

Coarse-graining space and time: kinetic Monte Carlo simulations

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[arXiv:0904.2556](https://arxiv.org/abs/0904.2556)

Outline

How to harness the power of DFT to solve real-world problems, e.g
in materials science, chemical engineering,.. ?

Reliability \Leftrightarrow Relevance

numerical accuracy	length and time scales
(e.g. of energy barriers)	quality of statistics

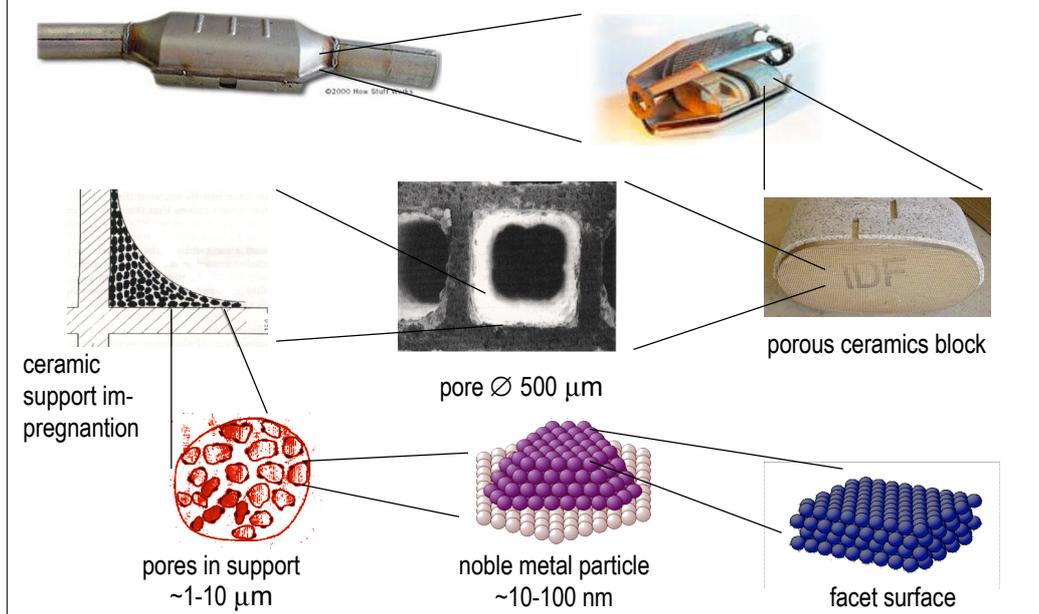
- **coarse-graining in space:** symbolic dynamics ('sites' instead of 'coordinates')
- **coarse-graining in time:** event-driven, variable-time step simulations

Non-equilibrium physics and chemistry of materials

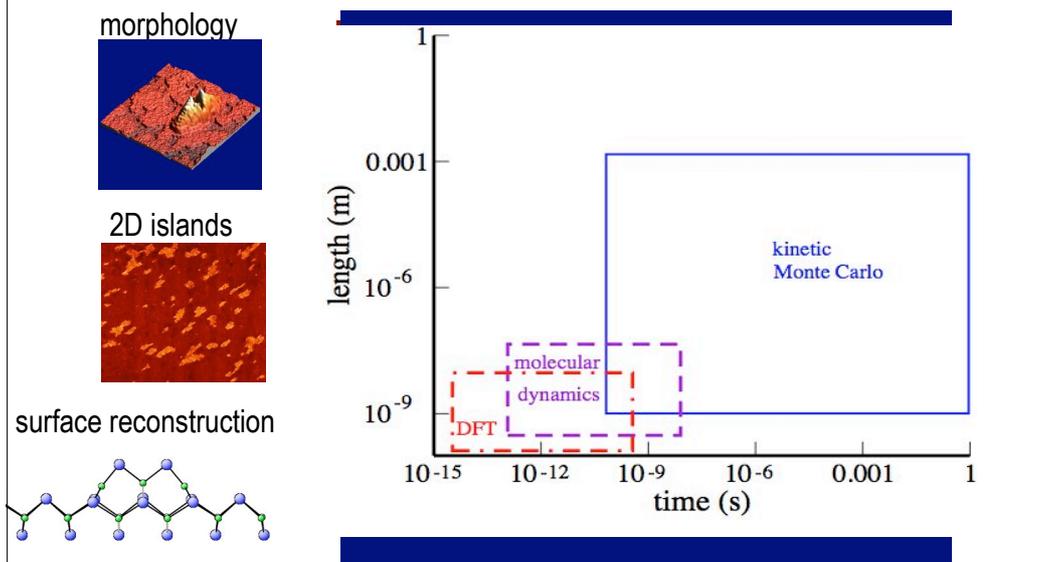
Examples:

- heterogeneous catalysis (fluctuations in a reactive environment)
- epitaxial growth of semiconductors (metastable nanostructured materials)

Example: car exhaust catalyst



Example: Time and length scales in epitaxy

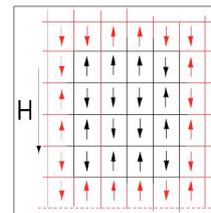


Methods of Statistical Physics (equilibrium)

Discrete models in Statistical Physics

- Ising model (magnetism)

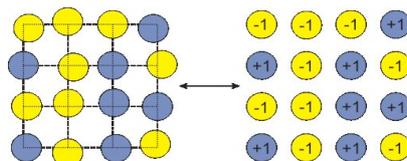
$$H(s) = -J_q \sum_i \sum_{j \in n(i)} s_i s_j - \mu_B B \sum_i s_i$$



- Lattice-gas interpretation $c_i = 0, 1$ $s_i = 2c_i - 1$

$$H = -4J_q \sum_i \sum_{j \in n(i)} c_i c_j + 2(qJ_q - \mu_B B) \sum_i c_i - N(qJ_q - \mu_B B)$$

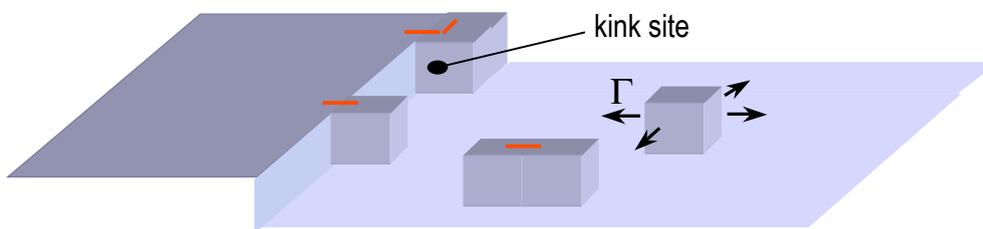
- **Goal:**
Calculation of thermal averages



A discrete model for epitaxy: solid-on-solid (SOS) model

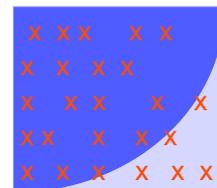
- Atoms are symbolized by little cubes placed on a lattice.
- The growth surface has no voids, no “overhangs”.
- Atoms move by discrete hops with rate $\Gamma = \exp(-E/kT)$.
- The binding energy is determined by the # of neighbors n

$$E = E_D + n E_B$$



Stochastic sampling

- Calculating the partition function $Z = \int \frac{d^{3N}q d^{3N}p}{N!(2\pi\hbar)^{3N}} \exp\left(-\frac{H(\mathbf{q}, \mathbf{p})}{k_B T}\right)$ in many-particles systems requires evaluation of **high-dimensional integrals**.
- Choosing the sampling points in an (almost) **random** way is a good strategy, in particular in high dimensions !
- Even better: **importance sampling** -- density of sampling points proportional to local value of the integrand
- Idea: create a **stochastic process** that achieves importance sampling.
- Application: **Metropolis algorithm** (this afternoon)

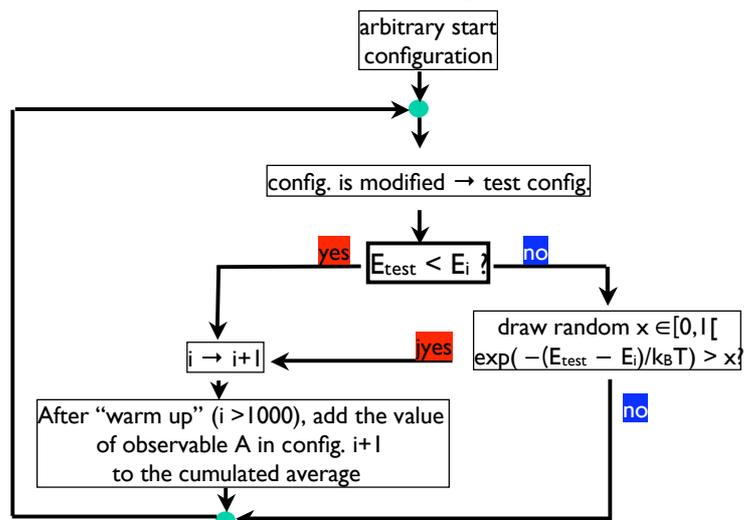


$$\pi/4 = 0.78 \dots \approx 20/25 = 0.8$$

Metropolis Sampling

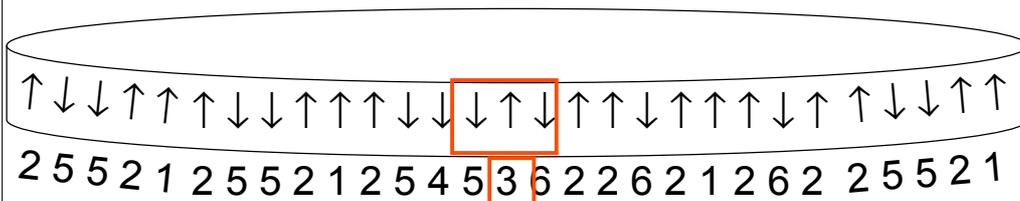
- **Solution:** Importance Sampling with $w(\mathbf{q}) = \frac{\exp(-V(\mathbf{q})/(k_B T))}{Z'}$
- Generate random support points, distributed according to $w(\mathbf{q})$, i.e., out of total K points, $k_i = Kw(\mathbf{q})$ in the unit volume around \mathbf{q}_i
- The expectation value of an observable is calculated as $\langle A \rangle \approx \frac{1}{K} \sum_{i=1}^K k_i A(\mathbf{q}_i)$
- The Metropolis algorithm generates, starting from \mathbf{q}_0 , successively a sequence of K configurations \mathbf{q}_i , distributed according to $w(\mathbf{q})$.
- Even so we don't know Z' , this is possible, because it is just the correct **relative** probabilities that matter:
 - accept new config. \mathbf{q}_{i+1} , if $\exp\left(-\frac{V(\mathbf{q}_{i+1}) - V(\mathbf{q}_i)}{k_B T}\right) > \text{rnd}$
 - else reject. $\text{rnd} \in [0,1[$
- This assures that $\frac{w(\mathbf{q}_{i+1})}{w(\mathbf{q}_i)} = \exp\left(-\frac{V(\mathbf{q}_{i+1}) - V(\mathbf{q}_i)}{k_B T}\right)$

Metropolis algorithm



From MC to kMC: the N -fold way (non-equilibrium)

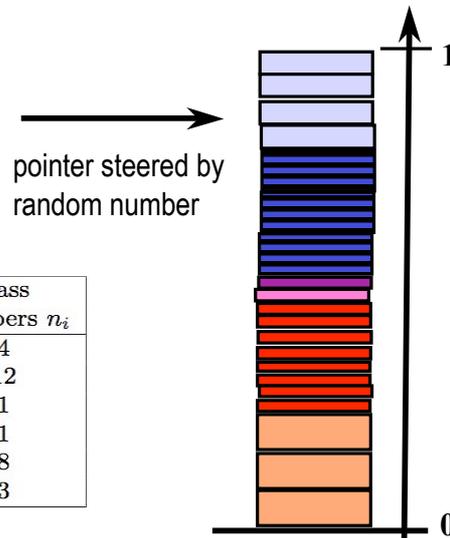
Classification of spins according to their neighborhood



class	central spin	neighbors	class members n_i
1	↑	↑, ↑	4
2	↑	↑, ↓	12
3	↑	↓, ↓	1
4	↓	↓, ↓	1
5	↓	↑, ↓	8
6	↓	↑, ↑	3

The N -fold way algorithm in MC

- processes are chosen with a probability proportional to their rates
- no discarded attempts (in contrast to Metropolis)



class	central spin	neighbors	class members n_i
1	↑	↑, ↑	4
2	↑	↑, ↓	12
3	↑	↓, ↓	1
4	↓	↓, ↓	1
5	↓	↑, ↓	8
6	↓	↑, ↑	3

Simulations of non-equilibrium processes: kinetic MC

- While being aware of all processes possible at an instant of time, we need a way of (randomly) selecting one process with the **appropriate** relative probability.
- An internal **clock** keeps track of the advancement of **physical** time.
 - If the processes are clearly separated in time, i.e. processes are uncorrelated on the time scale *during which* the processes takes place, the waiting time for each individual process has Poissonian distribution. (K. A. Fichthorn and W.H. Weinberg, J. Chem. Phys. 95, 1090 (1991))
- We need to update the list of **all** possible processes according to the new situation after the move.

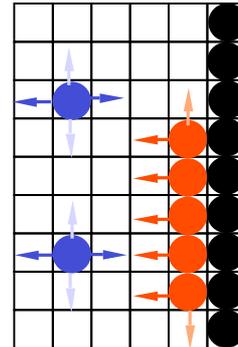
Specific algorithms:

- process-type list algorithm
- binary-tree algorithm
- time-ordered-list algorithm

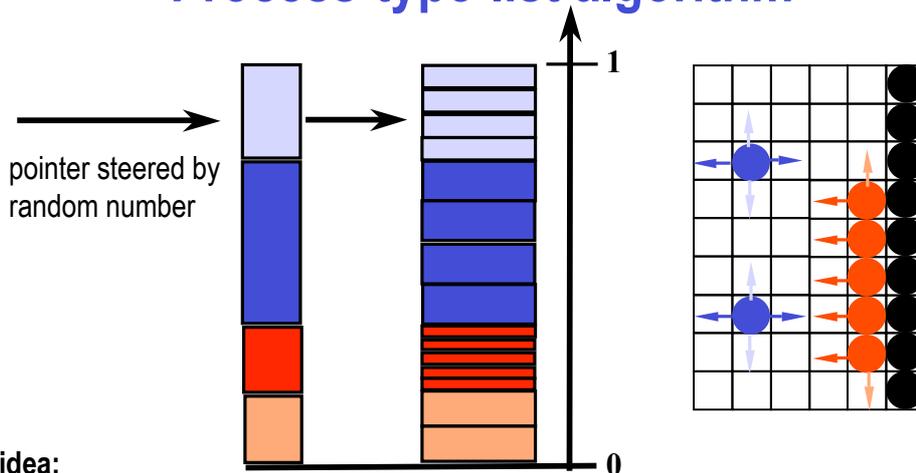
Application to a lattice-gas model

- example: lattice $L_x \times L_y$
- ~~fool's algorithm: first select one particle, then select one move of that particle~~
- **the correct solution:** cumulated partial rates

$$r_k = \sum_{i=1}^k \Gamma_i$$
, normalized to the total rate $R = r_N$
- **selection process:** draw a random number ρ and compare it to all the r_k/R sequentially; as soon as ρ exceeds r_k/R , execute process k
- **problem:** we need to compare ρ to many (in the worst case all) of the r_k/R
- **note:** Selecting a process with the right probability requires that we **can enumerate** all N processes.



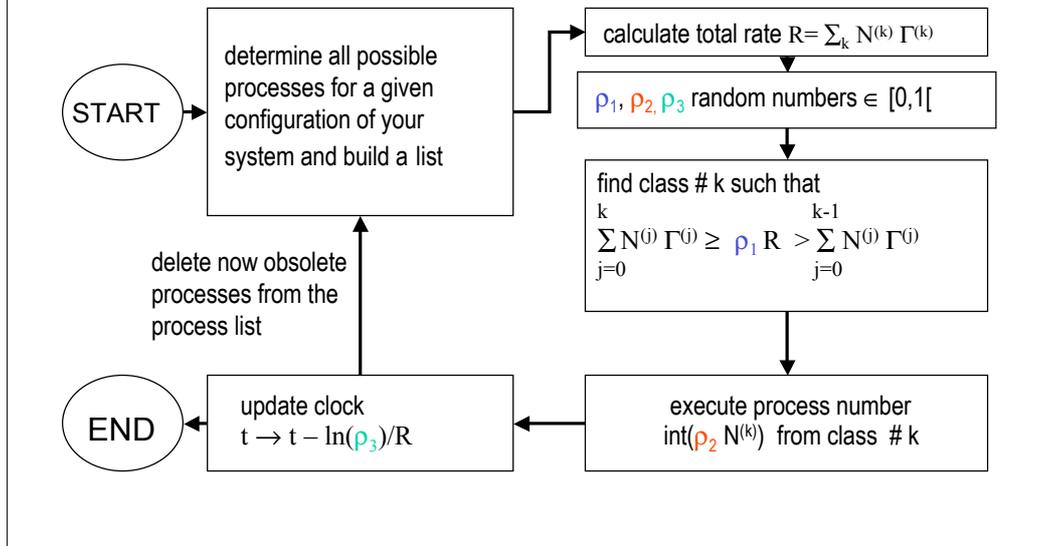
Process-type-list algorithm



idea:

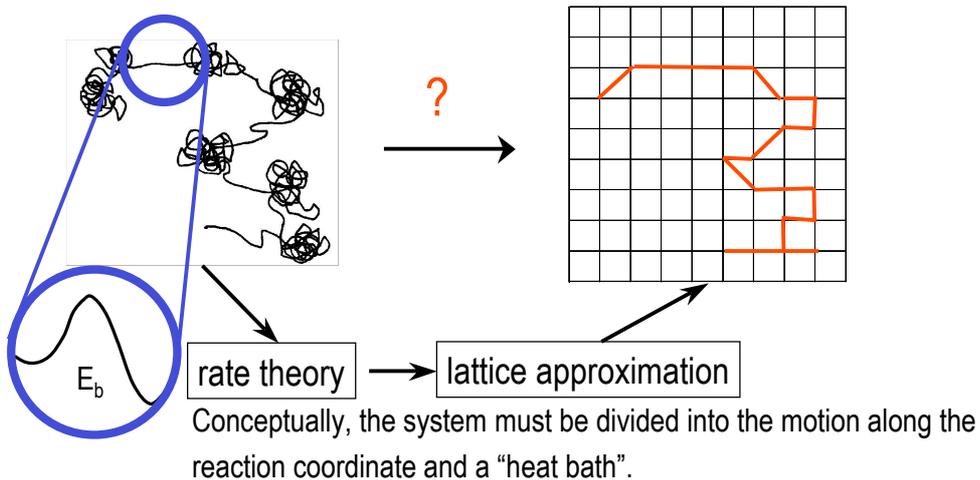
for p process types, we need to compare only to the p numbers $N^{(k)} \Gamma^{(k)}$, $k=1,p$, rather than to **all** r_k/R (which are much more numerous)

flow chart for a kMC algorithm



From molecular dynamics to kinetic Monte Carlo

From molecular dynamics to kinetic Monte Carlo



Transition State Theory (1-dim)

- Kramer's rate theory

$$\Gamma = \frac{\lambda}{\omega_b} \left(\frac{\omega_0}{2\pi} \exp\left(-\frac{E_b}{kT}\right) \right) \quad \lambda = \left(\frac{\gamma^2}{4} + \omega_b^2 \right)^{1/2} - \frac{\gamma}{2}$$

γ : friction due to coupling to the heat bath

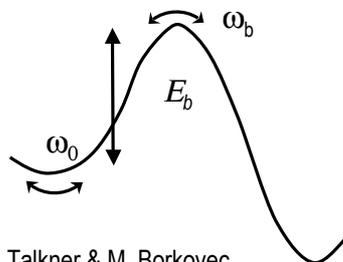
- high-friction limit

$$\Gamma = \frac{\omega_0 \omega_b}{2\pi\gamma} \exp\left(-\frac{E_b}{kT}\right)$$

- 'medium' friction \rightarrow transition state theory

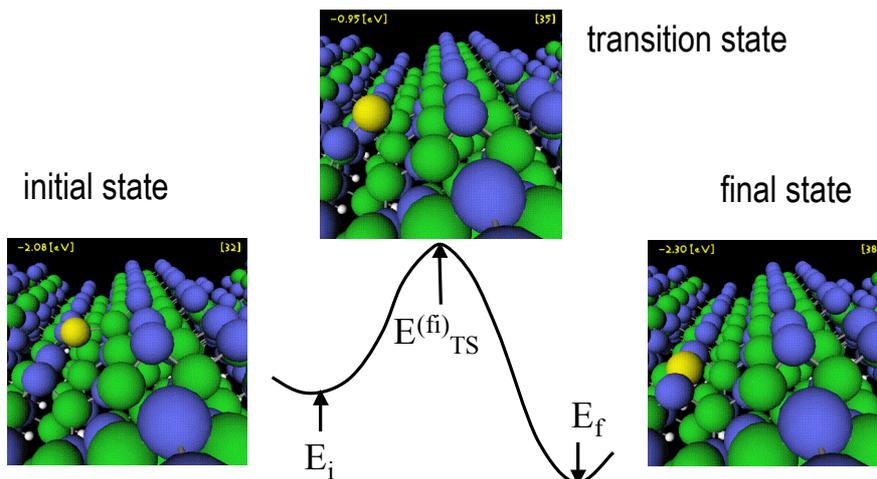
$$\Gamma = \frac{\omega_0}{2\pi} \exp\left(-\frac{E_b}{kT}\right)$$

P. Hänggi, P. Talkner & M. Borkovec,
Rev. Mod. Phys. **62**, 251 (1990)

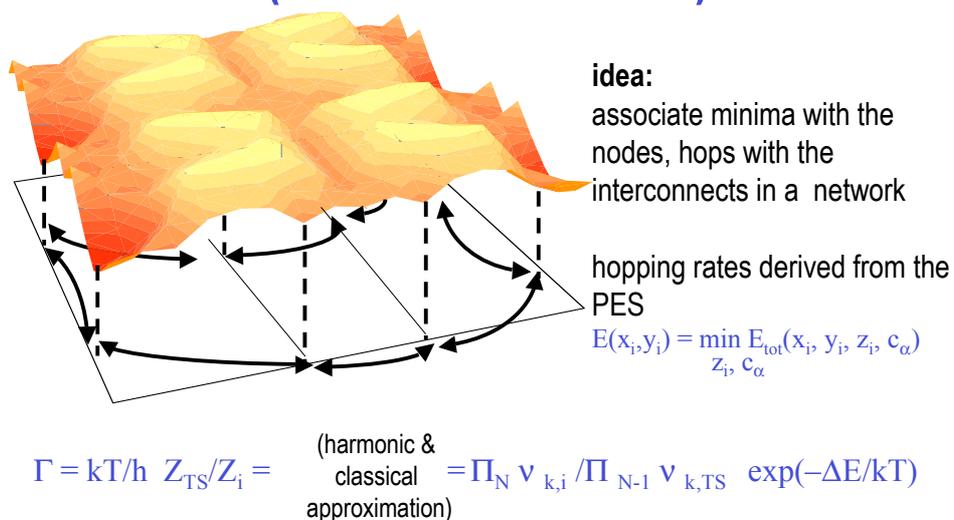


Rates from first-principles calculations

$$\Gamma^{(k)} = W(f,i) = \Gamma^{(fi)}_0 \exp(- (E^{(fi)}_{TS} - E_i)/kT)$$



From the PES to rate constants Γ (multi-dimensional)

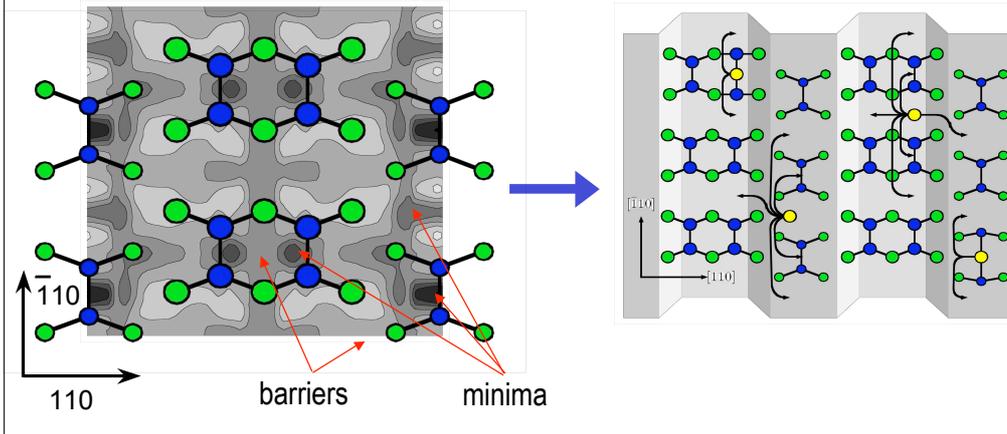


Surface diffusion on GaAs(001): mapping of PES to network graph

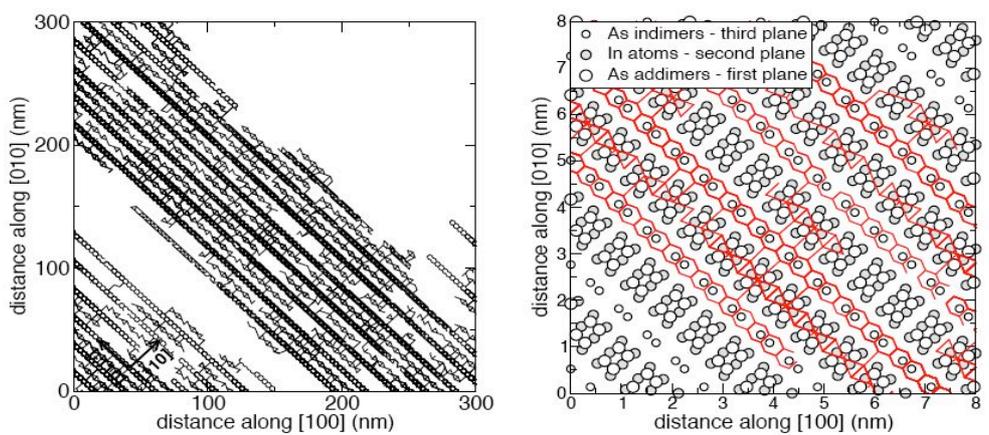
PES from DFT calculations



network of hops



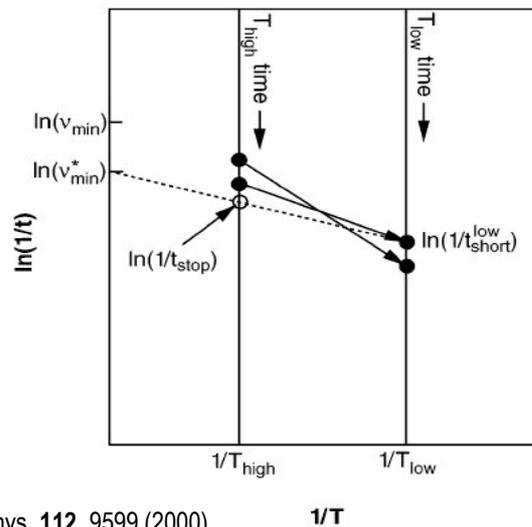
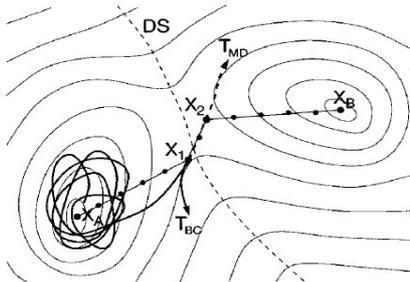
Surface diffusion in lattice-kMC



M. Rosini, M.C. Righi, R. Magri, P. Kratzer, Phys. Rev.B **79**, 075302 (2009)

Temperature-accelerated dynamics (TAD)

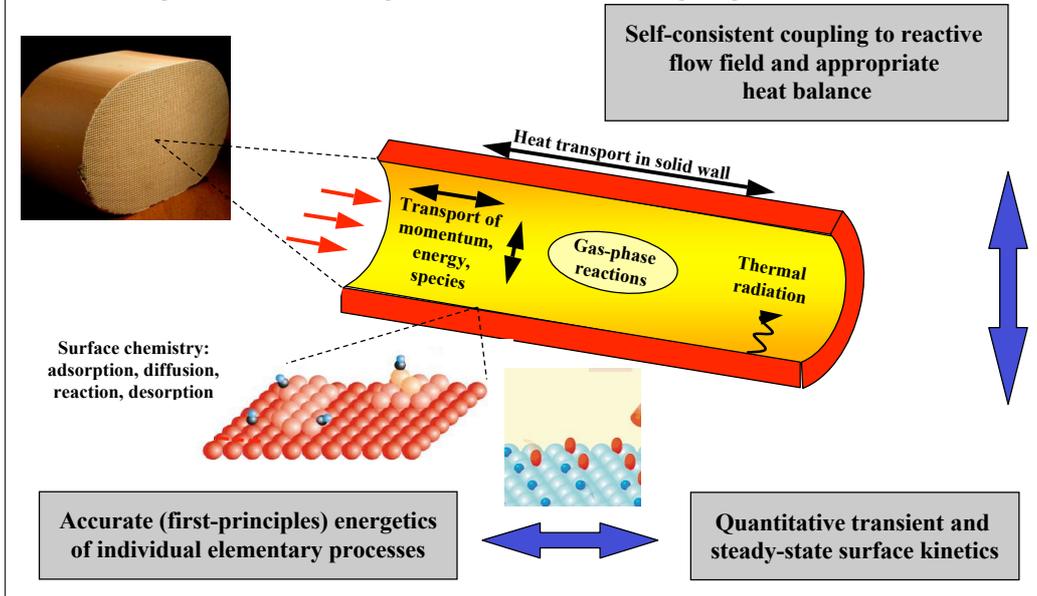
Event is observed at T_{high} , but its rate is extrapolated to T_{low} (using the TST rate law).



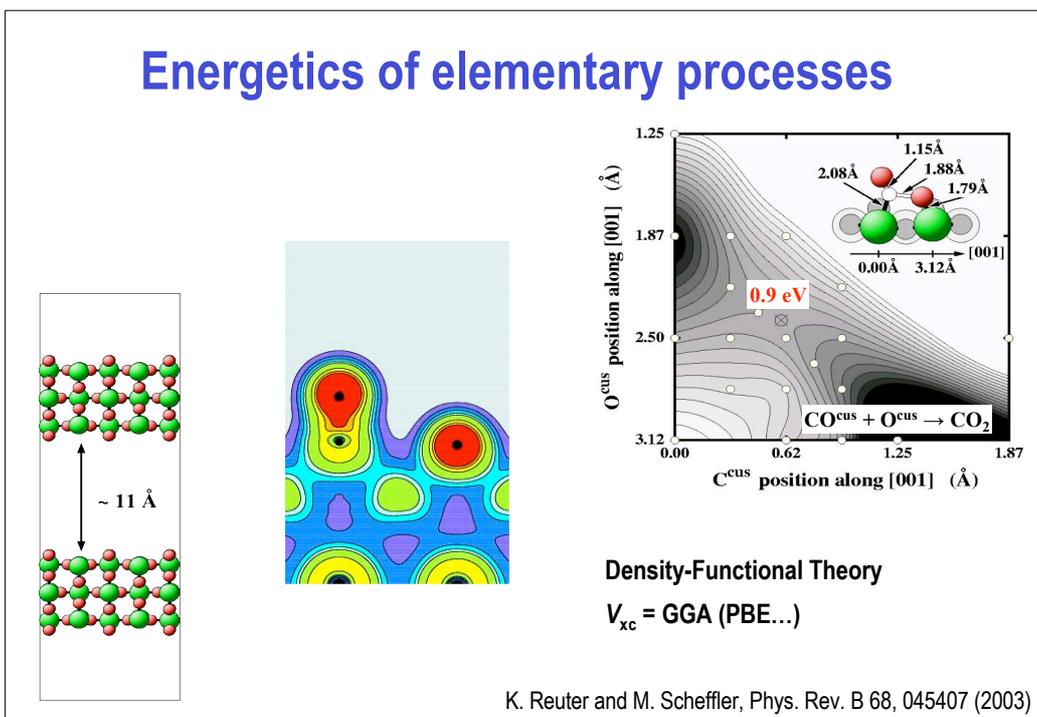
M.R. Sørensen and A.F. Voter, J. Chem. Phys. **112**, 9599 (2000)

Applications to real-world problems

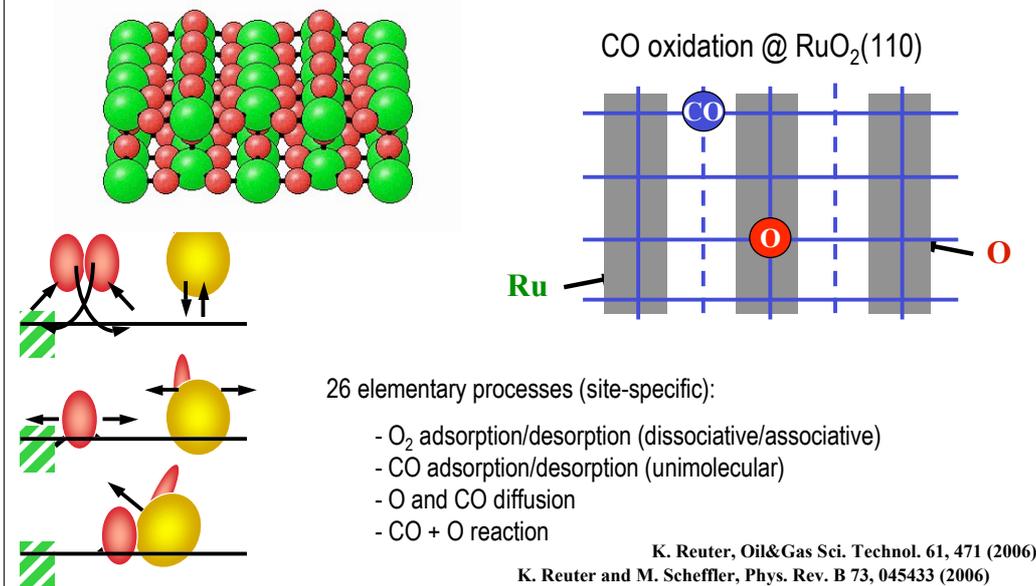
Catalysis: Challenges for scale-bridging simulations



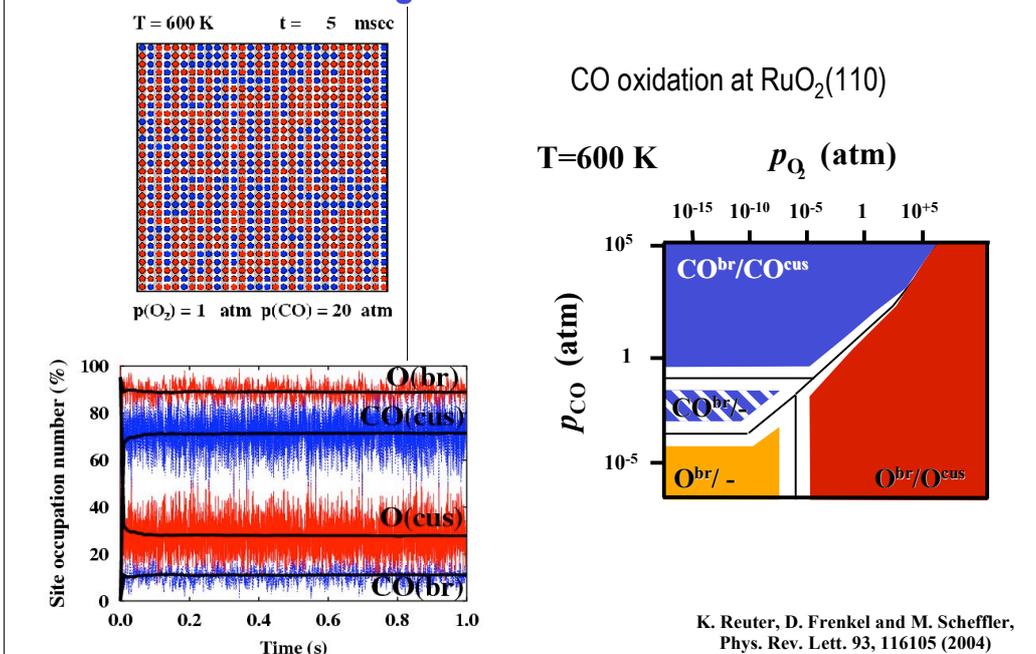
Energetics of elementary processes



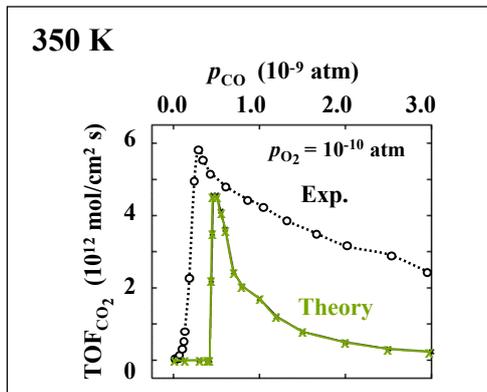
Building a first-principles kinetic Monte Carlo model



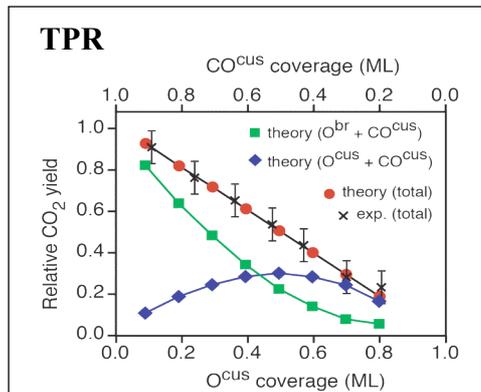
Surface coverage in the reactive environment



Steady state, and parameter-free transient turnover frequencies

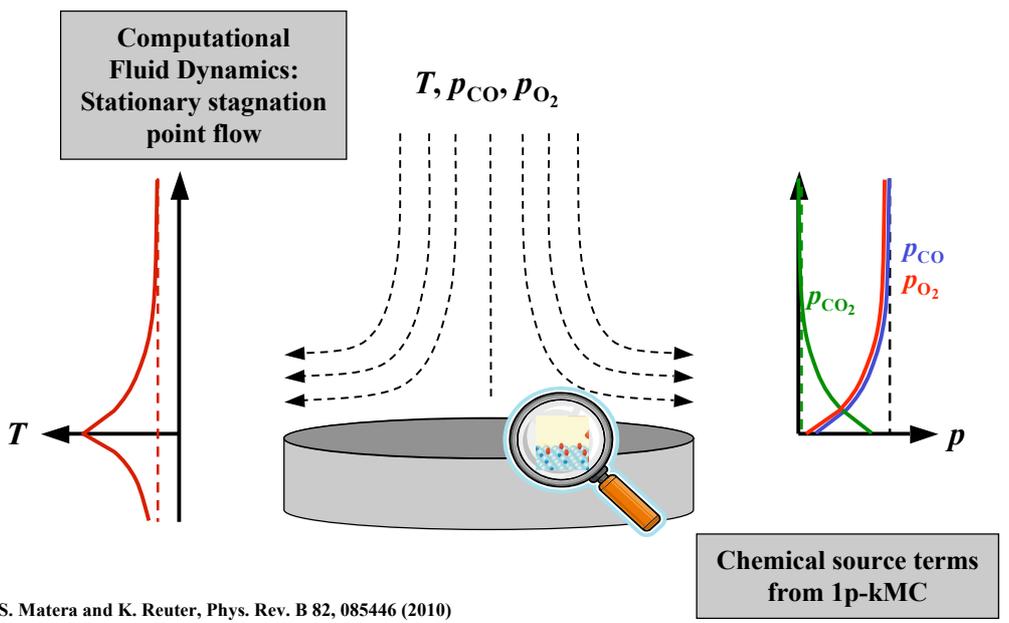


K. Reuter and M. Scheffler,
Phys. Rev. B 73, 045433 (2006)

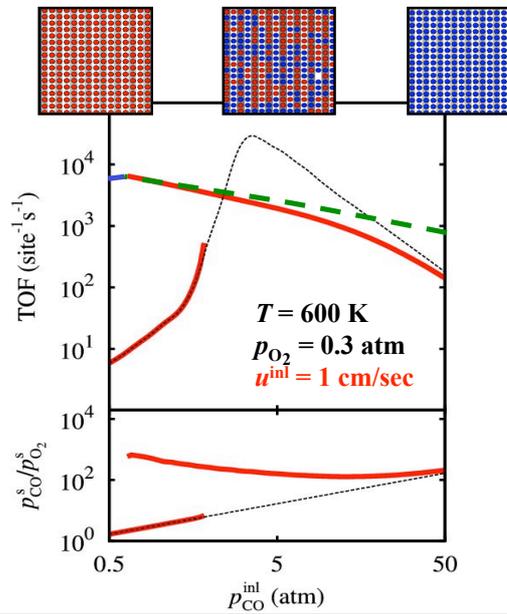
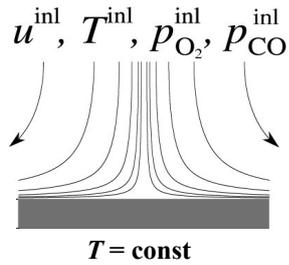


M. Rieger, J. Rogal, and K. Reuter,
Phys. Rev. Lett. 100, 016105 (2008)

Macro-scale: Heat and mass transfer

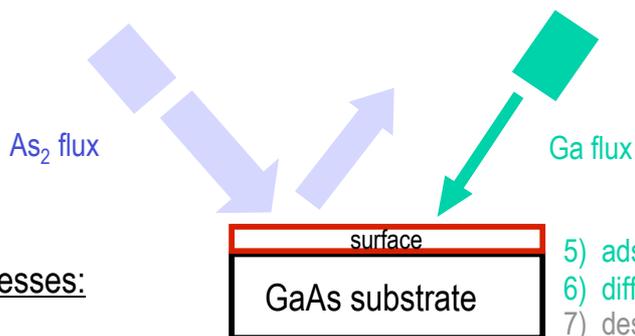


Isothermal limit: Mass transfer limitations



S. Matera and K. Reuter, Catal. Lett. 133, 156 (2009)

Example II: MBE growth of III-V semiconductors



Processes:

- 1) adsorption of As₂
- 2) dissociation of As₂
- 3) diffusion of As
- 4) desorption of As₂

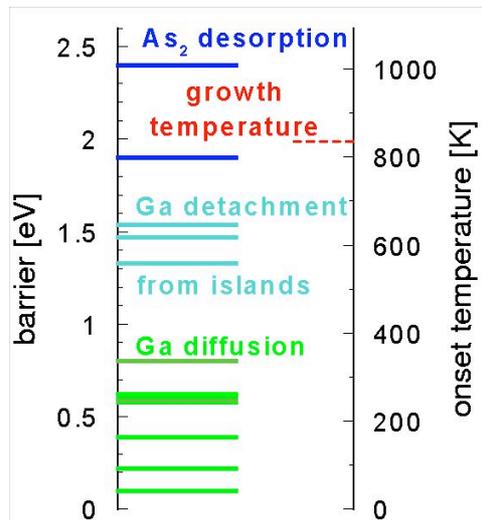
- 5) adsorption of Ga
- 6) diffusion of Ga
- 7) desorption of Ga

- 8) island nucleation
- 9) growth

What is the interplay of these processes for a given temperature and flux ?

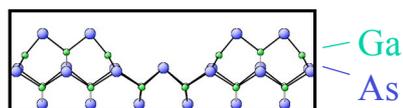
Kinetic Monte Carlo simulations of GaAs epitaxy

- 32 microscopically different Ga diffusion processes, and As_2 adsorption/desorption are included explicitly
- computational challenge: widely different time scales (10^{-12} sec to 10 sec)
- simulation cell
160 x 320 sites
(64 nm x 128 nm)

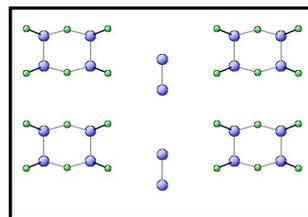


kinetics of island nucleation and growth

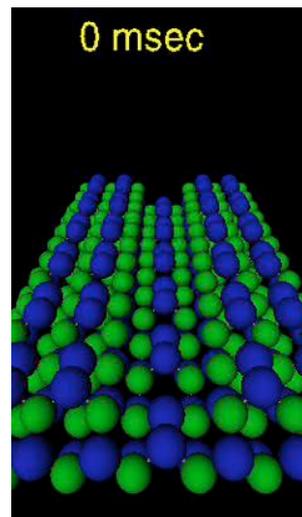
side view



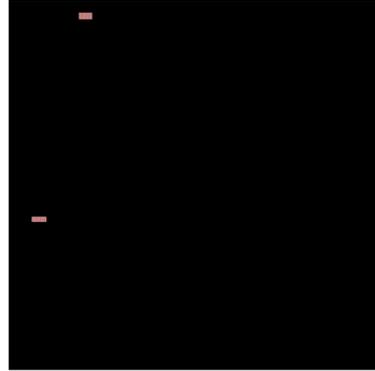
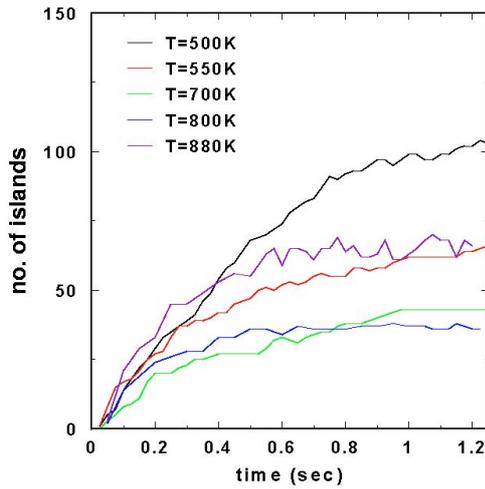
top view



1/60 of the full simulation cell
 As_2 pressure = 0.85×10^{-8} bar
 Ga deposition rate = 0.1 ML/s
 $T = 700$ K

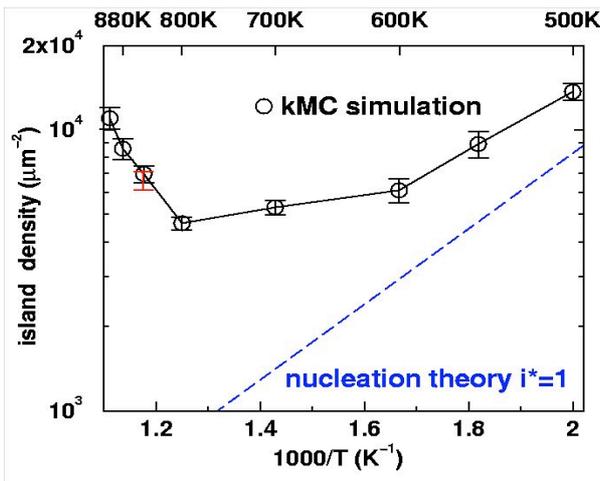


island density



deposition rate
0.1 ML Ga per second, III/V
ratio 1:1000, T=700K

scaling with temperature ?



'conventional'
nucleation theory

$$N_{is} = \eta (R/D)^{i^*/(i^*+2)}$$

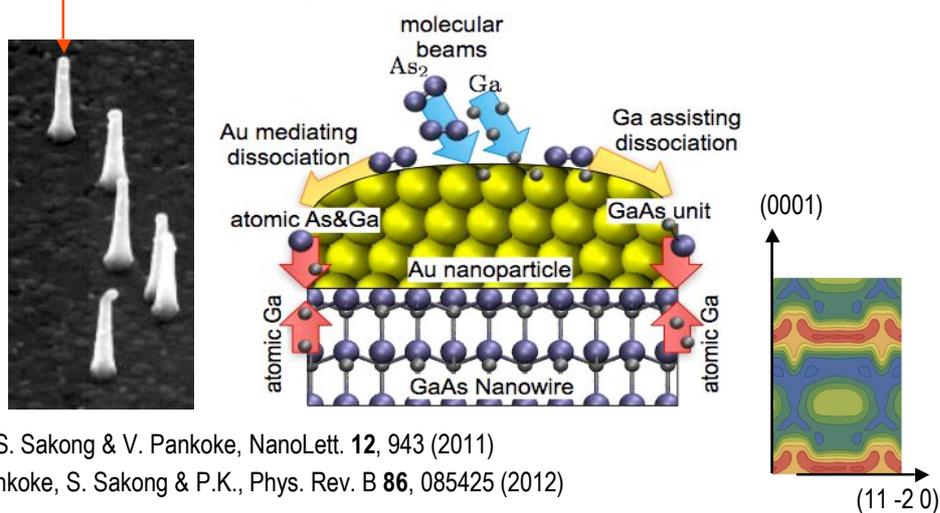
N_{is} island density
 D diffusion constant
 R deposition flux
 η numerical const.
 i^* critical nucleus

simulation: P. Kratzer and M. Scheffler,
Phys. Rev. Lett. **88**, 036102 (2002)

experiment: G.R. Bell et al.,
Surf. Sci. **423**, L280 (1999)

Current work: GaAs nanowire growth

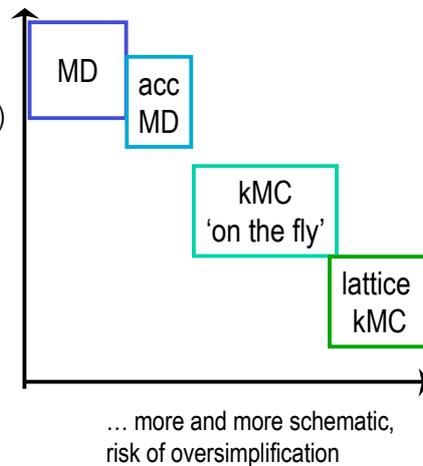
- unidirectional growth of hexagonal GaAs along (0001) direction
- Au droplet acts as catalyst

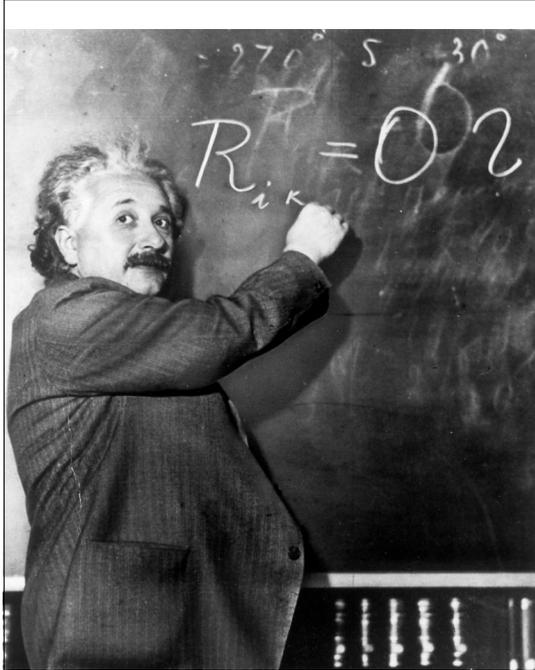


Summary: Bridging the time-scale gap

- molecular dynamics (Car-Parrinello method)
- accelerated molecular dynamics
 - using a boost potential (Voter, Fichthorn,...)
 - temperature-accelerated MD (Montalenti et al. PRL **87**, 126101 (2001))
- kinetic Monte Carlo with transition state search on the fly (avoids both lattice approximation and pre-defined rate table)
- lattice kinetic Monte Carlo, N-fold way (Voter PRB **34**, 6819 (1986))

computational effort





"Keep things as simple as possible, but not more simple .."

**Thank you for your
attention !**

Summary: [arXiv:0904.2556](https://arxiv.org/abs/0904.2556)