

# Mechanical Testing Issues of Advanced Materials for GenIV, ADS and Fusion Systems

*Peter Hähner*

**JRC-IE – Institute for Energy**

*Petten - The Netherlands*

<http://ie.jrc.ec.europa.eu/>

<http://www.jrc.ec.europa.eu/>

## **Materials issues associated with current reactor design**

- Tensile strength
- Toughness
- Aqueous corrosion resistance
- Radiation damage tolerance

## **Materials issues with GenIV**

- Strength
- Toughness
- Fatigue: thermal & mechanical
- Creep & creep-fatigue interaction
- New aspects of coolant compatibility (SCWR, VHTR & GFR, SFR & LFR, MSR)
- Radiation damage @ high dose levels
- Fabrication routes and joining techniques

Damage interaction, transients associated with DBA and NDBA become more critical issues, as materials are taken closer to their limits.

=> Importance of (non-standard) thermo-mechanical, environmental testing

- **Small specimen geometries** for mechanical testing
- Application of the CEN **Small Punch Creep Testing** Code of Practice to a representative repair welded P91 pipe
- **Thermal Fatigue** Studies for Nuclear Piping Components
- Strain-Controlled **Thermo-Mechanical Fatigue**: Research into Best Practices

# Small specimen geometries for mechanical testing



## **Variety of mechanical tests types foreseen in GETMAT project**

Tensile, Impact, Fracture toughness, Fatigue, Creep, Creep-Fatigue

## **Limited material availability due to the fabrication process**

14Cr-ODS (PM): 45kg bars of 12mm, or plates of 3mm in thickness

9Cr-ODS (PM): 45kg plates of 6mm thickness

9Cr-ODS (EMS): 200kg plates of 10mm thickness

## **Material quantities requested from participants exceed the available quantities**

Optimised use of test material required



Small specimen testing

(M. Serrano, CIEMAT)

## International symposia

- ASTM SSTT symposia (initiated in 1986, last in 2007)
- IEA Symposium on SSTT for fusion materials (1995 and 1996)

## Review papers

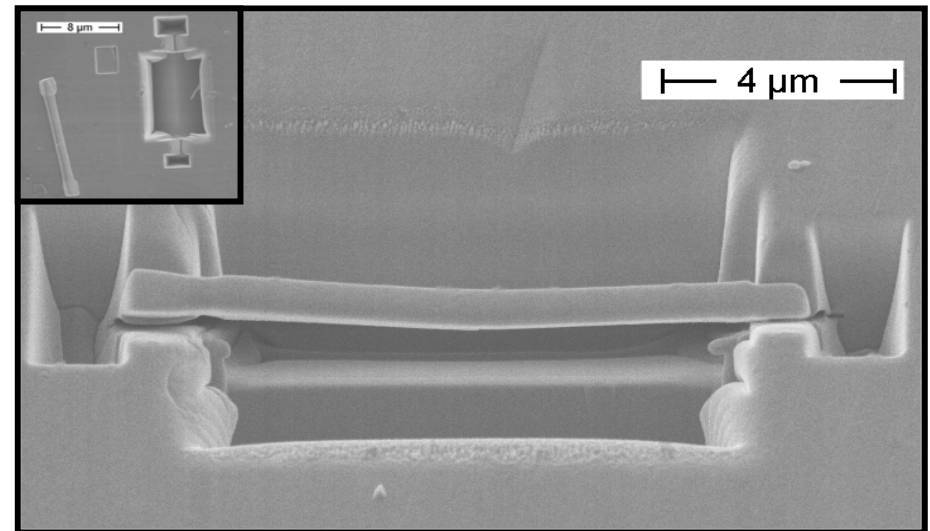
- Lucas, G.E., Odette, G.R., Matsui, H., Moslang, A., Spatig, P., Rensman, J., Yamamoto, T. “The role of small specimen test technology in fusion materials development” (2007) Journal of Nuclear Materials, 367-370 B (SPEC. ISS.), pp. 1549-1556

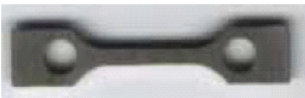



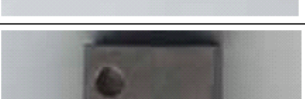

**Reactor pressure vessel surveillance** – Fracture toughness from pre-cracked charpy specimens

**Fusion community** – IFMIF irradiation

⇔ *Micro-mechanical testing*

Focused ion beam (FIB) milled  
three-point bend specimen in Fe  
(S. Roberts, U Oxford)



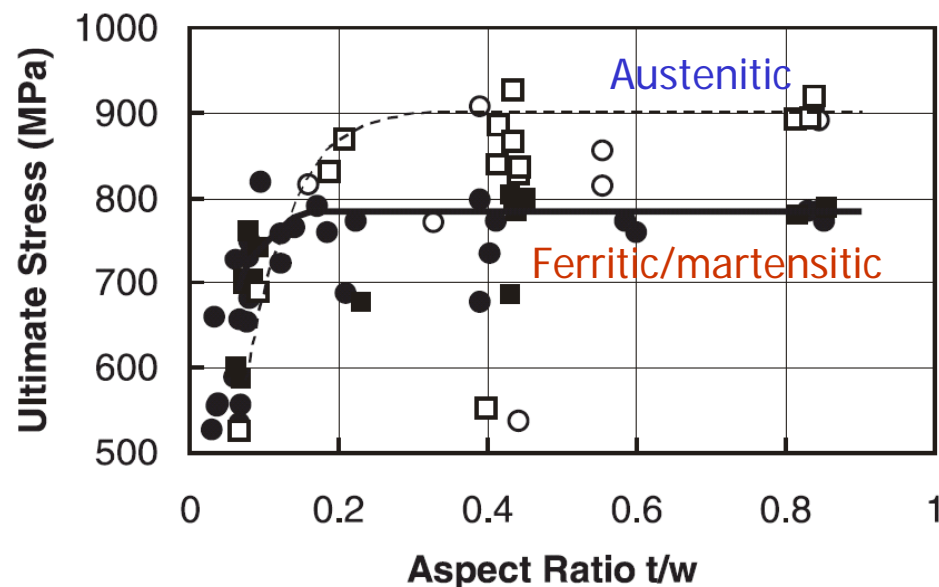
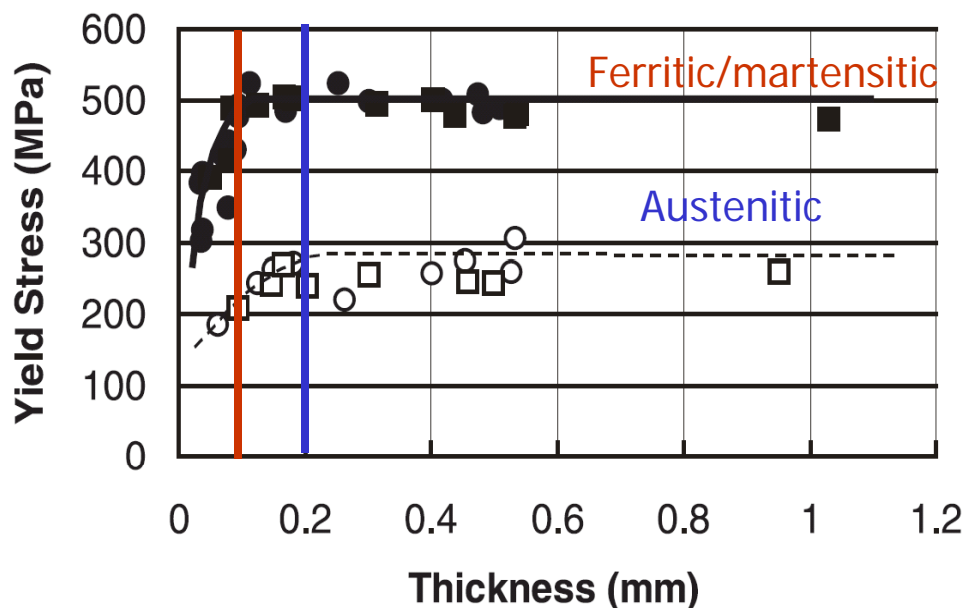
| Specimen type      | Present geometry  | Comments                              |
|--------------------|---|---------------------------------------|
| Tensile            |    | developed                             |
| Fatigue            |    | developed                             |
| Bend/Charpy DFT    |    | Standard achieved;<br>R&D ongoing     |
| Creep              |    | Miniaturization<br>needs verification |
| Crack growth       |   | International R&D<br>ongoing          |
| Fracture toughness |  | International R&D<br>ongoing          |

Significant volume reduction  
(20-125 times)  
compared to conventional  
standards

There are “**Developed**”  
specimen shapes, but are  
yet to be “**Standardized**” to  
be used for designing and  
licensing purposes

## Size effect on tensile properties *Y. Kohno et al. JNM 283-287 (2000)*

- Yield strength is thickness independent for thicknesses larger than a critical value.
- Ultimate tensile stress depends on specimen aspect ratio for aspect ratios smaller than a critical value (affected by irradiation).

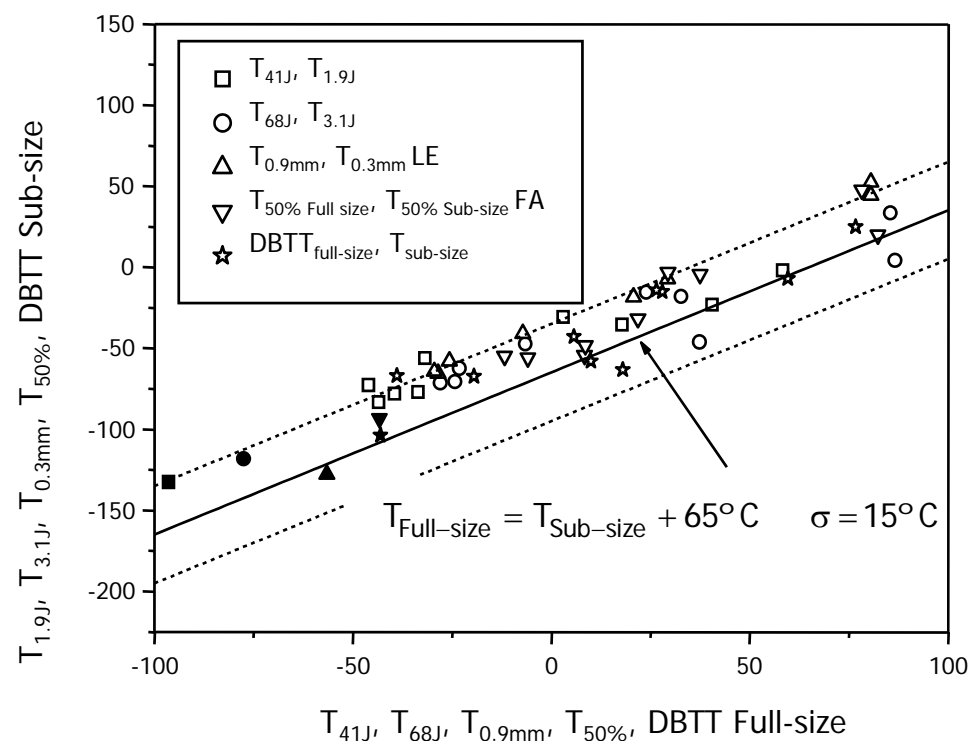
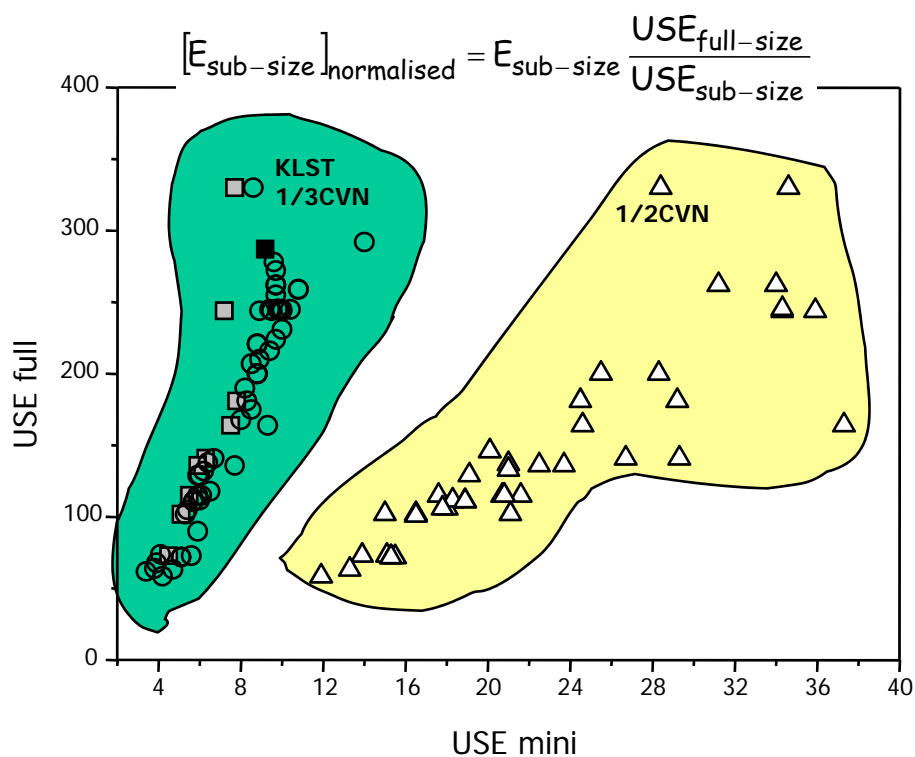




Reduction on specimen size reduces both Upper Self-Energy (USE) and Ductile to Brittle Transition Temperature (DBTT), due to change in stress state.

KLST (DIN) and 1/2 size Charpy V geometries (ESIS) are widely used.

Empirical correlations exist for 9Cr steels, e.g.



## ECCC RECOMMENDATIONS - VOLUME 3 Part III [Issue 4] for gauge section length $L_0$ and diameter $d_0$ :

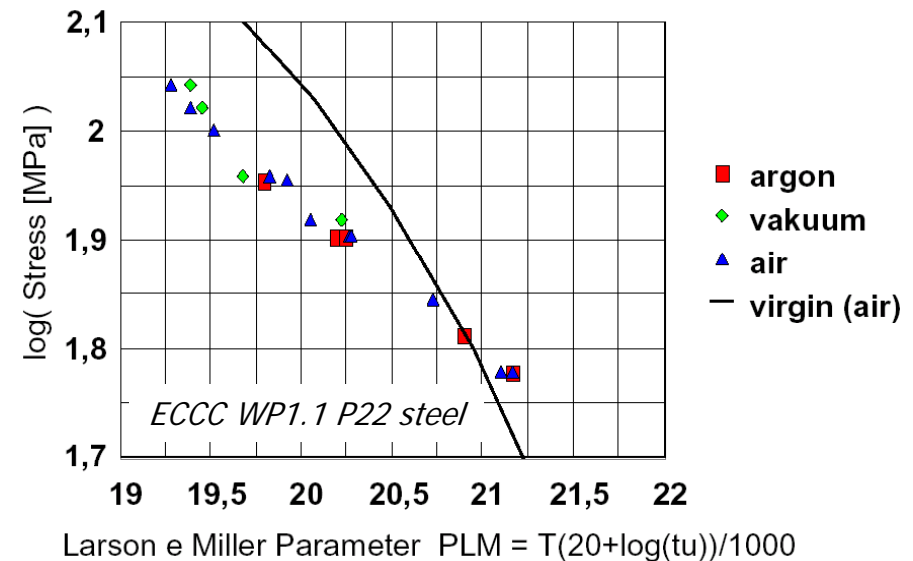
Full-size specimens:  $L_0/d_0 > 3$

Sub-size specimens :  $3 \text{ mm} < d_0 < 5 \text{ mm}$

Miniature specimens  $d_0 < 3 \text{ mm}$

- Generally good agreement between the creep rupture results of miniature specimens and those of conventional sized ones
- However, diffusion processes may induce a size effect, which tends to be more pronounced at lower stress levels [K. Krompholz, JNM 305 (2002) 112–123]
- Size sensitivity may also be associated with oxidation, bending misalignment.

Micro-specimens (Ar and Vakuum) and standard specimens



## Size effect for fatigue life is twofold:

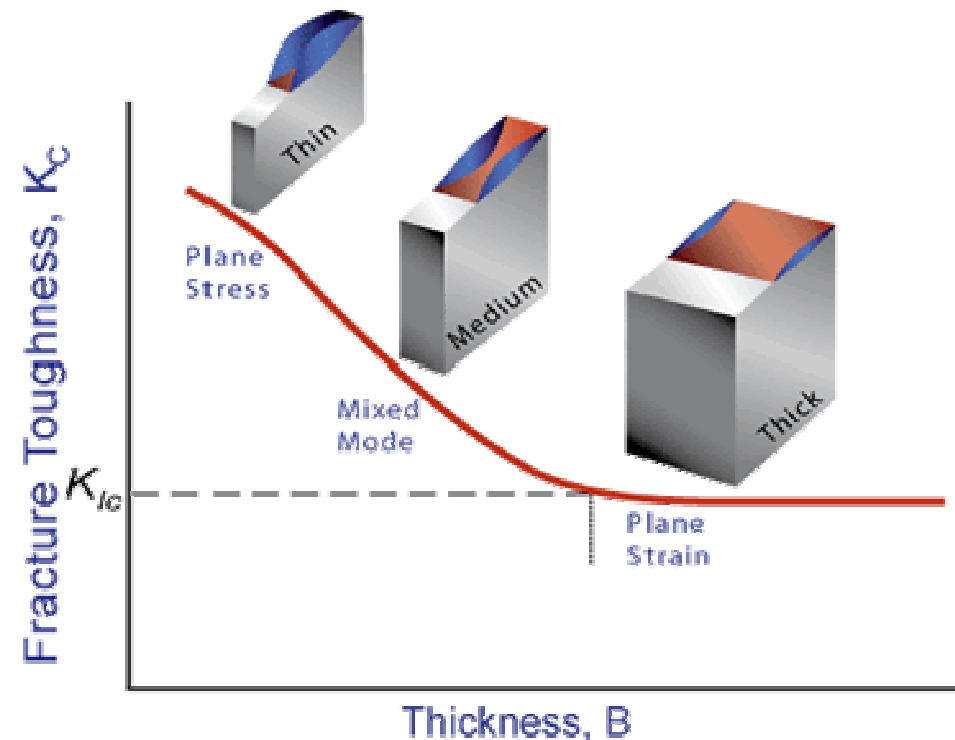
- **Statistical** for fatigue crack **initiation** (probability of fatigue crack initiators in the gauge section increases with size).
- **Deterministic** for fatigue crack **growth** until certain load drop level (ligament size)

**Recommended standard specimens, e.g. ASTM E606**

Fracture toughness is usually defined for **small scale yielding conditions** (plastic zone size  $\ll$  specimen dimensions).

Otherwise, fracture toughness is size dependent due to loss of constraint.

- ⇒ **Size limitations** of fracture toughness specimens included in the standards.
- ⇒ Relevance for fracture mechanics testing of thick components (e.g. RPV)



**Application of the CEN Small Punch Creep Testing  
Code of Practice  
to a representative repair welded P91 pipe**

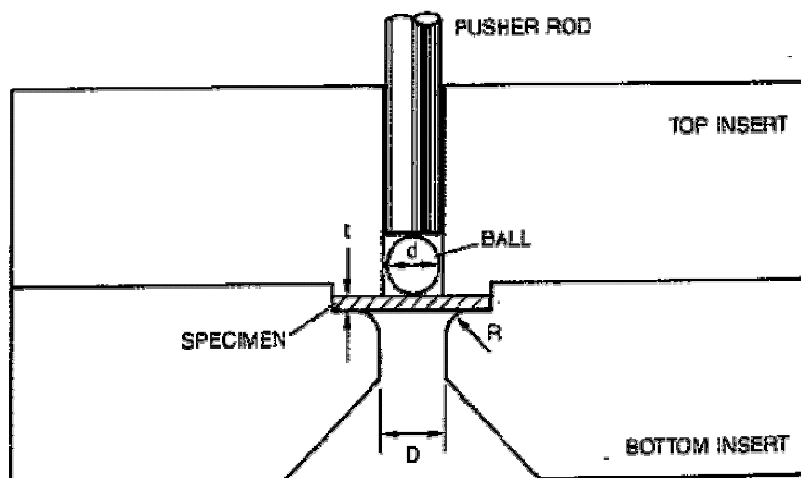
Developed by MTI in 1981, explored in US & Japan in the '80s, SP was introduced in Europe in 1992, in testing of steels.

CEN Workshop Agreement, CWA 15627, “Small Punch Test Method for Metallic Materials” developed on

Part A: A Code of Practice for Small Punch **Creep** Testing

- Time dependence of the deflection in the centre of the specimen is recorded
- Relationship between the SP load and uniaxial creep stress from the stretching membrane theory

Part B: A Code of Practice for Small Punch Testing for **Tensile** and **Fracture** Behaviour.



## Advantages

- Capable of generating tensile, toughness, and creep strength data from very small specimens
- Ability to sample from large component, relatively non-destructive (no repair is needed afterwards)
- Test itself is rather simple to perform and inexpensive
- Useful tool in assessing structural component integrity, for instance in weldments, clads and other anisotropic or critically weak regions

## Open Questions

- Representativeness of such small specimens
- Reliability of correlation with uniaxial creep data

ICS 77.040.10

English version

## Small Punch Test Method for Metallic Materials

This CEN Workshop Agreement has been drafted and approved by a Workshop of representatives of interested parties, the constitution of which is indicated in the foreword of this Workshop Agreement.

The formal process followed by the Workshop in the development of this Workshop Agreement has been endorsed by the National Members of CEN but neither the National Members of CEN nor the CEN Management Centre can be held accountable for the technical content of this CEN Workshop Agreement or possible conflicts with standards or legislation.

This CEN Workshop Agreement can in no way be held as being an official standard developed by CEN and its Members.

This CEN Workshop Agreement is publicly available as a reference document from the CEN Members National Standard Bodies.

CEN members are the national standards bodies of Austria, Belgium, Cyprus, Czech Republic, Denmark, Estonia, Finland, France, Germany, Greece, Hungary, Iceland, Ireland, Italy, Latvia, Lithuania, Luxembourg, Malta, Netherlands, Norway, Poland, Portugal, Romania, Slovakia, Slovenia, Spain, Sweden, Switzerland and United Kingdom.

The application of the SP test for creep has gained significant interest in the last decade, also as a result of research coordinated by the European Pressure Equipment Research Council (EPERC) to develop a CEN Code of Practice for the application and use of the small punch test for both creep rupture and tensile and toughness properties.



EUROPEAN COMMITTEE FOR STANDARDIZATION  
COMITE EUROPEEN DE NORMALISATION  
EUROPAISCHES KOMITEE FÜR NORMUNG

Management Centre: rue de Stassart, 36 B-1050 Brussels



## 1. INTEGRITY Project of repair welds

BM (P91 Base material)

WM (Weld material)

HAZ material

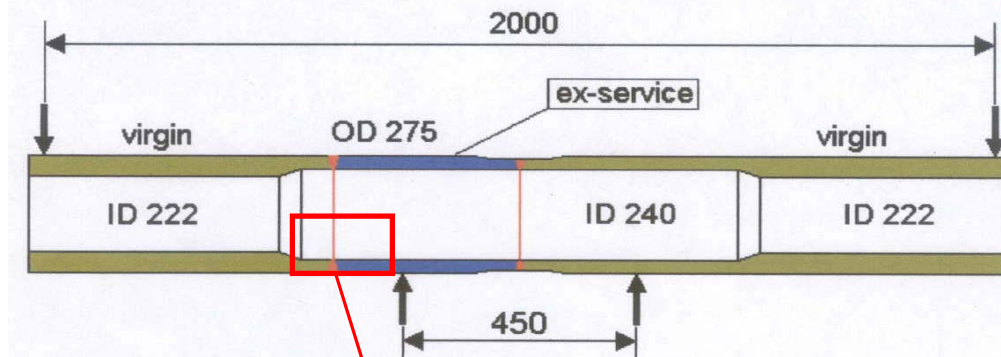
SE (Service exposed\*)

\*Service exposed conditions: 60kh at 565°C under pressure of 250 bar.

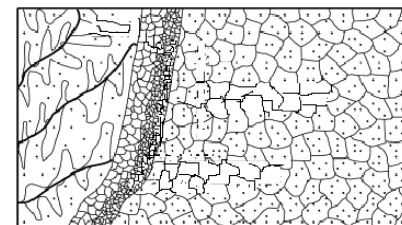
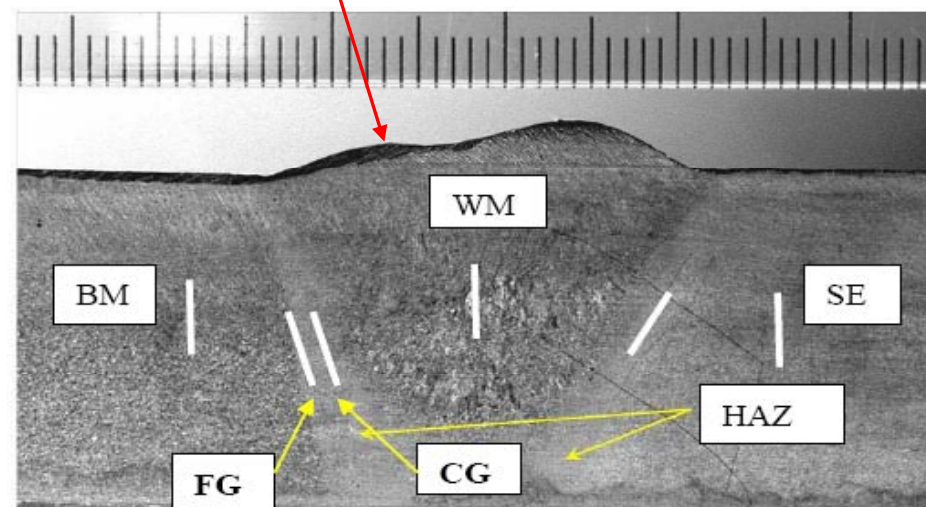
## 2. BRITE-EURAM LICON Project

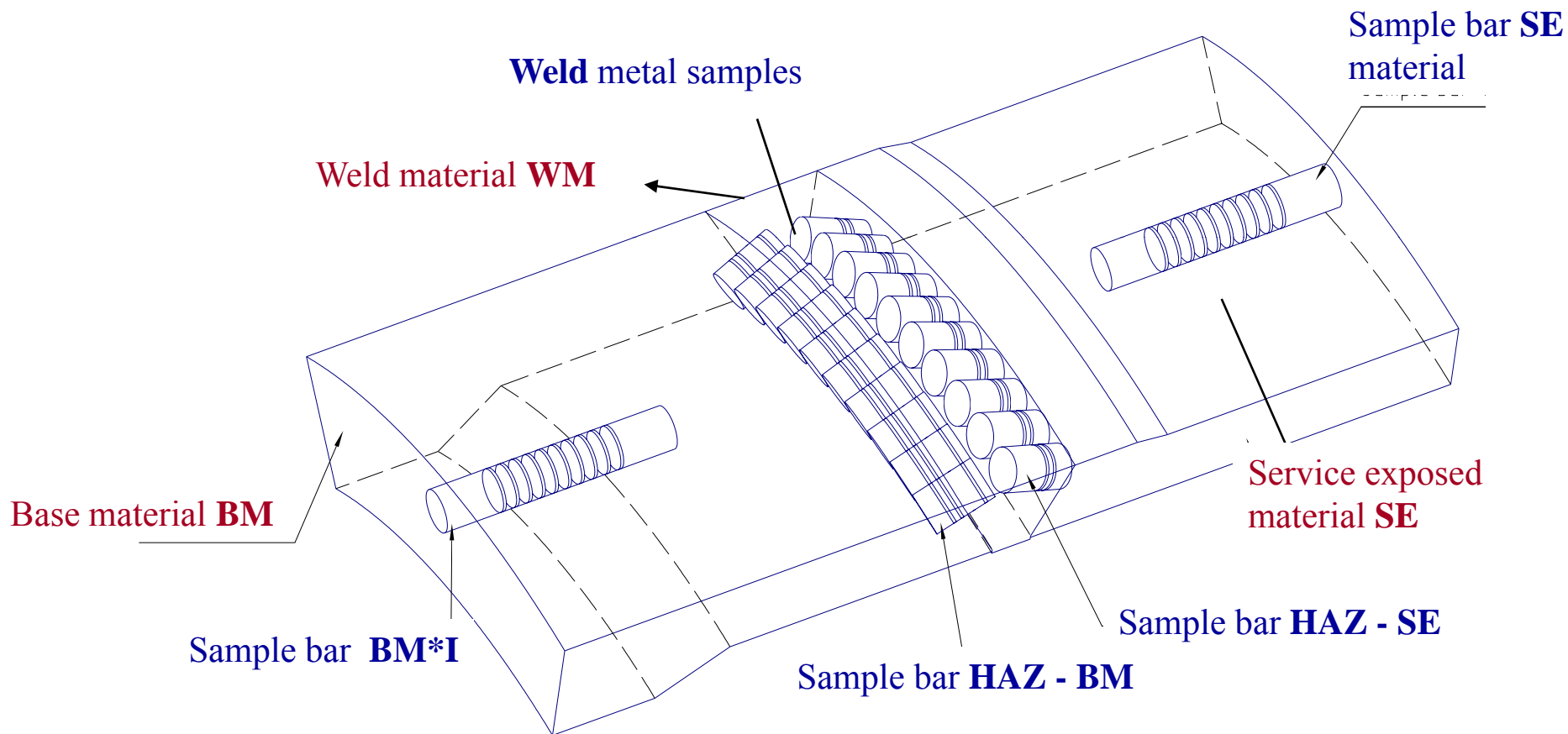
Virgin P91 material

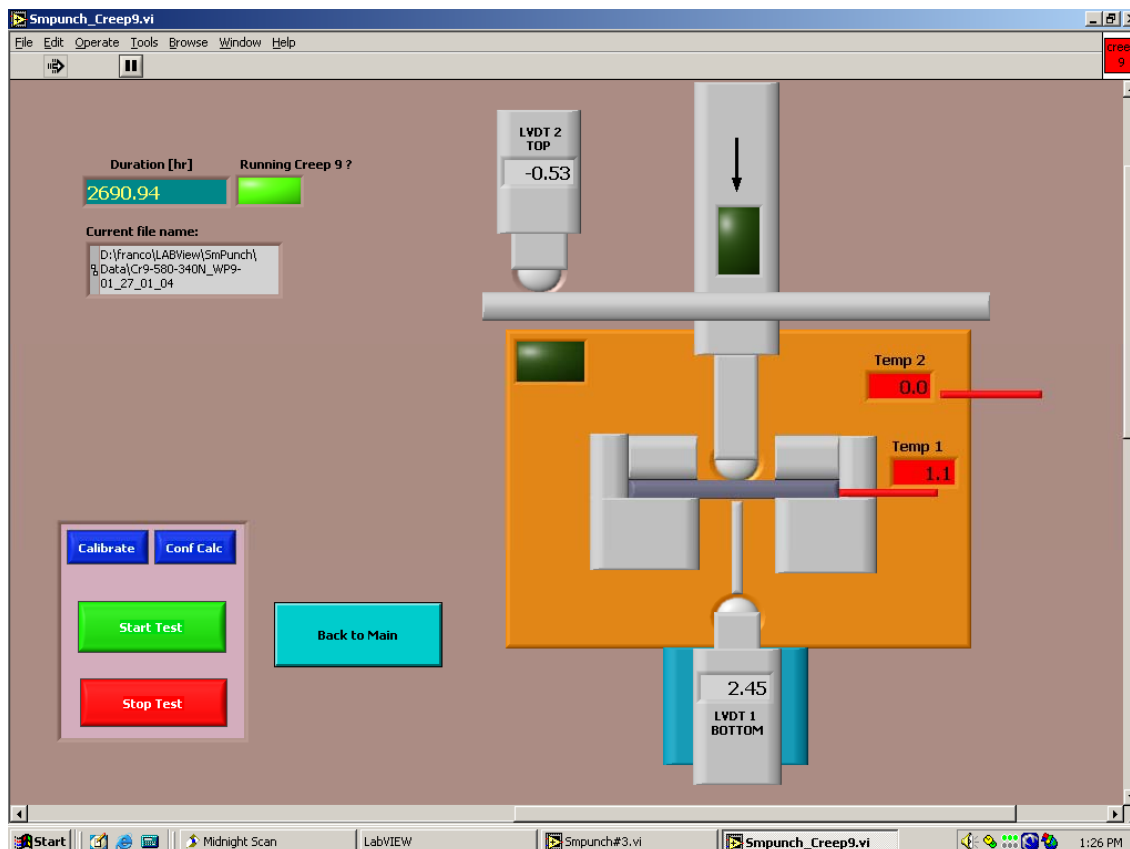
(D. Blagoeva & R. Hurst)



**INTEGRITY Pipe**

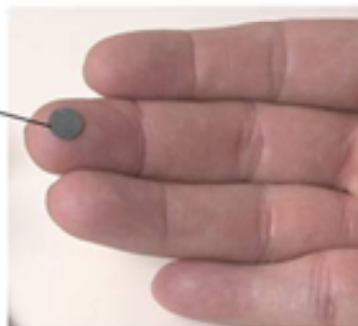
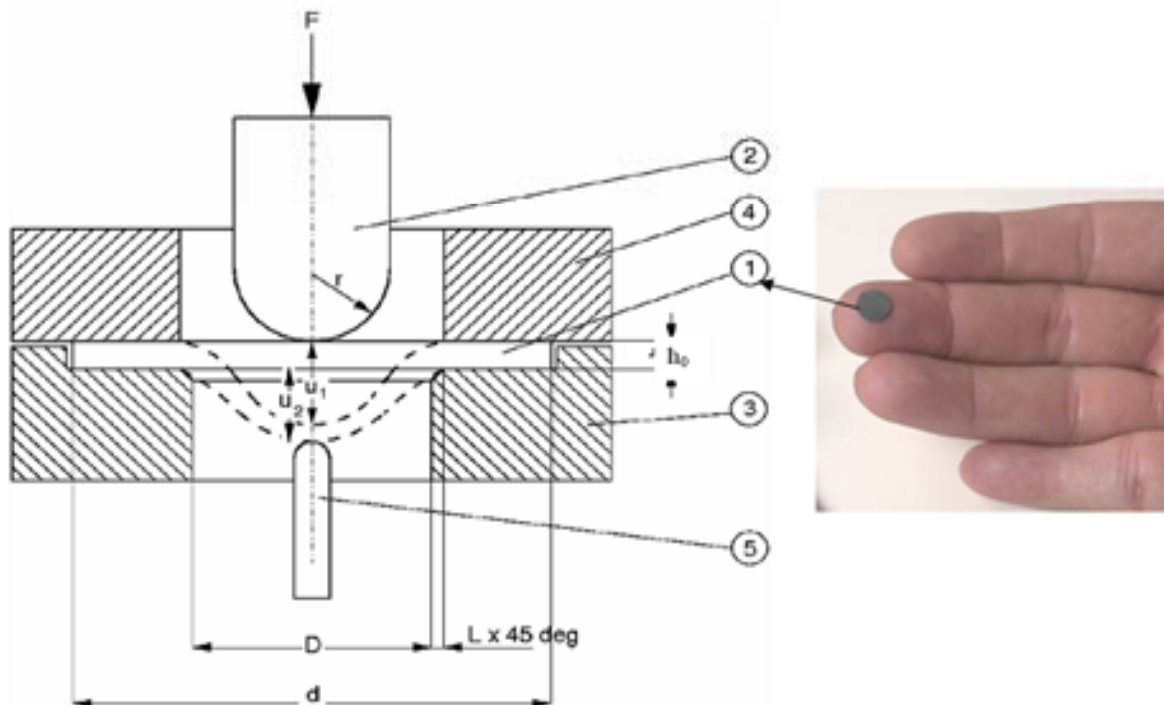






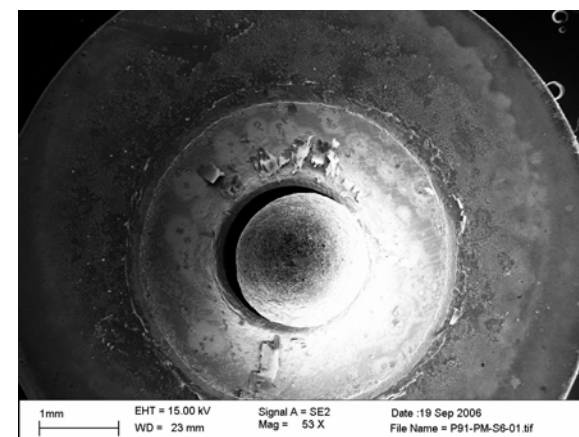
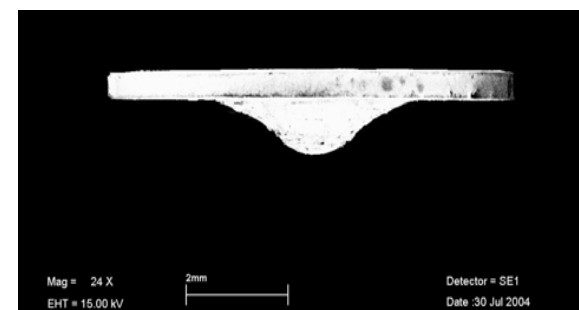
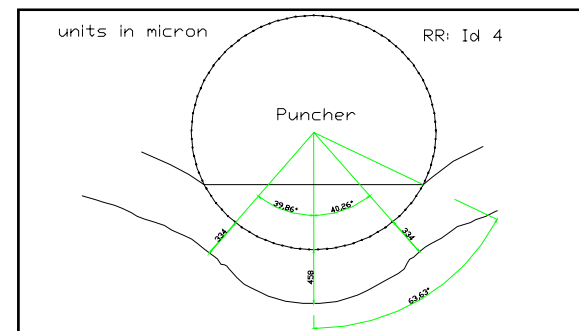
SP control interface

SP creep test rigs at JRC-IE

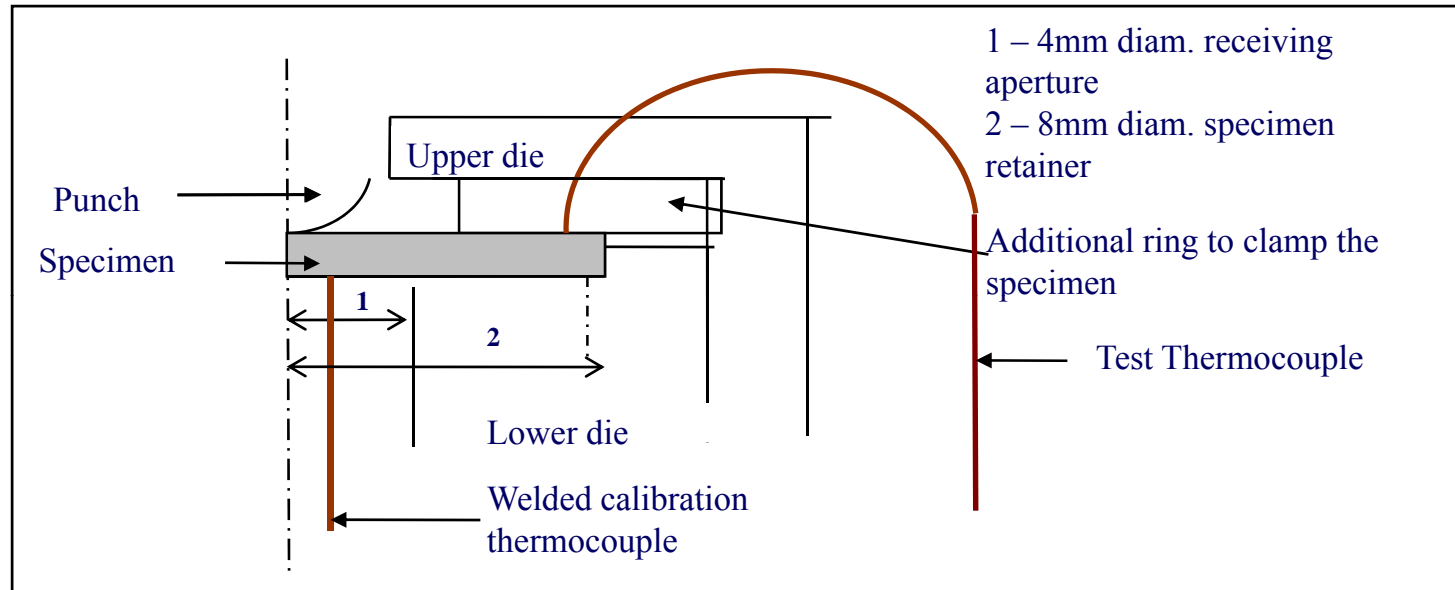


Key  
 1. SP disc test-piece  
 2. Hemispherical ended punch  
 3. Lower die  
 4. Upper die  
 5. Dilatometer push rod

- Sample: Disc of  $\varnothing d = 8$  mm, and thickness  $h = 0.5$  mm
- Hemispherical Puncher: radius  $r = 1.0$  mm
- Receiving hole:  $R = 4$  mm
- Protective atmosphere of Argon



## 1. Accurate temperature control

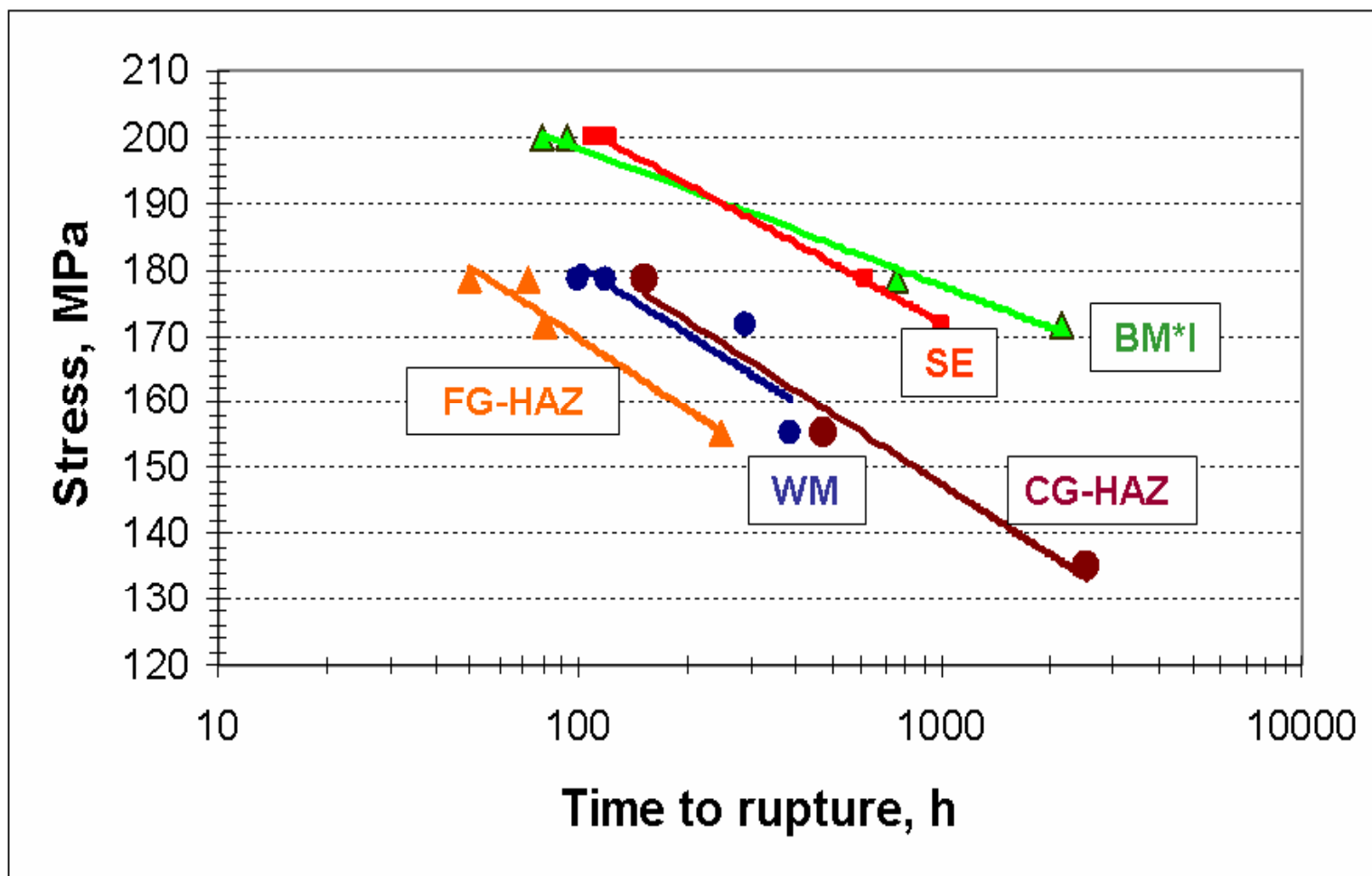


## 2. Correlating forces and stresses

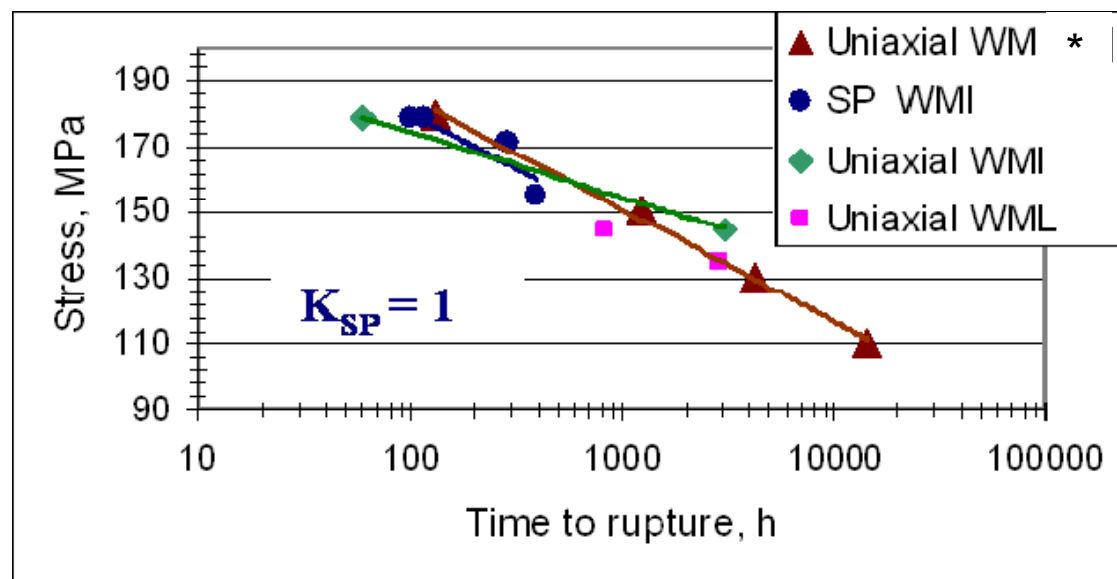
$$F/\sigma = 3.33 K_{SP} R^{-0.2} r^{1.2} h$$

- r radius of the punch indenter
- h specimen thickness
- R radius of the receiving hole
- $K_{SP}$  ductility constant for the material under test

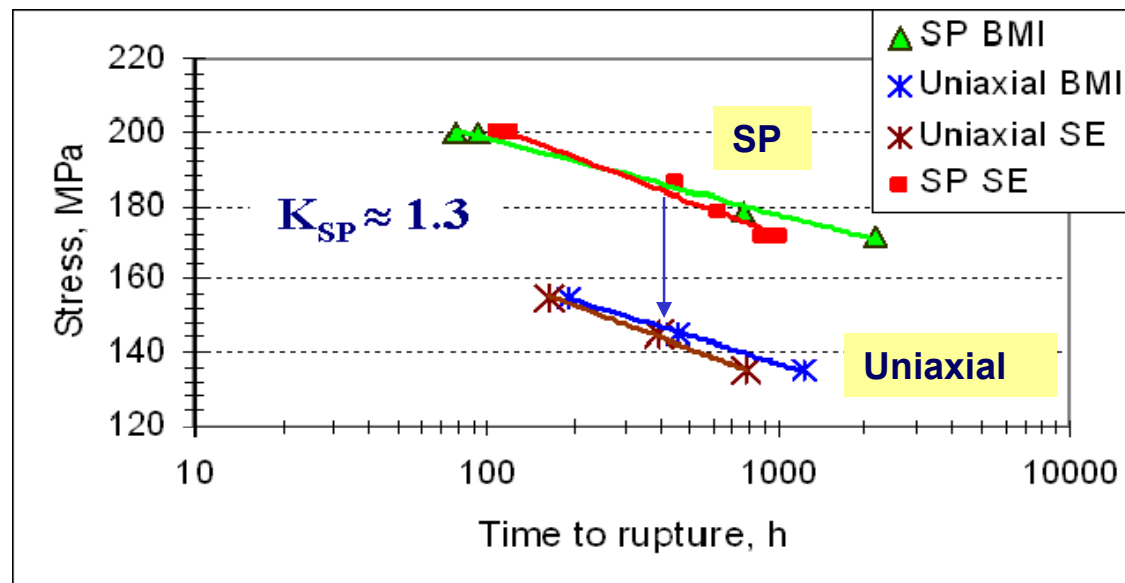
Comparison of SP creep rupture results for different zones from the P91 weldment



Comparison between uniaxial & SP stress rupture data

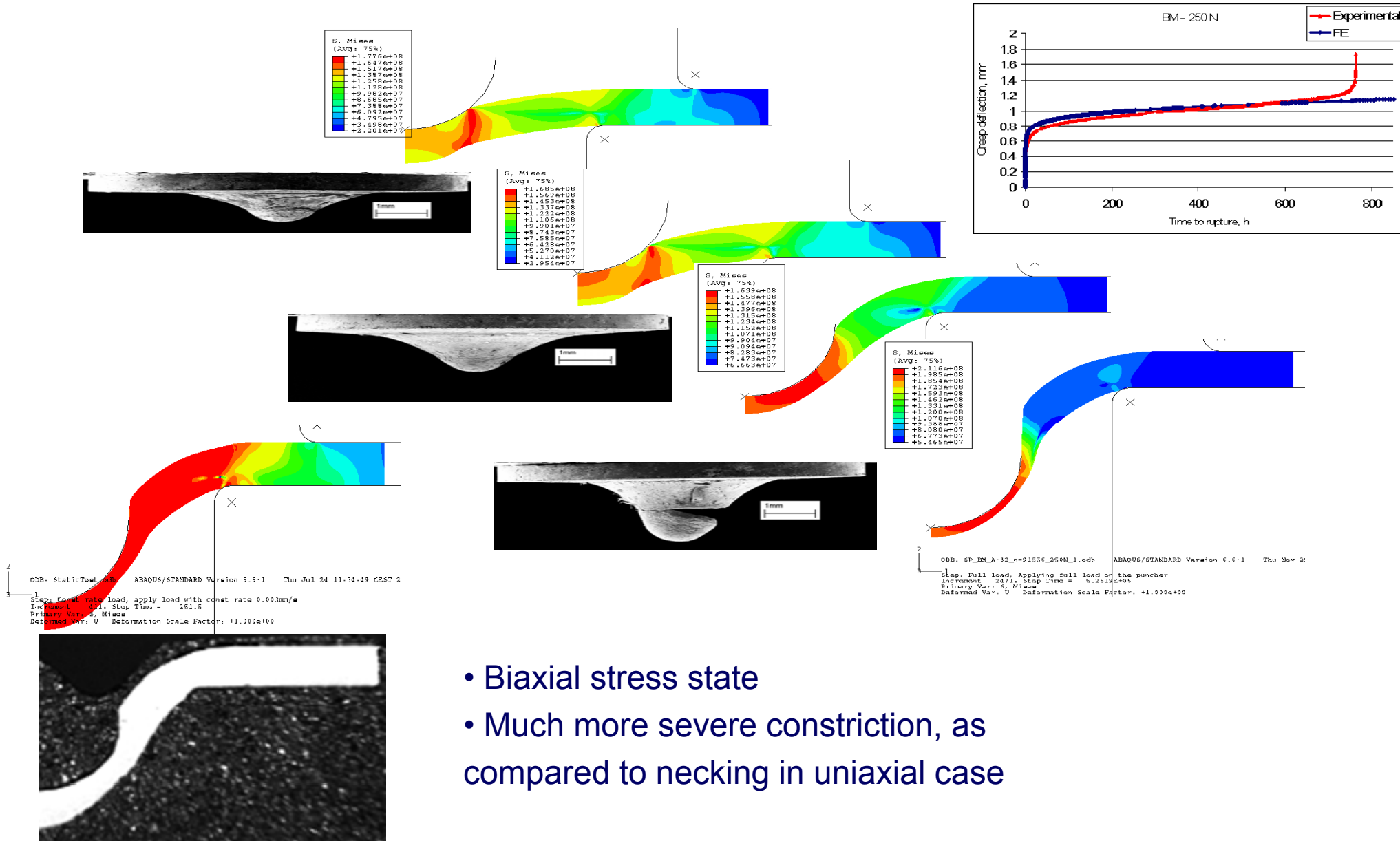


$$F_{SP}/\sigma = 3.33 K_{SP} R^{-0.2} r^{1.2} h$$



\* R.Sandström et al. uniaxial creep data for P91 at 600°C





- Biaxial stress state
- Much more severe constriction, as compared to necking in uniaxial case



- The SP test is a reliable method to depict creep behaviour of alloys. SP creep curves are similar to conventional uniaxial creep curves. A correlation between the SP load and creep stress has been demonstrated.
- The SP test leads to differentiation of creep curves and stress rupture properties for BM, weld and HAZ's.
- The SP results clearly indicate that the weld metal is weaker than both the BM and SE and the CG HAZ zone behaves similarly to the WM. The FG HAZ is shown to be the weakest component of the weldment.
- In terms of stress rupture behaviour there appears to be straightforward correlation between uniaxial and SP creep as regards WM tests (default ductility factor from the CoP,  $K_{SP} = 1$ ). However,  $K_{SP} \neq 1$  for BM & SE creep results due to larger ductility.

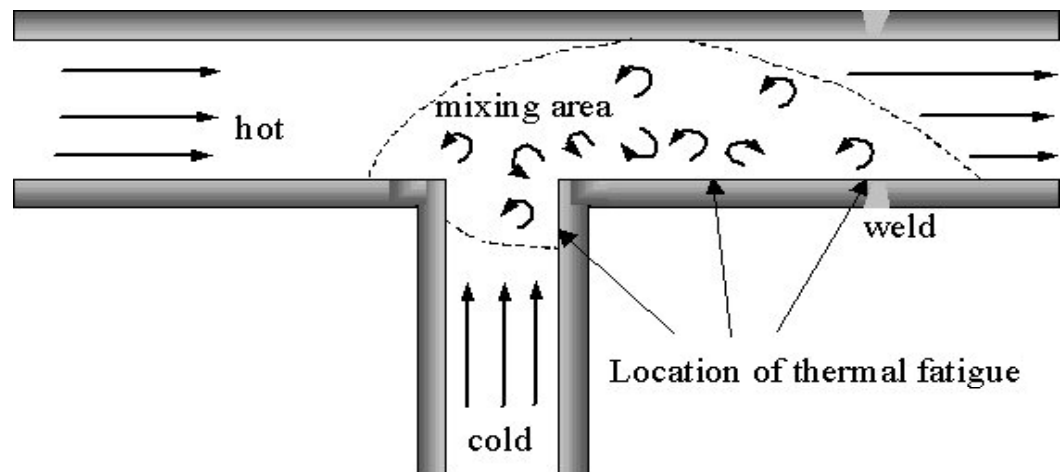
# **Thermal Fatigue Studies for Nuclear Piping Components**

Thermal fatigue due to stratification and mixing of hot and cold water continues to be of importance for the life assessment of the current NPP piping systems.

A specific TF phenomenon, known as ‘mixing tee scenario’, results from turbulent mixing of two fluids at different temperatures.

Here TF damage is influenced by:

- Flow characteristics
- Material properties
- Pipe geometry



Issue: Development of improved practical methods for predicting thermal fatigue damage

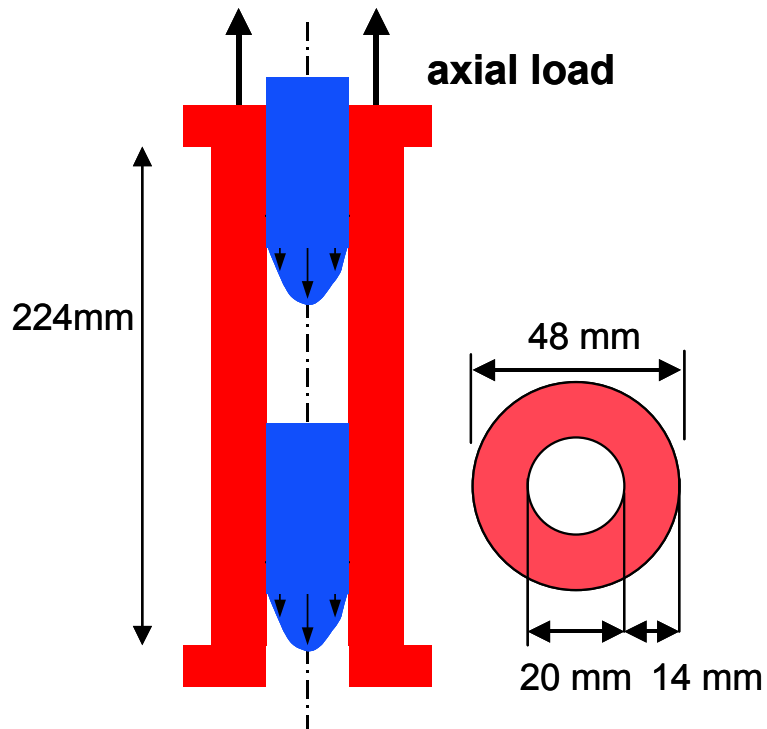
(E. Paffumi)

- Thermal striping fatigue damage due to incomplete mixing of fluid streams at different temperature has the potential to occur in a number of areas if sufficiently good heat transfer exists between fluid and component.
- In GEN IV: potential degradation mechanisms for components in liquid-metal cooled fast breeder reactor (LMFBR) and gas cooled fast reactors (GFR).
- Due to large temperature differences between liquid metal emerging from the core and breeder sub-assemblies, thermal striping can arise in LMFBR, in the structures located above the core.
- Other areas of potential occurrence include piping systems in pressurized (PWR) and boiling water (BWR) reactors:
  - at the Residual Heat Removal System (RHRS) of the Civaux NPP Unit-1 1998;
  - at the excessive extraction piping at the Genkai Nuclear Power Station Unit-2 2007

**This work is being performed within the framework of  
NULIFE Thermal Fatigue Network of Excellence**



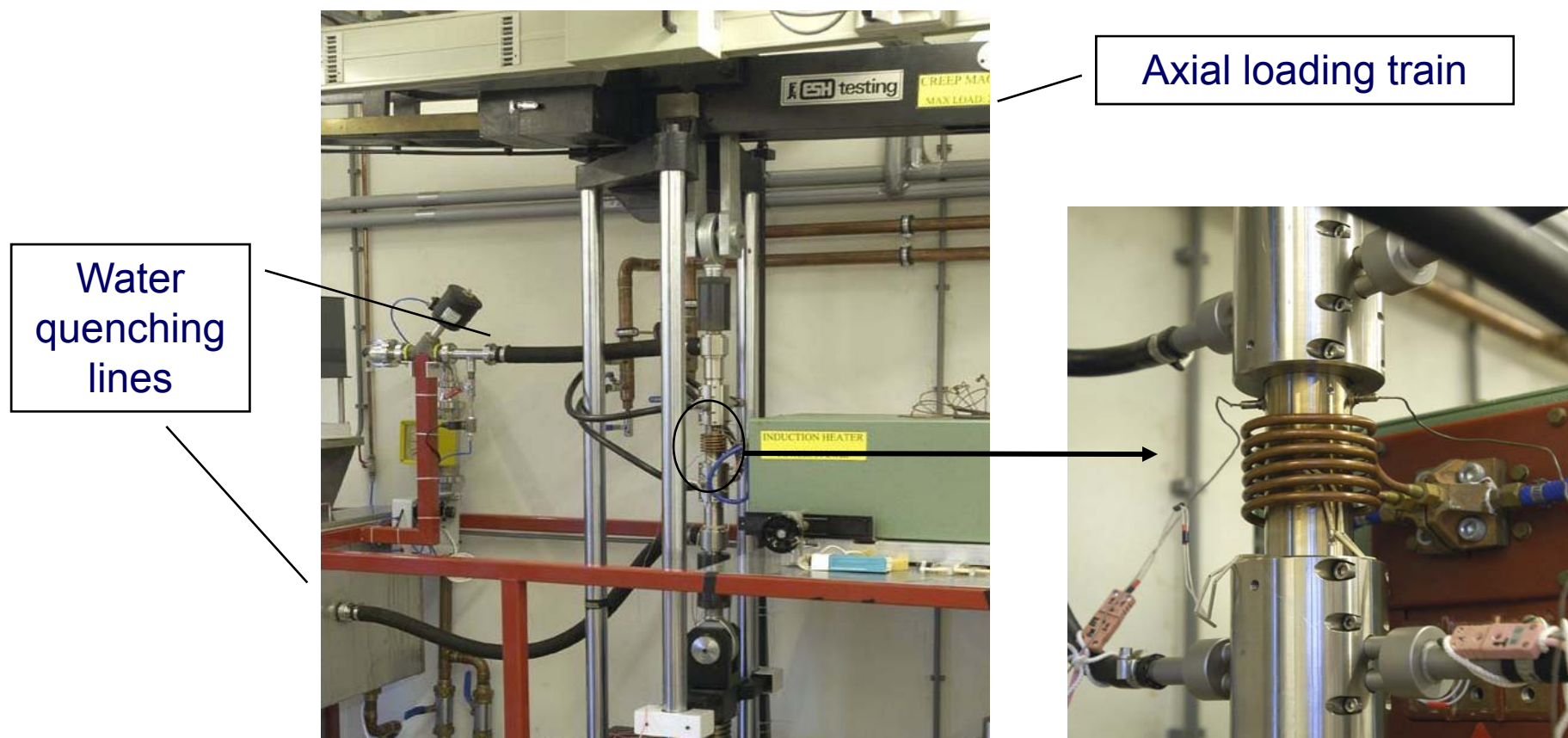
## Thermal shock experiments on cylindrical specimens from 316L stainless steel



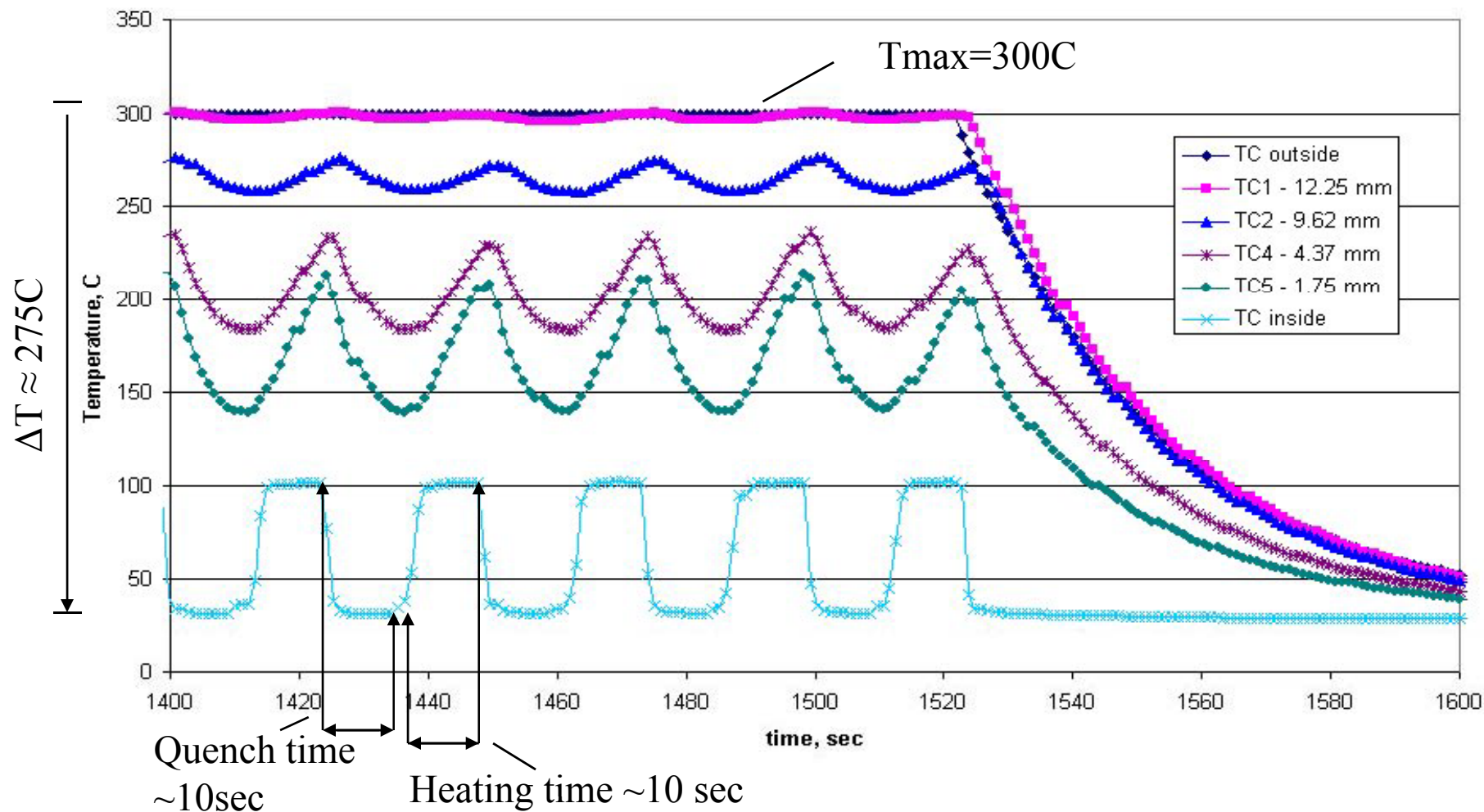
External diameter: 48 mm  
Internal diameter: 20 mm

Length: 224 mm  
Wall thickness: 14 mm

The specimen is heated by electromagnetic induction from outside and quenched internally with room temperature water.





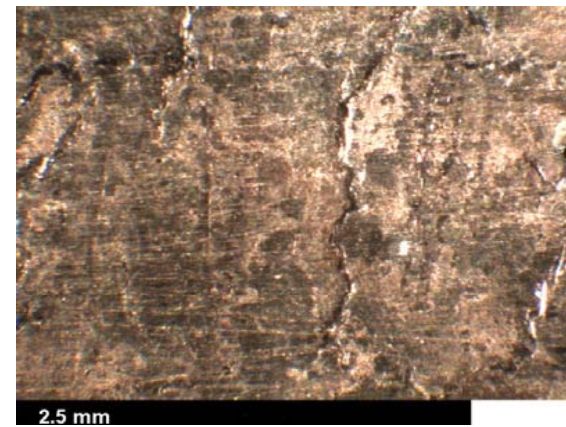
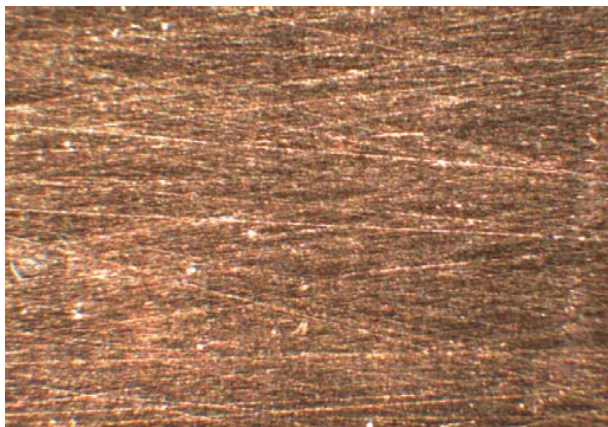


## Experimental techniques for fatigue damage detection by test interruption at pre-selected numbers of cycles

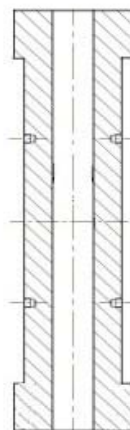
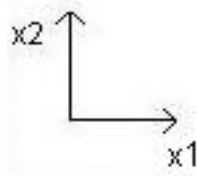
- **Crack initiation:**
  - Cellulose acetate replica technique for the detection of surface cracking (crazing cracks)
  - Radiography with panoramic view to provide complementary information about crack distribution revealed by replica technique
- **Crack growth:**
  - Ultrasound time of flight diffraction technique (TOFD) for the through-wall crack characterization - location, length and depth of cracks



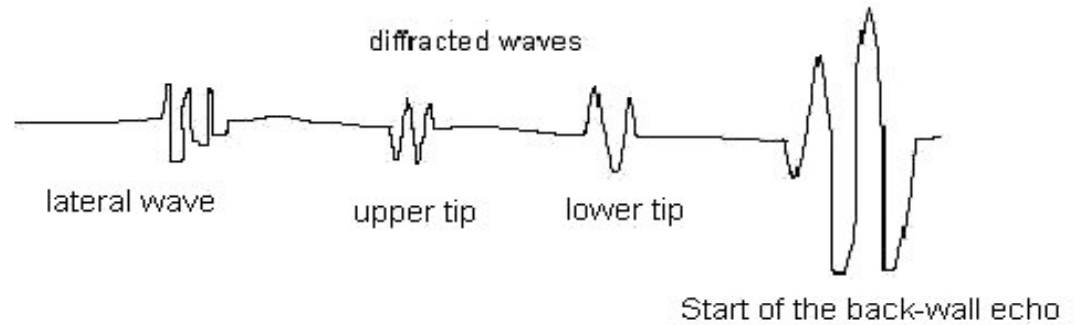
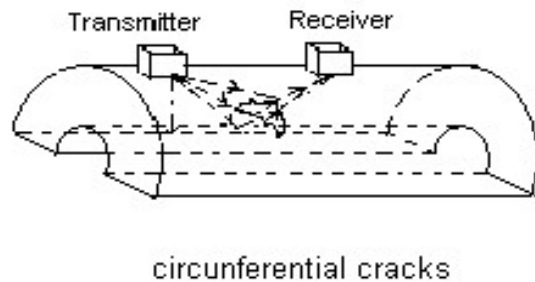
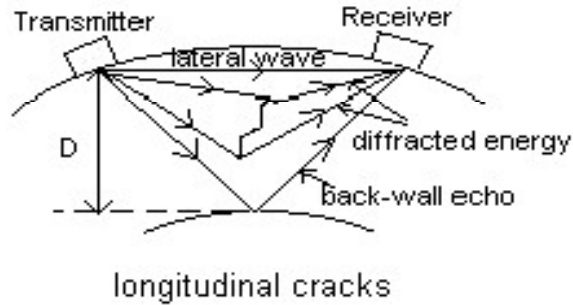
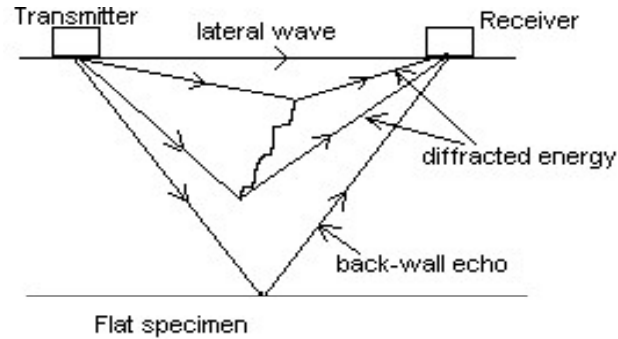
Examples of cracks detected by replica – Test TF1 –  $T_{max}=300^{\circ}\text{C}$



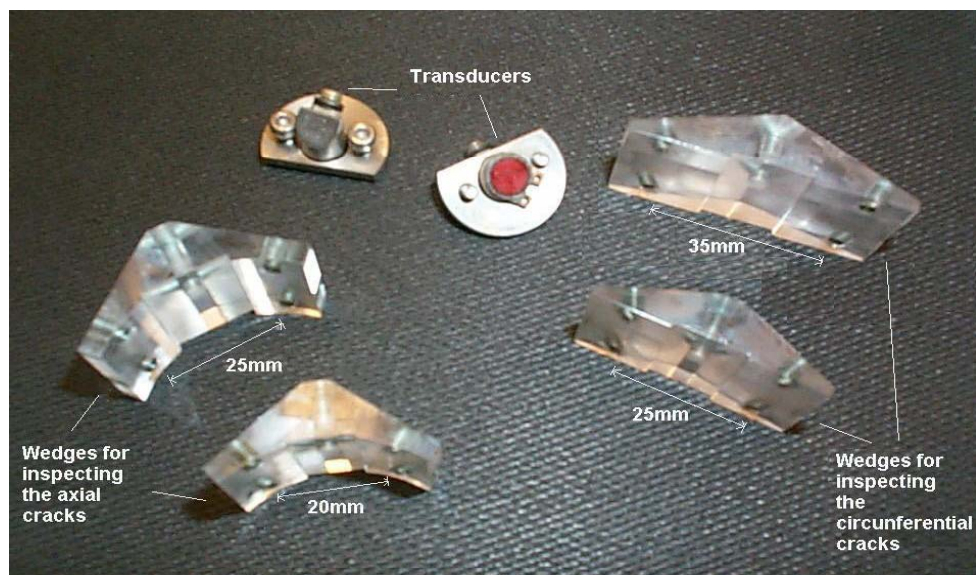
Inner surface before  
testing



Inner surface after 55600  
quenching cycles 25-300°C  
Time quenching: 5sec    Time  
heating: 40 sec



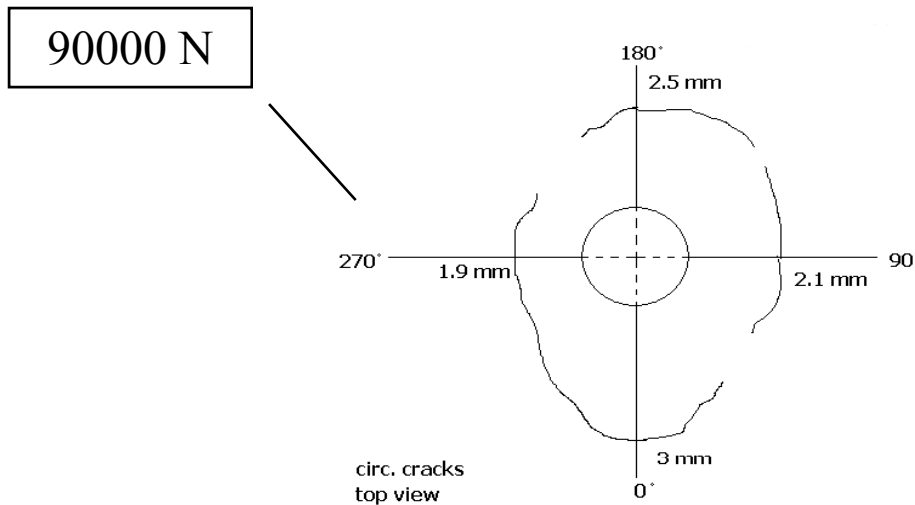
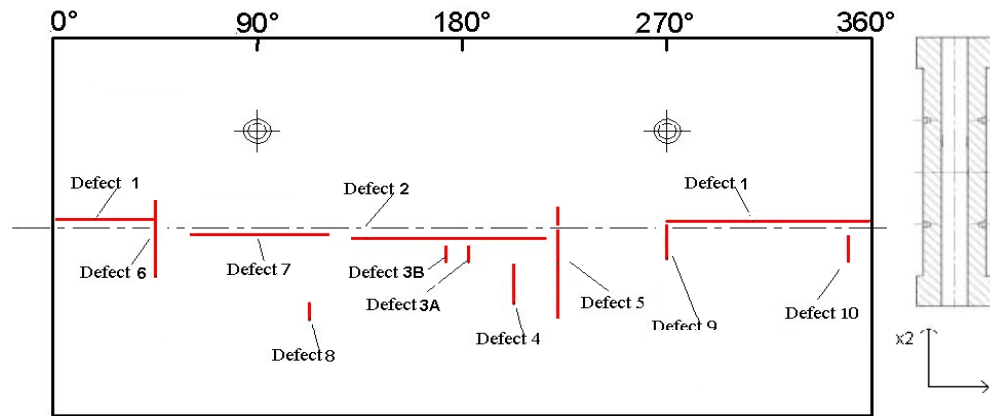
Typical pattern of pulses from an internal flaw



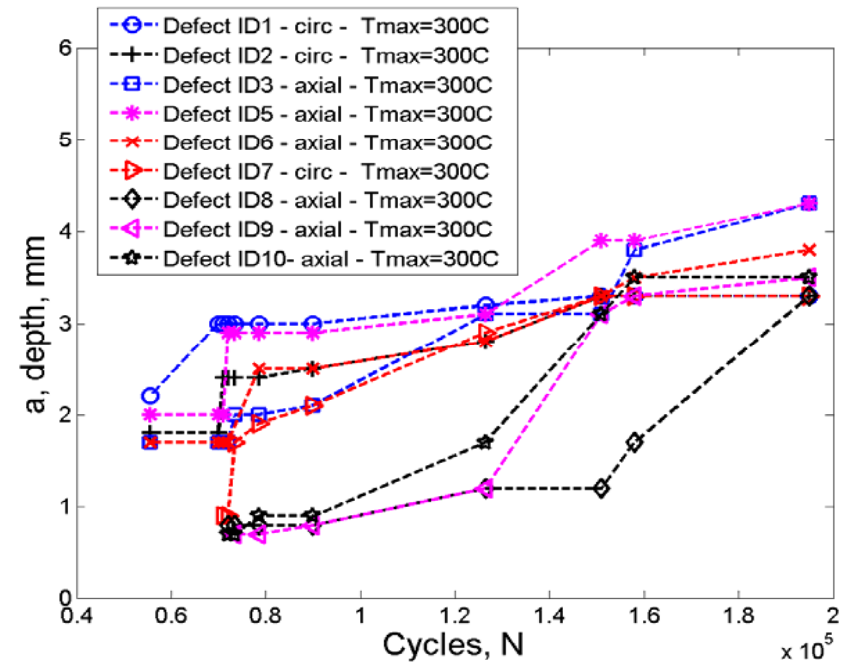
Transducers and wedges used during the inspections

Transducers on a wedge used for inspecting axial cracks





Crack depth versus number of quenching cycles for the defects of the specimen TF1



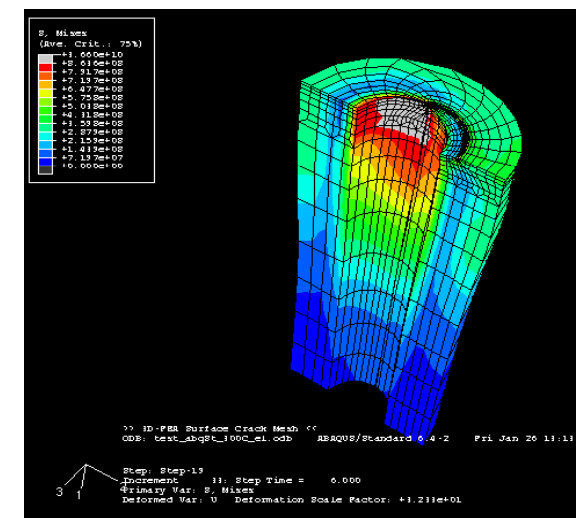
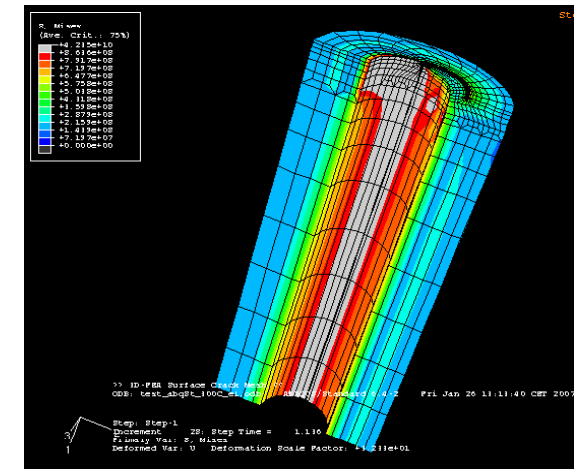
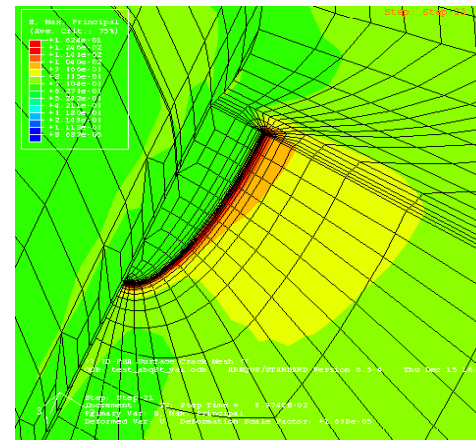
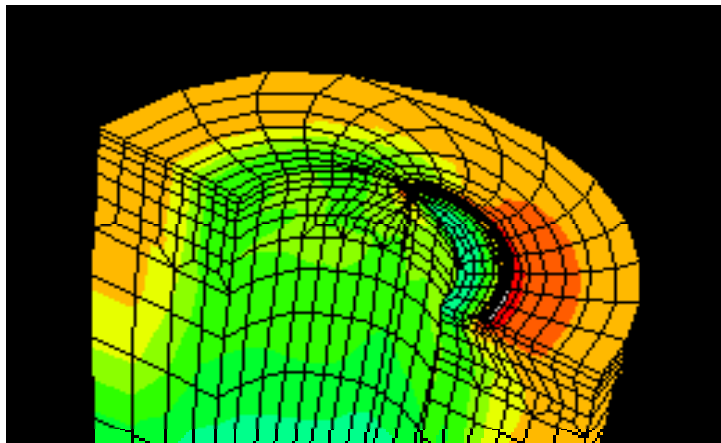


## FE simulation of semi-elliptical cracks under thermal fatigue loading

Combination of detailed 2-D and 3-D simulations and experimental data to assess experimental findings

Cracked body analysis based on

- Short (< 1 mm) crack model
- Elastic-plastic long crack fracture mechanics model



- A series of thermal fatigue tests has been carried out on tubular specimens in 316L(N) steel, with test durations up to 200000 cycles. A customized TOFD measurement system allows assessment of the distribution and depth of thermal fatigue cracks.
- Combining FE-generated thermal stress fields with simplified models for semi-elliptical cracks provides a conservative description of the crack growth behaviour.
- The results highlight the importance of well-documented experimental benchmarks for thermal fatigue assessment methods, such as the proposed European procedure being developed in the NULIFE network. This can be used to demonstrate the damage tolerance of existing and future reactor designs.

# **Strain-Controlled Thermo-Mechanical Fatigue: Research into Best Practices**

## ***Thermo-Mechanical Fatigue (TMF):***

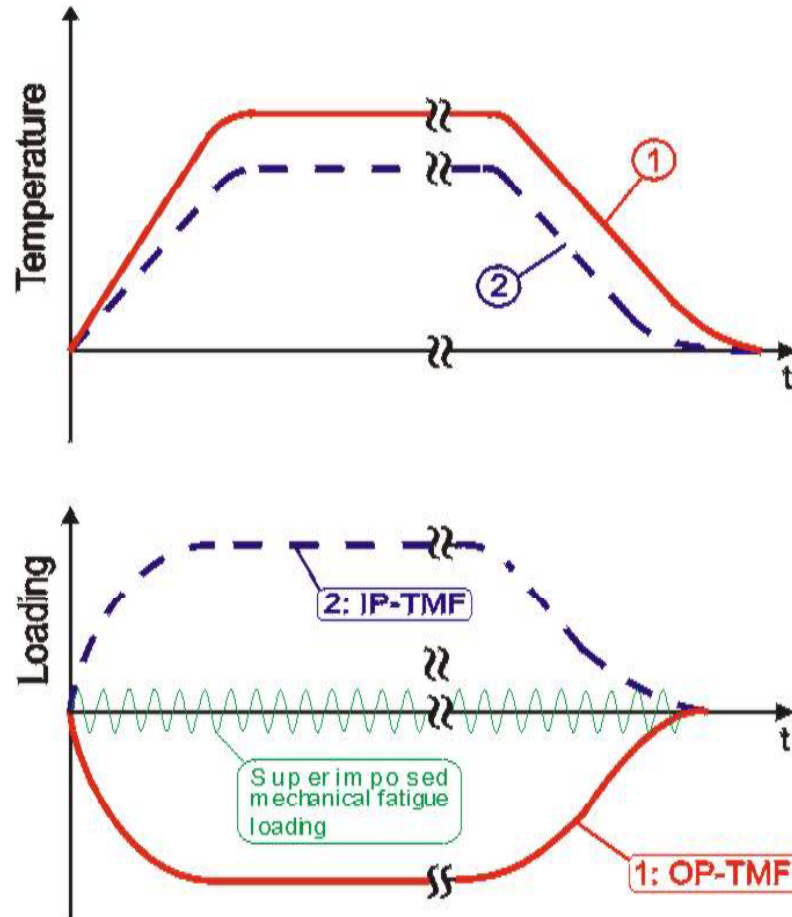
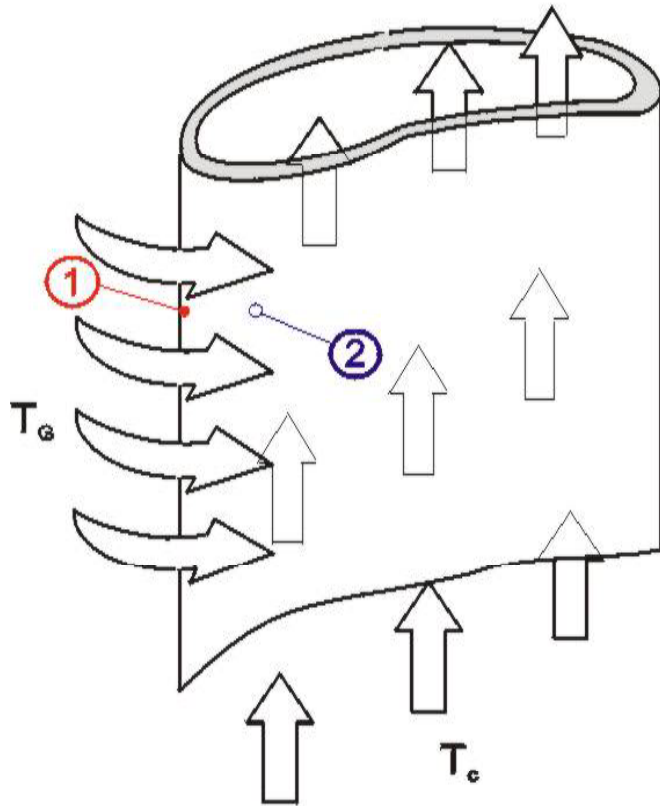
**Thermal fatigue** of components results from **internal thermal constraints** associated with cyclic temperature gradients.

Alternatively, thermal constraints of a representative volume element can be simulated by *uniform temperature and mechanical strain* fields cyclically imposed on a specimen with uniform gauge section.

⇒ strain-controlled **thermo-mechanical fatigue** test

Design & residual life analysis of safety critical components exposed simultaneously to thermal & mechanical loads (gas turbines, aero engines, automotive & process industries)



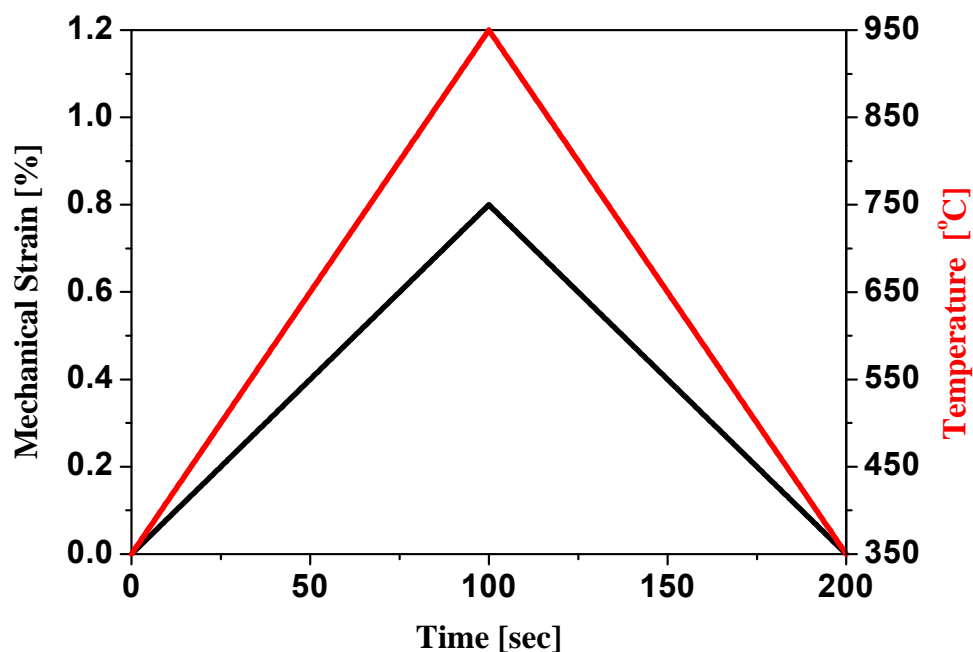


temperature:  $T = T_0 + \delta T F(\omega t)$ , mech. strain:  $\epsilon_m = \epsilon_{m,0} + \delta \epsilon F(\omega t - \varphi)$

phase angle  $\varphi$ ,  $F(x) = F(x + 2\pi)$ ,  $\forall x$

## Triangular TMF waveforms (in strain control without dwell):

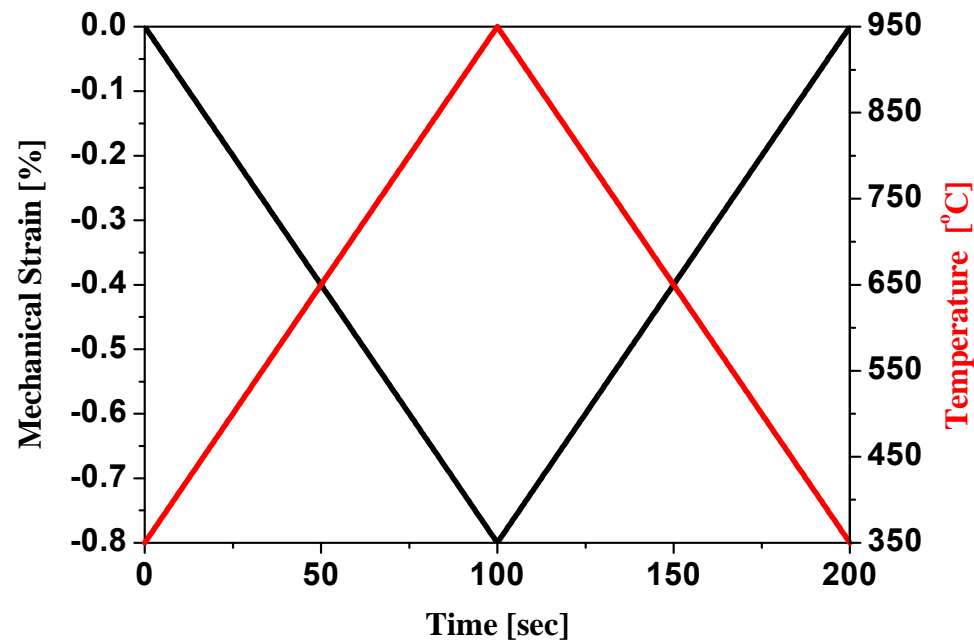
in phase  $\varphi = 0$ ,  $R_\varepsilon = 0$



*mech. strain ratio:*

$$R_\varepsilon = \varepsilon_{m,\min} / \varepsilon_{m,\max}$$

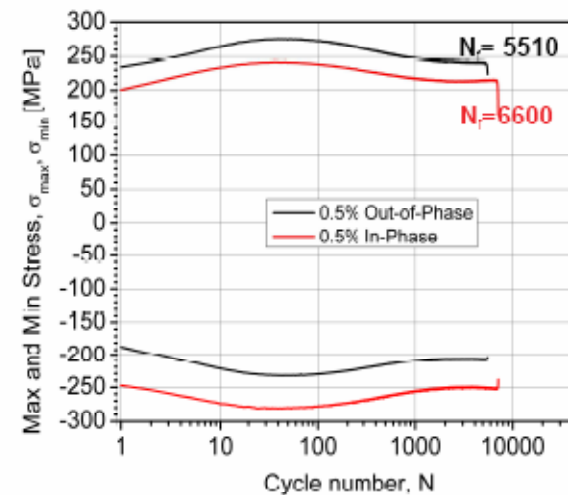
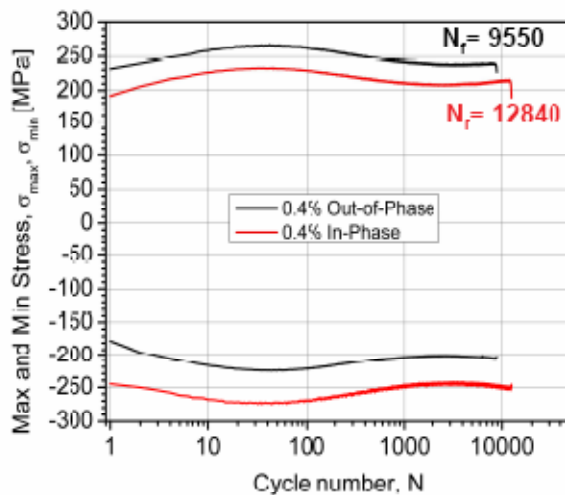
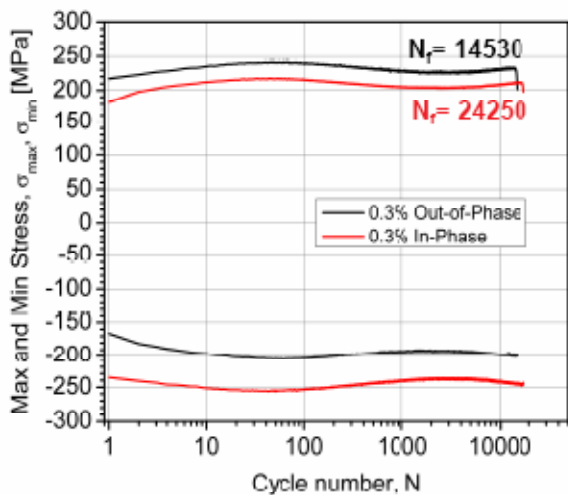
out-of-phase  $\varphi = 180^\circ$ ,  $R_\varepsilon = -\infty$



| TP (Code) | C     | Si   | Mn   | P     | S     | Cr   | Mo    | Ni    | Co    | Cu    | N      | Nb    | Ti    |
|-----------|-------|------|------|-------|-------|------|-------|-------|-------|-------|--------|-------|-------|
| 304L (H)  | 0.024 | 0.35 | 1.49 | 0.026 | 0.005 | 17.9 | 0.247 | 10.00 | 0.088 | 0.305 | 0.059  | 0.001 | 0.001 |
| 316L (B)  | 0.021 | 0.26 | 1.69 | 0.033 | 0.003 | 17.5 | 2.15  | 11.14 | 0.093 | 0.273 | 0.0601 | 0.012 | 0.003 |
| 321 (R)   | 0.073 | 0.20 | 1.87 | 0.021 | 0.004 | 18.0 | 0.334 | 10.20 | 0.055 | 0.182 | 0.0098 | 0.001 | 0.517 |
| 347 (K)   | 0.058 | 0.20 | 1.72 | 0.025 | 0.006 | 17.4 | 0.441 | 10.40 | 0.077 | 0.268 | 0.045  | 0.571 | 0.001 |

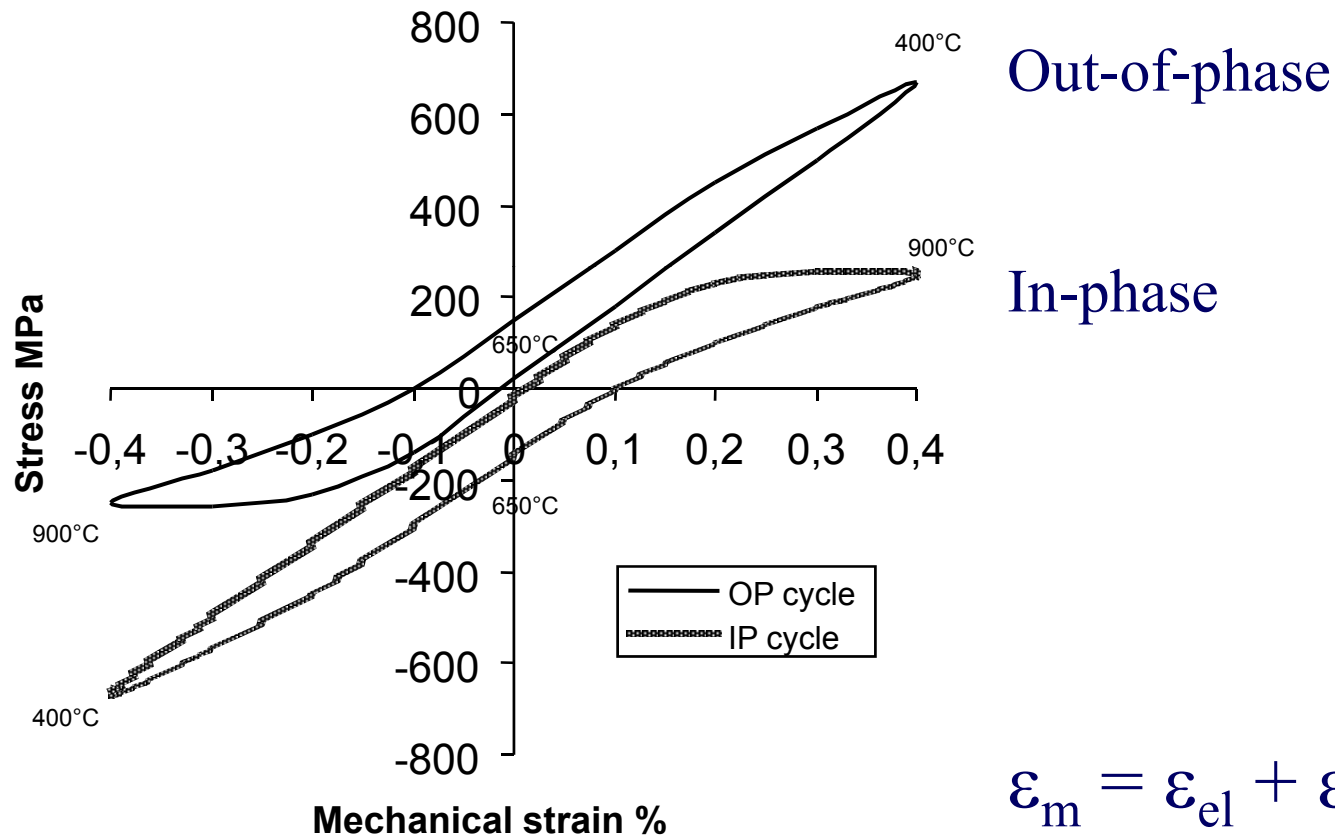
## Out-of-Phase vs. In-Phase TMF

Material TP 347,  $T = 100 - 340^{\circ}\text{C}$ ,  $\nu = 0.03\text{Hz}$ ,  $\epsilon_m = 0.3, 0.4, 0.5\%$



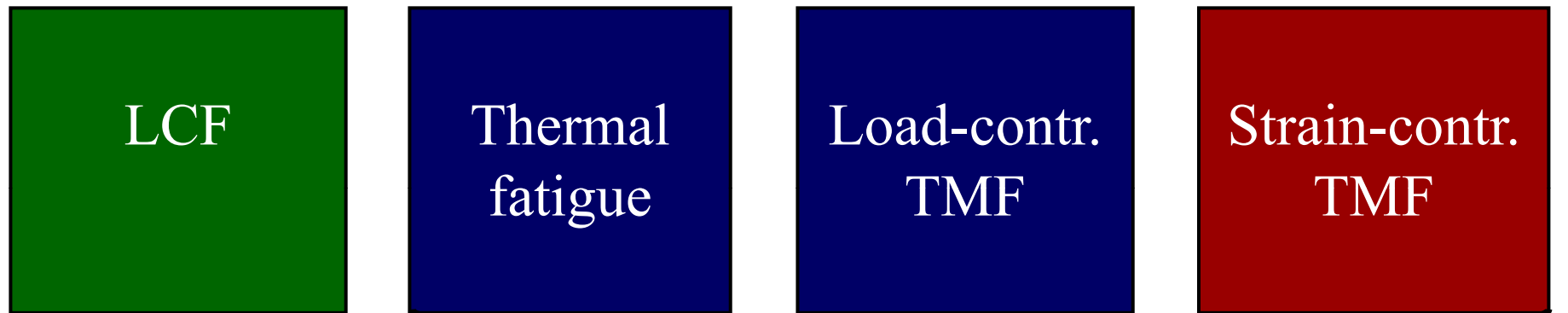
OP cycles result in shorter TMF lives

## Hysteresis loops:



*Effect on Number of TMF Cycles to Failure*

- Load train alignment
- Precision extensometry



**Dynamic temperature measurement & control**

Therm. – mech. load phasing

- Pre-cycling, start-up procedures
- Thermal strain compensation

Time based:

$$\varepsilon_{\text{tot}}(t) = \varepsilon_{\text{m}}(t) + \varepsilon_{\text{th}}(t)$$

Temperature based:

$$\varepsilon_{\text{tot}}(t, T) = \varepsilon_{\text{m}}(t) + \varepsilon_{\text{th}}(T)$$

## **TMF-Standard Project GRD2-2000-30014**

### **Thermo-mechanical fatigue – the route to standardisation**

(funded within the GROWTH Programme of the EC)

**Objectives:** to establish a TMF testing platform in Europe  
to issue a *validated* code-of-practice  
to disseminate and exploit the results

**Duration:** 48 months (2001 – 2005)

**Budget:** ~ 1.800.000 Euro (~ 1.080.000 EC contr.)

**Consortium:** 10 principal contractors  
9 assistant contractors, 1 external participant

**No. of tests:** ~ 400 incl. 120 validation tests



## ***WP 3: Pre-normative research on TMF procedures & tolerances***

Task 1: Pre-cycling, start, interrupt, restart procedures

### **Task 2: Dynamic $T$ measurement and control**

Task 3: Thermal strain comp., deviations from nominal  $T$

Task 4:  $T$  gradient effects in 3 different sample geometries  
(solid cylindrical, hollow cyl., solid rectangular)

Task 5: Deviations from nominal phase angle

### ***Challenges in TMF temperature measurement and control:***

- Temperature rates (heating & cooling) up to 50°C/s to be controlled with sufficient accuracy ( $\pm 5^\circ\text{C}$  or 1% of  $\Delta T$ )
  - > specimen design, heating system, fast temperature control system
- $T$  measurement must not affect crack initiation and TMF life
  - > Avoid any microstructural damage, e.g. by thermocouple attachment
- Heating method (induction, radiation, ...) must not influence temperature readout in the absence of thermal equilibrium
  - > no cold spot at thermocouple, nor direct over-heating of thermocouple
- Long-term stability of dynamic temperature fields must be ensured
  - > no drift of  $T$  profile



## Issues affecting dynamic $T$ measurement and control:

- heating method
- specimen geometry
- type of  $T$  measurement device: thermocouple (TC), pyrometer
- method of TC attachment

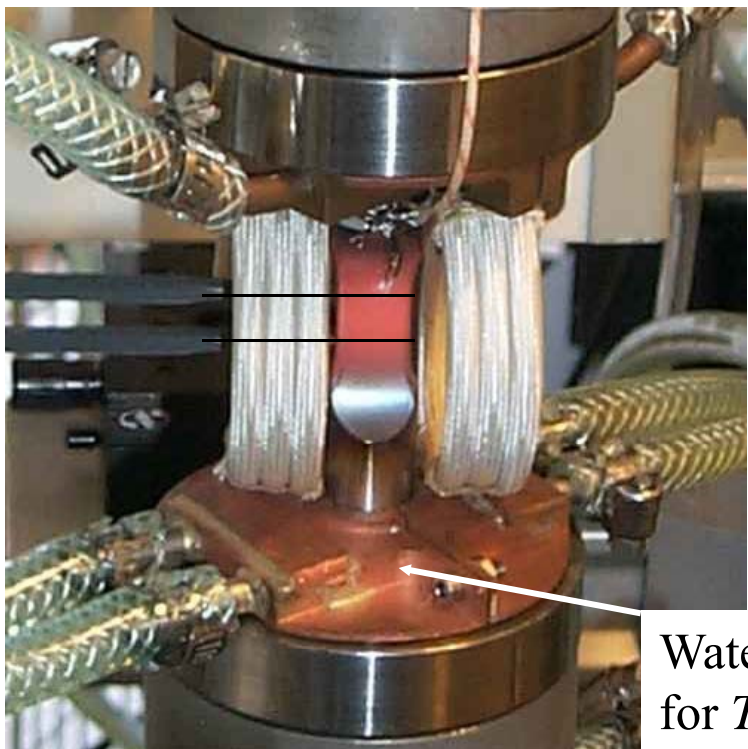
These issues are interrelated and cannot be optimized independently !

*Resistance furnace: too slow*

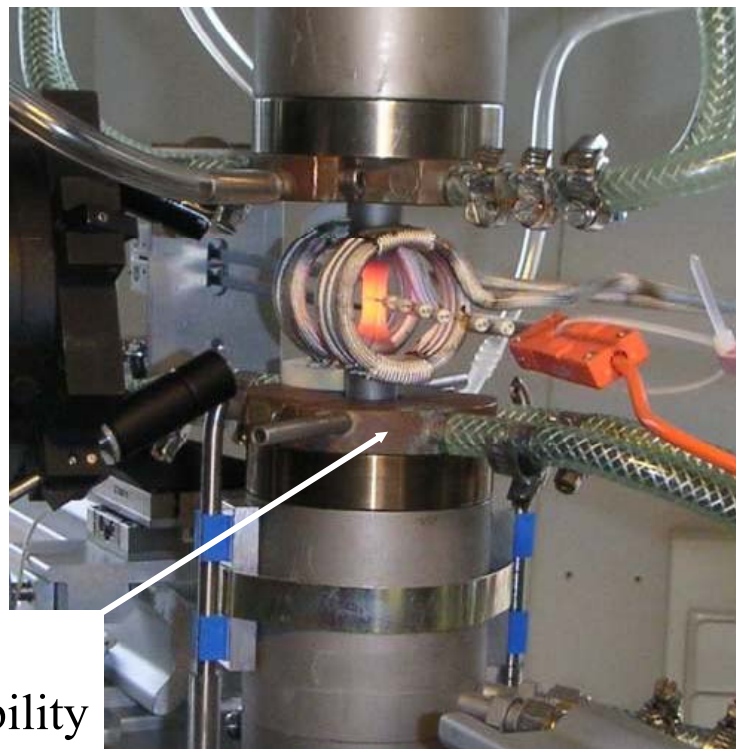
- *Direct ohmic heating: practicable for miniaturized, e.g. thin-walled tubular, specimens only*
- *radiation (bulb) furnace: appropriate, if*
  - excessive radial temperature gradients are avoided
  - issue of reflectivity differences: specimen – TC is addressed
- *Induction heating: most commonly used in TMF*
  - skin effect to be kept sufficiently low
  - “cold spot” to be avoided by proper TC type and attachment method

- heating of subsurface layer: Skin effect  $d = (4\pi\sigma\mu\nu)^{-1/2}$
- ensure very good thermal contact of TC (not heated directly)

TC spot-welded outside GL



Ribbon TC wrapped around centre

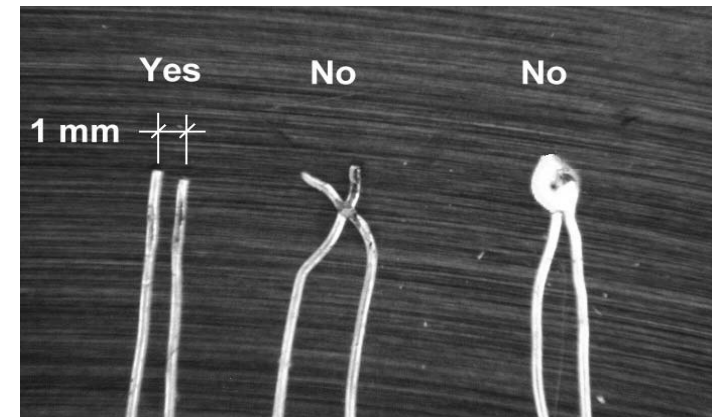


T control by **spot-welded TC outside** the gauge length:

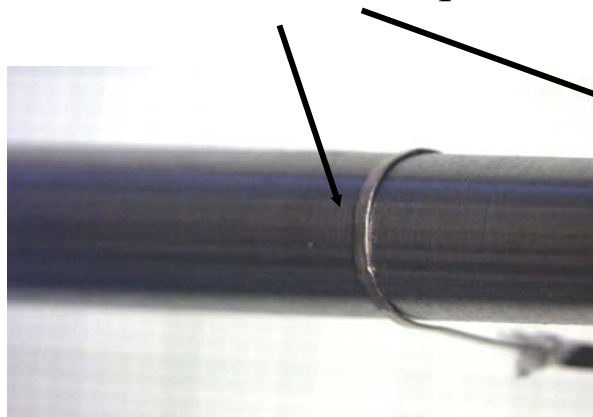
- + Stable and good thermal contact
- Reference temperature in the centre of the GL
- “Cold spot” can be an issue

T control by **ribbon TC in** the gauge length:

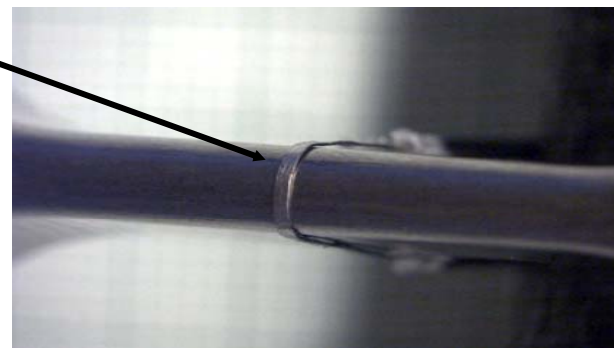
- + Direct measurement in GL w/o damage
- + No “cold spot”, as contacting length sufficiently long
- Thermal contact to specimen surface may be insufficient and prone to variation by oxidation, surface rippling, micro-cracking...



*T* measurement point



Cylindrical specimen



Flat specimen

## TMF-Standard CoP on strain-controlled TMF

- contains the “essence” of 4 years work of 20 European laboratories
- has been validated by ~ 120 TMF tests (OOP and IP)
- comprises a lot of informative material and practical recommendations
- has contributed to improving & harmonizing TMF practices
- provides **underpinning base to the new ISO 12111 standard:**
  - ✓ sign convention of phase angle
  - ✓ recommended specimen geometries
  - ✓ max. allowable temperature deviation
  - ✓ max. allowable temperature gradients (axial, radial, circumferential)
  - ✓ max. thermal hysteresis
  - ✓ Young’s modulus verification

While consensus has been reached as to the max. allowable temperature deviations and temperature gradients, there is still no generally accepted, optimized method of dynamic temperature measurement & control:

- awareness of critical issues remains important;
- use of complementary methods is necessary to ensure accuracy of technique

However, what does „accuracy“ mean in the absence of appropriate methods of

### **Dynamic Temperature Calibration**

???

## **In situ calibration of *dynamic* temperature measurement and control using a phase transformation:**

### **Issues to be addressed:**

- Temperature ramping at  $\pm 1 \dots 10^\circ \text{C/s}$
- No thermal equilibrium between TC & specimen: means of TC attachment
- Little direct heating of TC, mainly indirectly via specimen surface

Will the TC read the right temperature ?

### **Possibilities for establishing reference temperatures:**

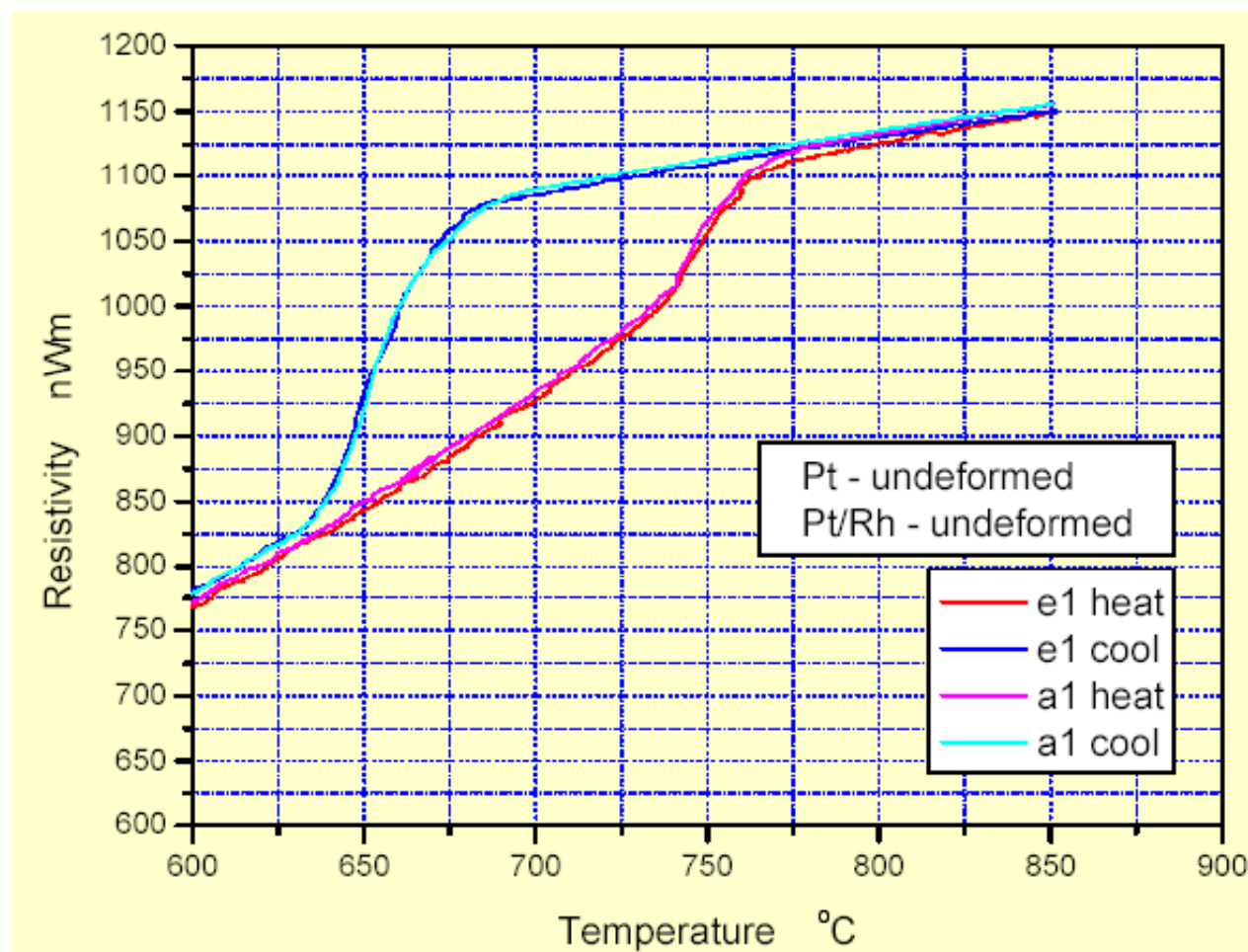
1. NPL: displacive phase transformation of eutectoid steel
2. JRC-IE: solid-liquid phase transition of Al-Cu



**Eutectoid steel (0.8% C) – phase change on heating at  $\sim 740^\circ\text{C}$**

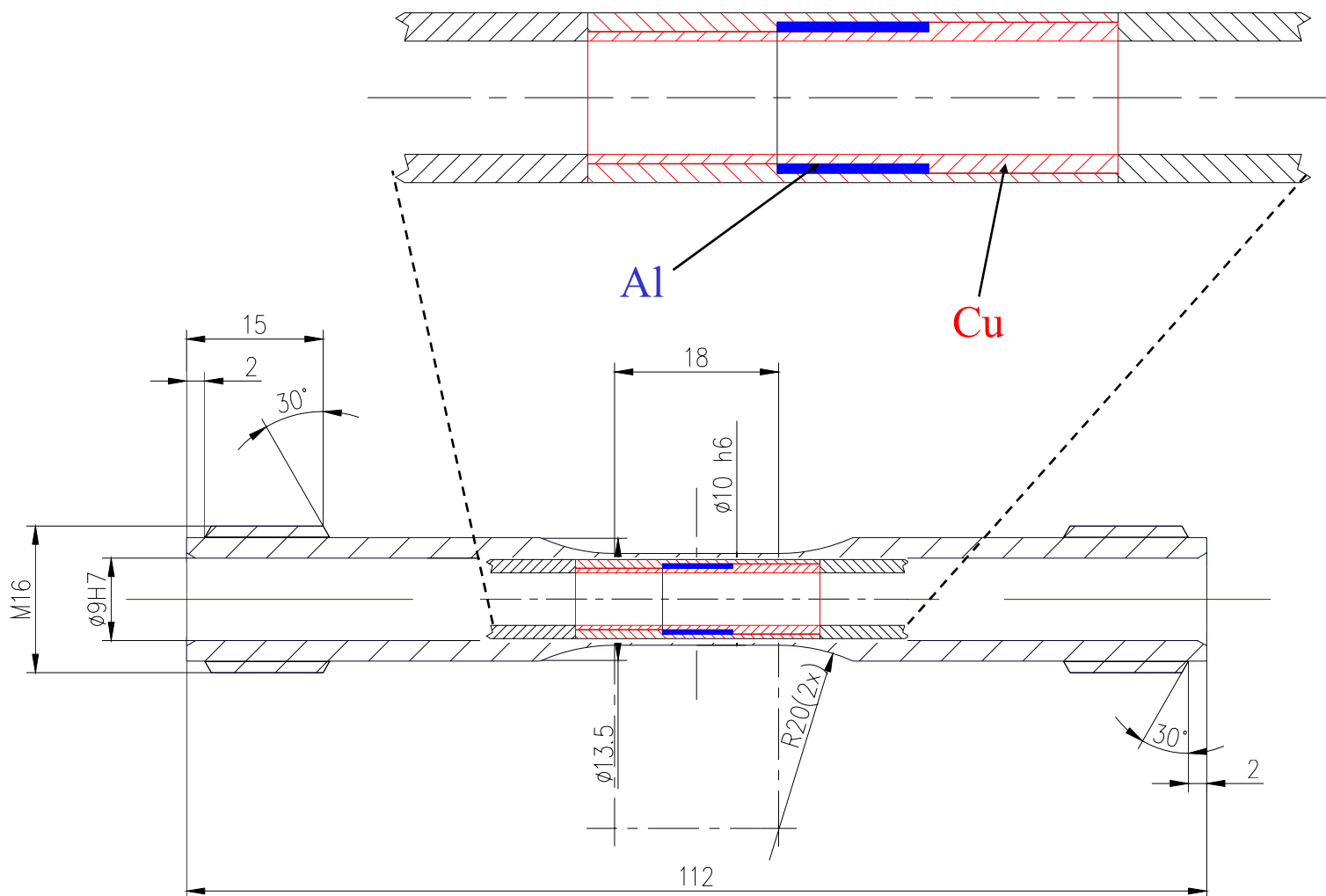
Pearlitic ( bcc Fe-C / Fe<sub>3</sub>C ) -> austenitic ( fcc Fe-C )

Phase change on cooling  $\sim 680^\circ\text{C}$  (under-cooling  $> 50^\circ\text{C}$ )





## Al ring soldered into Cu case inserted into tubular specimen



eutectic point

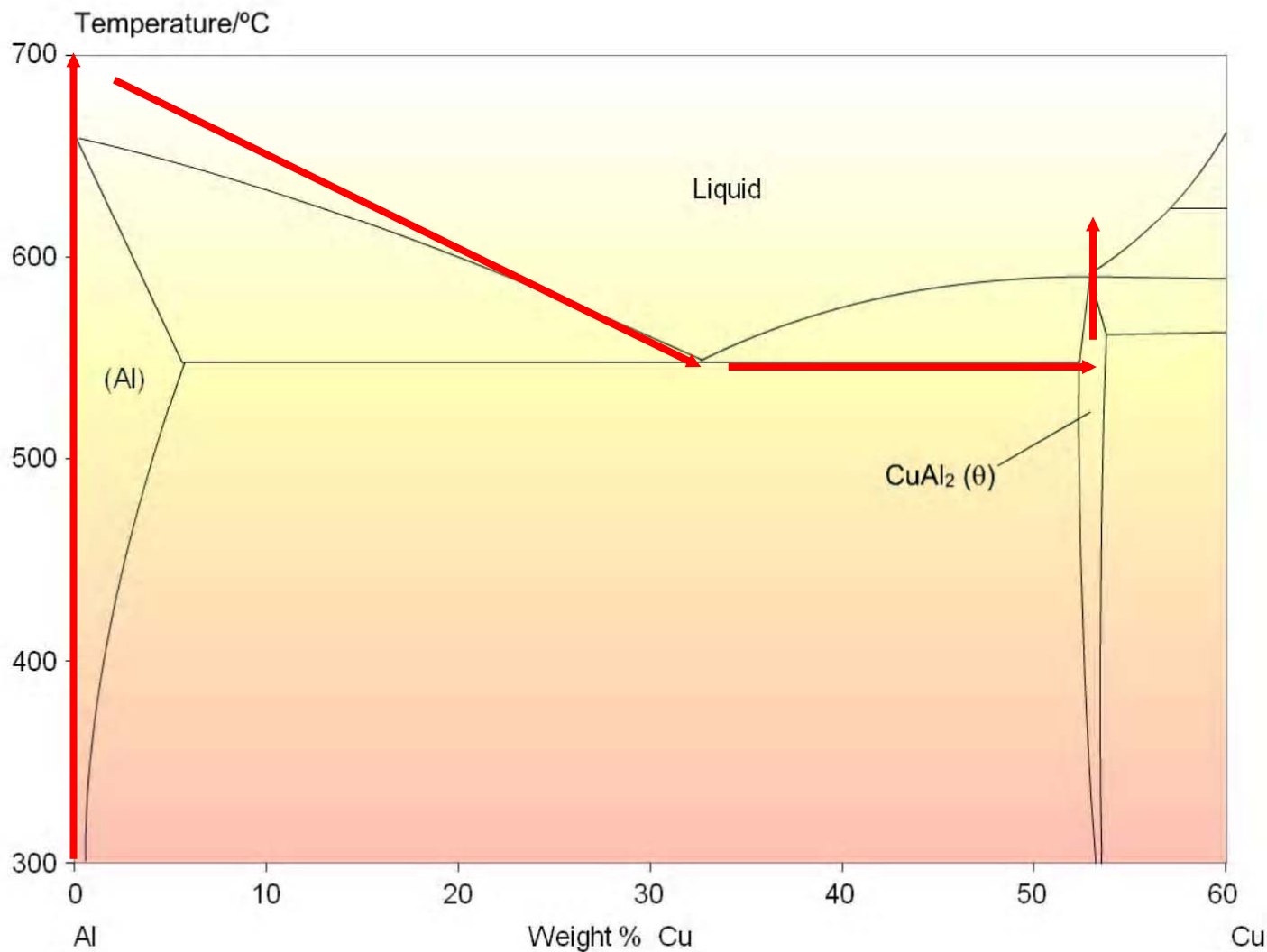
@

548°C, 33 wt% Cu

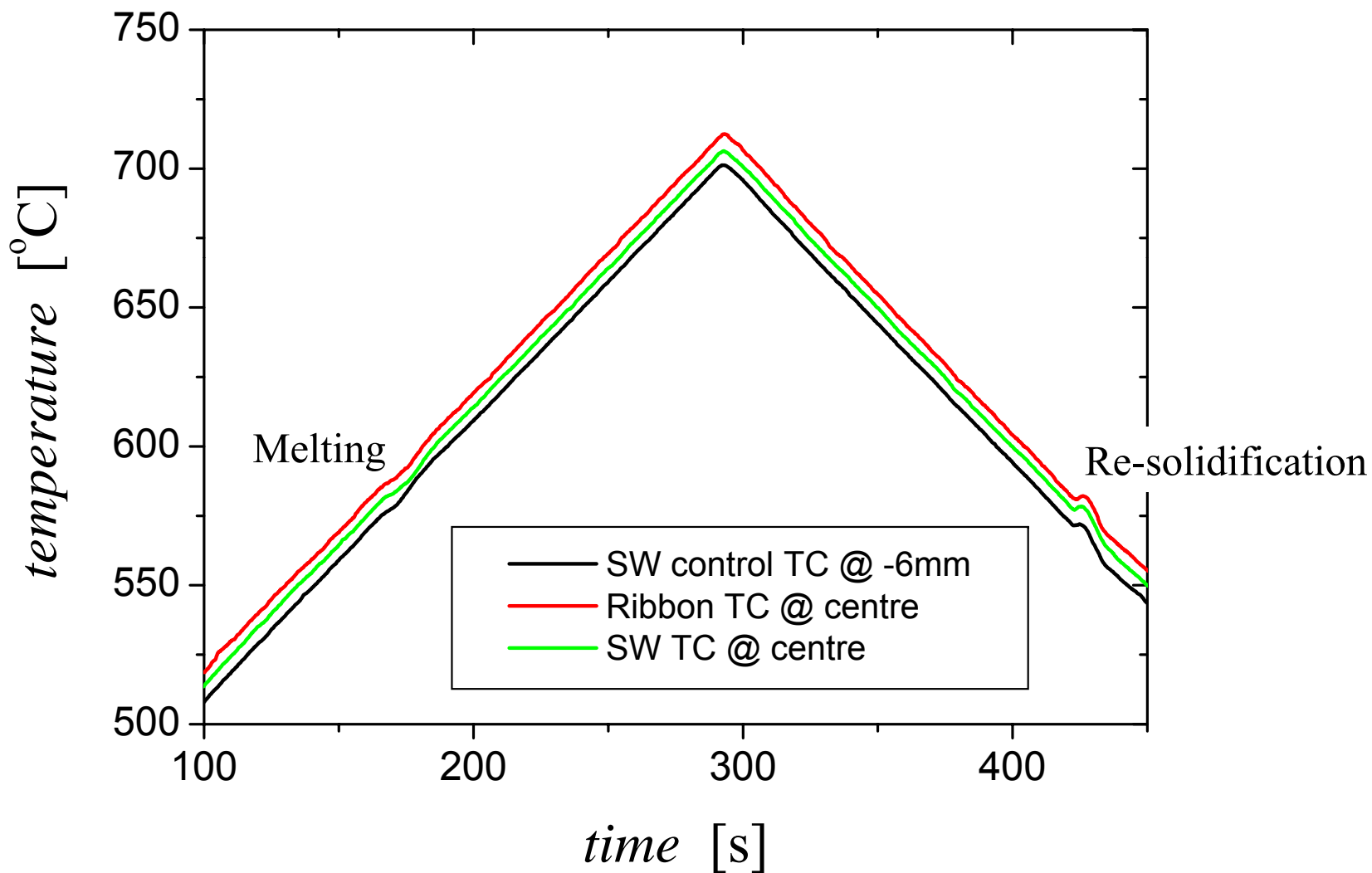
$\theta$  phase ( $\text{CuAl}_2$ )

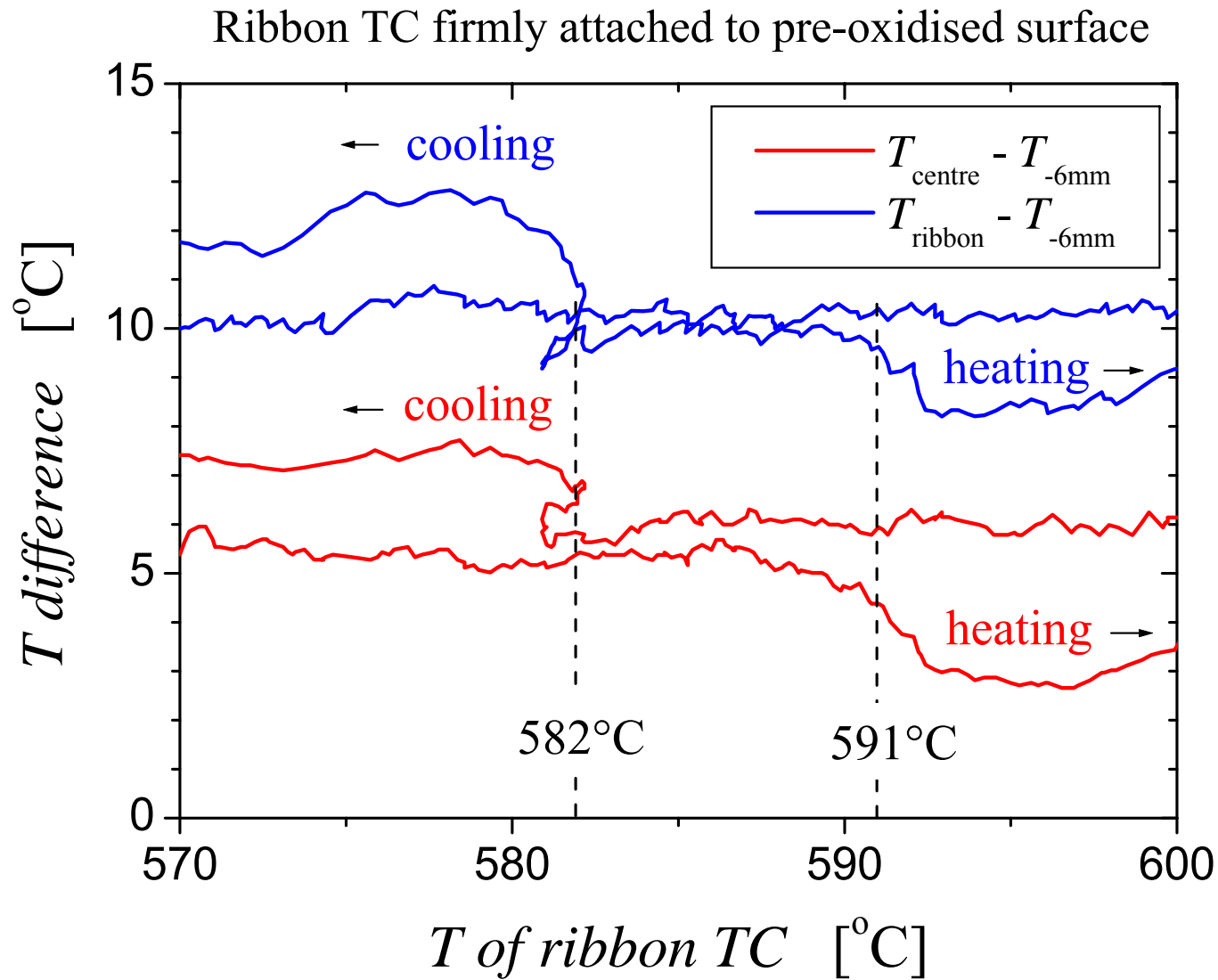
@

591°C, 53 wt% Cu



Ribbon TC firmly attached to pre-oxidised surface





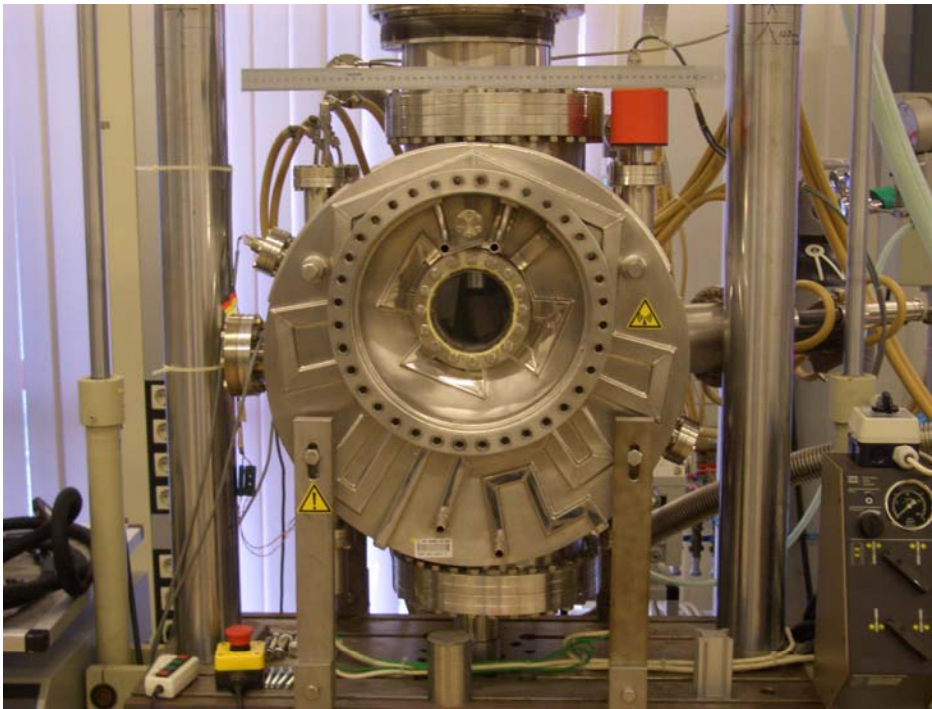
## ***Advantages of Al-Cu system***

- No need to alloy intermetallic  $\theta$  phase with 53 wt% Cu
- Detectable  $\theta$  phase develops already during 1<sup>st</sup> cycle
- Sharp solid-liquid phase transition upon heating
- Less hysteresis (under-cooling  $< 10^{\circ}\text{C}$ )
- $> 50$  cycles possible
- Not ferromagnetic

(cf. induction heating circuit affected by permeability)

TMF performance in

- vacuum/inert atmosphere
- corrosive environments
- hydrogen



Setups for low pressure (left) and high pressure (right) environmental LCF/TMF tests