSe





ADVANTAGES for high pressure experiments:

Focusing optics

Lack of mechanical movement

Direct visualization

Domain of energy: 250 - 700 eV above the edge

FIRST NEIGHBOR DISTANCE DETERMINATION

Absolute resolution: ~ 0.5 - 1 %

Relative resolution: 0.1 - 0.2 %

$$\chi(k, T) = \frac{1}{k} S_0^2(k) A(k) \int \frac{\rho(r, r)}{r^2} \exp \frac{-2r}{\lambda} \sin(2kr + \phi(k)) dr$$

ONE SHELL

$$R(T) = \langle r \rangle = \frac{1}{N} \int r \rho(r, r) dr$$

$$\sigma^2(T) = \langle (r - R(T))^2 \rangle_T$$

$$C_3(T) = \langle (r - R(T))^3 \rangle_T$$

$$\chi(k, T) = \mathcal{A}(k, T) \sin \Psi(k, T)$$

$$\mathcal{A}(k, T) = S_0^2 \frac{A}{R^2} N \exp -2 \frac{R}{\lambda} \exp[-2 \sigma^2 k^2]$$

$$\Psi(k, T) = 2k \left[R - \sigma^2 \left(\frac{1}{R} + \frac{2}{\lambda} \right) \right] - \frac{4}{3} C_3 k^3 + \phi(k)$$

$\uparrow_{T \ll 0}$
 $\uparrow$
 $\uparrow$

- $\phi(k)$

- S_0^2 MANY BODY EFFECTS

$$0.7 \rightarrow 0.9$$

$$|L\rangle = \varphi_{1s^2} \varphi_{2s^2} \varphi_{2p^6} \dots$$

$$|L'\rangle = \varphi_{1s^2}^* \varphi_{2s^2}' \varphi_{2p^6}^* \dots$$

$$\langle \varphi_{2s^2} / \varphi_{2s^2}' \rangle \lesssim 1$$