

SCHOOL ON SYNCHROTRON RADIATION AND APPLICATIONS
In memory of J.C. Fuggle & L. Fonda

19 April - 21 May 2004

Miramare - Trieste, Italy

1561/19

Photoelectron Diffraction: Applications and Developments

Juerg Osterwalder

Photoelectron Diffraction: Applications and Developments

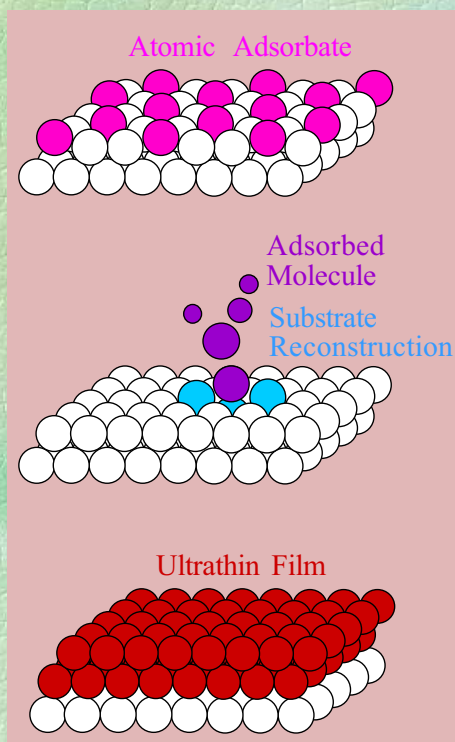
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<http://www.physik.unizh.ch/groups/grouposterwalder/>

Lecture 6

- Typical Surface Structure Problems
- Photoelectron Diffraction Patterns as Fingerprints
 - High-Temperature Superconductors
 - Tartaric Acid on Cu(110): Chiral Molecules
 - Na/Al(111): Atomic Adsorbates
- Application to Large Adsorbed Molecules: C₆₀
- Application to Heterogeneous Catalysis: O/Rh(111)
- Applications to Materials Science: *h*-BN/Ni(111)

Surface Structure Problems ... and their Solution by X-Ray Photoelectron Diffraction



Fingerprinting:

- Characteristic XPD pattern
- Forward Scattering Maxima
- Backscattering Cones
- 1st-Order Interference Fringes

Holography:

- Atomic-Resolution Image

Chemical Intuition:

- Reasonable Bond Lengths
- Coordination Number

Structural Refinement:

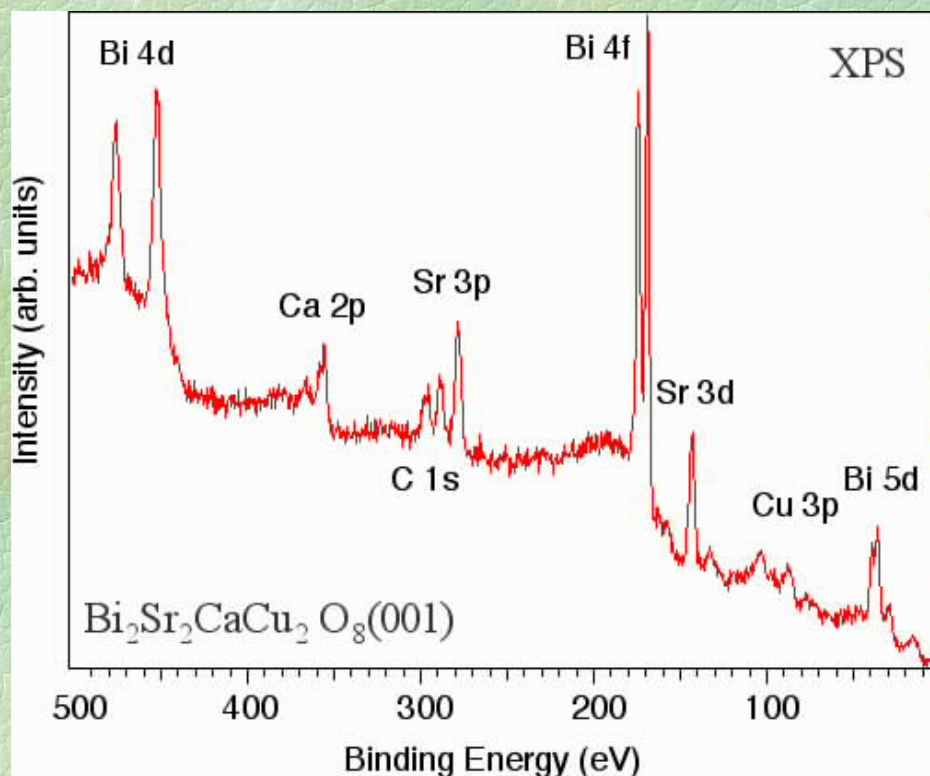
- Model Calculations (SSC, MSC)
- R-Factor Analysis

Resolution:

1.0 - 0.5 Å

0.05 Å

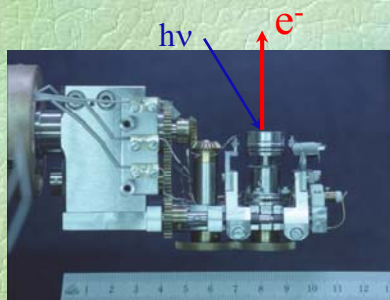
Exploiting the Elemental Sensitivity of XPS ...



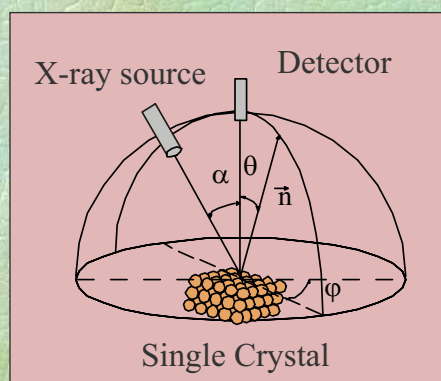
... for structural analysis

3

Full-hemispherical (2π) XPD patterns

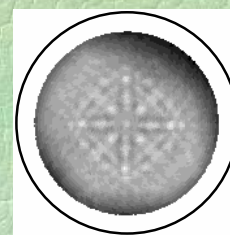


Fully automated two-axis
sample goniometer
(Fadley type)



Sequential measurement
of core level intensities
at ca. 4000 angular settings

e.g. Bi 4f

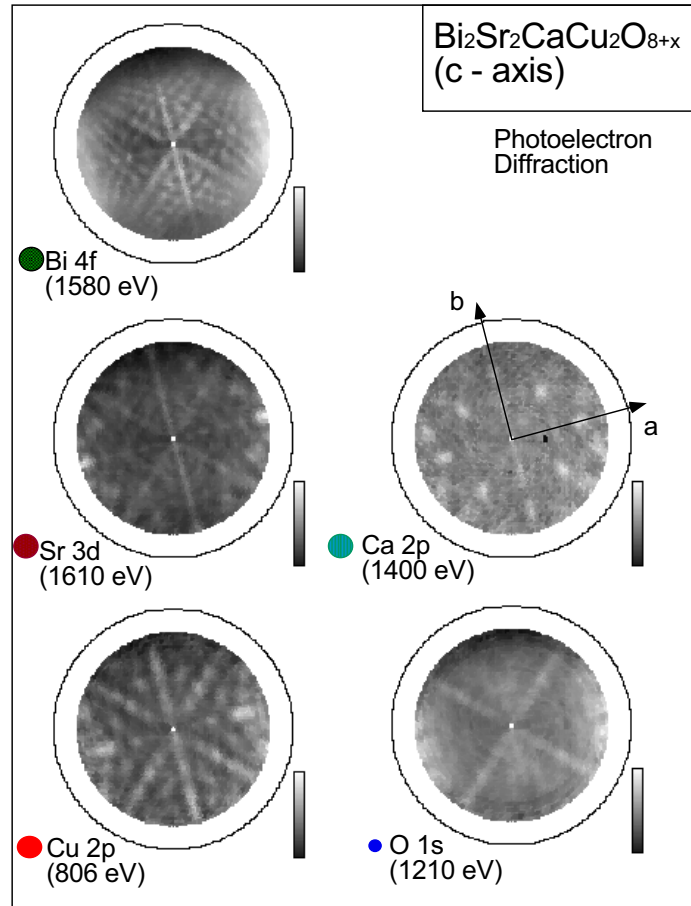
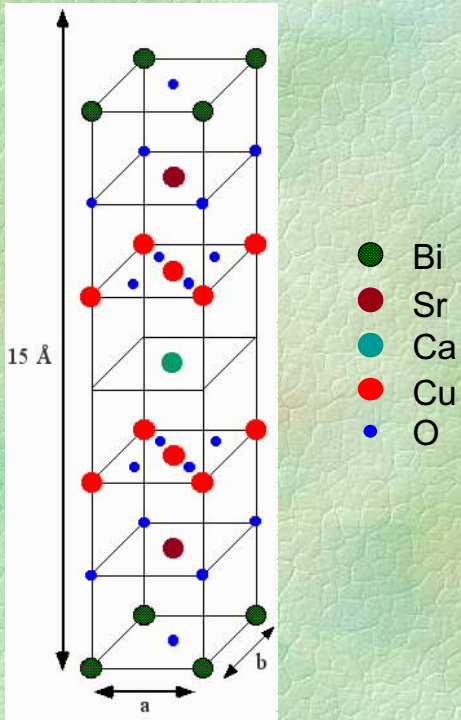


Intensities
in grey scale,
stereographic
projection

Measuring times for full 2π XPD pattern: 1 hour to 24 hours
(depending on count rates **and** on anisotropies)

4

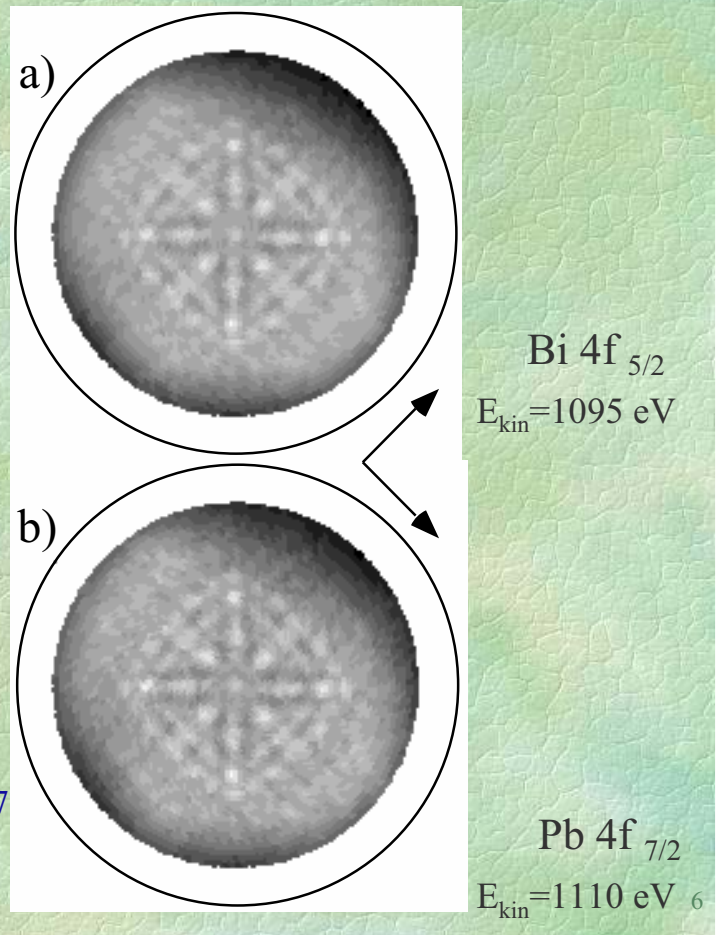
XPD Patterns of All Elements



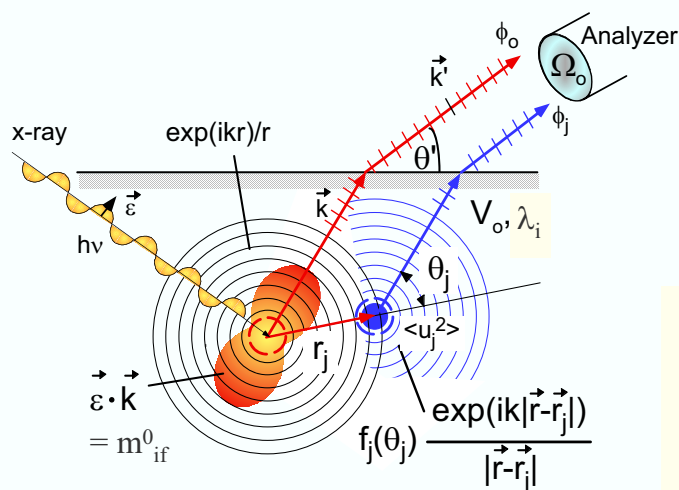
Determination of Dopant Sites



P. Schwaller et al., JES 76 (1995) 127



X-Ray Photoelectron Diffraction

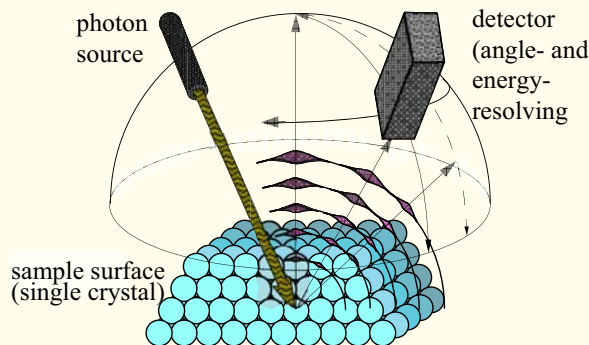


Photoemission matrix element

$$P_{if} \propto |\langle \psi_f(r, k_f) | \mathbf{\epsilon} \cdot \mathbf{r} | \psi_{nlm}(r) \rangle|^2$$

Scattering final state wave function

$$\psi_f(r, k_f) = \phi_0(r, k_f) + \sum_j \phi_j(r, l_j > k_f)$$



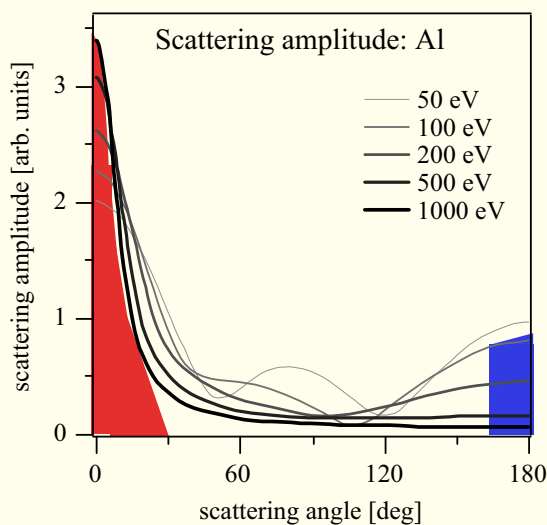
Single-scattering expression

$$I(k_f) \propto | m^0_{if}(k_f) \exp(-L_0/2 \lambda_i(E_f)) + \sum_j m^0_{if}(r_j)/r_j \exp(-L_j/2 \lambda_i(E_f)) W_j f_j(\theta_j) \exp(ik_f r_j (1 - \cos \theta_j)) |^2 + \text{TDS}$$

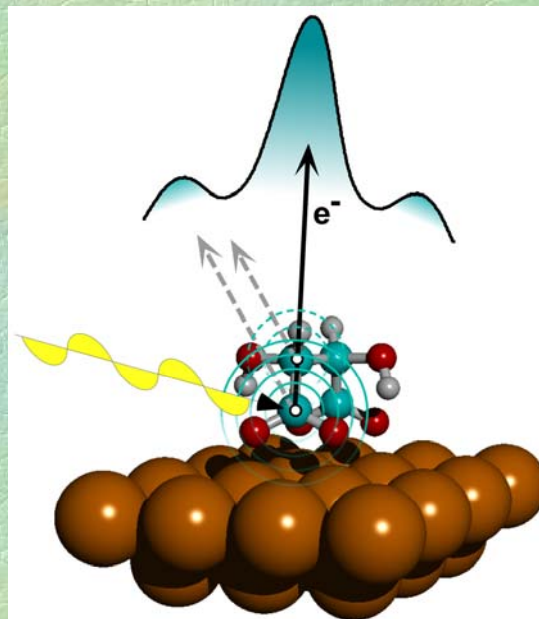
atomic scattering factor inelastic attenuation
geometrical and scattering phase shifts temperature effects

Forward Scattering in XPD

The atomic scattering factors have a characteristic dependence on the scattering angle θ_j .



High energies: forward scattering
Low energies: forward and backscattering



© R.Fasel
EMPA Dübendorf

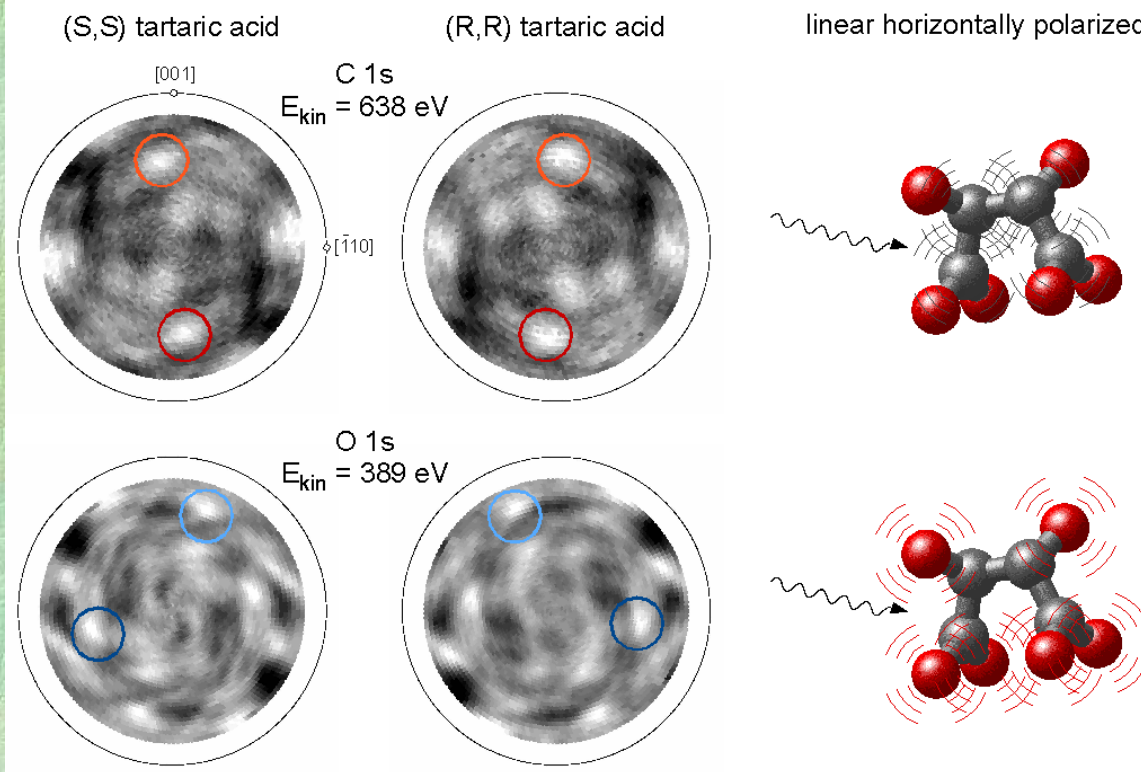
Example: Adsorption geometry of tartaric acid (chiral molecule) on Cu(110)

XPD Patterns for Two Enantiomers

x-rays:

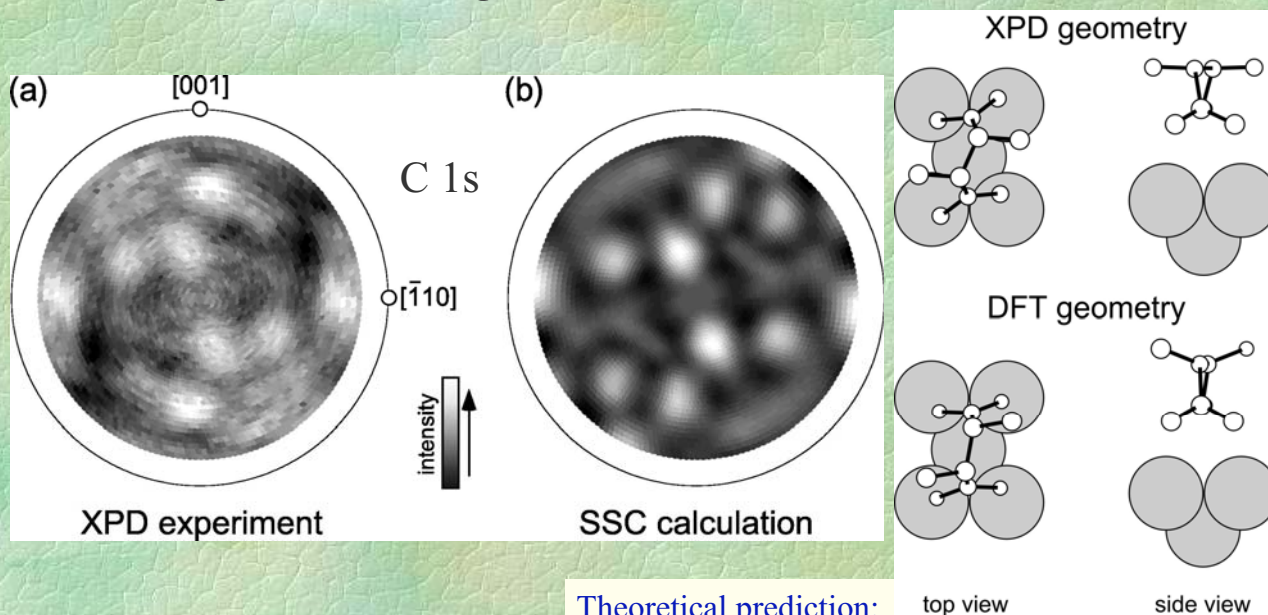
$\hbar\omega = 920 \text{ eV}$ SIM / SLS

linear horizontally polarized



R. Fasel, K.-H. Ernst, J. Wider, Ch. Quitmann,
T. Greber, *Angew. Chem.* (2004)

Refinement of Adsorption Geometry by Comparison with Single-Scattering Cluster Calculations



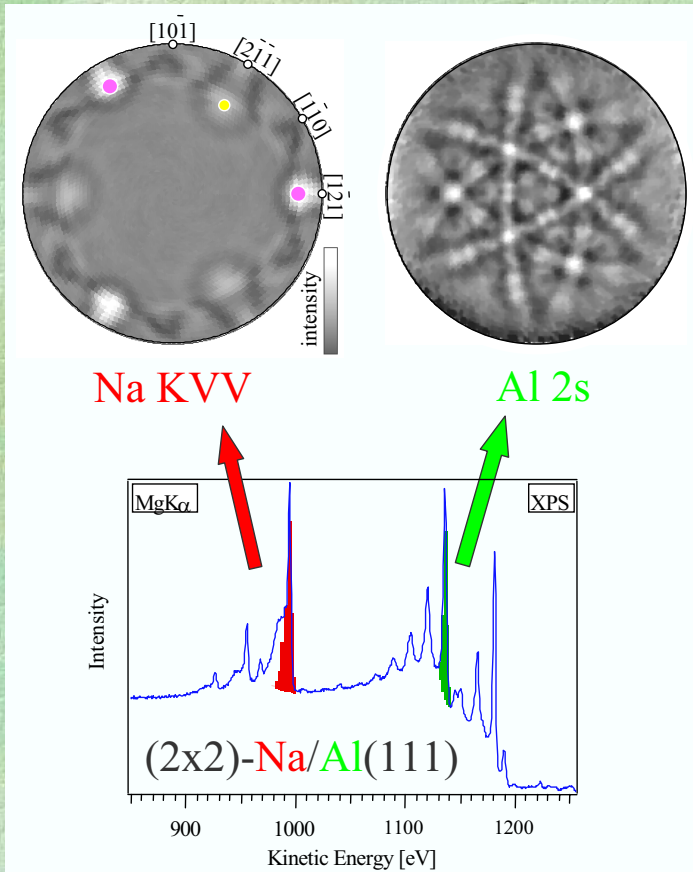
Theoretical prediction:

L.A.M.M. Barbosa, P. Sautet

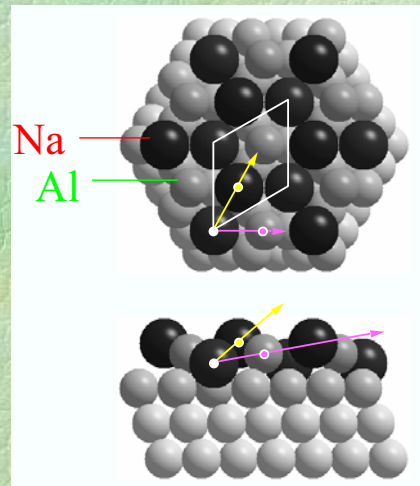
J. Am. Chem. Soc. 123 (2001) 6639

R. Fasel, K.-H. Ernst, J. Wider, Ch. Quitmann,
T. Greber, *Angew. Chem.* (2004)

Characteristic Substrate and Adsorbate Patterns

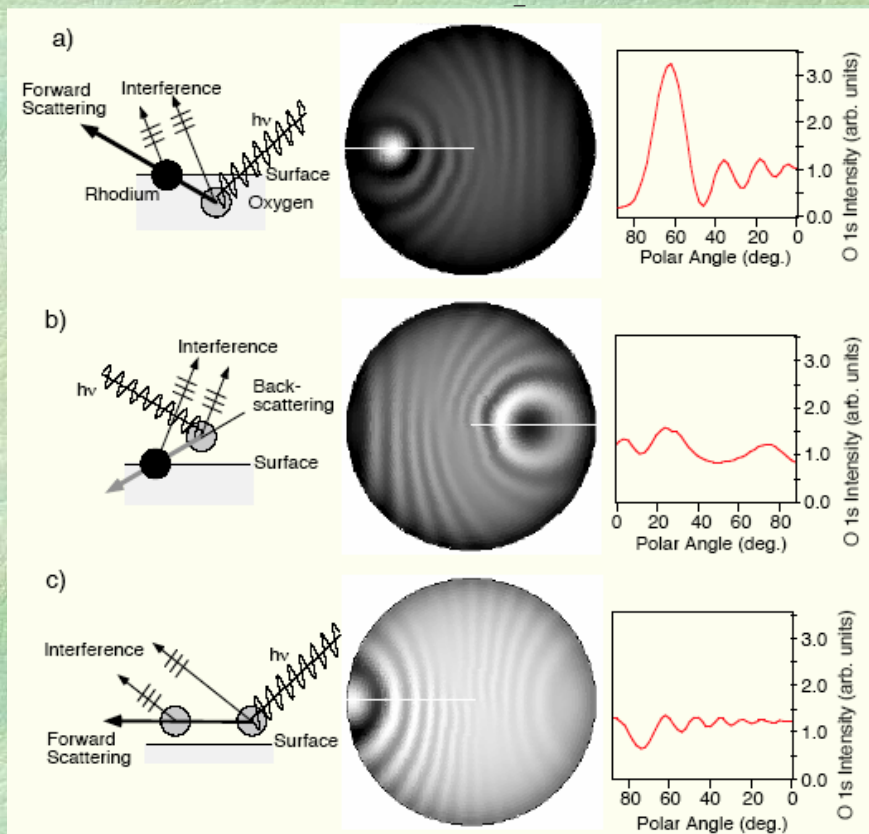


⇒ Structure determination by forward scattering peaks



Interpretation of XPD 2π diffraction patterns

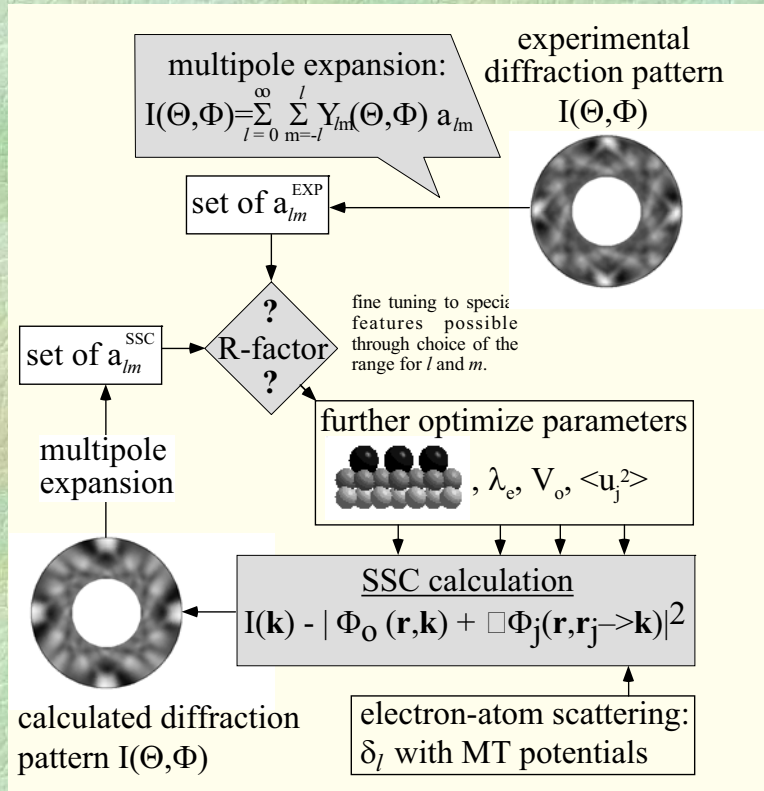
Adsorbate below top layer



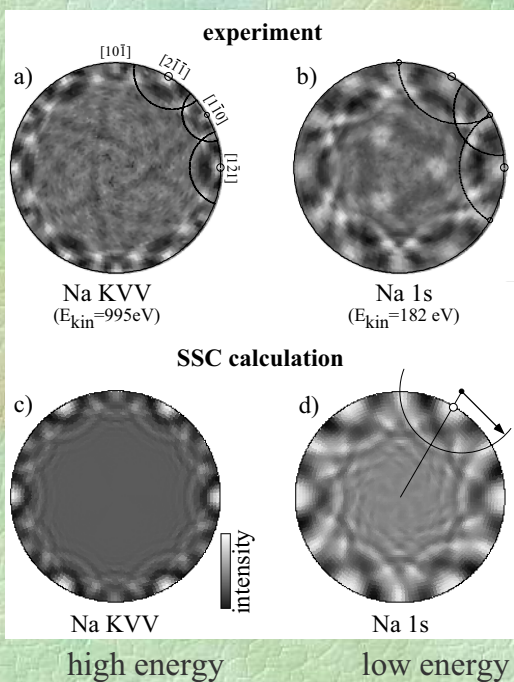
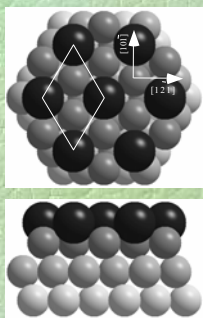
Adsorbate in top layer

Adsorbate-adsorbate in-plane scattering

SSC calculations and R -factor analysis

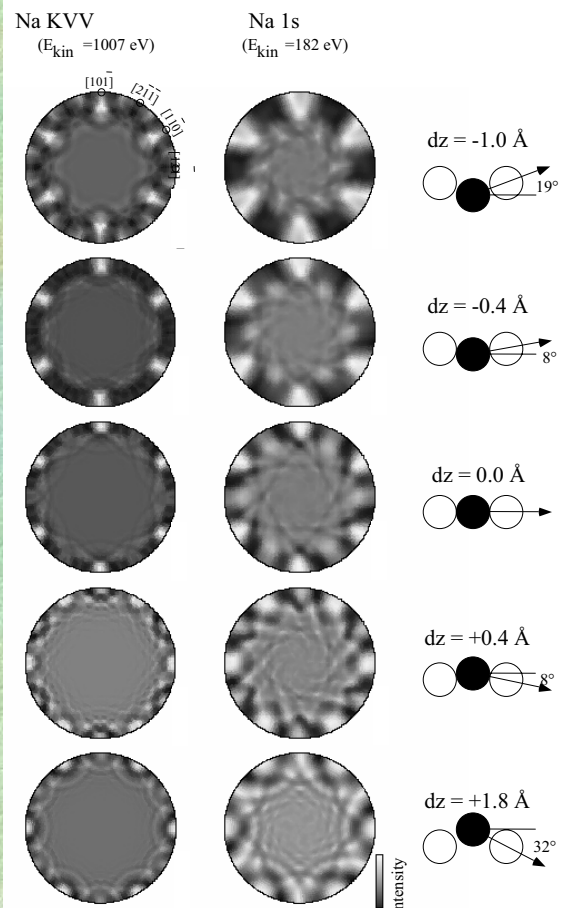


Example: $(\sqrt{3} \times \sqrt{3})R30^\circ$ -Na/Al(111)

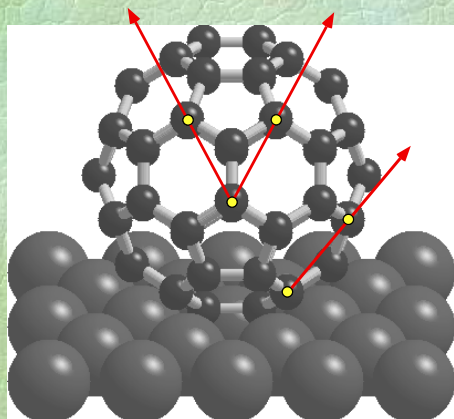


R. Fasel et al.,
Surf. Sci. 331-333 (1995) 80

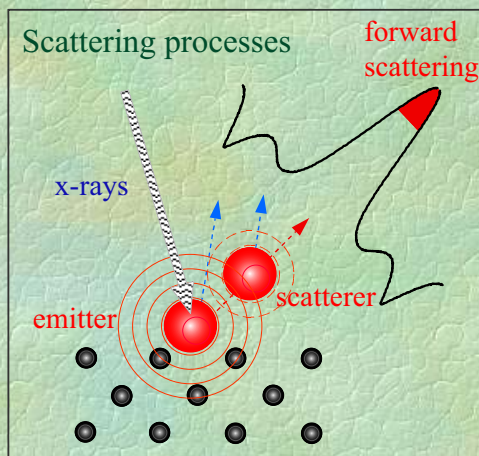
SSC calculations



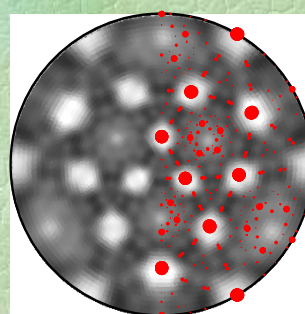
XPD pattern: Fingerprint of molecular orientation



C_{60} scattering situation

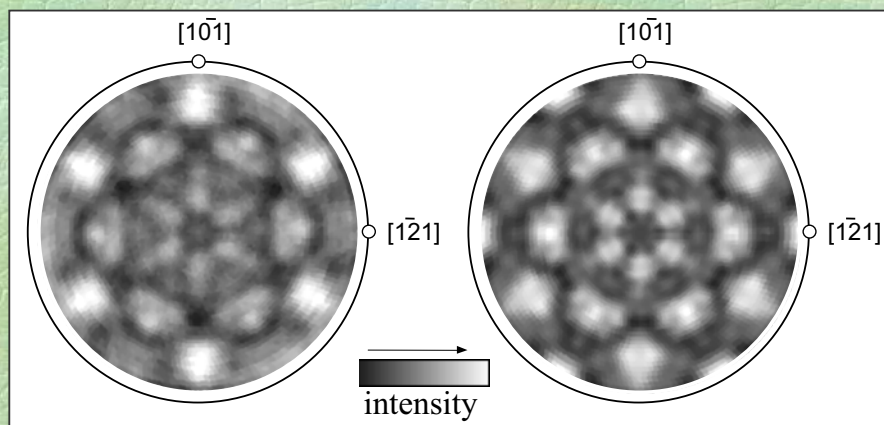


Single scattering cluster (SSC) calculation for C_{60} on 6-ring



Projections of interatomic (C-C) directions

XPD from $C_{60} / Cu(111)$



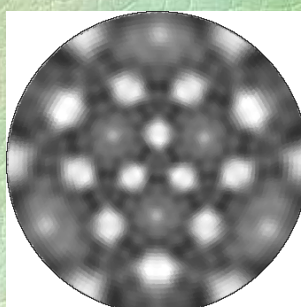
XPD Experiment

SSC Calc. 6-ring

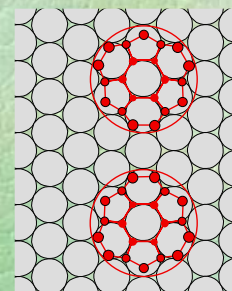


LEED: (4x4)

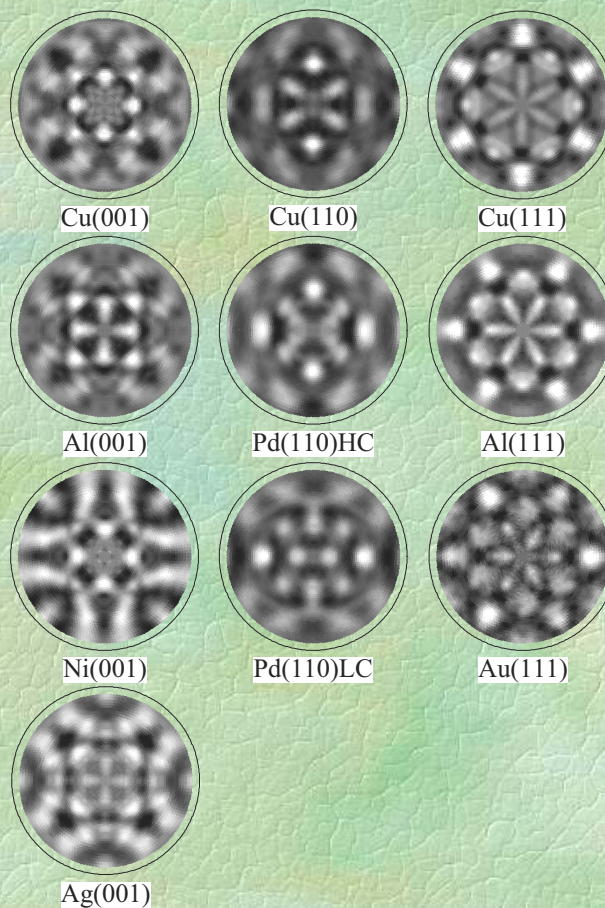
Single orientation:



6-ring (2 domains)

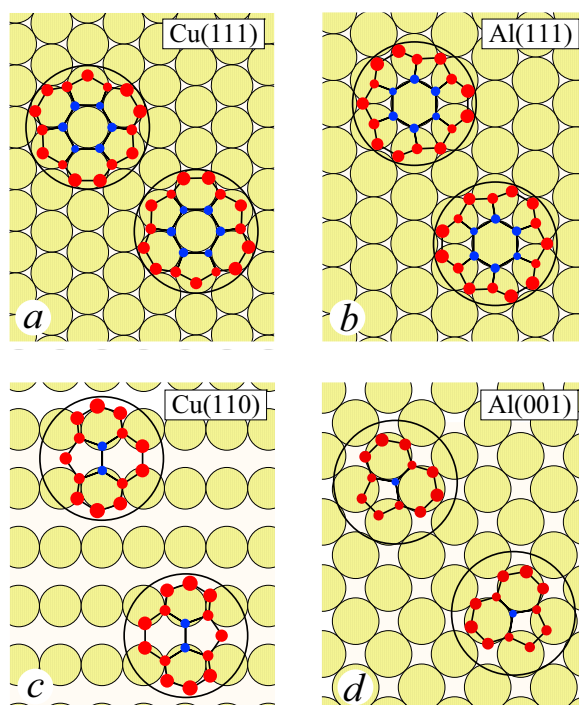


... one finds a variety of patterns for C_{60} monolayers adsorbed on different substrates !



⇒ one can determine the orientation(s) of the molecules by cumbersome trial and error methods

... Here Are Some Examples

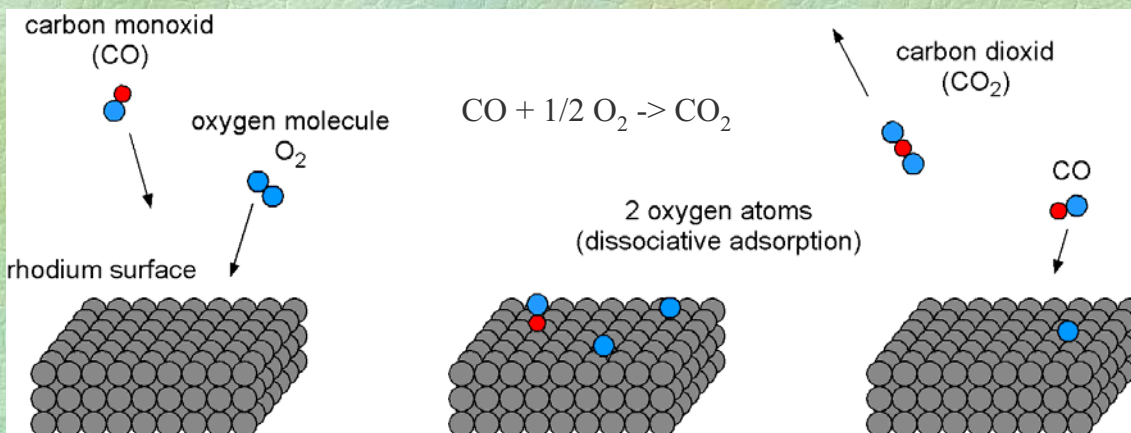


Usually there are several symmetry-equivalent orientation (depending on the symmetry of the substrate)

In some cases, there are 2 or more non-equivalent orientation which makes the analysis very hard !

Application of XPD to Catalysis Research

Example for heterogeneous catalysis:
Catalytic combustion of CO

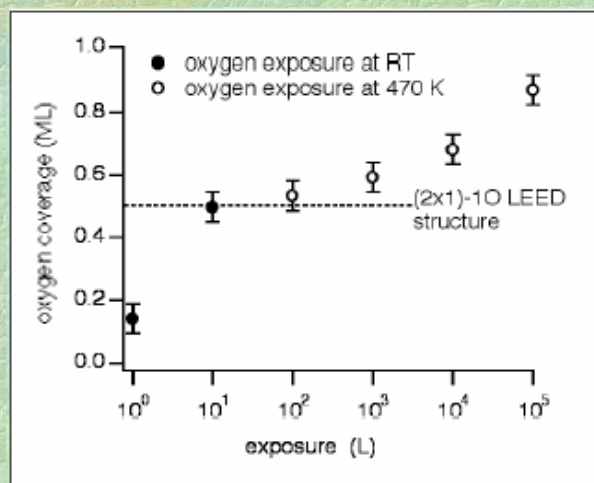
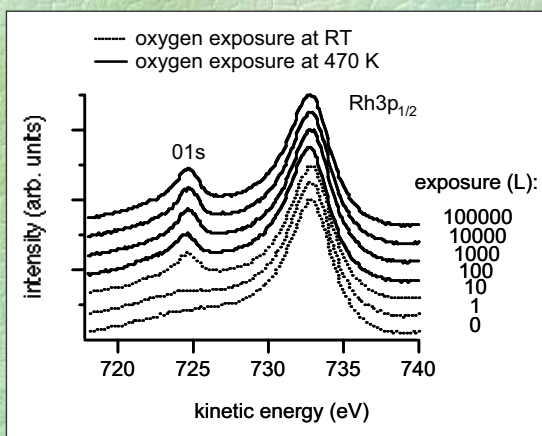


Reaction does not take place in the gas phase:
dissociation of O₂ costs too much energy (a few eV!).

... it is important where the adsorbed O sits !

Backscattering XPD from (2x1)-1 Oxygen on Rh(111)

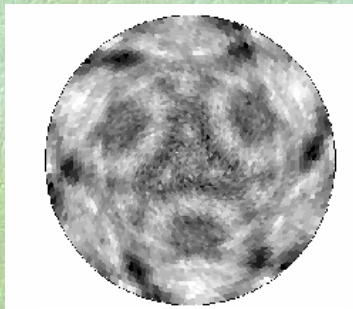
Coverage (XPS) vs. Exposure:



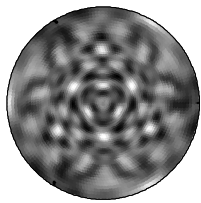
- (2x1)-1 adsorption site
- subsurface oxygen

The (2x1)-1 Oxygen Adsorption Site

Experiment

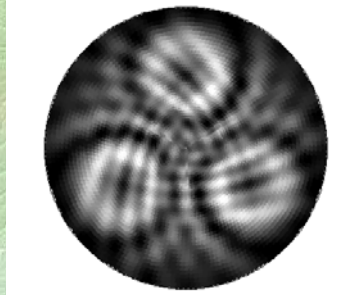
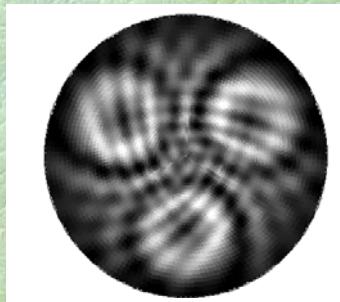


O 1s (723 eV)



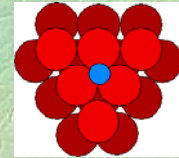
Rh 3d emission
($E_{kin} = 947$ eV)

SSC calculations



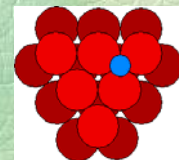
fcc site

Cluster
(top view)



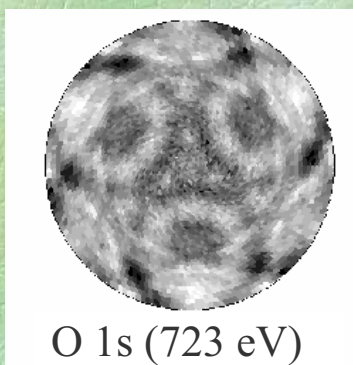
● Oxygen
● Rhodium

hcp site



... and the Bond Length

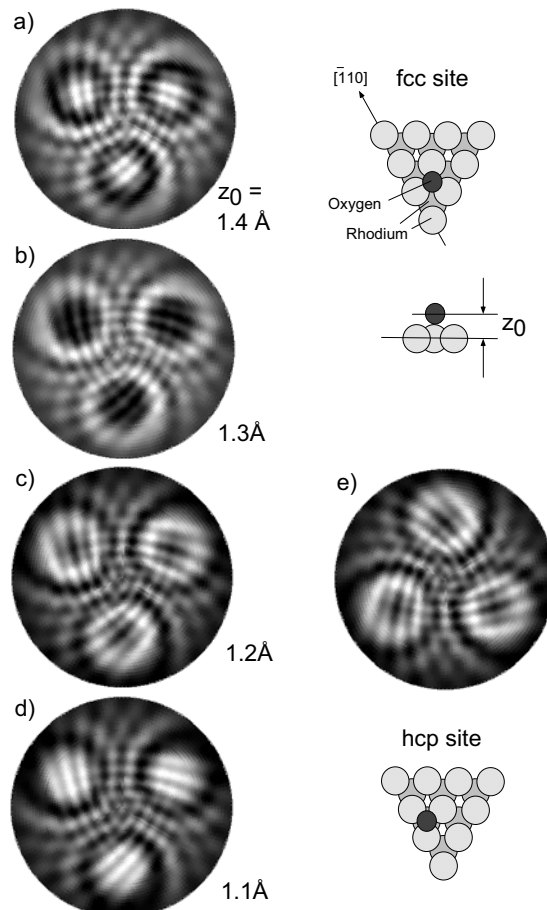
Experiment



O 1s (723 eV)

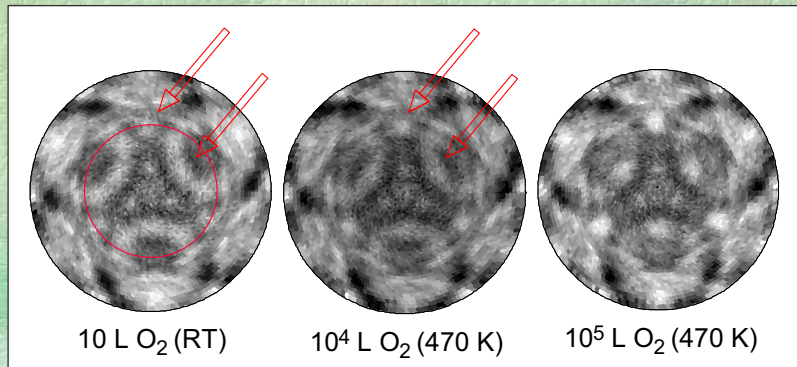
**=> A Sensitive
Electron Interferometer !**

T. Greber et al., PRL 81 (1998) 1654



Uncovering the Subsurface Oxygen in Rh(111)

O 1s emission
($E_{kin} = 723 \text{ eV}$)

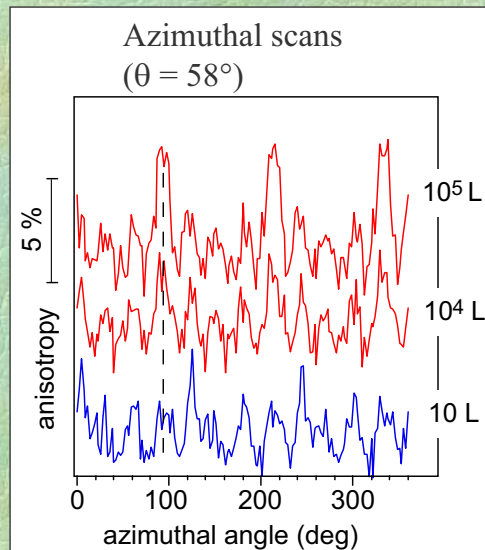


10 L O₂ (RT)

10⁴ L O₂ (470 K)

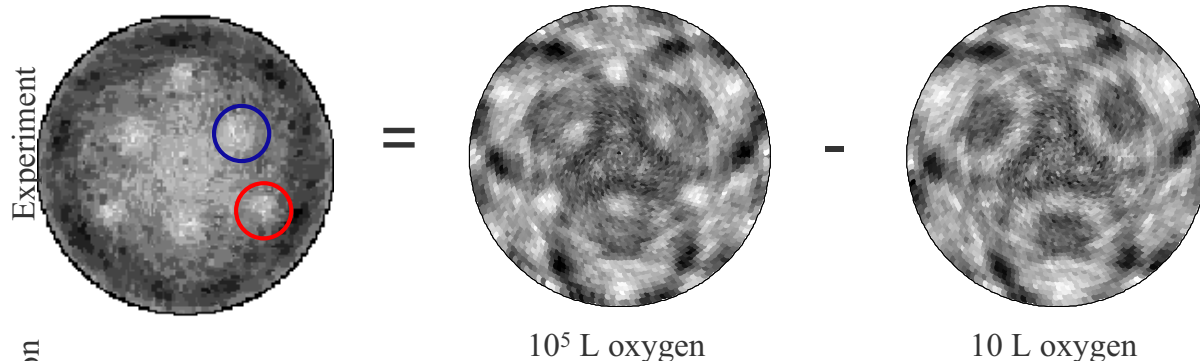
10⁵ L O₂ (470 K)

Exposure



⇒ Forward scattering peaks appear
⇒ **Subsurface oxygen (ca 5% ML) !**

Determination of the Subsurface Site

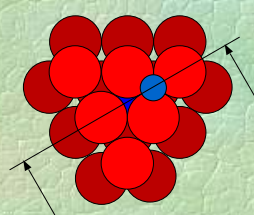


Experiment

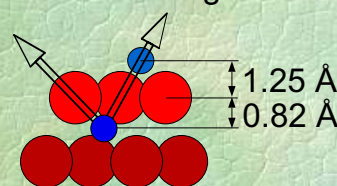
scattering calculation

10⁵ L oxygen

10 L oxygen

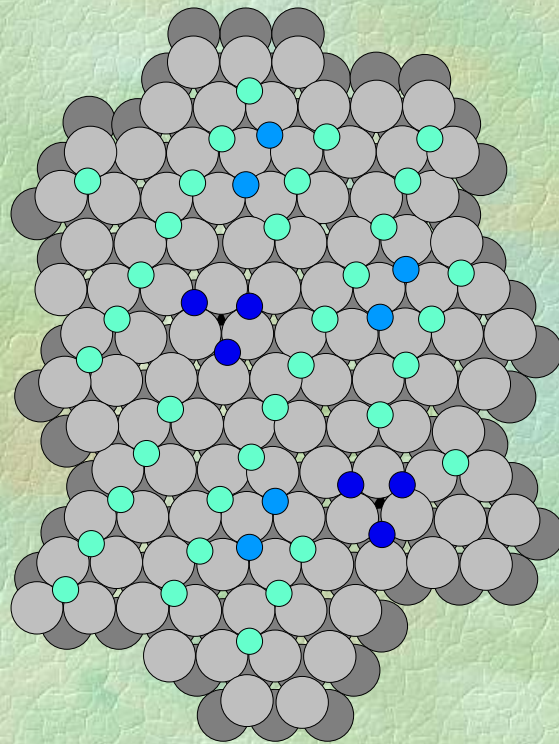


forward scattering



⇒ **Adsorption site switches to hcp !**

The Oxygen - Rh(111) System - Summary



RT-saturation

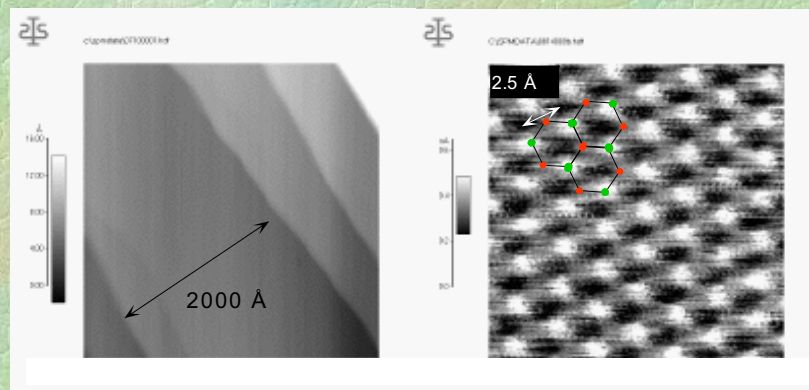
- fcc (2x1) - O
- Rh 1st layer
- Rh 2nd layer

High exposure (470 K)

- hcp - O
- fcc - O
- Subsurface oxygen

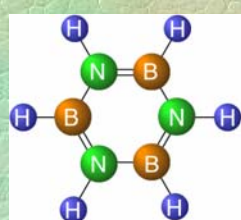
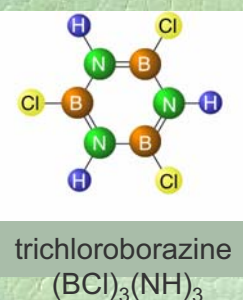
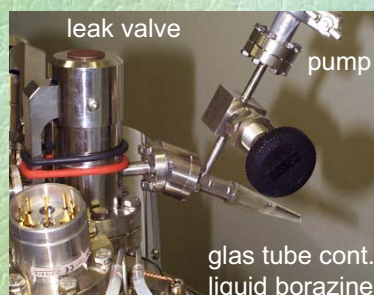
The Powerful Combination of XPD and STM: Hexagonal Boron-Nitride Monolayers on Ni(111)

... forms large and perfectly flat terraces !

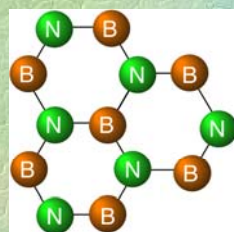


(1x1) translational symmetry
(from LEED and STM)

Growth of a *h*-BN monolayer on Ni(111)



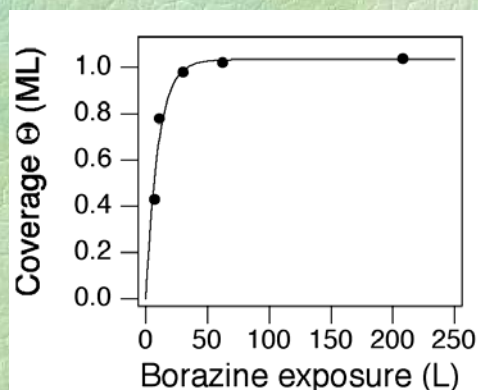
1050 K



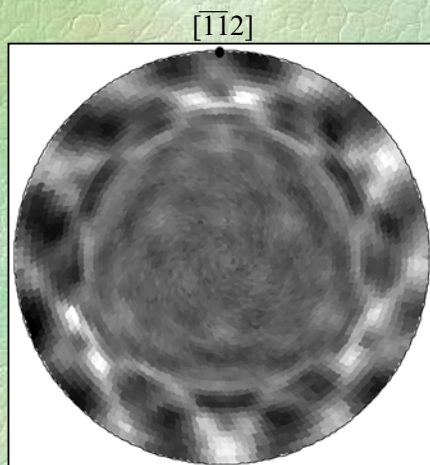
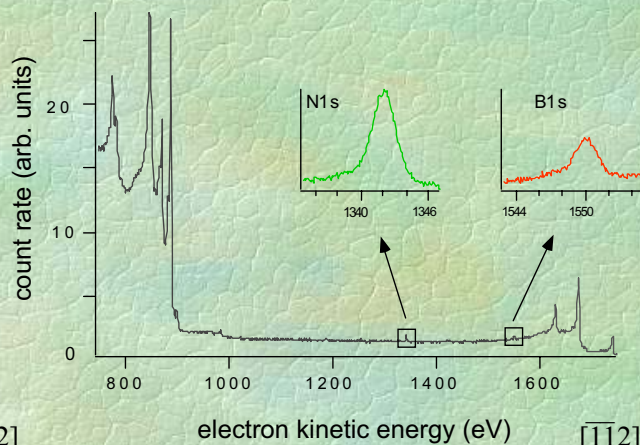
Borazine+Ni(111)

h-BN on Ni(111)+3H₂

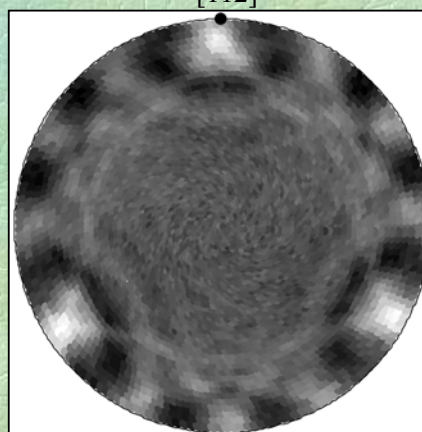
Nagashima et al., PRB 51, 4606 (1995)



Photoelectron Diffraction Experiments:

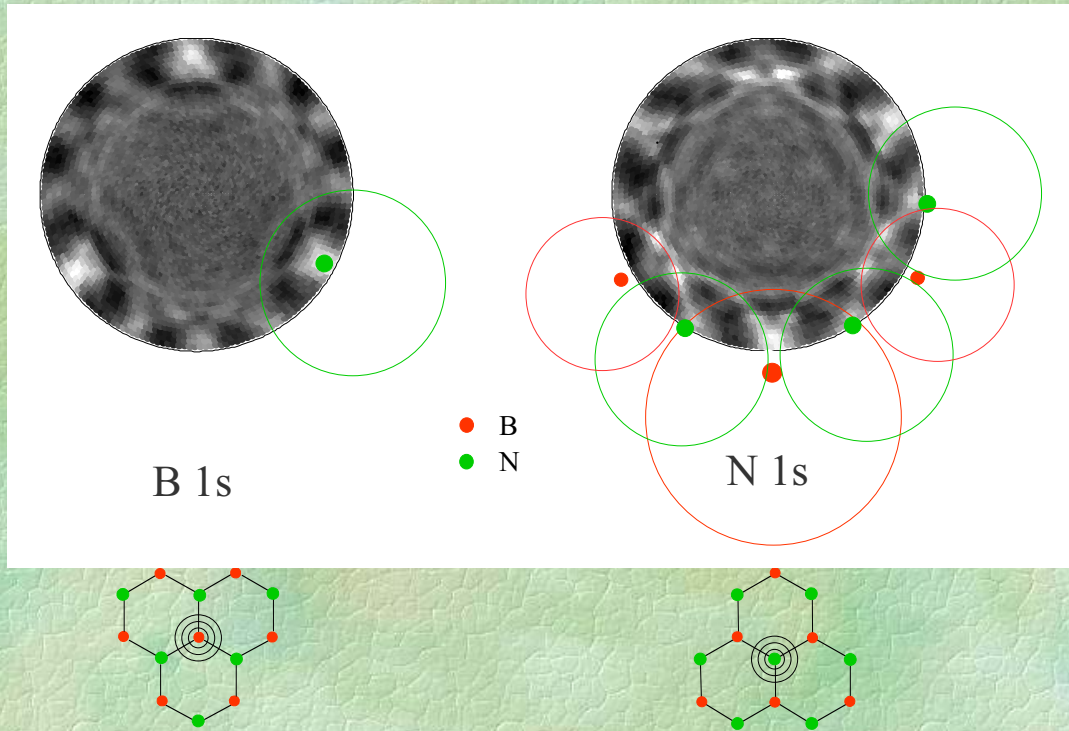


N 1s
(1342 eV)



B 1s
(1550 eV)

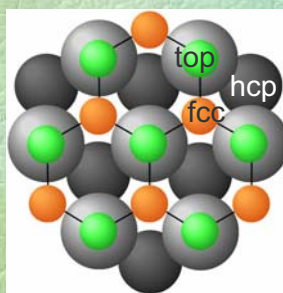
Forward scattering and interference fringes in B 1s and N 1s emission



=> Intuitive structural assignments !

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Summary of structural parameters:



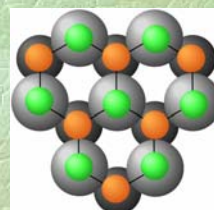
(N,B)=(top, fcc)

	Gamou (LEED)	this work (XPD)	Grad (DFT)
registry (N,B)	(top, fcc)	(top, fcc)	(top, fcc)
corrugat	0.18 Å	0.07 Å	0.11 Å
d N -Ni	2.22 Å	1.95 Å	2.19 Å
d Ni-Ni,1	1.98 Å	2.03 Å	2.03 Å

G. Grad *et al.*, submitted

Gamou *et al.*, Sci. Rep. RITU A 44, 221 (1997)

The structure is well understood ! ... but:

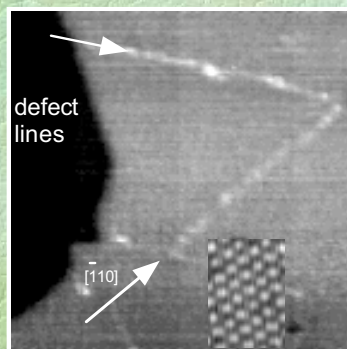


2nd stable DFT structure
+ 18 meV

(N,B)=(top, hcp)

30

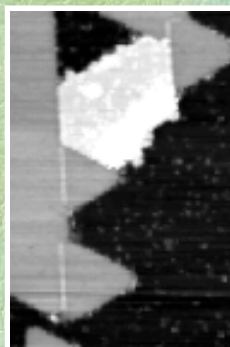
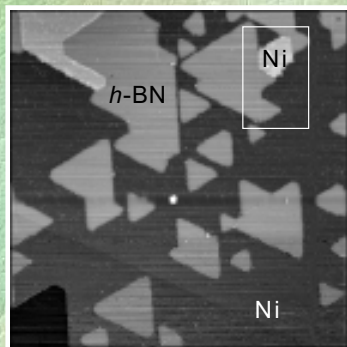
Defects and two-domain monolayers



Defect lines in 1 ML *h*-BN
(cf. Co deposits)

Where do they come from ?

425 Å

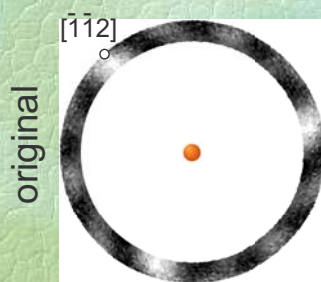


h-BN partial coverage:

- ⇒ Triangular islands of two orientations coexist on the surface.
- ⇒ Defect lines appear where they touch

400 nm

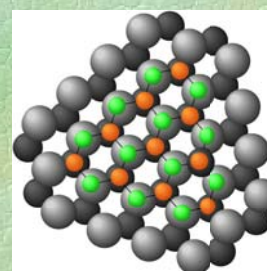
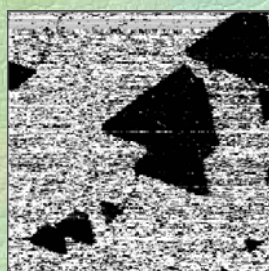
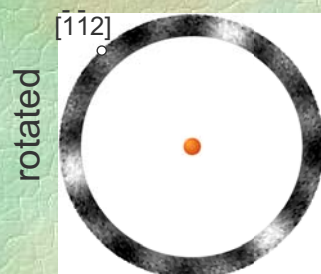
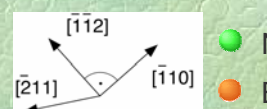
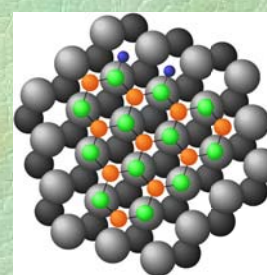
Different preparation can lead to predominantly one or the other domain!



B 1s XPD

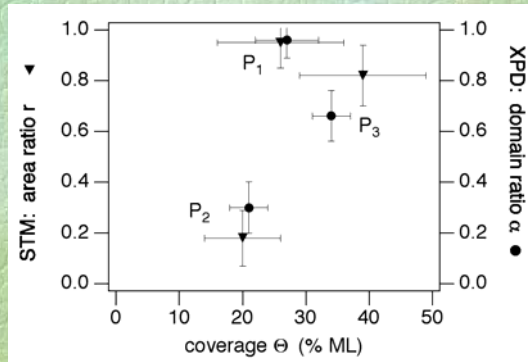


conductivity



⇒ XPD reveals the occurrence of the second most stable geometry from the DFT study ($\Delta E=18\text{meV}$)!³²

Quantitative comparison of XPD and STM Data



Compare areas of differently oriented islands

Determine domain ratios from B 1s forward scattering peaks

⇒ Island shape and bonding geometry correspond !

⇒ We know the atomic structure of the defect lines:

W. Auwärter et al.,
Surf. Sci. Lett. 545 (2003) L735

