



Claudia Draxl

Augmented - planewave methods

Kohn-Sham equation

$$[-\nabla^2 + V_{eff}(\mathbf{r})] \Psi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \Psi_{n\mathbf{k}}(\mathbf{r})$$

❖ Ritz variational principle

$$\Psi_{n\mathbf{k}}(\mathbf{r}) = \sum_{\mathbf{G}} C_{n\mathbf{k}}(\mathbf{G}) \phi_{\mathbf{k}+\mathbf{G}}(\mathbf{r})$$

❖ Matrix form

$$\sum_{\mathbf{G}'} (H_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'} - \varepsilon_{n\mathbf{k}} S_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'}) C_{n\mathbf{k}}(\mathbf{G}') = 0$$

Kohn-Sham equation

$$\sum_{\mathbf{G}'} (H_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'} - \varepsilon_{n\mathbf{k}} S_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'}) C_{n\mathbf{k}}(\mathbf{G}') = 0$$

★ Hamiltonian matrix

$$H_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'} \equiv \langle \phi_{\mathbf{k}+\mathbf{G}} | -\nabla^2 + V_{eff} | \phi_{\mathbf{k}+\mathbf{G}'} \rangle_{\Omega}$$

★ Overlap matrix

$$S_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'} \equiv \langle \phi_{\mathbf{k}+\mathbf{G}} | \phi_{\mathbf{k}+\mathbf{G}'} \rangle_{\Omega}$$

Augmented-planewave methods

- ★ APW
- ★ LAPW
- ★ SLAPW(3), SLAPW(4), ...
- ★ LAPW+LO
- ★ APW+lo
- ★ APW+lo+LO



"**LAPW**"

All these methods linearize
the eigenvalue problem

Dual basis for WF, density, potential, ...

- ❖ **Atomic spheres α, β**

Atomic-like basis functions

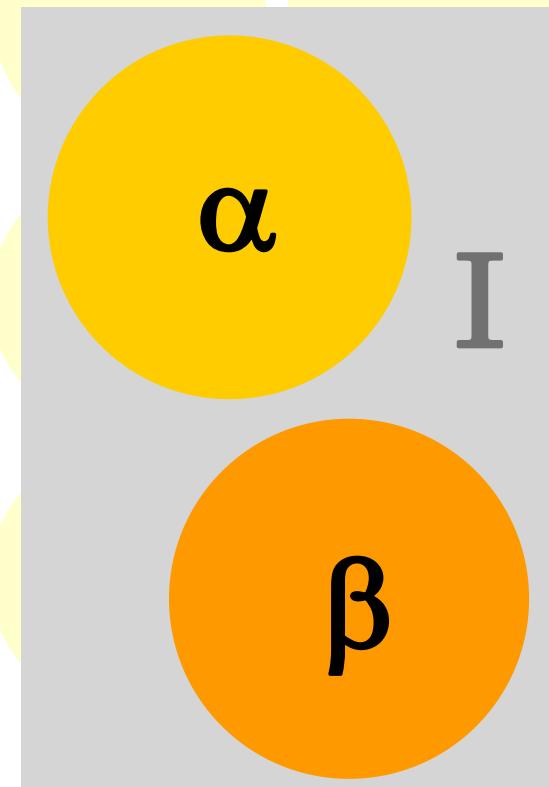
- ❖ **Interstitial**

Planewave basis

- ❖ **All-electron method**

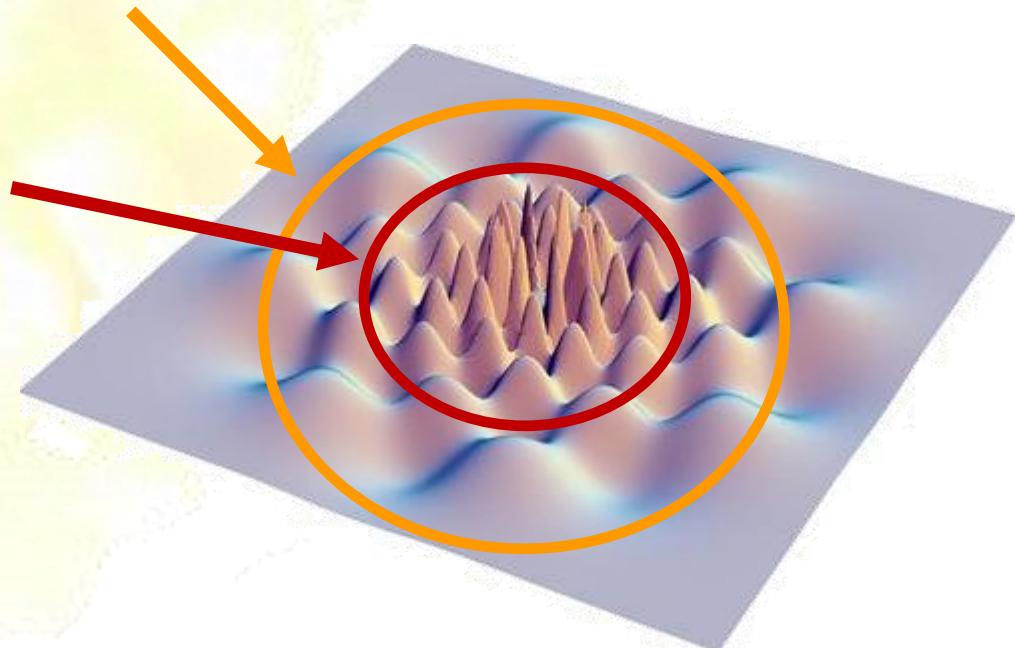
Can handle strong variations

Can explore the core region



What determines the size of the basis?

- Each planewave living in the interstitial is augmented by atomic-like functions inside each sphere
- Large muffin-tin sphere
few planewaves
- Small sphere size
many planewaves
- The product $R_{MT}G_{max}$
is a good measure for a converged basis



Potential and density: dual basis

★ The potential

$$V_{eff}(\mathbf{r}) = \sum_{\mathbf{K}} V_{\mathbf{K}} e^{i\mathbf{Kr}}$$

I

α

$$V_{eff}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{LM} V_{LM}^\alpha(r) Y_{LM}(\hat{\mathbf{r}})$$

★ The density

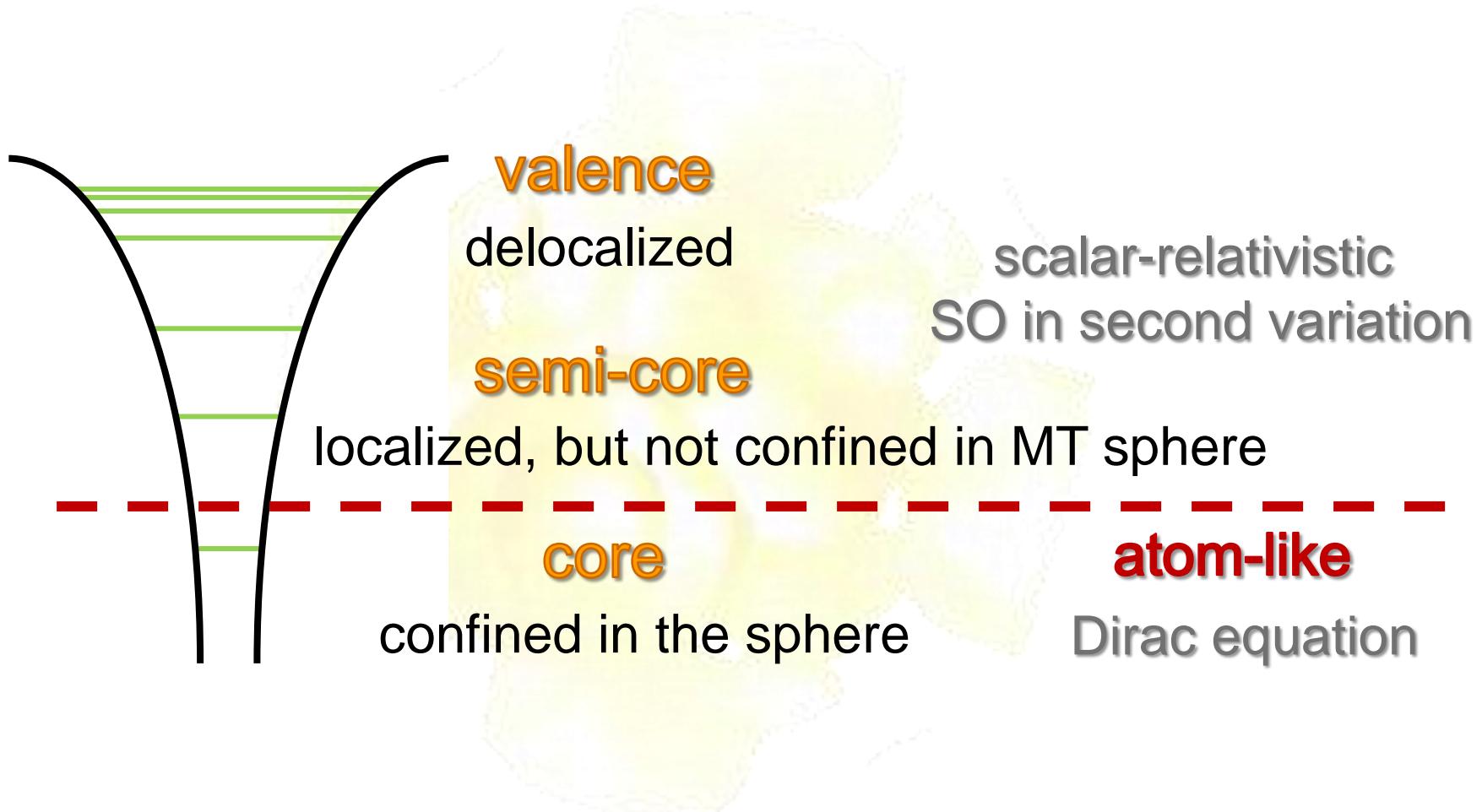
$$\rho_{eff}(\mathbf{r}) = \sum_{\mathbf{K}} \rho_{\mathbf{K}} e^{i\mathbf{Kr}}$$

I

α

$$\rho_{eff}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{LM} \rho_{LM}^\alpha(r) Y_{LM}(\hat{\mathbf{r}})$$

All-electron description



The APW basis set

J. C. Slater, Phys. Rev. 51, 834 (1937).
Adv. Quant. Chem. 1, 35 (1964).

Augmented Plane Waves

I

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{r}) = \frac{1}{\sqrt{\Omega}} e^{i(\mathbf{k}+\mathbf{G})\mathbf{r}}$$

basis continuous at sphere boundary R_{MT}

α

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{lm} A_{lm}^\alpha(\mathbf{k}+\mathbf{G}) u_l^\alpha(r, E) Y_{lm}(\hat{\mathbf{r}})$$

solutions of the radial Schrödinger equation

$$\left\{ -\frac{d^2}{dr^2} + \frac{l(l+1)}{r^2} + V^\alpha(r) - E \right\} r u_l^\alpha(r, E) = 0$$

APW: advantages & drawbacks

- ★ E has to be the exact KS eigenvalue
 E is a variational parameter
- ★ Energy-dependent basis set
- ★ Non-linear eigenvalue problem
- ★ Search for zeros of the determinant
 - No single diagonalization
 - Time-consuming
- ★ No full-potential method
- ★ **BUT** a true all-electron method!



The LAPW basis set

★ Linearized Augmented Plane Waves

O. K. Andersen, Phys. Rev. B 12, 3060 (1975).
D. D. Kölking and G. O. Arbman, J. Phys. F 5, 2041 (1975).

I

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{r}) = \frac{1}{\sqrt{\Omega}} e^{i(\mathbf{k}+\mathbf{G})\mathbf{r}}$$

α

basis continuous in value and slope at R_{MT}

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{lm} [A_{lm}^\alpha(\mathbf{k} + \mathbf{G}) u_l^\alpha(r, E_l) Y_{lm}(\hat{\mathbf{r}})]$$

energy parameter

energy derivative

LAPW: advantages & drawbacks

- ★ E_i is a fixed parameter
- ★ Energy-independent basis set
- ★ Linear eigenvalue problem



$$\sum_{\mathbf{G}'} (H_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'} - \varepsilon_{n\mathbf{k}} S_{\mathbf{k}+\mathbf{G}, \mathbf{k}+\mathbf{G}'}) C_{n\mathbf{k}}(\mathbf{G}') = 0$$

- ★ No true all-electron method!
Only one principle quantum number per l



The problem

- ★ Example: Cu: $3p^6 3d^{10} 4s^1$
 - 3p states not confined in the muffin-tin sphere
 - 4p states needed to have a flexible basis set

Ways out

- ★ Two-window calculation
 - Orthogonality problems
- ★ A better basis set
 - Super-LAPW (SLAPW)
 - or better

SLAPW Method

★ SLAPW-3

α

basis continuous up to 2nd derivative

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{lm} [A_{lm}^\alpha(\mathbf{k}+\mathbf{G})u_l^\alpha(r, E_l) + B_{lm}^\alpha(\mathbf{k}+\mathbf{G})\dot{u}_l^\alpha(r, E_l) + C_{lm}^\alpha(\mathbf{k}+\mathbf{G})u_l^\alpha(r, E_{lo})]Y_{lm}(\hat{\mathbf{r}})$$

★ SLAPW-4

different energy parameters

★

SLAPW: Advantages & drawbacks

- ★ All-electron method
- ★ Kinetic energy continuos
No surface terms needed



- ★ Costly: more basis functions needed
Roughly 50% larger basis set
Why?

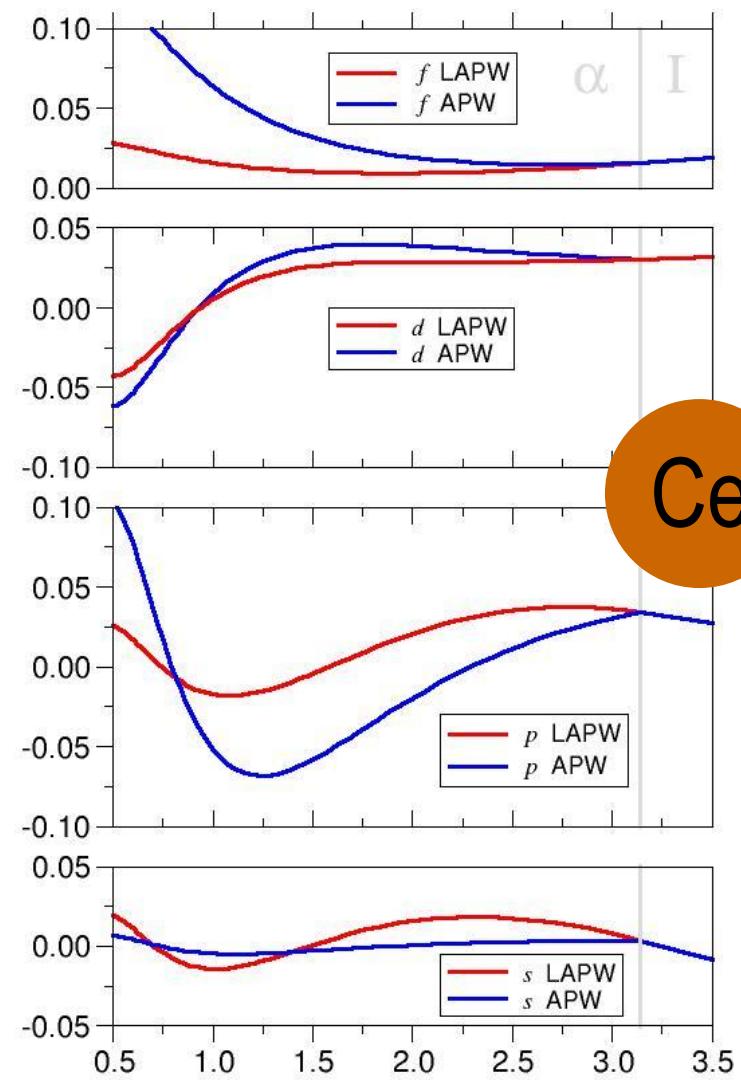


The radial functions in APW & LAPW

💡 They get distorted by boundary conditions

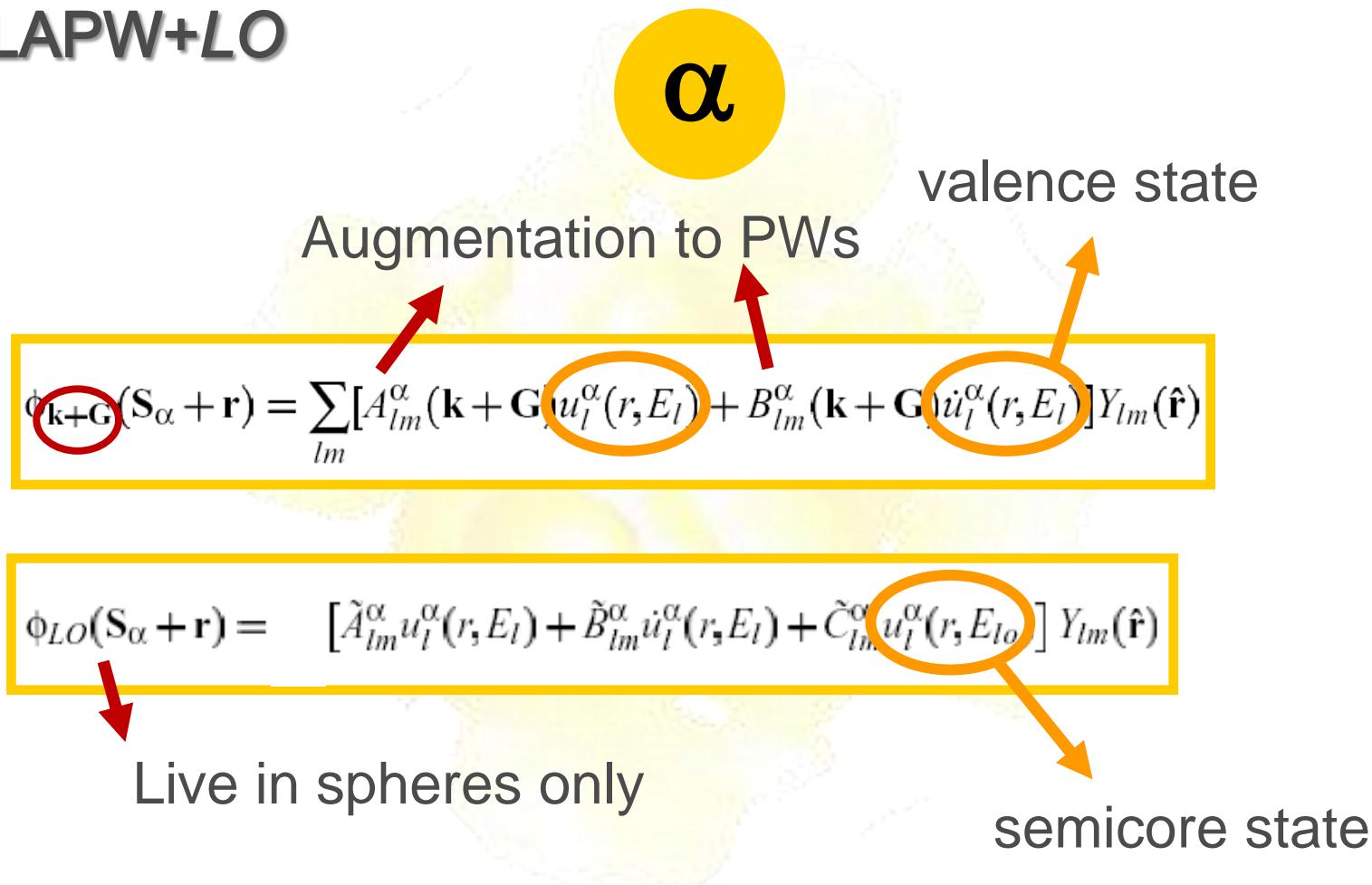
More basis functions needed in LAPW to describe the same behavior as in APW

💡 Same effect in SLAPW
Even more pronounced



The concept of local orbitals

★ LAPW+LO



Nearly no extra cost!

D. J. Singh, Phys. Rev. B 43, 6388 (1991).

Can we do better?

💡 ... *Is it possible to combine the advantages of the LAPW and the APW methods, i.e., to find an energy-independent basis that does not demand a noticeable higher planewave cutoff than the original APW functions? ...*

💡 **YES !!**

E. Sjöstedt, L. Nordström, and D. J. Singh, Solid State Commun. 114, 15 (2000).

The APW+lo basis set

💡 Forget classic LAPW

α

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{lm} [A_{lm}^\alpha(\mathbf{k}+\mathbf{G}) u_l^\alpha(r, E_l) + B_{lm}^\alpha(\mathbf{k}+\mathbf{G}) \dot{u}_l^\alpha(r, E_l)] Y_{lm}(\hat{\mathbf{r}})$$

💡 Use APW & *local orbital* at the same energy

$$\phi_{\mathbf{k}+\mathbf{G}}(\mathbf{S}_\alpha + \mathbf{r}) = \sum_{lm} A_{lm}^\alpha(\mathbf{k}+\mathbf{G}) u_l^\alpha(r, E_l) Y_{lm}(\hat{\mathbf{r}})$$

&

$$\phi_{lo}(\mathbf{S}_\alpha + \mathbf{r}) = [\tilde{A}_{lm}^\alpha u_l^\alpha(r, E_l) + \tilde{C}_{lm}^\alpha \dot{u}_l^\alpha(r, E_l)] Y_{lm}(\hat{\mathbf{r}})$$

fixed

Alternative way of linearization

APW+lo: only advantages - NO drawbacks

- ★ All-electron method
- ★ As accurate as LAPW
- ★ Can be supplemented by LOs

- ★ Very efficient

50% less basis functions compared to LAPW

Saves a factor of ~5 for large cells

- ★ Different augmentations can be combined

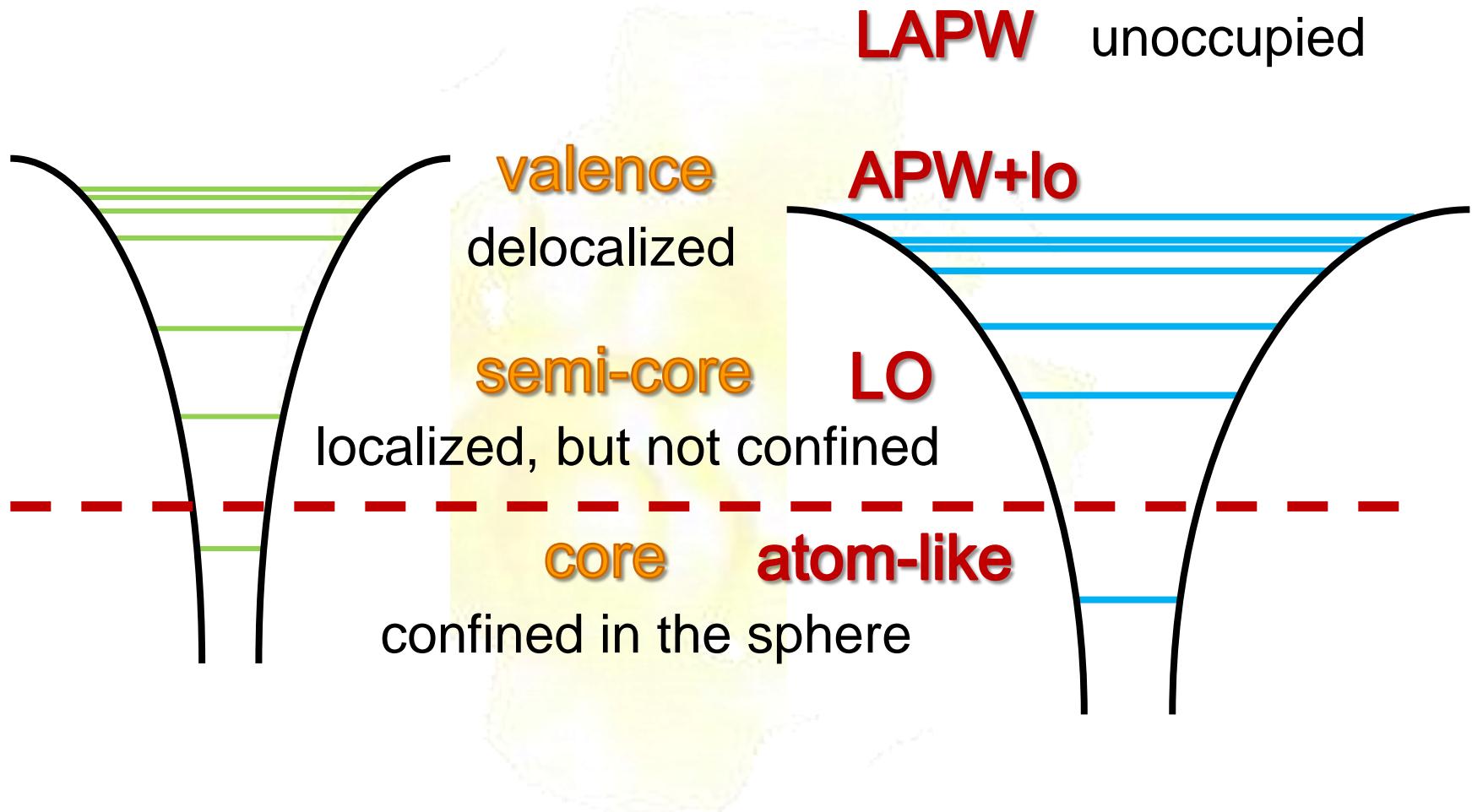
APW+lo for *relevant* valence states only

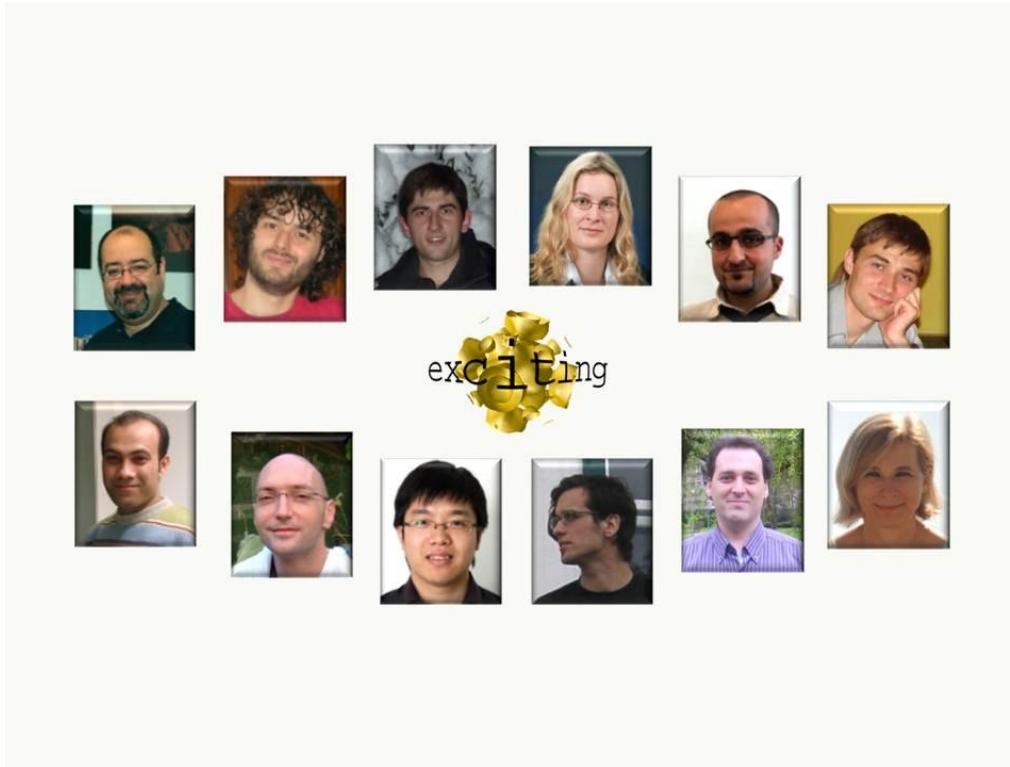
LAPW-like augmentation for all other / states



G. Madsen et al., PRB 64, 195134 (2001)

Our LAPW flavor of choice





<http://exciting-code.org>

Most general implementation in exciting

★ Species generator

All types of augmentation possible

Allows for an individual setup

Default basis: APW+lo

★ Species file for each atom

Written in XML

Defines

R_{MT} , radial mesh

core / semicore / valence
matching order

Features

★ State-of-the-art ground-state package

Band structure, DOS

Structure optimization

Various functionals (& libxc), including EXX

Phonons (supercell approach)

Thermodynamic properties

Link to cluster expansion: ATAT@exciting

★ Excitation spectra

Time-dependent DFT

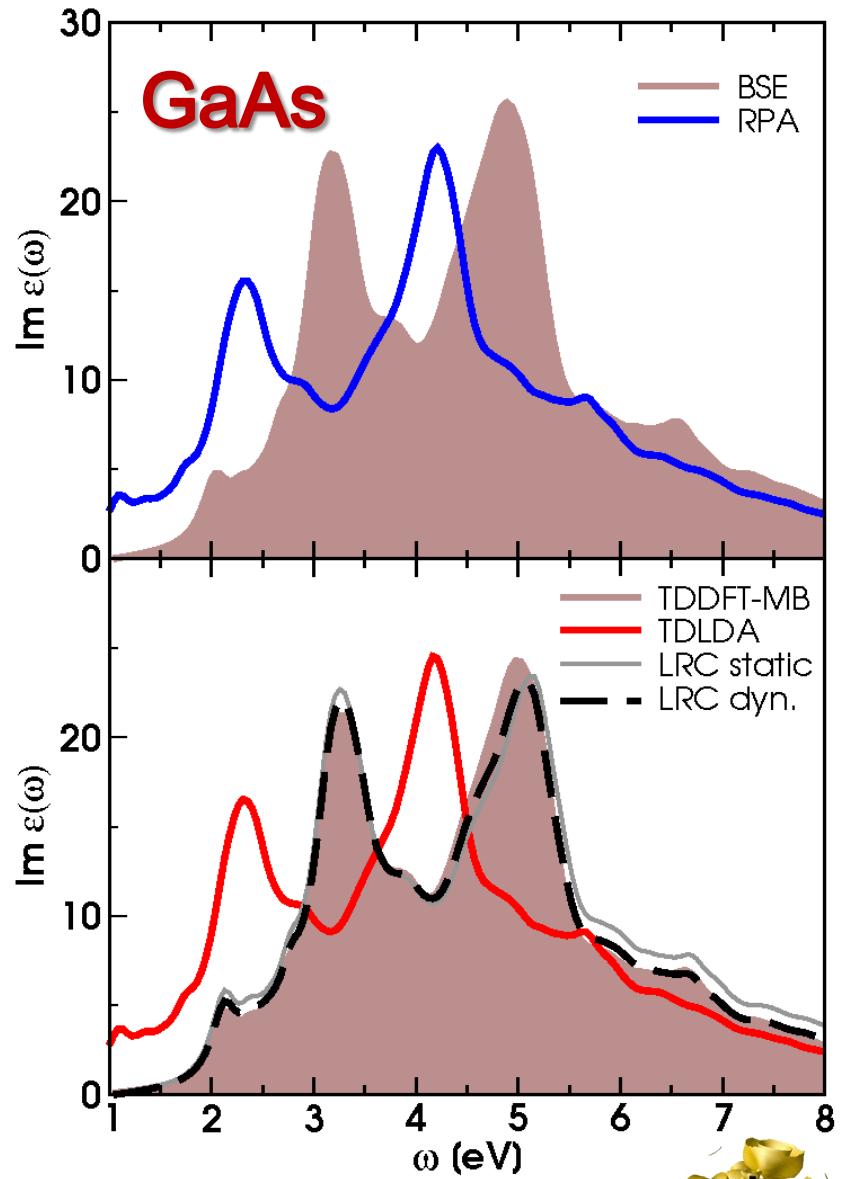
Many-body perturbation theory

GW & Bethe Salpeter equation (BSE)



Excitation spectra

- ★ BSE & TDDFT
 - For Solids
- ★ Allows comparison on equal footing
 - Same KS states, potential, density, matrix elements, ...

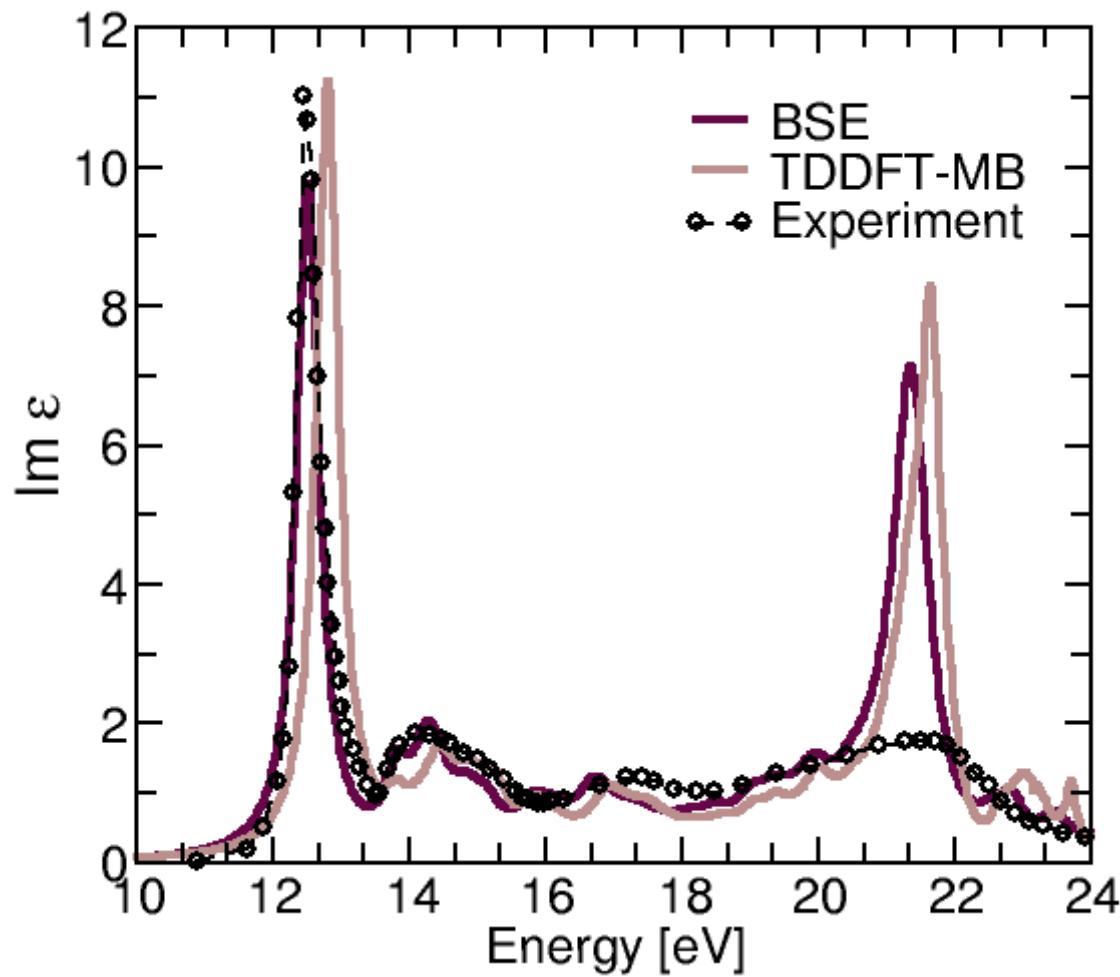


S. Sagmeister and CAD, PCCP 11, 4451 (2009).



Excitation spectra

LiF



Why all-electron methods?

⚠ A striking example

Results with the *FHI-gap* code: ***GW@WIEN2k***

Strongly improved version implemented in ***exciting***

R. Gómez-Abal, X. Li, M. Scheffler, and CAD, PRL 101, 106404 (2008).

X. Li, R. Gómez-Abal, H. Jiang, CAD, and M. Scheffler, New J. Phys. 14, 023006 (2012).

The G_0W_0 @LDA gap of Si

PP

1.19 – 1.29 eV

PAW

1.00 eV

AE- G_0W_0

0.85 eV - 0.95 eV

PP- G_0W_0 *all-electron like*

1.04 eV

LAPW

1.00 – 1.05

M. S. Hybertsen and S. G. Louie,

B. Arnaud and M. Alouani, PRB 62, 4464 (2000).

W. Ku and A. G. Eguiluz, PRL 89, 126401 (2002).

T. Kotani and M. van Schilfgaarde,
Solid State Comm. 121, 461 (2002).

M. L. Tiago, S. Ismail-Beigi,
and S. G. Louie, PRB 69, 125212 (2004).

C. Friedrich, A. Schindlmayr, S. Blügel, and
T. Kotani, PRB 74, 045104 (2006).

R. Gomez-Abal, X. Z. Li, M. Scheffer,
and C. Ambrosch-Draxl, PRL 101, 106404 (2008).



The manybody concept

★ Quasiparticle equation

$$\left[T + V_{ext}(\mathbf{r}) + V_H(\mathbf{r}) \right] \psi_i^{QP}(\mathbf{r}) + \int \boxed{\Sigma(\mathbf{r}, \mathbf{r}', \epsilon_i)} \psi_i^{QP}(\mathbf{r}') d^3\mathbf{r}' = \epsilon_i^{QP} \psi_i^{QP}(\mathbf{r})$$

★ Kohn Sham equation

$$\left[T + V_{ext}(\mathbf{r}) + V_H(\mathbf{r}) + \boxed{V_{xc}(\mathbf{r})} \right] \psi_i^{KS}(\mathbf{r}) = \epsilon_i^{KS} \psi_i^{KS}(\mathbf{r})$$

★ G_0W_0

$$\epsilon_{n\mathbf{k}}^{QP} = \epsilon_{n\mathbf{k}}^{KS} + \left\langle n\mathbf{k} \left| \Sigma(\epsilon_{n\mathbf{k}}^{QP}) - V_{xc}^{DFT} \right| n\mathbf{k} \right\rangle$$

LDA



The G_0W_0 @LDA band gap of Si ...

★ Why the big difference?

**Cancelation between vertex corrections
& lack of self-consistency?**

W. Ku and A. G. Eguiluz, PRL 89, 126401 (2002).

**Cancelation between missing core electrons
& lack of self-consistency?**

K. Delaney, P. Garca-Gonzalez, A. Rubio, P. Rinke, and R.W.Godby,
PRL 93, 249701(2004).

Bad convergence with unoccupied states?

★ Remaining discrepancy small: 0.1 eV

★ Materials with semicore states more problematic

Energy correction

$$\epsilon_{n\mathbf{k}}^{\text{qp}} = \epsilon_{n\mathbf{k}}^{\text{KS}} + \langle \varphi_{n\mathbf{k}} | \Re \left[\Sigma(\mathbf{r}, \mathbf{r}'; \epsilon_{n\mathbf{k}}^{\text{qp}}) \right] - V^{\text{xc}}(\mathbf{r}) | \varphi_{n\mathbf{k}} \rangle$$

Exact $\Sigma(\{\varphi_{n\mathbf{k}}, \varphi_{\text{core}}\}, \epsilon_{n\mathbf{k}}^{\text{qp}})$ $V^{\text{xc}}(n_{\text{val}} + n_{\text{core}})$

Pseudo-potential approximation

Pseudoization of potential and KS wavefunctions

Core-valence partitioning

$$V^{\text{xc}}[n(\mathbf{r})] = V^{\text{xc}}[n_{\text{core}}(\mathbf{r})] + V^{\text{xc}}[n_{\text{val}}(\mathbf{r})]$$

Frozen core

$$\Sigma(\{\tilde{\varphi}_{n\mathbf{k}}\}, \epsilon_{n\mathbf{k}}^{\text{qp}}) \quad V^{\text{xc}}(\tilde{n}_{\text{val}})$$

Three levels of approximation

>All electron (full potential)

$$\Delta\epsilon_{n\mathbf{k}}^{\text{AE}} = \Re \left(\langle \varphi_{n\mathbf{k}} | \Sigma \left(\{ \varphi_{n\mathbf{k}}, \varphi_{\text{core}} \}, \epsilon_{n\mathbf{k}}^{\text{qp}} \right) | \varphi_{n\mathbf{k}} \rangle \right) - \langle \varphi_{n\mathbf{k}} | V^{\text{xc}} (n_{\text{val}} + n_{\text{core}}) | \varphi_{n\mathbf{k}} \rangle$$

All electron – frozen core

$$\Delta\epsilon_{n\mathbf{k}}^{\text{AE-FC}} = \Re \left(\langle \varphi_{n\mathbf{k}} | \Sigma \left(\{ \varphi_{n\mathbf{k}}^{\text{FC}}, \varphi_{\text{core}}^{\text{FC}} \}, \epsilon_{n\mathbf{k}}^{\text{qp}} \right) | \varphi_{n\mathbf{k}} \rangle \right) - \langle \varphi_{n\mathbf{k}} | V^{\text{xc}} (n_{\text{val}}^{\text{FC}} + n_{\text{core}}^{\text{FC}}) | \varphi_{n\mathbf{k}} \rangle$$

All electron – valence

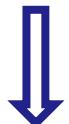
$$\Delta\epsilon_{n\mathbf{k}}^{\text{AE-V}} = \Re \left(\langle \varphi_{n\mathbf{k}} | \Sigma \left(\{ \varphi_{n\mathbf{k}}^{\text{FC}} \}, \epsilon_{n\mathbf{k}}^{\text{qp}} \right) | \varphi_{n\mathbf{k}} \rangle \right) - \langle \varphi_{n\mathbf{k}} | V^{\text{xc}} (n_{\text{val}}^{\text{FC}}) | \varphi_{n\mathbf{k}} \rangle$$

Pseudopotential

$$\Delta\epsilon_{n\mathbf{k}}^{\text{PP}} = \Re (\langle \tilde{\varphi}_{n\mathbf{k}} | \Sigma(\{ \tilde{\varphi}_{n\mathbf{k}} \}, \epsilon_{n\mathbf{k}}^{\text{qp}}) | \tilde{\varphi}_{n\mathbf{k}} \rangle) - \langle \tilde{\varphi}_{n\mathbf{k}} | V^{\text{xc}} (\tilde{n}_{\text{val}}) | \tilde{\varphi}_{n\mathbf{k}} \rangle$$

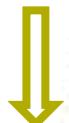
Three levels of approximation

⬇️ All electron



frozen-core approximation

⬇️ All electron – frozen core



core-valence partitioning

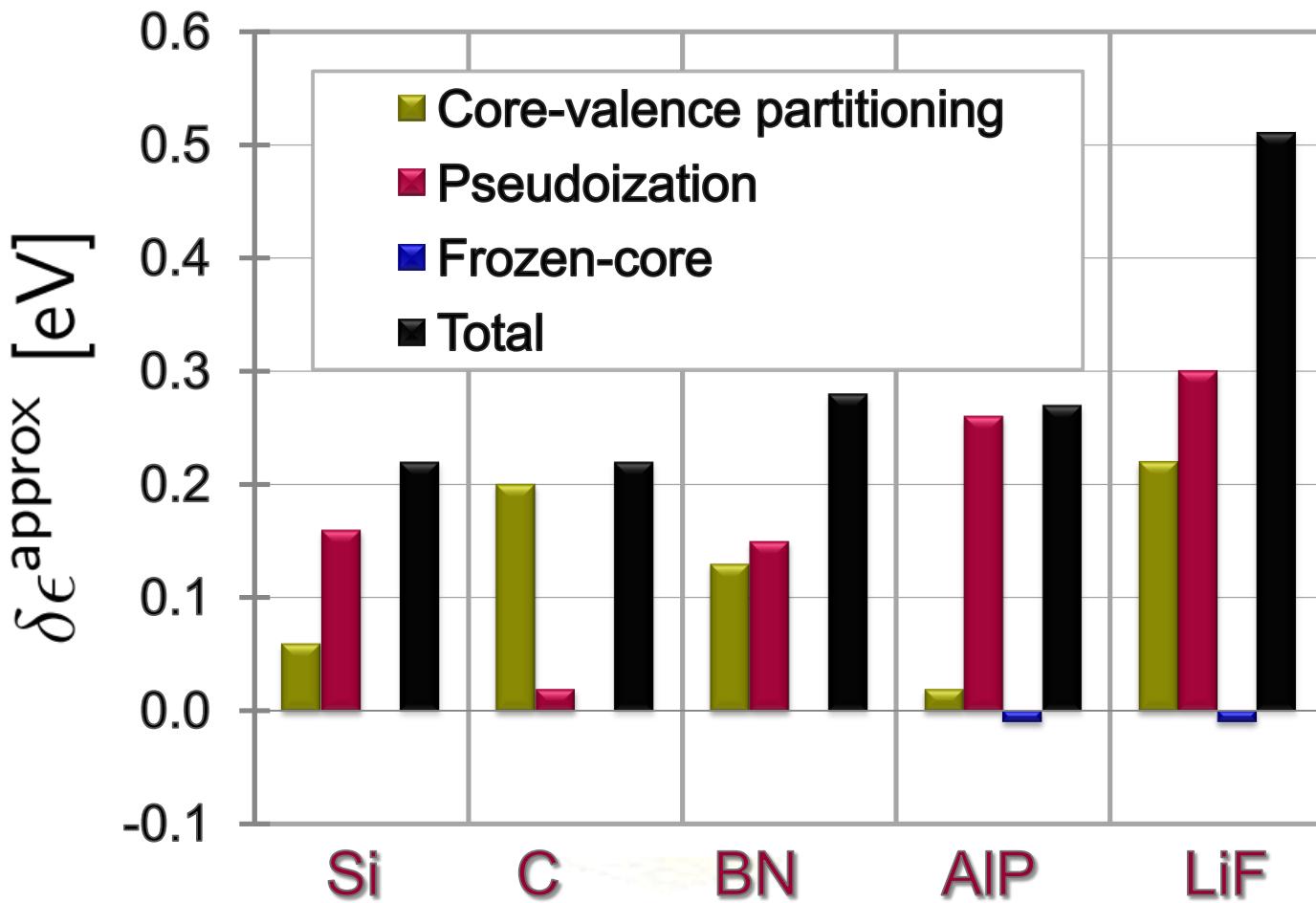
⬇️ All electron – valence



pseudoization

⬇️ Pseudopotential

sp semiconductors

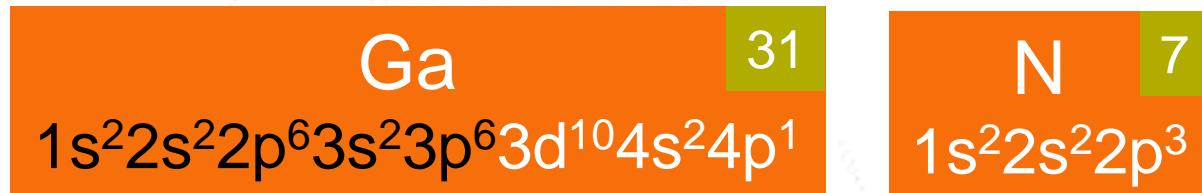


	$\epsilon_{\text{LDA}}^{\text{gap}}$	Σ^c	Σ^x	V^{xc}	$\Sigma^x - V^{xc}$	$\Delta\epsilon^{\text{gap}}$	$\epsilon_{G_0W_0}^{\text{gap}}$
AE							
Γ_v		1.42	-16.08	-14.07	-2.01	-0.59	
X_c		-3.96	-5.27	-9.40	4.13	0.17	
Δ	1.44	-5.38	10.81	4.67	6.14	0.76	2.20
AE-FC							
Γ_v		1.42	-16.10	-14.07	-2.03	-0.61	
X_c		-3.95	-5.31	-9.40	4.09	0.14	
Δ	1.44	-5.37	10.79	4.67	6.12	0.75	2.19
AE-V							
Γ_v		1.40	-14.08	-12.03	-2.05	-0.65	
X_c		-3.94	-4.48	-8.54	4.06	0.12	
Δ	1.44	-5.34	9.60	3.49	6.11	0.77	2.21
PP							
Γ		1.31	11.93	-11.82	-2.27	-0.93	
Δ	1.47	-5.16	9.52	3.33	6.19	1.03	2.50
core-valence partitioning							

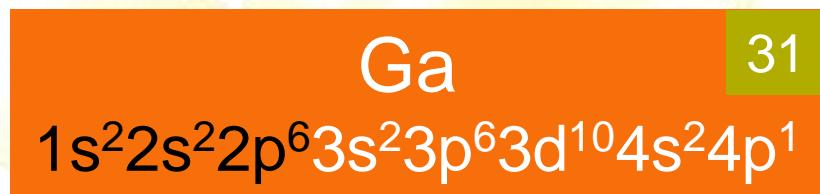


GaN: 3 configurations

⚠ AE-V-1



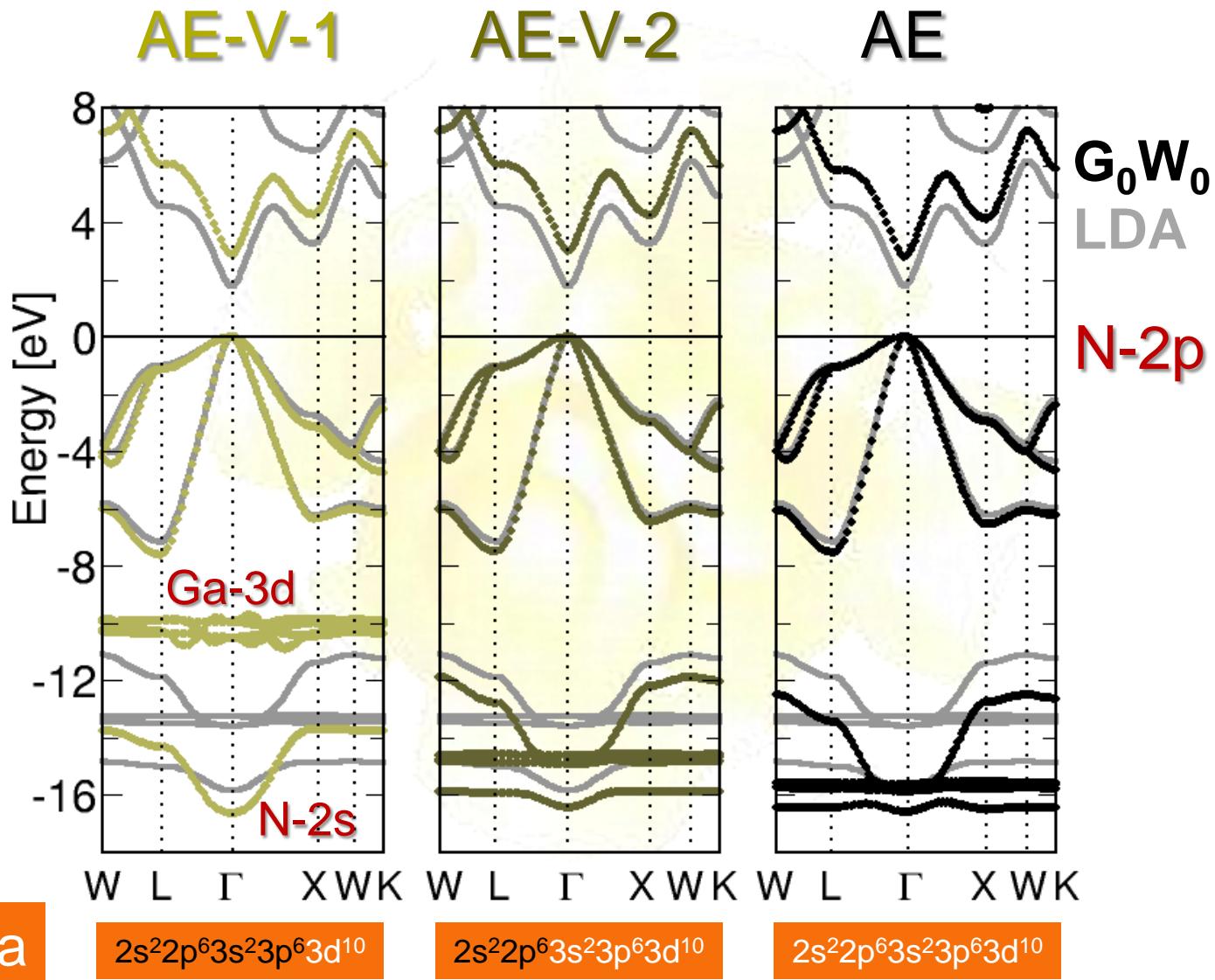
⚠ AE-V-2



⚠ AE



GaN: core-valence partitioning



Core-valence partitioning

	$\Delta\epsilon_{\text{LDA}}^{\text{VBM}}$	Σ^c	Σ^x	V^{xc}	$\Sigma^x - V^{xc}$	$\Delta\epsilon^{\text{corr}}$	$\Delta_{G_0 W_0}^{\text{VBM}}$
AE-V1	$2s^2 2p^6 3s^2 3p^6 3d^{10}$						
Γ_{VBM}		3.16	-21.60	-18.65	-2.95	0.21	
Γ_d		8.51	-38.28	-33.28	-5.00	3.51	
Δ	13.26	-5.35	16.68	14.63	2.05	-3.30	9.96
AE-V2	$2s^2 2p^6 3s^2 3p^6 3d^{10}$						
Γ_{VBM}		3.22	-22.62	-19.21	-3.41	-0.19	
Γ_d		10.23	-51.90	-40.11	-11.79	-1.56	
Δ	13.26	-7.01	29.28	20.90	8.38	1.37	14.63
AE	$2s^2 2p^6 3s^2 3p^6 3d^{10}$						
Γ_{VBM}		3.19	-23.88	-20.48	-3.40	-0.21	
Γ_d		10.50	-56.81	-43.76	-13.05	-2.55	
Δ	13.26	-7.31	32.93	23.28	9.65	2.34	15.60

All-electron G_0W_0 calculations reveal ...

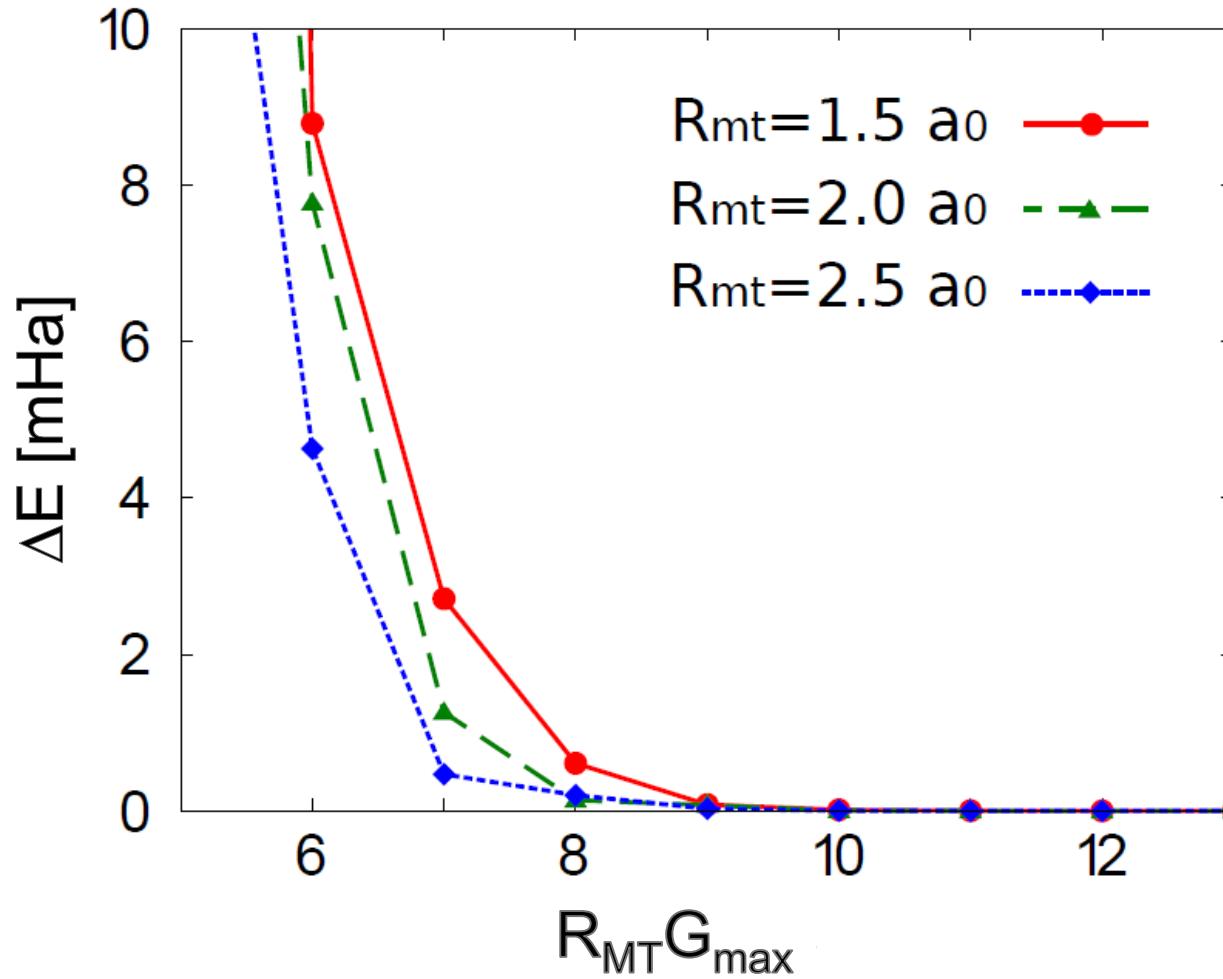
- ★ Errors in matrix elements can be of the order of several eV
- ★ Main problems in PP calculations due to
 - core-valence partitioning
 - pseudoization
- ★ Cancelation effects on different levels
 - Σ^x and V^{xc}
 - exchange and correlation
 - valence and conduction states
- ★ Cancelation incomplete for semicore
- ★ AE calculations needed as benchmark
 - but be aware of issues concerning unoccupied states!

LAPW and related methods ...

- ◆ are more complicated and more costly than many other methods
- ◆ but highly precise
 - ... if used properly!
- ◆ can handle any material
 - irrespective of the atoms involved
- ◆ allow for exploring the core region
- ◆ can be regarded as benchmark for other methods

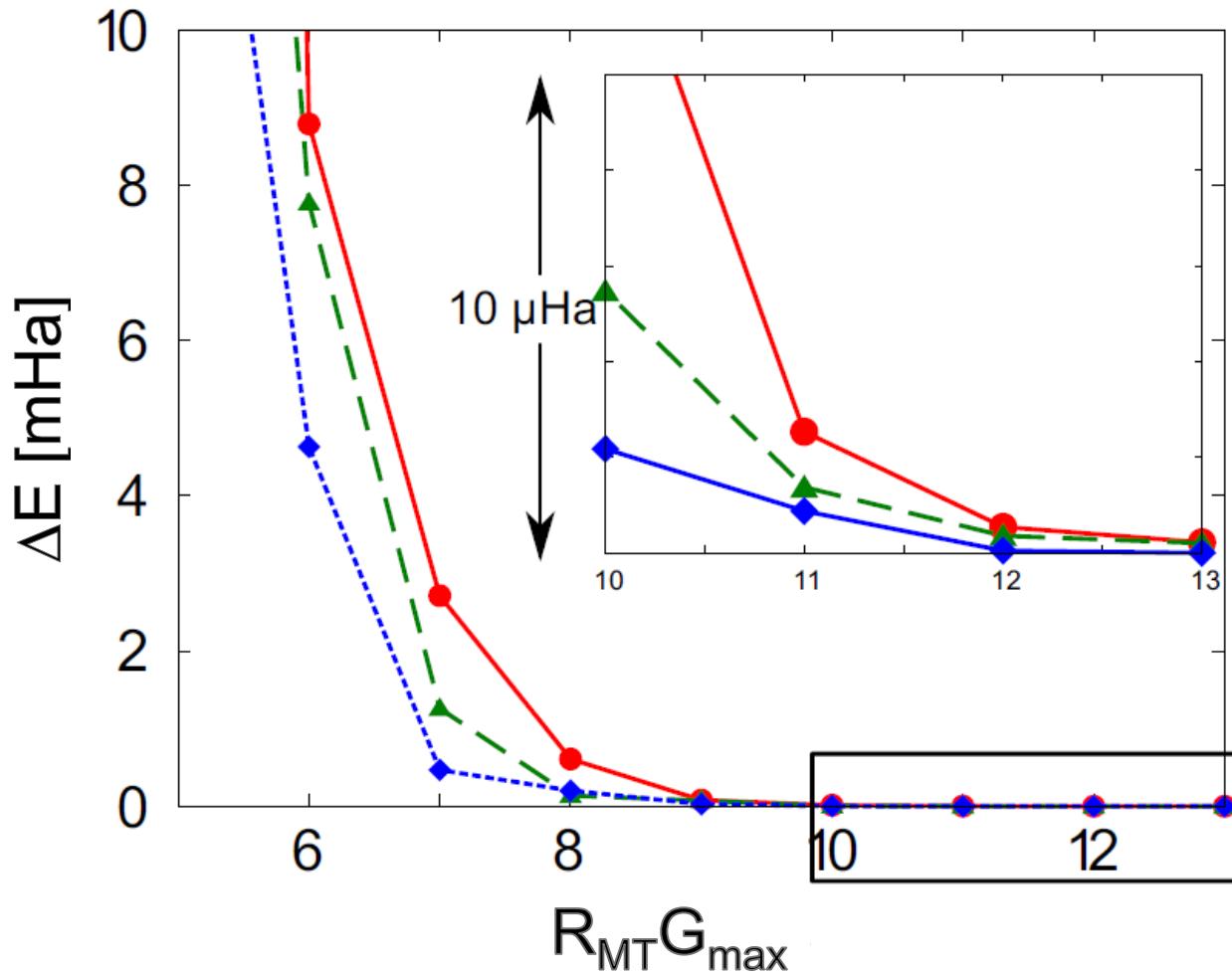


Example: Ar atom



★ Same holds for diamond, GaN, ...

Example: Ar atom



★ Same holds for diamond, GaN, ...

The exciting team

💡 in the excited state

Thanks for your attention!

