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**WORKING PARTY ON  
MECHANICAL PROPERTIES OF INTERFACES**

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***"High Resolution Electronmicroscope  
Studies on Interfaces"***

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***These are preliminary lecture notes, intended only for distribution to participants.***

# **ATOMIC SCALE INFORMATION AND MECHANICAL PROPERTIES OF INTERFACES IN TiAl INTERMETALLICS AND Al MATRIX COMPOSITES REINFORCED BY WHISKERS**

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## **OUTLINE**

### **Introduction**

#### **The Atomic Scale Techniques For Interfaces**

#### **Interfaces And Ductility/Toughness of TiAl**

#### **Interfacial Reaction Products And Mechanical Properties in Al composites**

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### **1. Introduction**

- GB, IB and Interfaces can affect properties to a great extent
  - ▶ Nanocrystal--high toughness
  - ▶ Multilayer superlattice--"supermodulus effect"
  - ▶ Ni-Al, Ti-Al intermetallics--B segregation at GB and interface
  - ▶ Composites--bonding status of interfaces
- Progress in characterization of interfaces
  - ▶ Simple theoretical reasoning
  - ▶ Crystal defect theory and experimental observations
  - ▶ Computer and computation techniques

- ▣ Demands in scientific frontiers
  - Surface engineering
  - Interface engineering for composites
  - Interface design for microelectronic devices
- ▣ Progress in studies on interface structure
  - Metallic G.B.
  - Diffraction contrast EM & HREM
  - Ceramics G.B.
  - Polyphase materials interface boundaries

## 2. Atomic Scale Techniques for Interface Studies

### 2.1 CTEM & HREM

- ◆ Instruments are similar
- ◆ Imaging principle is quite different
  - CTEM--Amplitude contrast
  - HREM--Phase contrast

### 2.2 Basic Requirement for Interface HR Images

- ◆ At least one but preferably both crystals along a zone-axis with the boundary seen end-on
- ◆ The boundary as 2-D medium should be projected to a 1-D image:
  - All displacements are confined in the plane perpendicular to the observation axis and does not depend on the Z coordinate.
  - The distances between atomic columns are larger than resolution limit.  
(Pure tilt G.B.)

### 2.3 Chemical Lattice Imaging

- ◆ A-priori information:
  - The atomic species which are involved are already known but their exact location is to be found.
  - The relationship between the lattice constant and the composition is supposed to be known,
  - The image pattern should be uniquely and of possible linearly dependant on the chemical composition.

### 2.4 Limitations

- ◆ Projection problem
- ◆ Relaxation at interfaces
- ◆ Radiation damage

#### Main application:

- Periodicity at the interface
- Coherency or the loss of coherency with or without interfacial dislocations
- Atomic modeling and positioning at the interfaces
- Roughness of boundaries: steps; facets; chemical gradients

## 2.5 Study Cases of Atomic Structures in Materials Interfaces

- ♦ N.C. G.B.
- ♦ Interface boundaries in metallic materials  
     $\delta'$ ,  $S'$  and Al matrix
- ♦ Interfaces in ceramics materials  
    TiC/TiB<sub>2</sub>
- ♦ Heterophase  
    Cu-Pd metallic multilayer film with  $\lambda=1.4, 2.3$ , and  $3.4$  nm on mica substrate and Cu buffer showing partial dendrite crystal structure
- ♦ Composite interface  
    TiC<sub>p</sub>/Ti

## 2.6 Advantages and Limitations of The Atomic Scale Techniques

Technique	Resolution (struct.)	Resolution (chemical)	Probed thickness	Material limitat.	Specimen require.
HREM	0.16-0.2nm	indirec. 1nm	over 20 nm	Radiation damage in ionic materials	20 nm thin foil with boundary end-on
AP-FIM	0.3 nm	0.4 nm atomic sensitive	Surface peeling layer by layer	Best in non-ducti. materials	Wire with tip radius = tens nm, high density of boundary
STM	0.2nm L 0.005nm N	0.2 nm	Surface	Conductors	Flat & clean surface with B. end-on

## 3. Interface and Ductility/Toughness of TiAl

- ♦ TiAl intermetallic compounds are the potential materials for use at high-temperature because they exhibit a desirable combination of high modulus retention, creep and oxidation resistance with low density.

### 3.1 Lamellar Structure

- ♦ The alloys are usually composed of a lamellar mixture of TiAl( $\gamma$ -phase) and Ti<sub>3</sub>Al ( $\alpha_2$ -phase) and always exhibit superior mechanical properties in comparison with the single  $\gamma$  or  $\alpha_2$  phases. It is reasonable to suggest that the interfaces between the  $\gamma$ - and  $\alpha_2$ - phases play an important role in improving the ductility and toughness.
- ♦ The hierarchy of lamellar mixture structures  
    Twins and dislocations in the coarse  $\gamma$  phase
- ♦ Terraced interfaces in the coarse lamellar with micrometer scale
- ♦ Flat interfaces in the fine lamellar with 20 nm scale

- 3.2 Fully Lamellar Structure and Mechanical Properties
- ◆ Elongation increases with decreasing of the lamellar spacing
  - ◆ Toughness decreases with coarsening of the lamellar spacing
- 3.3 Interfaces Are the Easily Deformable Regions
- ◆ In the case of the less plastic deformation (2-3%) one set of twins formed and the initial deformation began at  $\gamma/\alpha_2$  interfaces.
  - ◆ Larger plastic deformation (6-7%) will induce the second set of twins and the latter causes interfaces twist.
  - ◆ Finer lamellar, more interfaces and easier to be deformed
- 3.4 Strengthening of the  $\gamma$  Matrix as the Lamellar Becomes Thinner
- ◆ 0.7% expansion in the thin  $\gamma$  slice in comparison with the coarse  $\gamma$  matrix
  - ◆ Composite defect structure inside the thin  $\gamma$  slices
  - ◆  $\gamma$  matrix strengthening may degrade toughness (K1c)
- 3.5 Atomic Scale Information Provided by HREM Might Be Localized, A full Understanding of Mechanical Properties of Interfaces Should Use Micro-scale Information As Well.
- 3.6 Phase Transformation Induced By Deformation In The TiAl(Cr) Intermetallics During The Deformation
- ◆ The interface regions became difficult to be identified
  - ◆  $\gamma$ - $\alpha_2$  transformation induced by deformation at interfaces
4. Interfacial Reaction Products and Mechanical Properties in Al Composites
- ◆ The whisker-reinforced Al composites have been of much interest for their high specific strength, high modulus, high wear resistance and thermal stability.
- 4.1 Materials
- ◆ SiCw/6061Al
  - ◆  $\text{Si}_3\text{N}_4$ w/6061Al
  - ◆  $\text{Al}_{18}\text{B}_4\text{O}_{33}$ w/6061Al
  - ◆  $\text{K}_2\text{Ti}_6\text{O}_3$ w/6061Al
- 4.2 Bending Strength, Micro-Hardness and the Effect of T6 Treatment
- ◆ The influence of T6 treatment on bending strength was similar to that on micro-hardness
  - ◆ Strength increased after T6 for SiC and  $\text{Si}_3\text{N}_4$  whiskers, however it decreased slightly for oxide whiskers.
- 4.3 HREM Observations of Whisker/Matrix Interfaces
- ◆ There was no chemical reaction which damaged whiskers during T6 treatment for SiCw and  $\text{Si}_3\text{N}_4$ w/Al
  - ◆ Chemical reaction between whisker and Al alloy during T6 treating
- 4.4 Mg Segregation May Play An Important Role For Interfacial Reaction in Whisker/Al Composites

